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Quasiparticle energies and optical excitations of 3C-SiC divacancy from math xmlns="http://www.w3.org/1998/Math/MathML">mrow>mi >G/mi>mi>W/mi>/mrow>/math> and math xmlns="http://www.w3.org/1998/Math/MathML">mrow>mi >G/mi>mi>W/mi>/mrow>/math> plus Bethe-Salpeter equation calculations Weiwei Gao, Felipe H. da Jornada, Mauro Del Ben, Jack Deslippe, Steven G. Louie, and James R. Chelikowsky

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Quasiparticle energies and optical excitations of 3C-SiC divacancy from GW and GW plus Bethe-Salpeter equation calculations

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Excitons localized around point defects in semiconductors are promising candidates for longlived and photon-addressable qubits. However, their microscopic origin is difficult to characterize due to the computational complexity of studying large systems with defects. Here we study the quasiparticle and optical absorption spectrum of the divacancy defect in 3C-SiC, a prototypical defect for quantum information applications, by means of large-scale GW and GW plus Bethe-Salpeter equation calculations. Despite the presence of localized unoccupied quasiparticle states in the gap, we find that the low-energy excitonic states are made primarily of transitions from occupied defect states to continuum conduction states from SiC, especially from the X point of the Brillouin zone. The mixed character of defect states and bulk states of these low-energy exciton states is in contrast with the NV⁻ center in diamond and the divacancy in 4H-SiC, where the deep defect levels are well separated from bulk states. Our calculations provide a quantitative prediction of the defect quasiparticle energy levels and a physical understanding of the zero-phonon absorption. They highlight the important role of frontier conduction bands in the optical properties and formation of low-energy excitons in 3C-SiC divacancy.

I. INTRODUCTION

Optically accessible deep-level defects in solids have the potential to function as quantum bits for quantum computing or quantum sensing applications^{1,2}. Among numerous point defects, the negatively charged nitrogenvacancy (NV⁻) center is one of the most studied systems, demonstrating many key properties, including long spin coherence time at room temperature and a robust spinphoton interface 3,4 . However, the high cost of processing diamond, integrating it with established semiconducting fabrication practices, and scaling up the number of coherently coupled NV⁻ centers still pose huge challenges for practical applications⁴. Recently, research efforts are shifting to defects in alternative semiconductors, such as divacancies in SiC polymorphs^{5–8}, silicon-vacancy centers in diamond $^{9-12}$, and rare-earth ions in solids $^{13-15}$ and so on^{16-18} . Among them, experiments show that divacancies of 3C-SiC and 4H-SiC fulfill a few key requirements, such as single-defect addressability and high-fidelity spin-tophoton interfaces, which are critical for physical implementation of qubits⁶. Notably, the divacancy in 3C-SiC possesses a record-long Hahn-echo spin coherence time of one millisecond⁶. In addition to these advantages, SiC is widely used in industry and can be fabricated in high quality with lower costs than diamond¹⁹.

The ground-state properties, such as formation energies, hyperfine coupling tensors and zero field splittings, of SiC divacancies have been thoroughly studied by experiment and faithfully explained by theoretical calculations based on density functional theory (DFT).^{20–23} On the other hand, accurate theoretical calculations of excited-states properties, including the quasiparticle energies of defect states and optical excitations, are important for a quantitative understanding of the interactions between the defects and photon. As for divacancies in common SiC polymorphs, the 4H phase is the most studied one by both experiments and calculations. Excited-states properties of 4H-SiC divacancies, such as zero-phonon emission lines and the single-particle energies of deep defect states, were calculated using hybrid functionals^{18,20,22}. Furthermore, the energies of manybody excited states with different spin multiplicities of 4H-SiC divacancy were calculated with active-space configuration interaction calculations^{24,25}. However, despite many recent experimental efforts on exploiting the optical properties of 3C-SiC divacancies for quantum photonic applications²⁶⁻²⁸, accurate and thorough computational studies on the excited-states properties of 3C-SiC divacancy are still lacking, e.g., no conclusive computations for quasiparticle energies of defect states and the characters low-energy optical transitions exist.

In this context, DFT in principle can yield accurate ground state energy and density, but it cannot accurately vield excited-states properties. Recently, several methods designed specifically for excited-states properties, such as active-space configuration interaction and the GW approximation, have been applied to study defects^{17,18,24,25,29–32}. Among them, GW approximation and GW plus Bethe-Salpeter equations (GW-BSE), which are based on wellestablished many-body perturbation theories, can give accurate descriptions for the single-particle energy levels and spin-conserving excitations^{33,34}. Unfortunately, the cubic to quartic scaling and non-trivial convergence behavior of typical GW implementation $^{35-38}$, as well as the N^6 -scaling of BSE hinder wide applications of these methods to defect systems.^{39,40} Moreover, in practical calculations, defects are typically modeled with a finitesize nanocluster or a periodic supercell. To minimize artifacts from periodic images, large supercells containing hundreds to thousands of atoms are required, posing further difficulties for GW and GW-BSE calculations. Due to these reasons. GW and GW-BSE calculations are not routinely carried out for defects in solids, while DFT calculations based on generalized-gradient approximation (GGA) or hybrid functionals, which has relatively low computational costs, are often used for the calculation of defect levels and zero phonon optical transition lines neglecting electron-hole transitions. ^{3,16,18,20,22,41–44}.

Here we study quasiparticle energies and spin-conserved optical absorption spectra of the divacancy defect in 3C-SiC with large-scale GW and GW-BSE calculations. In the study of the optical properties, we treat the system as two spin fluids, and neglect all spin multiplet effects for both the ground and excited states. Our calculations explicitly evaluate the full frequency dependence of the dielectric matrix without resorting to plasmon-pole approximations⁴⁵. Different from the 4H-SiC divacancy and diamond NV⁻ center, our calculations reveal that unoccupied defect quasiparticle states in 3C-SiC are relatively shallow. Our GW-BSE calculations yield accurate zerophonon line energy within the two-spin-fluid formalism and show the lowest-energy exciton states involve transitions between localized defect and bulk continuum states, while these low-energy excitons are still fairly localized near the divacancy.

