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# Crystal structure and magnetic properties of the layered van der Waals compound $VBr_3$

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KEYWORDS: van der Waals, magnetism, crystal structure

## **Abstract**

Single crystals of the layered, van der Waals compound  $VBr_3$  were synthesized by a vapor transport technique, and characterized by single crystal x-ray diffraction, temperature- and field-dependent magnetization, and temperature-dependent specific heat measurements.  $VBr_3$  has a honeycomb-based  $BiI_3$ -type structure with space group  $R\bar{3}$  (No. 148) at 100 K; analogous to the structure of other honeycomb trihalides. Structural stacking faults were observed as evidenced by the streaking of reflections along  $c^*$  in the diffraction data.  $VBr_3$  goes through a structural phase transition at 90.4 K and a subsequent antiferromagnetic phase transition at 26.5 K, below which the magnetic moments are primarily aligned along the  $c$ -axis.

## Introduction

Binary transition metal trihalides have been known for several decades and used as model systems for the study of magnetism in layered compounds<sup>1,2</sup>. This interest has been revived recently through the experimental observation of a monolayer ferromagnet,<sup>3,4</sup> possible due to the van der Waals type crystalline structure of these materials, which makes them favorable for cleaving and subsequent heterostructure fabrication. The magnetic properties of  $\text{RuCl}_3$  are also of recent interest as promising system to study the Kitaev model<sup>5</sup>. Despite the fact that many binary halides are known, their actual crystal structures and physical properties have not been fully characterized<sup>1</sup>. Among these poorly characterized compounds, the recent rediscovery of ferromagnetism in  $\text{VI}_3$  has been of interest as it expands the available ferromagnetic halides beyond  $\text{CrI}_3$ <sup>6-8</sup>. For the chromium trihalides, both bromide and iodide are ferromagnetic, with  $\text{CrI}_3$  having a higher phase transition temperature<sup>1,7,9</sup>. Vanadium trihalides (vanadium neighbors chromium in the periodic table) share many structural similarities with the chromium family, and thus are of interest for the systematic characterization of their crystal structures and magnetic properties. In this paper, we focus on  $\text{VBr}_3$ .

$\text{VBr}_3$  was reported to exist in a  $R\bar{3}$  crystal structure<sup>10,11</sup>, but no detailed crystal structure is reported in the International Crystal Structure Database (ICSD). In terms of magnetism,  $\text{VBr}_3$  was briefly mentioned to order antiferromagnetically at 35 K with an effective moment of 2.29  $\mu_B/\text{V}$  at room temperature<sup>12</sup>. Further, it is a reported semiconductor with an optical band gap of slightly larger than 1 eV<sup>13</sup>. In this paper, we present a detailed crystal structure determination at 100 K via single crystal x-ray diffraction, and physical property characterization via temperature- and field-dependent magnetization and specific heat measurements.

## Experimental

$\text{VBr}_3$  single crystals were synthesized by using the vapor transport technique. The as-received vanadium bromide (III) from Alfa Aesar was in fact nearly pure  $\text{VBr}_2\text{O}$ . Starting materials of  $\text{VBr}_2\text{O}$  and VO (Alfa Aesar) were mixed in a 1:1 molar ratio in an argon glove box, and sealed in a silica tube under vacuum. The VO acts as a reducing agent, forming  $\text{V}_2\text{O}_3$  during the crystal growth process. The silica tube was placed in a horizontal tube furnace with the hot end kept at 480 °C and the cold end kept close to the edge of the tube furnace. Typical vapor transport

growth took 4 days. Dark, soft, plate-like single crystals of  $\text{VBr}_3$  were collected at cold end of the tube. The vapor pressures of  $\text{VBr}_3$  and the vanadium oxybromides ( $\text{VBr}_2\text{O}$  and  $\text{VBrO}$ ) are different, which makes the growth of  $\text{VBr}_3$  single crystals possible out of an oxygen-containing environment. A small amount of yellow  $\text{VBrO}$  can sometimes be found at the cold end of the silica tube, and can be easily detached from the dark  $\text{VBr}_3$  single crystals mechanically.  $\text{VBr}_3$  single crystals are air-sensitive, similar to  $\text{VI}_3$ , and thus are handled inside of the glove box as much as possible. Attempts to measure the resistivity of  $\text{VBr}_3$  with silver paint and platinum wires were not successful at room temperature due to the extremely large resistance of the material. This is consistent with the semiconducting nature of  $\text{VBr}_3$ <sup>13</sup>.

