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Tensile-Strain Dependent Spin States in Epitaxial LaCoO₃ Thin Films

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The spin states of $\mathrm{Co^{3+}}$ ions in perovskite-type LaCoO₃, governed by complex interplay between the electron-lattice interactions and the strong electron correlations, still remain controversial due to the lack of experimental techniques which can detect directly. In this letter, we revealed the tensile-strain dependence of spin states, *i.e.* the ratio of the high- and low-spin states, in epitaxial thin films and a bulk crystal of $\mathrm{LaCoO_3}$ via resonant inelastic soft x-ray scattering. The tensile-strain as small as 1.0% was found to realize different spin states from that in the bulk.

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Charge, spin, and orbital degrees of freedom activated by strong electron correlation realize emergent physical phenomena such as superconductivity, metalinsulator transition, and charge ordering [1]. Perovskitetype cobalt oxide LaCoO₃ is one of the most intriguing materials due to its versatile electron degrees of freedom. Since the spin state of LaCoO₃ is sensitive to the crystal field, it shows a spin crossover from the low-spin (LS) to high-spin (HS) state with increasing temperature [2]. In addition, the spin states can be modified by external stimuli such as a magnetic field and pressure. For example, the possibility of an excitonic insulating state under a high magnetic field was theoretically proposed in the bulk crystal and has also attracted considerable interest these days [3–5]. Epitaxial strain can also influence the spin states as observed for ferromagnetism in LaCoO₃ thin films at lower temperatures ($\lesssim 85 \text{ K}$) [6–13].

The crystal structure of bulk LaCoO₃ is a rhombohedral perovskite-type with corner-sharing CoO₆ octahedra, in which the valence of Co is 3+ with $3d^6$ configuration. The spin state of LaCoO₃ is considered to take LS state with $e_q^0 t_{2q}^6$, intermediate-spin (IS) state with $e_q^1 t_{2q}^5$, or HS state with $e_q^2 t_{2q}^4$. The population of HS state gradually increases as the temperature increases from around 100 K and the HS state becomes dominant above 500 K [2]. Although the spin state and its temperature dependence of LaCoO₃ bulk have already been studied by using various experimental and theoretical approaches [14–21], the spin states have not been completely determined vet. Especially, the spin states in the intermediate temperature region remain controversial, i.e. the mixed HS/LS state or the IS state. On the other hand, in ferromagnetic LaCoO₃ thin films, it is indicated that the ferromagnetism originates from the spin-state, orbital, and

spin orderings [11–13]. From the resonant x-ray diffraction, modulation vector $\mathbf{q} = (1/4, -1/4, 1/4)_{pc}$ (the suffix pc stands for the pseudo-cubic setting) and $\mathbf{q} = (1/6, 1/6, 1/6)_{pc}$ were observed in the thin films grown on LSAT(110) and LSAT(111) substrates, respectively [LSAT: (LaAlO₃)_{0.3}(SrAl_{0.5}Ta_{0.5}O₃)_{0.7}]. Then, the spin states are considered to depend on the tensile-strain. In these thin films, some possible models of spin state orderings were reported in Ref. [11, 12]. The proposed relationships between the strain and the estimated ratio of spin states per unit cell are shown in TABLE I. The change of lattice constant and symmetry induced by the tensile-strain are the main factors which determine the spin state.

In previous studies, the spin states were conjectured from the orderings of 3d electrons on the basis of resonant x-ray diffraction [11–13]. However, direct observations of the electronic structures are needed to clarify the spin states. Since it is difficult to determine the HS/LS ratio by conventional methods such as x-ray absorption spectroscopy (XAS), we performed resonant inelastic soft x-ray scattering (RIXS) by using Co $2p \rightarrow 3d \rightarrow 2p$ process (L-edge). The RIXS is one of the most powerful techniques to clarify the spin states by observing the d-d excitations [22–24]. A higher-energy-resolution (a few hundreds of meV) is required to distinguish the d-d excitations from the elastic scattering [16].