II. COMPUTATION DETAILS

We use the Perdew-Burke-Ernzerhof (PBE) exchangecorrelation functional⁴⁶ for collinear spin density functional theory (DFT) calculations. The structures are optimized with the real-space DFT code PARSEC^{47,48} using real-space grid of 0.285 bohr spacing. The optimized lattice constant of bulk 3C-SiC is 4.379 Å, agreeing well with the experimental value⁴⁹. The divacancy is modelled with a supercell. Here we keep lattice constant as that of bulk SiC and relax atomic coordinates until residue forces on each atom are less than 1×10^{-4} Ryd/Bohr. By



Figure 1. Convergence of single-particle transition energies calculated using (a) DFT and (b) G_0W_0 with supercell sizes. The shadow areas highlight the energy ranges of 40 meV around the data calculated with the 998-atom supercell. The labels of defect states are explained in section III.

testing the convergence of the DFT results with respect the supercell size, we find the energy separation between the conduction band minimum (CBM) and spin up state with the e symmetry converge within 50 meV with a 510-atom supercell and 2 meV with a 998-atom supercell, as shown in Figure 1 (a).

GW and GW-BSE calculations are carried out with the BerkelevGW package^{33,34,50}. The starting points for GW-BSE calculations are the mean-field wave functions calculated with PBE functional 46 and SG15 Optimized Norm-Conserving Vanderbilt pseudopotentials^{51–53} using Quantum Espresso package^{54,55}. The mean-field wavefunctions are calculated with an 50 Ryd cut-off and sampled at Γ point. We perform "one-shot" GW calculations $(G_0W_0 \text{ approximation})$ at the full-frequency level within the contour deformation formalism 56,57. We also tested the convergence of G_0W_0 calculations with respect to supercell sizes. Figure 1 (b) shows that the G_0W_0 results calculated with 510-atom and 998-atom supercells only differ by up to 20 meV, which suggests a reasonably good convergence. In the following discussions, the GW calculation results for the divacancy are performed with 998-atom supercell, unless otherwise mentioned. The frequency dependent parts of dielectric functions are obtained within the static-subspace approximation $^{45,58-60}$. The staticsubspace approximation method is a low-rank approximation method for calculating the frequency-dependent dielectric matrix. In brief, we first calculate the eigenvalues and eigenvectors of the static dielectric matrix.



Figure 2. (a) Quasiparticle band structure of pristine 3C-SiC computed within DFT using Perdew-Burke-Ernzerhof (PBE) functional and within the G_0W_0 approximation. (b) Comparison between the imaginary part of dielectric function calculated with GW-BSE and with non-interacting particle approximation. A Gaussian smearing of 0.05 eV is used for calculating dielectric function.

Then we pick the eigenvectors with large eigenvalues as the basis set functions to represent dielectric matrices of finite frequencies. This method has been discussed in detail in reference⁴⁵. To converge the G_0W_0 calculation with a 998-atom supercell, we employed up to 18×10^3 eigenvectors for the static-subspace basis expansion. For frequency integration, 16 imaginary frequencies and 34 frequencies on the real axis up to a maximum of 8 eV. With the sum-over-bands GW implementation used in our work, the equations of irreducible polarizability and self-energy include a summation over all empty states. To reduce the cost for summing over many empty states, we employ the stochastic pseudobands method. With this method, one combines several high-energy orbitals into a single unnormalized states to represent high-energy empty states 45,61 . The stochastic pseudobands allow us to effectively include the whole Hilbert space for the evaluation of the polarizability and self energy. A similar idea was also reported in reference⁶². For the GW-BSE calculation of 998-atoms supercell, we consider the Γ point and the transitions between 40 conduction bands and 40 valence bands for each spin channel. Our tests show that these number of bands can converge the absorption spectrum up to $\omega = 2.6$ eV. For pristine bulk 3C-SiC, we considered 3 conduction bands, 3 valence bands, and a $24\times24\times24$ q-points grid for optical absorption spectrum calculations.

III. RESULTS AND DISCUSSION

Before studying the divacancy, we briefly discuss the electronic structure of bulk 3C-SiC. As shown in Figure 2 (a), pristine 3C-SiC has an indirect band gap between valence band maximum (VBM) at Γ and CBM at X. There are three equivalent X point on the edge of the Brillouin Zone (BZ), so the CBM is triply degenerate. Our DFT-PBE calculation predicts an indirect band gap of 1.4 eV. With the GW approximation at the G_0W_0 level, the quasiparticle self-energy effect opens up the band gap



Figure 3. (a) The structure of divacancy in 3C-SiC⁶⁴ (b) Top view of a divacancy. The isosurface of the net spin charge density $|\rho_{\uparrow}(\mathbf{r}) - \rho_{\downarrow}(\mathbf{r})|$ is shown in yellow.

to 2.3 eV, which agrees well with the experimental value of 2.36 eV⁶³. In Figure 2 (b), we compare the optical absorption spectrum of 3C-SiC calculated with GW-BSE and a non-interacting particle approximation. Within the non-interacting particle approximation, the lowest single-particle transition, which corresponds to the direct transition at X, has an GW energy of 5.64 eV. In comparison, the energy of the lowest optically active exciton calculated with GW-BSE is 5.45 eV, which corresponds to an exciton binding energy of 0.19 eV.