The reported crystal structure of  $\text{VBr}_3$  was confirmed and refined by single-crystal x-ray diffraction. The diffraction data were collected at 100 K on a single crystal with a Kappa Apex2 CCD diffractometer (Bruker) using graphite-monochromated Mo- $K\alpha$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ). The raw data were corrected for background, polarization, and Lorentz factor and multi-scan absorption corrections were applied. Finally, the structure was analyzed by the Intrinsic Phasing method provided by the ShelXT structure solution program<sup>14</sup> and refined using the ShelXL least-squares refinement package with the Olex2 program<sup>15</sup>. The ADDSYM algorithm in program PLATON was used to double check for possible higher symmetry<sup>16</sup>.

Both magnetization and specific heat data were measured using a Quantum Design (QD) physical property measurement system (PPMS) Dynacool. Anisotropic magnetization was measured using the vibrating sample magnetometer (VSM) function from 1.8-300 K and from 0-90 kOe. Single crystalline samples were manually aligned in preferred orientation in a plastic capsule provided by QD. The applied field for the temperature dependent measurements was 90 kOe, and magnetic susceptibility is defined as  $M/H$ . Zero-field specific heat data were measured down to 1.8 K using the two-tau relaxation method on a collection of single crystals.

## Results and discussion

The crystal structure of  $\text{VBr}_3$  measured at 100 K is schematically shown in Fig 1. Similar to  $\text{VI}_3$ <sup>6</sup>,  $\text{VBr}_3$  has a  $\text{BiI}_3$ -type honeycomb layer structure, space group  $R\bar{3}$ , No.148. In this structure,  $\text{VBr}_6$  octahedra are connected via edge-sharing within each plane, forming a honeycomb lattice, and are separated by a van der Waals gap in between planes. Each V-Br octahedron is only slightly

distorted. As shown in Fig1c, the vanadium atom is slightly displaced from the center of the octahedron along the  $c$ -axis, causing a minimal difference in V–Br bond lengths: 2.53 and 2.54 Å. Detailed crystal structure information can be found in Table 1 and Table 2.

In the structure of  $VBr_3$ , the vanadium atoms are found in two sites. The majority of the vanadium atoms form a honeycomb lattice, similar to many other transition metal trihalides, such as  $CrI_3$ ,  $VI_3$ , and  $TiCl_3$ <sup>1</sup>. The central normally vacant site in the honeycomb lattice is occupied in a minor but not insignificant fashion in the average structure, which if the occupancy was not due to stacking faults, would lead to a triangular lattice such as in  $VBr_2$  if fully occupied<sup>17</sup>. We attribute this partial occupancy in the average structure not to actual site occupancy in the honeycomb layer but to the presence of stacking faults as is commonly seen in van der Waals layered materials. In the case of  $VI_3$ , these stacking faults are the reason for the 4% average occupancy of the central atom site in the honeycomb<sup>6</sup>. In the case of  $VBr_3$ , however, the stacking faults are found at a much larger fraction (Table 2). This can be illustrated by the comparison of diffraction patterns in  $k$ -space for  $VI_3$  and  $VBr_3$ , shown in Fig 2. It is clear that the  $VI_3$  sample gives “cleaner” spot-like diffraction peaks, showing little streaking. In contrast,  $VBr_3$  shows both twinning and the streaking of reflections along  $c^*$  that is characteristic of layered materials with stacking faults<sup>18</sup>. The twinning can be accounted for during crystal structure refinement, and the streaking gives rise to the partial occupancy of the central atom site in the average structure, because the streaking comes from local deviations of the stacking sequence from A-B-C to more complex non-A-B-C forms. This kind of streaking pattern appears to all of the  $VBr_3$  samples that were measured. During crystal structure refinements, the stacking faults lead to a significant scattering density in the ideally vacant centers (the V2 sites) of the V-based honeycombs (the V1 sites). Hence, to specify the crystal structure averaged over the honeycomb layers and the stacking fault layers, vanadium was introduced into our crystal structure refinement in the V2 site. Unconstrained refinement of the occupancies of the V1 and V2 sites leads to 0.864(10) and 0.312(19) respectively, which adds up to only  $V_{1.02}Br_3$ , within error of  $VBr_3$ . Thus for the final structural model, the sum of the V1+V2 occupancies was constrained to yield the formula  $VBr_3$ ; a fully  $V^{3+}$  material.