The $LaCoO_3$ epitaxial thin films with the thickness of 30 nm were fabricated on LSAT(110) and LSAT(111) substrates by pulsed laser deposition technique. By using the same substrates with different orientations, we can obtain the pure strain effects on XAS and RIXS spectra. In addition, the LSAT substrate has no structural phase transition and thus suitable for the tem-

perature dependent measurements. Details for the sy theses and the characterizations were described on R [11, 12]. The lattice constant of LSAT substrate is 3.8 Å, while that of LaCoO₃ bulk is 3.804 Å[25], meani that the LaCoO₃ epitaxial thin films grown on LS. substrates are distorted by tensile-strain. The tensi strains from LSAT(110) and LSAT(111) substrates a 1.0\% and 0.5\%, respectively (strains are defined as t ratio of the cubic root of the unit cell volume) [11, 1 The experiments of XAS and RIXS were performed BL07LSU HORNET, SPring-8 [26]. Before RIXS m surements, we obtained the XAS spectra at 300 K total electron yield (TEY) in order to determine the citation energy of RIXS. The RIXS measurements w performed at 40 K and 300 K by using soft x-ray from 770 eV to 810 eV (Co $L_{3,2}$ edge). In the range of the ray energy, energy resolution $\Delta E \sim 300$ meV, which v determined via full width of half maximum (FWHM) the elastic peak. The charge coupled device (CCD) tector was set at 90° relative to the incident x-ray w horizontal polarization to suppress the elastic scatteri (see inset in Fig. 1(b)).

First, we measured Co $L_{3,2}$ XAS by TEY mode LaCoO₃ thin films at 300 K. The spectra are shown Fig. 1(a). The peaks around 779 eV and 794 eV cor spond to the Co L_3 and L_2 edges, respectively. Althou the magnitude of the tensile-strain and modulation v tor q are different between LSAT(110) and LSAT(11 the spectral shapes are quite similar. To investigate t detailed electronic structures from d-d excitations, performed RIXS measurements. As the excitation ergy of RIXS, we selected the energy of A: $L_3 - 2.9 \epsilon$ B: $L_3 - 1.5$ eV, C: $L_3 - 1.0$ eV, D: L_3 , E: $L_3 + 1.7$ ϵ and F: L_2 . The Co $L_{3,2}$ RIXS spectra of the LaCoO₃ tl films are shown in Fig. 1(b). These spectra are norm ized by the intensity of the highest peak. In this figu the thick solid line shows the center of elastic scatt ing peaks and the arrows indicate the fluorescence on Other peaks from 0 eV to 4 eV correspond to the a excitations and the peak around 5.0 eV may be a char transfer (CT) excitation. In the spectra of A and B, t peaks of d-d excitations are clearly observed. We fi that the d-d excitations are obviously different between the thin films on LSAT(110) with 1.0% tensile-strain a LSAT(111) with 0.5% tensile-strain; this indicates the the spin states of LaCoO₃ change drastically according the magnitude of the tensile-strain. On the other hand, in the spectra excited by higher energy (C, D, E, and F),

TABLE I: The estimated ratio of spin states per unit cell at the lowest temperature in the previous studies. A positive value of strain indicates a tensile-strain.

	Strain (%)	LS:	IS:	HS	Ref.
$LaCoO_3/LSAT(110)$	1.0	0	3	1	[11]
$LaCoO_3/LSAT(111)$	0.5	2	0	1	[12]
LaCoO ₃ bulk	0	1	0	0	[2]