In Figure 3, we show the structure of a divacancy $V_C V_{Si}$ in 3C-SiC, where the carbon vacancy V_C and silicon vacancy V_{Si} are placed along the z axis (z axis is along the (111) direction). Similarly to the diamond NV⁻ center, a divacancy in 3C-SiC has the C_{3v} point group symmetry, and the divacancy in 3C-SiC induces sizable distortions to the positions of nearby atoms. In particular, the nearest neighboring carbon atoms shift away from the silicon vacancy V_{Si} , making the distance d_{C-C} between nearest carbon atoms 3.298 Å, larger than the bulk value of 3.096 Å. The positions of the nearest neighboring silicon atoms show smaller distortions than carbon atoms. The relaxed distance d_{Si-Si} between nearest neighboring Si is 3.092 Å, close to the bulk value. DFT calculations within GGA using PBE functional show a spin-polarized ground state with a magnetic moment of $2\mu_{\rm B}$ per each divacancy. We show the isosurface of the net spin charge density $|\rho_{\uparrow}(\mathbf{r}) - \rho_{\downarrow}(\mathbf{r})|$ in Figure 3(b), which also demonstrates the C_{3v} symmetry of the system. The spin density is mostly localized at the neighboring carbon atoms and has the symmetry of p orbitals.

Next, we investigate the electronic structure of 3C-SiC divacancy in terms of the single-quasiparticle energy levels. A divacancy $V_{Si}V_C$ in 3C-SiC creates six dangling bonds, which form 6 defect states per each spin channel within this limited Hilbert space. These localized defect states are classified into either the non-degenerate a_1 states or doubly-degenerate e states according to their symmetry properties. To identify all the localized states from our full calculation, we evaluate the inverse participation ratio (IPR), defined as IPR = $\int |\phi(\mathbf{r})|^4 d^3r$, for states $\phi(\mathbf{r})$; a larger IPR is an indication that ϕ is more localized in

real space. As shown in Figure 4 (a), we plot the IPRs of states close to the band gap from a DFT-GGA calculation using a supercell of 998 atoms and a Γ -point sampling. The states inside the shaded region have IPRs ranging from $\sim 10^{-5}$ to 8×10^{-5} and are recognized as localized defect states. We identify one a_1 state and two pairs of e states, which are labelled as $1e^{\sigma} = \{1e_x^{\sigma}, 1e_y^{\sigma}\}$ and $2e^{\sigma}=\{2e^{\sigma}_x,2e^{\sigma}_y\}$ according to their energy ordering, for two spin channels $\sigma = \downarrow, \uparrow$. Two more a_1 states are located out of the plotting range and mixed with valence bands. Conforming with previous group-theory analysis⁶⁵, our results show that e states have higher energies than the a_1 state. Among all localized defect states identified, the spin-down a_1^{\downarrow} state and the spin-up $1e^{\uparrow}$ states are located deep inside the band gap, while the others are located either below the VBM or above the CBM as resonances.

In addition to these localized defect states, we also find a pair of spin-down states, denoted as X_1^{\downarrow} and X_2^{\downarrow} , display large IPRs that are one order-of-magnitude larger than typical extended bulk states, as shown in Figure 4 (a). X_1 and X_2 are close to CBM and show strong hybridization between spin-down $1e^{\downarrow}$ and conduction bands. We plot the isosurface of the charge density of selected defect states in Figure 4 (b). One can clearly see that the hybridization makes the $1e^{\downarrow}$ states more delocalized in real space than a_1^{\downarrow} and $1e^{\uparrow}$, which are inside the band gap and have minor hybridization with bulk states.

We obtain a deeper understanding on the nature of states X_1^{\downarrow} and X_2^{\downarrow} by computing the matrix elements $\langle \phi_1 | R | \phi_2 \rangle = \int_{\Omega_{\text{cell}}} \phi_1^*(\mathbf{r}) R \phi_2(\mathbf{r}) d^3 \mathbf{r}$, where $R = |\mathbf{r}|$ is the distance to the center of a divacancy and Ω_{cell} is the realspace domain of a supercell. We find the expectation value $\langle \phi | R | \phi \rangle$ is an intuitive measure to distinguish localized and extended states. Related measures were also $used^{66,67}$ in the literature. A larger expectation value $\langle \phi | R | \phi \rangle$ suggests that $\phi(\mathbf{r})$ is more extended in real space. We find $\langle X_1^{\downarrow} | R | X_1^{\downarrow} \rangle = \langle X_2^{\downarrow} | R | X_2^{\downarrow} \rangle \approx 14$ Bohr, which is quite close to $\langle 1e|R|1e \rangle \approx 12$ Bohr. For comparison, the expectation value of R for typical bulk states range from 18 to 22 Bohr, which is dictated by the size of the supercell used. When then diagonalize the matrix $\langle \phi_1 | R | \phi_2 \rangle$ including four states that are nearly degenerate within our DFT calculations (i.e., the manifold of $X_{1,2}^{\downarrow}$ and the two spindown $1e^{\downarrow}$ states), we find two pairs of doubly degenerate eigenstates. One pair of degenerate eigenstates has a small expectation value of $\langle R \rangle = 6.5$ Bohr and, apart from the stronger localization, are very similar to the two $1e^{\downarrow}$ states. The remaining degenerate eigenstates have a large expectation values of $\langle R \rangle = 19.6$ Bohr and their wavefunctions resemble that of the CBM at the Xpoint of bulk SiC. This strongly suggests that the two $X_{1,2}^{\downarrow}$ and two $1e^{\downarrow}$ states are hybrid states made mostly out of extended CBM states and localized defect orbitals, respectively. These states are parts of the defect resonance from two otherwise localized states if they were in the gap; and there would be more X-like states (in the same energy range) as the number of k-points are increase in



Figure 4. (a) The inverse participation ratio for states close to the Fermi energy; (b) The isosurface of $sgn(\phi(\mathbf{r}))|\phi(\mathbf{r})|^2$ of a_1^{σ} and $1e^{\sigma} = \{1e_x^{\sigma}, 1e_y^{\sigma}\}$ states for two spin channels. Each isosurface includes 50% of the charge density contributed by the corresponding orbital.

the sampling in the supercell calculation.