It is worth noting that this stacking fault, or stacking sequence disorder, also occurs in other transition metal trihalide systems such as in  $RuCl_3$ <sup>19</sup>. In  $VBr_3$ , such stacking faults, although

present in significant amounts, do not seem to affect the magnetic and structural phase transition temperatures, as was observed in  $\text{RuCl}_3$ <sup>20</sup>. As will be discussed in the following sections, both magnetic and structural phase transitions of  $\text{VBr}_3$  are clear and sharp. Elemental analysis was conducted on several pieces of  $\text{VBr}_3$  single crystals. A typical scanning electron microscopy (SEM) image is shown in Fig. 1b. In the SEM image, clear terraces can be observed for this layered material, with the  $c$ -axis perpendicular to the flat surfaces. Elemental analysis yields a Br to V ratio of  $2.9 \pm 0.2$ , which is consistent with single crystal diffraction refinement result.

Now we look into the magnetic properties of  $\text{VBr}_3$ . Fig. 3 shows its temperature- and field-dependent magnetization. The magnetic susceptibility diverges as temperature decreases, indicating a clear Curie-Weiss behavior, due to the vanadium magnetic moments. In the paramagnetic state, the magnetization is weakly anisotropic with fitted Curie-Weiss temperatures of  $\Theta_c = -20$  K and  $\Theta_{ab} = 1$  K. The difference in anisotropic Curie-Weiss temperatures, albeit small, may reflect competition between in-plane and out-of-plane magnetic interactions. The effective moment was determined to be approximately  $2.6 \mu_B/\text{V}$  by fitting from 150 to 300 K, which is close to the expected value of a spin-1 vanadium atom ( $2.8 \mu_B/\text{V}$ ), i.e. the formally  $\text{V}^{3+}$  in  $\text{VBr}_3$ .

There are two phase transitions observable in the temperature-dependent magnetization data. The higher temperature phase transition appears at  $\sim 88$  K. The magnetization for both  $H//c$  and  $H//ab$  dips slightly at the phase transition. For  $H//c$ , the feature is stronger than that in  $H//ab$ . This is similar to the case in  $\text{VI}_3$ . It is possible that the feature in the  $H//ab$  data comes from a small misalignment of the sample being measured. This high-temperature phase transition may be of a structural origin. At lower-temperature,  $\sim 30$  K, there is another phase transition. Both magnetization values drop at this phase transition, which suggests that the transition is due to the onset of an antiferromagnetic phase at lower temperature. The magnetization value for  $H//c$  decreases significantly more than that for  $H//ab$ , suggesting that the vanadium magnetic moments have a larger  $c$ -axis component in the ordered state. Both phase transitions barely shift in temperature upon the application of axial magnetic field up to 9 T. More detailed neutron scattering measurements are needed to determine the detailed magnetic structure of  $\text{VBr}_3$  at low temperature.

The isothermal magnetization data shown in Fig 3b is consistent with the temperature-dependent data. At room-temperature, the data show a clear paramagnetic behavior. Below the structural

phase transition, at 50 K, the magnetic anisotropy remains small, consistent with a paramagnetic state in this temperature range. Below the magnetic phase transition temperature, at 1.8 K, magnetic anisotropy develops, with both values at 90 kOe much smaller than the expected saturation moment of a vanadium spin-1 system ( $2 \mu_B/V$ ), implying that more metamagnetic phase transitions may occur at higher applied magnetic fields.