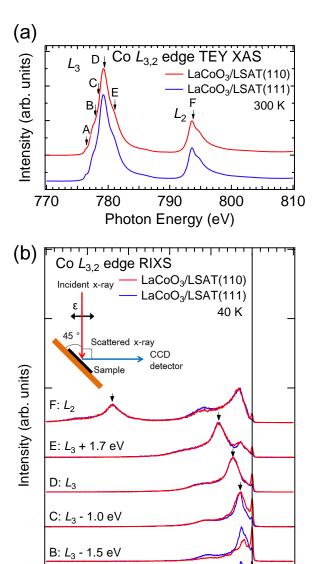


FIG. 1: (Color online) Spectra measured at Co $L_{3,2}$ edge by (a) TEY-XAS and (b) RIXS spectra of LaCoO₃ thin films grown on LSAT(110) and LSAT(111) substrates. The arrows and letters in XAS indicate the excitation energies for RIXS. All RIXS spectra are normalized by the intensity of the highest peak and the arrows indicate the fluorescence peaks. The inset shows the schematic diagram of the experimental setup in our RIXS measurements.

10

Energy Loss (eV)

15

5

0

-5

A: *L*₃ - 2.9 eV

20

25

the peaks of d-d excitations are not clear because of the larger fluorescence. Since the d-d excitations were buried in the fluorescence at the excitation energies near L_3 and L_2 edges, we selected the excitation energy A: $L_3 - 2.9$ eV. Then, we compared the RIXS spectra of the thin films with those of bulk single crystal in order to discuss

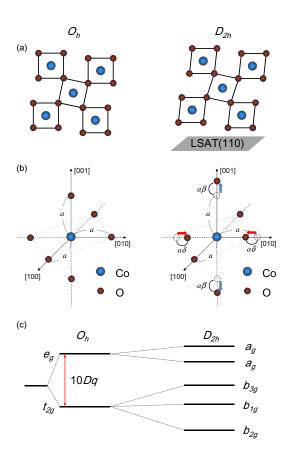


FIG. 2: (Color online) Schematic view of (a) the crystal structure and (b) $[CoO_6]^{9-}$ cluster in O_h and D_{2h} symmetry. "a" correspond to the bond distance between Co and O ions in LaCoO₃ bulk, while " δ " and " β " were defined as the scaling factors of the crystal distortion. (c) The energy level diagram of O_h and D_{2h} symmetry in our calculations. 10Dq indicates the crystal field splitting in O_h symmetry.

the relationship between the spin states and the epitaxial strain.

In the analysis, we carried out impurity Anderson model calculations in O_h and D_{2h} local symmetry. Figure 2(a) show the schematic figures of the crystal structures in bulk and thin film on LSAT(110). The symmetry of bulk is nearly O_h , while that of the thin film on LSAT(110) and LSAT(111) are D_{2h} and D_{3d} , respectively. In the film on LSAT(111), the strain is so small that we consider its symmetry as O_h . Therefore, we performed the calculations in O_h symmetry for the LaCoO₃

TABLE II: Parameter values used in our calculations based on the impurity Anderson model. For details of these parameters, see main text. Units are in eV.

	Δ	Δ^*	10Dq	$(pd\sigma)$	δ	β
$HS(O_h)$	1.1	0.4	1.38	-1.96	0	0
$LS(O_h)$	1.8	0.7	1.40	-2.00	0	0
$HS(D_{2h})$	0.2	0.1	1.40	-2.00	0.010	0.005

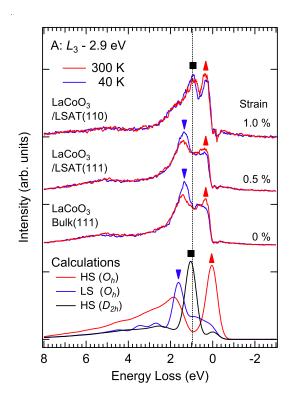


FIG. 3: (Color online) Temperature dependence (300 K & 40 K) of the RIXS spectra excited at A: $L_3-2.9$ eV (776.5 eV) in comparison with the impurity Anderson model calculations in O_h and D_{2h} symmetries. These calculations were performed in the scattering geometry for LaCoO₃/LSAT(110) and the polarization dependence is shown in Supplementary information. The experimental spectra were normalized by their area after subtracting the elastic scatterings by Gaussian fitting. The triangles, squares, and inverted triangles indicate the main peak of the $\mathrm{HS}(O_h)$, $\mathrm{HS}(D_{2h})$, and $\mathrm{LS}(O_h)$ states, respectively.

