Ti interstitial flows giving rutile TiO_{2} reoxidation process enhancement in (001) surface

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Phys. Rev. Materials **3**, 125801 — Published 2 December 2019

DOI: 10.1103/PhysRevMaterials.3.125801

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(Dated: November 5, 2019)

We revisited ab initio evaluations of the barrier energies along the possible diffusion paths of the defects in rutile TiO_2 by using diffusion Monte Carlo method. We found that Ti interstitials hopping along the c-axis are identified as the major diffusion directing to (001) surface, contradicting any of the previous DFT studies. Our finding reasonably explains recent experiments reporting that the photocatalytic activity in (001) surface is superior to that in (110) surface: The faster Ti diffusion directing to (001) surface leads to the better self-compensation ability and maintains its photocatalytic activity.

I. INTRODUCTION

TiO_2 is a representative transition metal oxide with various applications such as white paints, photovoltaic cells, and rechargeable batteries. [1–5] Its photo-catalysis ability is especially useful for water splitting and anti-pollution/bacteria coating. [6] During the photo-catalysis reaction, O ions are easily detached from the surface, [7] and hence one may anticipate the depression of the photo-catalysis ability. Yet in reality, the surface gets O ions from the atmosphere, and the photo-catalysis ability is maintained. [7]

One of the most useful properties is the reoxidization of rutile surface state even in a vacuum keeping its stoichiometry. The property is promising for such applications in space as a coating over the solar panels of spaceships keeping its performance of photo reactions.[8] The reoxidization in a vacuum is explained to be caused by the possible ionic flows of Ti interstitials (Ti_i) and/or Oxygen vacancies (V_O) from within the bulk toward the surface compensating the stoichiometry kept unchanged. [7] However, a consensus on the diffusion process of point defects has yet to be established and controversy remains even for within a simple bulk structure. [9, 10]

Surveying over the controversy, the points to be clarified here would be summarized into two simple questions: (a) which defect (Ti_i or V_O) is the dominant, and (b) which diffusion path is dominating. An experiment of the reoxidization of the sputtered rutile TiO_2 (110) surface annealed in ultrahigh vacuum [11] reports a conclusion that Ti_i plays a major role in the process. This is also supported from *ab initio* studies using density functional theory (DFT), [9, 10] predicting lower energy barriers for Ti_i than V_O diffused in any directions. Taking Ti_i being superior to V_O, the controversy exists on which path gives faster diffusion, parallel ($c_\parallel$) or perpendicular ($c_\perp$) to the c-direction [parallel to the Ti-chain in the crystal]. While two old experiments [12, 13] report contradicting conclusions to each other, both of the previous DFT works [9, 10] support $c_\perp$ as the major diffusion process.

One of the major origin of the energy barrier required for a defect to move beyond is the interaction between the surrounding atoms. It is therefore sensitive to how the electronic distribution of a defect spreads to contact with the neighboring atoms. Here we remind that such a spreading is poorly estimated by the conventional type of DFT using LDA or GGA type exchange-correlation (XC) functionals. In these XCs, the cancellation of the self-interaction is incomplete, leading to a spurious delocalization of the charge distribution. [14, 15] The shortcoming is known to be recovered to some extent by using DFT+U methods [16, 17] mainly curing the self-interaction problem but also the description of electronic correlations. [16, 17] The method is reported [17] to achieve fairly well descriptions of the ground states in the systems with transition metal elements, which have been regarded as a representative challenge for the electronic correlations. The drawback for the method has been how to choose

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the adjustable parameters, U and J, by which the method is known to be sensitive in its predictions. [17, 18] There have been some advances on this matter by such approaches to determine U by some variational scheme. [19] In the scheme, however, J is practically fixed at zero while the choice of J can seriously be affecting the prediction. [18] The remedy by DFT+U would, therefore, be limited to some extent toward the perfect descriptions of the problem.

We hence revisit the evaluation of the energy barriers for defects diffusion applying diffusion Monte Carlo method (DMC). [20] The method is based on the variational principles in which the delicate balance between the exchange and correlation [21] can be handled satisfactorily without any arbitrary modeling to be required. The method has successfully been applied to the present TiO_2 system in previous works. [4, 5, 22, 23] We confirmed that Ti_i is the dominant defect to diffuse, contributing to the reoxidation process with an energy barrier lower than that for V_O, being consistent with previous DFT works. [9, 10] A striking finding we made is that the previous DFT prediction supporting $c_\perp$ is reverted into $c_\parallel$ when the cancellation of the self-interaction is considered by using '+U' or DMC. The results support a better reoxidation activity on (001) surface, consistent with experiments [24, 25] reporting that the said surface has almost the highest photocatalytic activity.