In Figure 5 (a)-(c), we compare the energy levels near the band edge of 4H-SiC divacancy, diamond NV⁻ center, and 3C-SiC divacancy, calculated within DFT with the PBE functional. Clearly, the defect states associated with low-energy and optically active transitions, such as the a_1^{\downarrow} and e^{\downarrow} of 4H-SiC divacancy and diamond NV⁻ center, are separated from bulk states, and the lowest-energy single-particle transition is $(a_1^{\downarrow} \rightarrow 1e^{\downarrow})$. In contrast to 4H-SiC and diamond, 3C-SiC has a much smaller band gap, so it is not surprising to see hybridization between certain localized defect states and bulk states, as discussed before. As shown in Figure 5 (c), the DFT-PBE states $1e^{\downarrow}$ are above the CBM and the lowest-energy band-toband transition of the 3C-SiC divacancy is $(1e_1^{\uparrow} \rightarrow \text{CBM}^{\uparrow})$. Compared to DFT-PBE results, the G_0W_0 approximation increases the separations between deep defect states and bulk states. In particular, states in the middle of band gap, namely a_1^{\downarrow} and $1e^{\uparrow}$, are further separated from the VBM. Notably, G_0W_0 self-energy effects pushes $1e^{\downarrow}$ lower than the conduction bands.

We compute the imaginary part of dielectric function $\epsilon_2(\omega)$ using the independent-particle approximation (with the G_0W_0 state energies) and GW-BSE approach, as shown in Figure 6 (a) and (b). Full relativistic effects and spin-flipping optical transitions are not considered in our calculations. To analyze the characters of low-energy optical excitations of the 3C-SiC divacancy within our formalism, we classify optical transitions according to the characters of the electron and hole states involved in a transition, as shown in Figure 6 (c). We define a subspace D of defect states: $D = \{a_1^{\sigma}, X_1^{\downarrow}, X_2^{\downarrow}, 1e^{\sigma}, 2e^{\sigma}\},$ where $\sigma = \uparrow, \downarrow$ and all the other states are recognized as bulk states. Under the non-interacting particle approximation (using G_0W_0 energies), the lowest-energy excitation corresponds to the transition from $1e^{\uparrow}$ to the CBM^{\uparrow} with the energy of 1.55 eV. The lowest-energy defect-to-defect transitions $(a_1^{\downarrow} \to 1e^{\downarrow})$ and $(a_1^{\downarrow} \to X_{1,2}^{\downarrow})$ are located at 1.67 eV and at 1.74 eV, respectively. Next, we perform first-principles GW-BSE calculations and find that excitonic effects lower the absorption edge to 1.06 eV, giving a large exciton binding energy of 0.49 eV. Interestingly, the calculated $\epsilon_2(\omega)$ in the low-energy range becomes smaller with the inclusion of excitonic effects. This observation is in stark contrast to common bulk or 2D materials, where excitonic effects usually enhance the low-energy absorption peaks. The reason behind this contrasting behavior is due to the different nature of the low-energy excitonic states compared to independent-particle transitions. As we discussed later, low-energy excitons have significant contributions from transitions to bulk states of different momenta, as finite-momenta transitions can mix with zeromomentum transitions since translational symmetry in SiC is broken due to the presence of a defect.

We also find the low-energy absorption spectra strongly depend on the polarization direction \mathbf{e} of the incident light. As shown in Figure 6 (a), the single-particle transitions $(a_1^{\downarrow} \rightarrow 1e^{\downarrow})$ and $(a_1^{\downarrow} \rightarrow X_{1,2}^{\downarrow})$ show strong absorption of transversely polarized light ($\mathbf{e} = [1, 0, 0]$), but become dark for longitudinally polarized light ($\mathbf{e} = [0, 0, 1]$). The linear dichroism can be qualitatively explained by dipole selection rules. Within the independent-particle approximation, the optical transition matrix element of a transition between a valence state v and a conduction state c is proportional to $|\mathbf{e} \cdot \langle v | \hat{\mathbf{v}} | c \rangle|^2$, where $\hat{\mathbf{v}} = \hat{\mathbf{v}}_x + \hat{\mathbf{v}}_y + \hat{\mathbf{v}}_z$ is the velocity operator. Since $[\hat{\mathbf{v}}_x, \hat{\mathbf{v}}_x]$, $[X_1^{\downarrow}, X_2^{\downarrow}]$, and $1e^{\downarrow} = [1e_x^{\downarrow}, 1e_y^{\downarrow}]$ transforms according to the e representation.

tation and a_1^{\downarrow} is fully symmetric under C_{3v} group, the matrix elements $\langle 1e_x | \hat{\mathbf{v}}_x | a_1 \rangle$ and $\langle 1e_y | \hat{\mathbf{v}}_y | a_1 \rangle$ are generally non-zero, while $\langle 1e_x | \hat{\mathbf{v}}_z | a_1 \rangle$ and $\langle 1e_y | \hat{\mathbf{v}}_z | a_1 \rangle$ vanish according to the dipole-selection rules. Consequently, $(a_1^{\downarrow} \rightarrow 1e^{\downarrow})$ and $(a_1^{\downarrow} \rightarrow X_{1,2}^{\downarrow})$ transitions are bright under transversely polarized light as we observed in independent-particleapproximation calculations. Similarly, when we solve the BSE, we observe that the lowest-energy excitons between 1.0 and 1.3 eV are bright under longitudinal polarization but dark under transverse polarization. The optical oscillator strength is contributed mainly by transitions between defect states.