To identify the bulk nature of both phase transitions, we also characterized the specific heat of  $VBr_3$ , as shown in Fig. 4. Consistent with magnetic measurements, we observed two clear phase transitions in the zero-field, temperature-dependent specific heat data, at 90.4 K and 26.5 K. The magnetic phase transition temperature ( $T_N$ ) determined in this study is smaller than the previously reported one of 35 K<sup>12</sup>. The estimated Debye temperature for  $VBr_3$  is about 157 K, larger than that of  $VI_3$  (143 K), which is consistent with the relative molecular weights of the two compounds. The numerical difference in the Debye temperatures, however, is smaller than expected based on a simple Debye model.

Comparing  $VBr_3$  to  $VI_3$ <sup>6</sup>, the structural phase transition temperature ( $T_s$ ) is higher while the magnetic phase transition temperature is lower (Fig. 4). The  $T_N$  of  $VBr_3$  is close to that observed for  $VBr_2$  at 29.5 K<sup>17</sup>. The structural phase transition of  $VBr_3$  at 90.4 K has not been previously reported. Unfortunately, we were not able to obtain the crystal structure below 90.4 K due to our experimental limitations. Additional study would be needed to reveal the low temperature crystal structure.

## Conclusion

We have reported the single crystal growth, crystal structure and magnetic characterization of the layered van der Waals compound  $VBr_3$ . Single crystal x-ray diffraction refinement of the average structure indicates that  $VBr_3$  has a  $BiI_3$ -type structure (space group R-3, No.148) above 90.4 K.  $VBr_3$  goes through two phase transitions that are both sharp: one that is likely a structural phase transition at 90.4 K and a lower temperature magnetic phase transition at 26.5 K, which is likely an antiferromagnetic phase transition below which the magnetic moment primarily aligns along the  $c$ -axis. The stacking fault disorder does not broaden these transitions substantially in temperature. The magnetization and heat capacity measurements are consistent in the

identification of these two bulk phase transitions. Future detailed studies may be of interest to reveal the low temperature crystal structure and magnetic structure of  $VBr_3$ .

### **Supporting Information.**

Crystallographic Information File (CIF) for  $VBr_3$ <sup>21</sup>.

### **Accession Codes**

CCDC number 1934916 contain the supplementary crystallographic data for this paper. This data can be obtained free of charge via [www.ccdc.cam.ac.uk/data\\_request/cif](http://www.ccdc.cam.ac.uk/data_request/cif), or by emailing [data\\_request@ccdc.cam.ac.uk](mailto:data_request@ccdc.cam.ac.uk), or by contacting The Cambridge Crystallographic Data Centre, 12 Union Road, Cambridge CB2 1EZ, UK; fax: + 441223336033.

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**Figure Captions:**

**Figure 1.** (a) Schematic drawing of the crystal structure of  $VBr_3$ . Central orange atoms are vanadium and dark spheres represent bromine atoms. The area of white color on the vanadium sites represents the percentage of vacancies on that site. (b) shows a SEM image of a  $VBr_3$  single crystal. The orange scale bar in the left corner indicates 50  $\mu\text{m}$  in length. (c) shows a  $VBr_6$  octahedron with V-Br bond distance labeled on the side.

**Figure 2.** Comparison of the observed single crystal x-ray diffraction patterns for crystals of  $VBr_3$  and  $VI_3$  in two different reciprocal lattice planes ( $hk0$  and  $0kl$ ).

**Figure 3.** (a) Anisotropic magnetization of  $VBr_3$  measured at 90 kOe. Inset shows the zoom-in view of the structural phase transition at  $\sim 90$  K. (b) Anisotropic magnetic isotherms measured at 300 K (solid lines), 50 K (open symbols) and 1.8 K (solid symbols).

**Figure 4.** Zero-field, temperature-dependent specific heat of  $VBr_3$ . The dashed line represents the zero-field specific heat data<sup>6</sup> from  $VI_3$  for comparison.