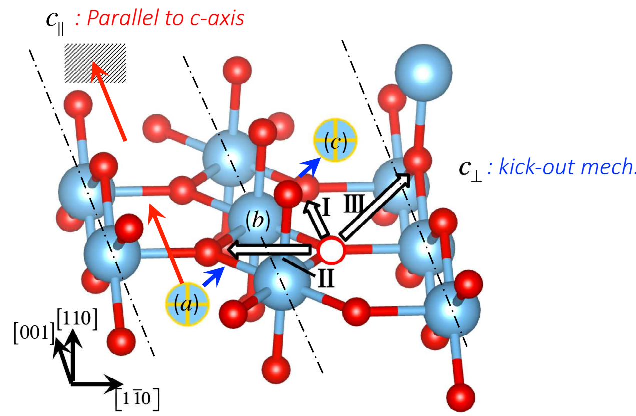

FIG. 1. Five possible paths for defect diffusions of Ti_i (blue and red arrows) and V_O (white arrows) in bulk rutile TiO_2. The large blue balls are Ti ions and the red small balls are oxygen ions. Ti atoms are located along the c-axis ([001]-direction). In $c_\perp$ diffusion (blue arrow), a Ti_i kicks a Ti on the axis out to make another Ti_i in opposite side (kick-out diffusion[26]), directing along [100] or [010] axis. The diffusion along the path $c_\parallel$ (red arrow) directs toward [001] surface as shown by a hatched square.

II. SYSTEM

The rutile structure of TiO_2 is shown in Fig. 1. It consists of Ti chains along the c-axis. Ti positions along the axis are shifted by 1/2 period between the neighboring chains. Ti_i is formed in the middle of Ti chains as shown in Fig. 1, [9] for which two possible diffusion paths ($c_\parallel$ and $c_\perp$) are of interest. [9] The hopping along $c_\perp$ is described as the 'kick-out mechanism'. [26] For V_O, three paths, I-III in Fig. 1, are considered. [9] We evaluated barrier energies along these five paths for fully positively charged defects ($Ti_i^{\bullet\bullet\bullet\bullet}$, $V_O^{\bullet\bullet}$), as summarized in Table I. Previous theoretical works [9, 27] predict only the possibility of getting $Ti_i^\times$,$Ti_i^{\bullet\bullet\bullet\bullet}$,$V_O^\times$,$V_O^{\bullet\bullet}$ depending on the Fermi level, where, for example, $Ti_i^{\bullet\bullet\bullet\bullet}$ represents there are +4 charges per Ti interstitial (less by 4 electrons per defect than the neutral state) and $V_O^\times$ represents there are ±0 charges per O vacancy. Experimentally, the charged defects are confirmed to be realized in surface, [28] and hence we took $Ti_i^{\bullet\bullet\bullet\bullet}$ and $V_O^{\bullet\bullet}$ as the defects to be investigated. The results by the neutral defect, $Ti_i^\times$, are also shown in Table I, which are referred only when we make further discussions. The descriptions henceforth are therefore about the $Ti_i^{\bullet\bullet\bullet\bullet}$ and $V_O^{\bullet\bullet}$ unless noted otherwise.

III. CALCULATION DETAILS

We made a simulation cell by putting a point defect into a 2×2×3 supercell of the ideal rutile TiO_2 unit cell. We optimized the crystal structures at the *edge* and the *saddle* points of the states along the diffusion paths using the PAW-DFT method implemented on VASP.[29] The optimizations are made to relax internal atomic positions within a cell under the fixed lattice constants at experimental values.[30] The energy cutoff is 700 eV and the spacing of the k-mesh sampling is denser than 0.50 Å^{-1}. Atomic positions are relaxed until the forces on any ions are suppressed less than 0.01 eV/Å. The structures at the saddle states are determined by the climbing nudged elastic band (c-NEB) method.[31] A diffusion path is expressed with 5 or 15 intermediate states between the edge states. Since one of the states must be converged to be the saddle state in c-NEB, [31] the number of states does not affect the barrier energy prediction but affect the convergence of the relaxation.