We analyze our results including electron-hole interactions by writing each excitonic state $|S\rangle$, within the Tamm-Dancoff approximation, as a linear combination of electron-hole pairs

$$|S\rangle = \sum_{v}^{\text{hole elec}} A_{vc}^{S} |v\rangle |c\rangle,$$

where v, c, and S are the index for occupied, unoccupied, and excitonic states, respectively. And A_{vc}^S are the eigenvectors of the BSE and satisfy the normalization relation $\sum_{vc} |A_{vc}^S|^2 = 1$.

The decomposition of excitonic states into different characters are shown in Figure 6 (d), where the length of each colored vertical lines at energy ω is the weight of the corresponding type of single-particle transition of the excitons within an energy range $[\omega - \delta/2, \omega + \delta/2]$, where $\delta = 0.02$ eV. For example, the length of red lines is the weight of defect-to-defect transitions, defined as

$$w_{\text{defect} \to \text{defect}}(\omega) = \frac{\int_{\omega-\delta/2}^{\omega+\delta/2} \sum_{S} \sum_{v,c \in D} \delta(E - \Omega_S) |A_{vc}^S|^2 dE}{\int_{\omega-\delta/2}^{\omega+\delta/2} \sum_{S} \delta(E - \Omega_S) dE}$$

where defect states D are defined before and Ω_S are the energy of excitonic states $|S\rangle$. For low-energy excitons below 1.3 eV, the dominant characters are defect-to-bulk transitions, such as $(1e^{\uparrow} \rightarrow \text{CBM}^{\uparrow})$, mixed with a small amount of defect-to-defect transitions. Some of the excitons between 1.3 to 1.8 eV have large components of defect-to-defect transitions. For excitons with energies ranging from 1.8 to 2.6 eV, transitions involving bulk states are dominant. We note that the bulk-to-bulk transitions contributing to low-energy excitons (less than 2.6 eV) originate from indirect transitions between a valence electron at Γ and an unoccupied state at the X point of the BZ in pristine 3C-SiC. These bulk-to-bulk transitions have negligible contributions to the optical transition matrix elements.

The zero-phonon line of 3C-SiC divacancy corresponds to the excitation from ground state ${}^{3}A_{2}$ to the first spintriplet excited state ${}^{3}E^{6}$. Within the single-particle terminology, this excitation corresponds roughly to promoting an electron from a_{1}^{\downarrow} to $1e^{\downarrow}$. In Table I, we compare the



Figure 5. Single particle energy levels of (a) diamond NV^- (b) 4H-SiC kk divacancy (There are two different lattice sites , namely, the hexagonal and cubic sites in 4H-SiC. And "kk divacancy" means both V_{Si} and V_C are located at cubic sites.) (c) 3C-SiC divacancy calculated with a DFT-PBE approach. (d) Defect energy levels of 3C-SiC divacancy calculated with G_0W_0 approximation. Horizontal lines represent localized defect states and grey shaded areas are extended bulk states.



Figure 6. Imaginary part of frequency-dependent dielectric functions (absorption spectra) of a 3C-SiC divacancy calculated with (a) independent particle approximation with the G_0W_0 energies and (b) GW-BSE approach. The defect is modelled with a 998-atom supercell and a Gaussian smearing of 0.035 eV is used for the calculation of delectric function. Characters of optical excitations calculated with (c) independent particle approximation with the G_0W_0 energies and (d) GW-BSE approach. Note that a vertical line in (c) and (d) may correspond to several optical excitations.

components of the $a_1^{\downarrow} \rightarrow \{X_{1,2}^{\downarrow}, 1e^{\downarrow}\}$ transitions in selected excitons. The excitons at 1.33, 1.43, 1.49 eV have large contributions from $(a_1^{\downarrow} \rightarrow 1e^{\downarrow})$ and $(a_1^{\downarrow} \rightarrow X_{1,2}^{\downarrow})$ transitions. Our analysis also shows that, while the lowest-energy exciton at 1.06 eV only has a small fraction of $a_1^{\downarrow} \rightarrow \{X_{1,2}^{\downarrow}, 1e^{\downarrow}\}$ transitions, its oscillator strength mainly originates from these defect-to-defect transitions. The electron-hole interactions tends to reduce the defectto-defect character of the lowest-energy exciton. Such a reduction of the defect-to-defect characters makes an

exciton optically weaker compared to low-energy defectto-defect transitions computed within the independentparticle picture. We plot the wave function $\Psi_S(\mathbf{r}_e, \mathbf{r}_h)$ of the lowest-energy exciton in Figure 7, with the hole position \mathbf{r}_h kept at the center of a divacancy. Despite the significant contributions from conduction states with extended features, the exciton wave function is still highly localized near the divacancy, showing the symmetry of $1e^{\downarrow}$ states.

We compare the vertical transition energy and zero-

Table I. Excitons with large contributions from single-particle transitions from a_1^{\downarrow} to $\{X_1^{\downarrow}, X_2^{\downarrow}, 1e^{\downarrow}\}$.