Figure 1

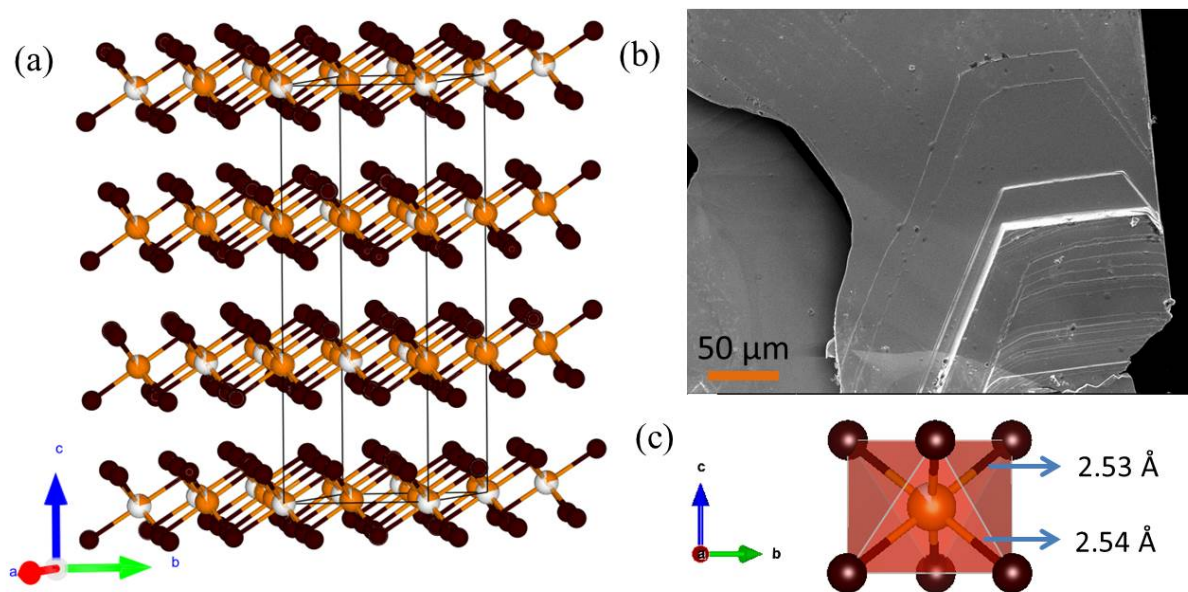


Figure 2