We applied DMC to evaluate the energies at the edge and saddle structures using QMCPACK. [32] We used Slater-Jastrow type trial wave functions. [20, 33] Orbital functions used in the Slater determinant are generated by LDA+U method implemented in Quantum Espresso. [34] We used a Hubbard correction value of U=4.86 eV from a previous work, [22] giving the best accessible nodal surface within this formalism, guaranteeing the lowest energy for TiO_2 from the variational principle. Core electrons in both Ti and O atoms were described by the use of a hard norm-conserving pseudopotentials developed to reproduce accurately all electrons results with the context of many-body theory and as described in previous works [22]. The orbitals are generated with a 300 Ry energy cutoff and the thermodynamic limit is reached with a 2×2×2 k-mesh size. The Jastrow factor consists of one, two, and three body terms amounting to 144 variational parameters in total, which are optimized by variational Monte Carlo calculations. [20, 33] The parameters are optimized by

TABLE I. Barrier energies of Ti_i ($c_\parallel$ and $c_\perp$) and V_O (I, II, and III) paths evaluated by various methods, including previous works. [9, 10] All the predictions are made for fully positively charged defects ($Ti_i^{\bullet\bullet\bullet\bullet}$, $V_O^{\bullet\bullet}$), except 'DMC ($Ti_i^{\times}$' (neutral) which is shown for a reference in discussions. The geometries to evaluate the barrier are optimized each to neutral and charged states, independently.

	Ti_i		V_O		
	$c_\parallel$	$c_\perp$	I	II	III
GGA-PW91[9]	0.37	**0.225**	1.77	0.69	1.1
GGA-PW91[10]	0.31	**0.23**	–	–	–
LDA+U	**0.54**	0.90	2.42	1.60	1.36
DMC	**0.4(1)**	0.9(1)	2.0(1)	0.9(2)	1.7(1)
DMC ($Ti_i^{\times}$)	2.6(4)	1.6(1)	–	–	–

the scheme to minimize a hybridization of energy and variance in 7:3. Twist averaging over the boundary conditions are taken into account with 2×2×2 grid. [35] We estimated a time-step bias by a linear extrapolation of the energies obtained at two time steps, dt = 0.020 and 0.005 a.u.$^{-1}$. It is confirmed that the time-step bias is proportional to dt in a range of dt < 0.020 a.u.$^{-1}$. We set a target population of walkers to be 4,000. Practically this size of target population is large enough to suppress a population control error.

IV. RESULTS AND DISCUSSION

Table I summarizes the results of the barrier energies along each path. Looking at the lowest barrier-energies (shown in bold), all methods, consistent with each other, predict Ti_i as the preferred diffusion carrier. The striking difference is found between our current result and the previous ones regarding Ti_i preferred diffusion path. Updated predictions by LDA+U and DMC supports $c_\parallel$ as the dominant flow, directing towards the (001) surface while $c_\perp$ directing towards the (100) or (010) surface. The prediction here may explain the experimental observation of the photocatalytic activity being enhanced at (001) surface compared to the (100) surface. [24, 25] We note that LDA+U and DMC give different predictions about the fastest diffusion path for V_O. Our final DMC prediction gives path II as the fastest path for oxygen vacancy diffusion (V_O). However, path II alone cannot produce any diffusion flows because sites in this path are disconnected from each other. For V_Os to diffuse globally in the bulk a combination of path I /III with path II is needed, otherwise V_Os will be constrained to the isolated sites in path II.

When compared to our DMC results, previous GGA-DFT calculations show a significant underestimation of barrier energies. Even using "the same fixed geometry relaxed with DFT+U" in GGA and DFT+U calculations, the trend of underestimation is confirmed. This can be attributed to that GGA generally underestimates a cohesive energy [36], since a defect is more weakly combined with the surrounding ions than reality, making its hopping easier.

As can be seen in Table I, evaluating the diffusion path of the neutral defect 'DMC ($Ti_i^{\times}$)', the most favorable diffusion path is $c_\perp$, opposite to what is found for a charged defect. This might be a clue to understanding why the present result is contradicting to the previous DFT works, as well as to understanding the contradiction in the earlier experiments: [12, 13] One of the dominant factor to determine the preferred diffusion path could be the ionic radius of the defects, which is reduced when they are positively charged to reduce accompanying electrons. The sensitive dependence on the choice of XC potentials in Table I could support this, because the estimation of the radius is known to be sensitive to how the self-interaction is carefully treated. [15] Poor treatments are expected to give a spurious delocalization of distribution leading to a larger radius. [14] The Hubbard '+U' correction is introduced to correct this, and hence corrects the radius smaller. Previous GGAs are therefore suspected to give overestimations of the radius, namely '*spuriously less positively charged defects*'. [14] Our Bader analyses using a scheme described in ref.[37] actually showed that PBE has larger volume than LDA+U, predicting 6.765 and 6.914 Å^3 for LDA+U and PBE, respectively.