Exciton	$ A_{vc}^S ^2$					
energy	$c = X_1^{\downarrow}$	$c = X_2^{\downarrow}$	$c = 1e_x^\downarrow$	$c = 1e_y^{\downarrow}$		
$\Omega_S \ (eV)$	$v = a_1^{\downarrow}$	$v = a_1^{\downarrow}$	$v = a_1^{\downarrow}$	$v = a_1^{\downarrow}$		
1.06	0.02	0.02	0.03	0.07		
1.11	0.07	0.00	0.04	0.00		
1.14	0.00	0.08	0.00	0.02		
1.33	0.13	0.19	0.04	0.08		
1.43	0.21	0.03	0.16	0.00		
1.49	0.03	0.15	0.06	0.15		
1.59	0.02	0.05	0.06	0.06		
1.80	0.09	0.04	0.12	0.04		
1.84	0.02	0.08	0.03	0.12		



Figure 7. The lowest-energy exciton wave function $\Psi_S(\mathbf{r}_e, \mathbf{r}_h)$, where \mathbf{r}_e and \mathbf{r}_h are the electron and hole coordinates, respectively. \mathbf{r}_h is fixed at the center of a divacancy, shown as a red cross here.

phonon line of 3C-SiC divacancy estimated with constrained DFT (cDFT) and first-principles GW-BSE approaches, as shown in Table II. We performed cDFT calculations in two different excited-states configurations. The first cDFT calculation depopulates a_1^{\downarrow} and populates one of $X_{1,2}^{\downarrow}$ states. In this calculation, our result underestimates the zero-phonon line by around 0.17 eV. Incidentally, the first cDFT and GW-BSE calculations yield the same vertical excitation energy. For the second cDFT calculation, we set the excited-states configura-

Table II. Comparisons between calculated excitation properties and experiment.

Properties (eV)	$\begin{vmatrix} \mathrm{cDFT} \\ a_1^\downarrow \to X_{1,2}^\downarrow \end{vmatrix}$	$\begin{array}{c} \mathrm{cDFT} \\ 0\rangle \rightarrow S_0\rangle \end{array}$	GW-BSE	Expt. ⁶
Anti-Stokes Shift	0.095	0.09	-	-
Stokes Shift	0.11	0.05	-	-
Vertical Excitation	1.06	1.17	1.06	-
Zero-phonon line	0.95	1.12	1.01	1.12

tion according to the lowest exciton $|S_0\rangle$ of 1.06 eV, such that the occupation of an originally occupied state $|v\rangle$ is changed to $1.0 - \sum_{c} |A_{vc}^{S_0}|^2$, while that of an unoccupied state $|c\rangle$ is set to $\sum_{v} |A_{vc}^{S_0}|^2$. In the second cDFT calculation we find a state $|c\rangle$ is set to $\sum_{v} |A_{vc}^{S_0}|^2$. tion, we find a zero-phonon line of 1.12 eV, which agrees well with the experimental data⁶. We also find a smaller Stokes shift energy of 0.05 eV, which is smaller than the first cDFT calculation. Such a smaller Stokes shift is due to the large contribution from conduction bands to the exciton wave function. Since the conduction bands are more delocalized in real-space, they tend to induce smaller atomic distortions to atoms near a divacancy than localized defect states and therefore a smaller Stokes shift. Combining the Stokes shift energy of 0.05 eV from the second cDFT calculation and exciton energy 1.06 eV, we get an ZPL energy 1.01 eV, which slightly underestimates the experimental value by 0.11 eV.

Finally, we note that, while current many-body perturbation theory approaches cannot be rigorously performed on degenerate ground states, as is the case with an openshell system, spin-preserving neutral excitations energies are typically well-described in such systems like diamond NV^- center by GW-BSE calculations⁶⁸. However, in order to accurately capture neutral excitations that involve a change in the total spin, one needs to resort to alternative techniques, such as a full configuration-interaction calculations based on model systems or parameterized extended Hubbard models²⁹ may provide a more faithful descriptions.

IV. CONCLUSIONS

We performed large-scale G_0W_0 and GW plus Bethe-Salpeter equation calculations for 3C-SiC divacancy, an emerging candidate for physical implementation of solidstates qubit. In contrast to the diamond NV⁻ center and the 4H-SiC divacancy, where the deep defect levels are well separated from bulk states, our GW calculations show the optically active spin-down e states are only 0.1 eV below the extended bulk states. Such a small separation lead to the emergence of excitons composed of transitions between defect-to-defect $(a_1^{\downarrow} \rightarrow 1e^{\downarrow})$, defect-to-bulk, and bulk-to-bulk transitions. Our independent-particle approximation and GW-BSE calculations show that the calculated absorption spectra display strong linear dichroism near the absorption edge, which is explained with dipole selection rules. Through GW-BSE calculations, we show that the lowest exciton has a large binding energy of 0.49 eV and observe that electron-hole interactions decrease the optical weight of the lowest-energy exciton. Substantial contribution from bulk states are observed in low-energy excitons, indicating the importance of bulk states in the understanding the optical properties of 3C-SiC divacancy. Finally, we estimate the zero-phonon line using cDFT and GW-BSE methods and obtain good agreement with the experimental value.