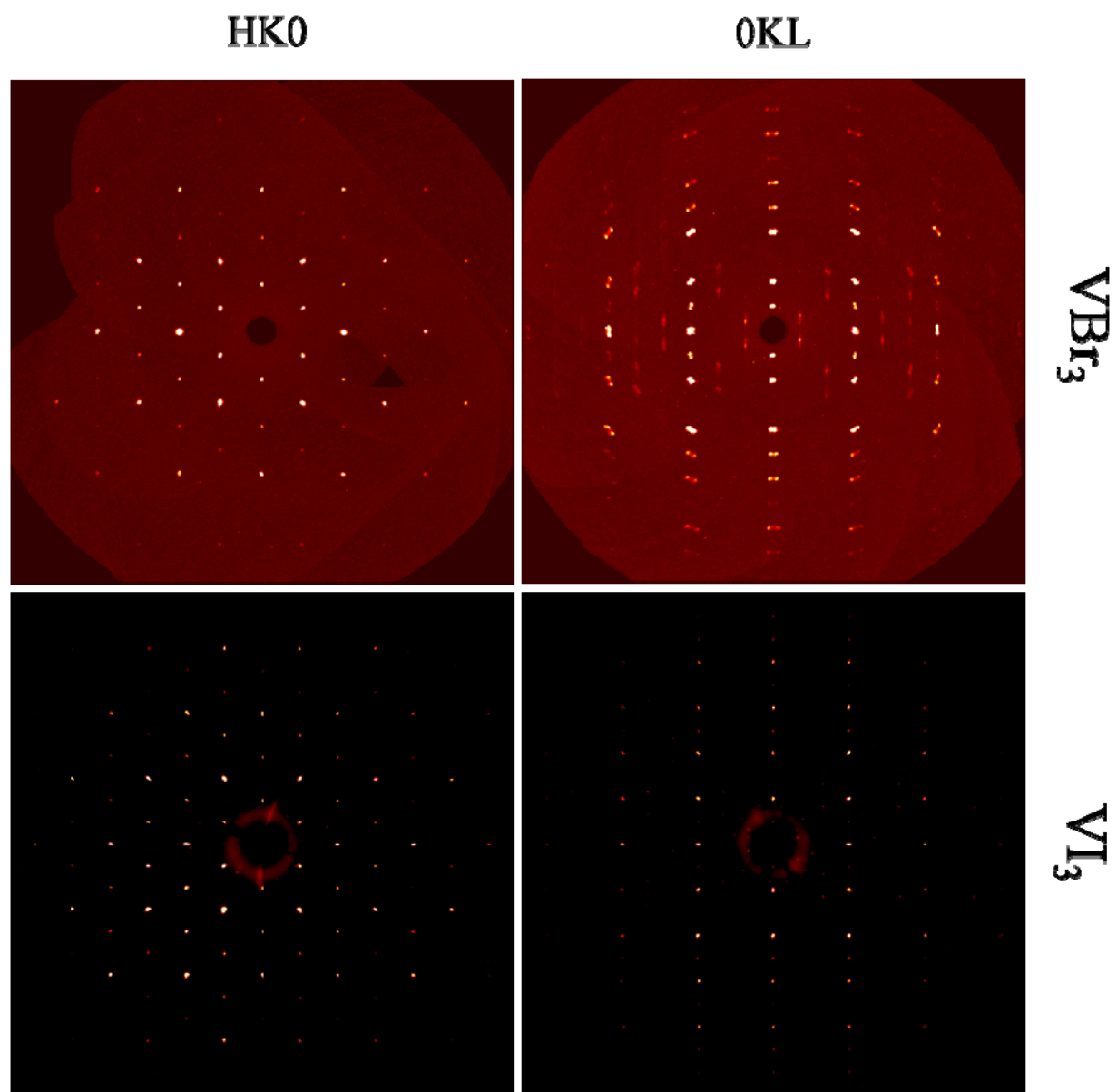
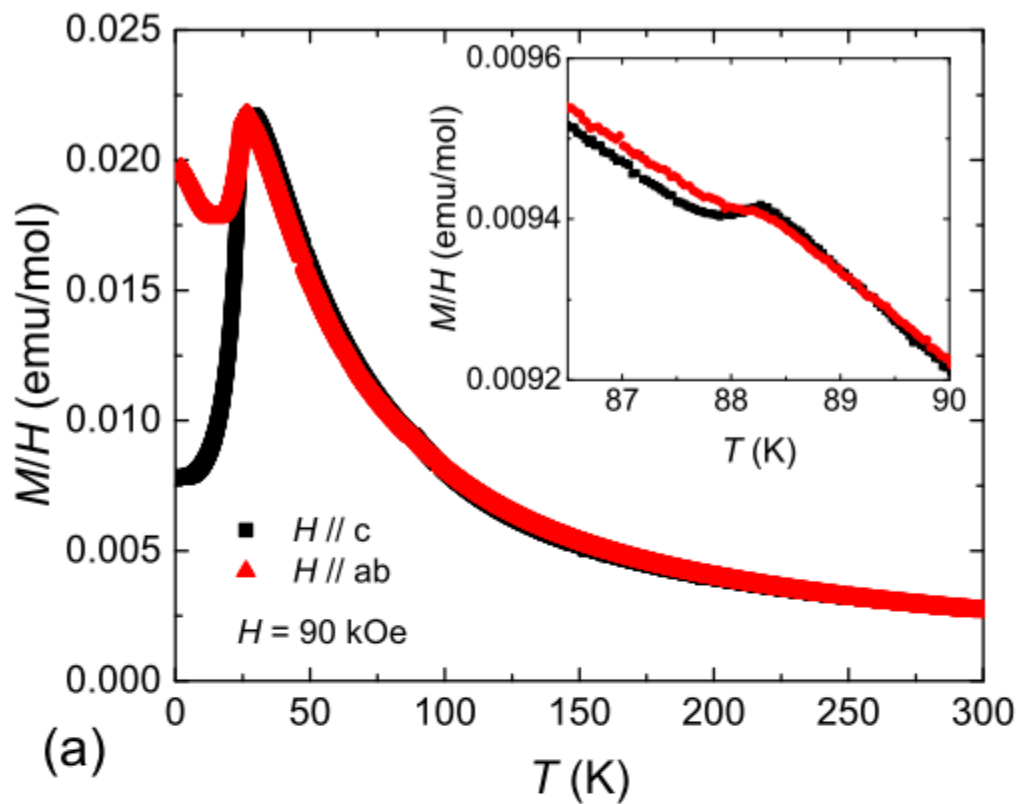