An earlier experiment [13] supporting $c_\perp$ as the preferred path was performed at high temperatures raging from 1000 to 1500 K. It is shown through simulation that the electronic distribution in the valence region is expanded with high temperatures. [38] The high temperature experiment suggests a less positively charged defect favoring the $c_\perp$ path. This behavior is confirmed by our DMC ($Ti_i^{\times}$) calculation on a neutral defect (see table I), which has larger Bader volume (7.690 Å^3) than $Ti_i^{\bullet\bullet\bullet\bullet}$ (6.765 Å^3).

The faster ionic flow, Ti_i, in [001] direction as our updated prediction would explain the experimental facts fairly reasonably as follows: In the photochemical reactions without any oxygen compensations such as those with Ag^+ ions in an aqueous solution, the enhanced reactivities are actually observed when using (001) surface [24, 25, 39]. Aiming to recover desorbed oxygens by catalytic reduction processes, Ti ions are required to flow from a surface into the bulk inside so that the stoichiometry at the surface can be kept to support the reactions. Having the surface being perpendicular to the faster axis would enhance such ionic flows, and then the reactions get to be accelerated. We also note that there are contradicting reports that the (001) surface gives less reactivity in some systems [39, 40]. When the roughness gets reduced to the atomic scale (~ 1nm), the (001) surface turns into less reactive than other surface directions. In this case, however, the reactivity also gets suppressed by a couple of orders [39, 40]. Under such a reduced reactivity, the desorptions of oxygen atoms become reluctant, and hence the self-compensation process would become a secondary factor not dominating the reaction anymore, being not contradicting our prediction.

V. CONCLUSION

In conclusion, we performed *ab initio* evaluations of the energy barriers for defects of Ti interstitials and Oxygen vacancies using LDA+U and DMC methods. Ti interstitials diffusing along the Ti-chains (c-axis) are predicted to give the lowest energy barrier, being the most likely origin of the atomic flow toward [001] surface supporting the surface reoxidizations. The result is consistent with the photocatalytic activity in (001) surface being superior to (100) as experimentally observed. [24, 25] The prediction is found to be sensitive to how carefully the cancellation of self-interactions is taken into account, not reproduced by the conventional DFT with non-hybrid XC functionals. [9, 10] The cancellation critically changes the radius of the defects interacting surrounding atoms, which was overestimated by the previous DFT works. [9, 10]

ACKNOWLEDGMENTS

An award of computer time was provided by the Innovative and Novel Computational Impact on Theory and Experiment (INCITE) program. This research used resources of the Argonne Leadership Computing Facility, which is a DOE Office of Science User Facility supported under Contract No. DE-AC02-06CH11357, and the Research Center for Advanced Computing Infrastructure (RCACI) at JAIST. T.I. is grateful for financial suport from Grant-in-Aid for JSPS Research Fellow (18J12653). A.B. is supported by the U.S. Department of Energy, Office of Science, Basic Energy Sciences, Materials Sciences and Engineering Division, as part of the Computational Materials Sciences Program and Center for Predictive Simulation of Functional Materials. K.H. is grateful for financial support from a KAKENHI grant (JP17K17762), a Grant-in-Aid for Scientific Research on Innovative Areas "Mixed Anion" project (JP16H06439) from MEXT, PRESTO (JPMJPR16NA) and the Materials research by Information Integration Initiative (MI^2I) project of the Support Program for Starting Up Innovation Hub from Japan Science and Technology Agency (JST). R.M. is grateful for financial supports from MEXT-KAKENHI (19H04692 and 16KK0097), from Toyota Motor Corporation, from I-O DATA Foundation, from the Air Force Office of Scientific Research (AFOSR-AOARD/FA2386-17-1-4049;FA2386-19-1-4015), and from JSPS Bilateral Joint Projects (with India DST). R.M. and K.H. are also grateful to financial supports from MEXT-FLAGSHIP2020 (hp190169 and hp190167).

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