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- ¹ J. R. Weber, W. F. Koehl, J. B. Varley, A. Janotti, B. B. Buckley, C. G. Van de Walle, and D. D. Awschalom, Proceedings of the National Academy of Sciences **107**, 8513 (2010).
- ² D. D. Awschalom, R. Hanson, J. Wrachtrup, and B. B. Zhou, Nature Photonics **12**, 516 (2018).
- ³ M. W. Doherty, N. B. Manson, P. Delaney, F. Jelezko, J. Wrachtrup, and L. C. Hollenberg, Physics Reports 528, 1 (2013), the nitrogen-vacancy colour centre in diamond.
- ⁴ A. Gali, Nanophotonics 8, 1907 (2019).
- ⁵ A. L. Falk, B. B. Buckley, G. Calusine, W. F. Koehl, V. V. Dobrovitski, A. Politi, C. A. Zorman, P. X.-L. Feng, and D. D. Awschalom, Nature Communications 4, 1 (2013).
- ⁶ D. J. Christle, P. V. Klimov, C. F. de las Casas, K. Szász, V. Ivády, V. Jokubavicius, J. Ul Hassan, M. Syväjärvi, W. F. Koehl, T. Ohshima, N. T. Son, E. Janzén, A. Gali, and D. D. Awschalom, Phys. Rev. X 7, 021046 (2017).
- ⁷ N. T. Son, C. P. Anderson, A. Bourassa, K. C. Miao, C. Babin, M. Widmann, M. Niethammer, J. Ul Hassan, N. Morioka, I. G. Ivanov, F. Kaiser, J. Wrachtrup, and D. D. Awschalom, Applied Physics Letters **116**, 190501 (2020).
- ⁸ K. C. Miao, J. P. Blanton, C. P. Anderson, A. Bourassa, A. L. Crook, G. Wolfowicz, H. Abe, T. Ohshima, and D. D. Awschalom, Science (2020), 10.1126/science.abc5186.
- ⁹ C. Hepp, T. Müller, V. Waselowski, J. N. Becker, B. Pingault, H. Sternschulte, D. Steinmüller-Nethl, A. Gali, J. R. Maze, M. Atatüre, and C. Becher, Phys. Rev. Lett. **112**, 036405 (2014).
- ¹⁰ J. N. Becker and E. Neu, in *Diamond for Quantum Appli*cations Part 1, Semiconductors and Semimetals, Vol. 103, edited by C. E. Nebel, I. Aharonovich, N. Mizuochi, and M. Hatano (Elsevier, 2020) pp. 201 – 235.
- ¹¹ B. L. Green, S. Mottishaw, B. G. Breeze, A. M. Edmonds, U. F. S. D'Haenens-Johansson, M. W. Doherty, S. D. Williams, D. J. Twitchen, and M. E. Newton, Phys. Rev. Lett. **119**, 096402 (2017).
- ¹² M. H. Metsch, K. Senkalla, B. Tratzmiller, J. Scheuer, M. Kern, J. Achard, A. Tallaire, M. B. Plenio, P. Siyushev, and F. Jelezko, Phys. Rev. Lett. **122**, 190503 (2019).
- ¹³ M. Zhong, M. P. Hedges, R. L. Ahlefeldt, J. G. Bartholomew, S. E. Beavan, S. M. Wittig, J. J. Longdell, and M. J. Sellars, Nature **517**, 177 (2015).
- ¹⁴ T. Utikal, E. Eichhammer, L. Petersen, A. Renn, S. Gtzinger, and V. Sandoghdar, Nature Communications 5, 1 (2014).
- ¹⁵ P. Siyushev, K. Xia, R. Reuter, M. Jamali, N. Zhao, N. Yang, C. Duan, N. Kukharchyk, A. D. Wieck, R. Kolesov,

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and J. Wrachtrup, Nature Communications 5, 1 (2014).

- ¹⁶ T. A. Abtew, W. Gao, X. Gao, Y. Y. Sun, S. B. Zhang, and P. Zhang, Phys. Rev. Lett. **113**, 136401 (2014).
- ¹⁷ H. Seo, M. Govoni, and G. Galli, Scientific Reports 6, 20803 (2016).
- ¹⁸ H. Seo, H. Ma, M. Govoni, and G. Galli, Phys. Rev. Materials 1, 075002 (2017).
- ¹⁹ P. J. Wellmann, Semiconductor Science and Technology 33, 103001 (2018).
- ²⁰ J. Davidsson, V. Ivády, R. Armiento, N. T. Son, A. Gali, and I. A. Abrikosov, New Journal of Physics **20**, 023035 (2018).
- ²¹ J.-I. Iwata, C. Shinei, and A. Oshiyama, Phys. Rev. B 93, 125202 (2016).
- ²² J. Davidsson, V. Ivády, R. Armiento, T. Ohshima, N. T. Son, A. Gali, and I. A. Abrikosov, Applied Physics Letters **114**, 112107 (2019).
- ²³ P. A. Schultz, R. M. Van Ginhoven, and A. H. Edwards, Phys. Rev. B **103**, 195202 (2021).
- ²⁴ H. Ma, M. Govoni, and G. Galli, npj Computational Materials 6, 85 (2020).
- ²⁵ M. Bockstedte, F. Schütz, T. Garratt, V. Ivády, and A. Gali, npj Quantum Materials 3, 31 (2018).
- ²⁶ Y. Wang, Q. Lin, and P. X.-L. Feng, Opt. Express 29, 1011 (2021).
- ²⁷ I. Chatzopoulos, F. Martini, R. Cernansky, and A. Politi, ACS Photonics 6, 1826 (2019).
- ²⁸ S. Castelletto, M. Barbiero, M. Charnley, A. Boretti, and M. Gu, Phys. Rev. Applied 14, 034021 (2020).
- ²⁹ S. Choi, M. Jain, and S. G. Louie, Phys. Rev. B 86, 041202 (2012).
- ³⁰ Y. Ma, M. Rohlfing, and A. Gali, Phys. Rev. B 81, 041204 (2010).
- ³¹ M. Gerosa, F. Gygi, M. Govoni, and G. Galli, Nature Materials 17, 1122 (2018).
- ³² M. Del Ben, F. H. da Jornada, A. Canning, N. Wichmann, K. Raman, R. Sasanka, C. Yang, S. G. Louie, and J. Deslippe, Comput. Phys. Commun. **235**, 187 (2019).
- ³³ M. S. Hybertsen and S. G. Louie, Phys. Rev. B **34**, 5390 (1986).
- ³⁴ M. Rohlfing and S. G. Louie, Phys. Rev. B **62**, 4927 (2000).
- ³⁵ B.-C. Shih, Y. Xue, P. Zhang, M. L. Cohen, and S. G. Louie, Phys. Rev. Lett. **105**, 146401 (2010).
- ³⁶ W. Gao, W. Xia, Y. Wu, W. Ren, X. Gao, and P. Zhang, Phys. Rev. B **98**, 045108 (2018).
- ³⁷ C. Friedrich, M. C. Müller, and S. Blügel, Phys. Rev. B 83, 081101 (2011).
- ³⁸ Z. Ergönenc, B. Kim, P. Liu, G. Kresse, and C. Franchini, Phys. Rev. Materials 2, 024601 (2018).
- ³⁹ D. Golze, M. Dvorak, and P. Rinke, Frontiers in Chemistry 7, 377 (2019).