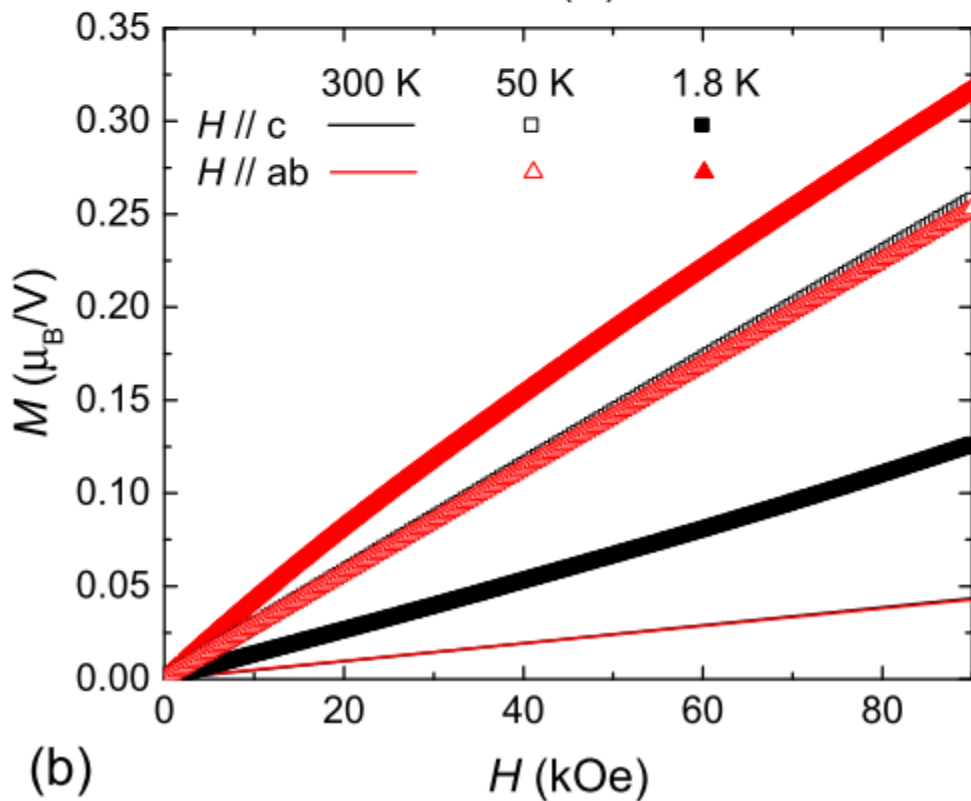


Figure 3

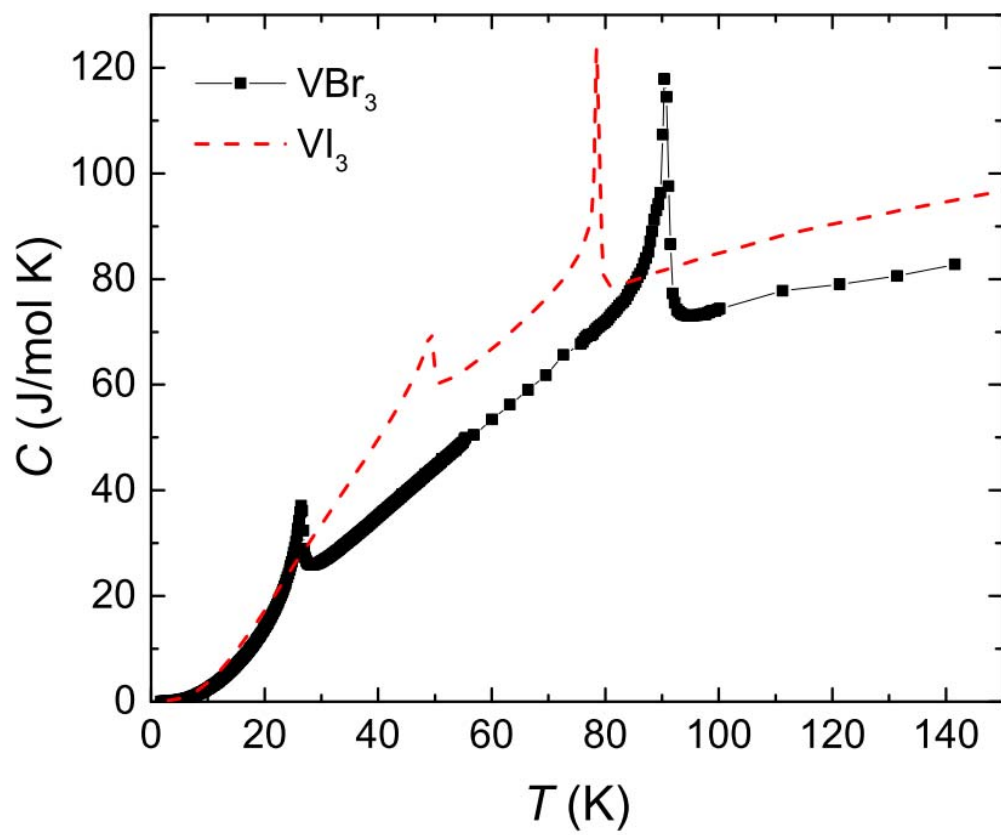


(a)



(b)

Figure 4



**Table 1.** Crystallographic Data for VBr<sub>3</sub> at 100 K.

<i>T</i> (K)	100(1)
<i>formula mass</i> (amu)	290.67
<i>crystal system</i>	trigonal
<i>space group</i>	<i>R</i> -3 (No.148)
<i>a</i> (Å)	6.3711(5)
<i>c</i> (Å)	18.3763(16)
<i>V</i> (Å <sup>3</sup> )	645.98(9)
<i>Z</i>	6
<i>ρ</i> (calcd)(g/cm <sup>3</sup> )	4.483
<i>λ</i> (Å)	0.71073
<i>F</i> (000)	768
<i>θ</i> (deg)	3.33-27.46
Cryst size (mm <sup>3</sup> )	0.032 × 0.076 × 0.194
<i>μ</i> (mm <sup>-1</sup> )	29.908
<i>Final R indices</i> ( <i>R</i> <sub>1</sub> / <i>ωR</i> <sub>2</sub> )	0.0368/0.0635
<i>R indices (all data)</i> ( <i>R</i> <sub>1</sub> / <i>ωR</i> <sub>2</sub> )	0.0467/0.0666
<i>Residual electron density</i> / (eÅ <sup>-3</sup> )	(-1.671) - 1.630
<i>Goodness of fit</i>	1.166

**Table 2.** Wyckoff positions, coordinates, occupancies, and equivalent isotropic displacement parameters respectively for VBr<sub>3</sub> at 100 K.

Atoms	Wyck.Site	x	y	z	S.O.F.	U <sub>eq</sub>
V1	6 <i>c</i>	0	0	0.33345(15)	0.897(4)	0.0050(5)
V2	3 <i>a</i>	0	0	0	0.207(8)	0.0050(5)
Br1	18 <i>f</i>	0.34914(14)	0.00106(15)	0.07924(4)	1	0.0081(2)

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