- ⁴⁰ S. G. Louie, Y.-H. Chan, F. H. da Jornada, Z. Li, and D. Y. Qiu, Nature Materials **20**, 728 (2021).
- ⁴¹ L. Gordon, A. Janotti, and C. G. Van de Walle, Phys. Rev. B **92**, 045208 (2015).
- ⁴² L. Torpo, T. E. M. Staab, and R. M. Nieminen, Phys. Rev. B 65, 085202 (2002).
- ⁴³ T. Kobayashi, K. Harada, Y. Kumagai, F. Oba, and Y.-i. Matsushita, Journal of Applied Physics **125**, 125701 (2019).
- ⁴⁴ N. T. Son, P. Carlsson, J. ul Hassan, E. Janzén, T. Umeda, J. Isoya, A. Gali, M. Bockstedte, N. Morishita, T. Ohshima, and H. Itoh, Phys. Rev. Lett. **96**, 055501 (2006).
- ⁴⁵ M. Del Ben, F. H. da Jornada, G. Antonius, T. Rangel, S. G. Louie, J. Deslippe, and A. Canning, Phys. Rev. B 99, 125128 (2019).
- ⁴⁶ J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. **77**, 3865 (1996).
- ⁴⁷ J. R. Chelikowsky, N. Troullier, and Y. Saad, Physical Review Letters **72**, 1240 (1994).
- ⁴⁸ L. Kronik, A. Makmal, M. L. Tiago, M. M. G. Alemany, M. Jain, X. Huang, Y. Saad, and J. R. Chelikowsky, physica status solidi (b) **243**, 1063 (2006).
- ⁴⁹ Z. Li and R. C. Bradt, Journal of Materials Science 21, 4366 (1986).
- ⁵⁰ J. Deslippe, G. Samsonidze, D. A. Strubbe, M. Jain, M. L. Cohen, and S. G. Louie, Computer Physics Communications 183, 1269 (2012).
- ⁵¹ D. R. Hamann, Phys. Rev. B 88, 085117 (2013).
- ⁵² M. Schlipf and F. Gygi, Computer Physics Communications 196, 36 (2015).
- ⁵³ P. Scherpelz, M. Govoni, I. Hamada, and G. Galli, Journal of Chemical Theory and Computation **12**, 3523 (2016).
- ⁵⁴ P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. B. Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni, N. Colonna, I. Carnimeo, A. D. Corso, S. de Gironcoli, P. Delugas, R. A. D. Jr, A. Ferretti, A. Floris, G. Fratesi, G. Fugallo, R. Gebauer, U. Gerstmann, F. Giustino, T. Gorni, J. Jia, M. Kawamura, H.-Y. Ko, A. Kokalj, E. Kkbenli, M. Lazzeri, M. Marsili, N. Marzari, F. Mauri, N. L. Nguyen, H.-V. Nguyen, A. O. de-la Roza, L. Paulatto, S. Ponc, D. Rocca, R. Sabatini, B. Santra, M. Schlipf, A. P. Seitsonen, A. Smogunov, I. Timrov, T. Thonhauser, P. Umari, N. Vast, X. Wu,

and S. Baroni, Journal of Physics: Condensed Matter **29**, 465901 (2017).

- ⁵⁵ P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. Dal Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, A. Smogunov, P. Umari, and R. M. Wentzcovitch, Journal of Physics: Condensed Matter **21**, 395502 (19pp) (2009).
- ⁵⁶ R. W. Godby, M. Schlüter, and L. J. Sham, Phys. Rev. B 37, 10159 (1988).
- ⁵⁷ S. Lebègue, B. Arnaud, M. Alouani, and P. E. Bloechl, Phys. Rev. B 67, 155208 (2003).
- ⁵⁸ M. Govoni and G. Galli, J. Chem. Theory Comput. **11**, 2680 (2015).
- ⁵⁹ T. A. Pham, H.-V. Nguyen, D. Rocca, and G. Galli, Phys. Rev. B 87, 155148 (2013).
- ⁶⁰ H.-V. Nguyen, T. A. Pham, D. Rocca, and G. Galli, Phys. Rev. B 85, 081101 (2012).
- ⁶¹ L. Hung, F. H. da Jornada, J. Souto-Casares, J. R. Chelikowsky, S. G. Louie, and S. Öüt, Phys. Rev. B 94, 085125 (2016).
- ⁶² W. Gao, W. Xia, X. Gao, and P. Zhang, Scientific Reports
 6, 36849 (2016).
- ⁶³ "Appendix c: Major physical properties of common sic polytypes," in *Fundamentals of Silicon Carbide Technology* (John Wiley & Sons, Ltd, 2014) pp. 521–524.
- ⁶⁴ K. Momma and F. Izumi, Journal of Applied Crystallography 44, 1272 (2011).
- ⁶⁵ J. R. Maze, A. Gali, E. Togan, Y. Chu, A. Trifonov, E. Kaxiras, and M. D. Lukin, New Journal of Physics 13, 025025 (2011).
- ⁶⁶ W. Gao and J. R. Chelikowsky, Journal of Chemical Theory and Computation 15, 5299 (2019), pMID: 31424933.
- ⁶⁷ T. Sommerfeld, Journal of Chemical Theory and Computation 9, 4866 (2013), pMID: 26583406.
- ⁶⁸ V. Ivády, I. A. Abrikosov, and A. Gali, npj Computational Materials 4, 76 (2018).