bulk and the thin film on LSAT(111), while in D_{2h} symmetry for the thin film on LSAT(110). As shown in Fig. 2(b), we consider the $[CoO_6]^{9-}$ cluster model with the scaling factors δ and β in order to describe the crystal distortion, i.e. shrinkage along [010] axis and elongation along [001] axis. In the present model, we consider the Co 3d orbitals of a central Co atom (core hole site) and appropriate linear combinations of Co 3d orbitals on neighboring sites. The anisotropic effect due to the lattice mismatch of substrate is also taken into account through the anisotropic hybridization and the crystal field for Co 3d states, where the local symmetry around the Co ion is approximately treated as D_{2h} . As shown in Fig. 2(c), the energy diagram in D_{2h} symmetry is different from that in O_h symmetry, where 10Dq is defined as the crystal field splitting between e_g and t_{2g} orbitals. In D_{2h} symmetry, the two energy levels $(e_q \text{ and } t_{2q})$ are split into five energy levels by reducing the symmetry, indicating that the new spin states can be realized. In our calculations, the charge-transfer energy from the O 2p

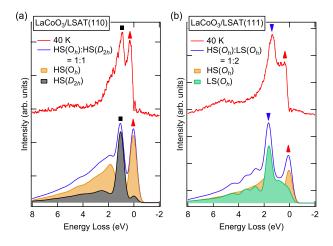


FIG. 4: (Color online) Comparison between the experimental RIXS and the linear combinations of theoretical spectra in (a) $LaCoO_3/LSAT(110)$ and (b) $LaCoO_3/LSAT(111)$. The triangles, squares, and inverted triangles indicate the main peak of the $HS(O_h)$, $HS(D_{2h})$, and $LS(O_h)$ ground states, respectively.

ligand band to the upper Hubbard band and that from the non-local band to the upper Hubbard band are defined as Δ and Δ^* , respectively. In the case of LaCoO₃, Ref. [20] shows the importance of the screening effects from neighboring Co ions. The Coulomb interaction between Co 3d states (U_{dd}) and that between Co 3d and 2p core-hole states $(-U_{dc})$ were set as $U_{dd} = 5.6$ eV and $U_{dc} = 7.0 \text{ eV}$, respectively. The meaning of these parameters were defined as in Ref. [27]. The Slater integrals and the spin-orbit coupling constant are calculated by Cowan's Hartree-Fock program [28] and then the Slater integrals are rescaled by 85%, as usual. The values of multiplets terms are shown in Supplementary information. $(pd\sigma)$ and $(pd\pi)$ are the transfer integrals between Co 3d and O 2p orbitals and we used the parametrization given in Harrison's rule as $(pd\sigma) \propto d^{-3.5}$ [29]. (d denotes the atomic distance.) In our calculations, the parameters were set as TABLE II. The spin crossover (HS \leftrightarrow LS) occur between 10Dq = 1.38 and 1.40 eV in the O_h symmetry. The value of δ and β in the case of the D_{2h} symmetry were obtained from lattice parameters [11]. Note that the calculations were performed in the wide range of parameters (10Dq: 0 - 4.00 eV, β : 0 - 0.040, δ : 0 - 0.040, and $(pd\sigma)$: -3.00 - -1.00 eV) and there is no parameter region where the IS state is the

The experimental RIXS spectra excited with A: L_3 – 2.9 eV (776.5 eV) measured at 40 K and 300 K are shown in comparison to the theoretical ones with several electronic states in Fig. 3. Although the scattering geometry is different among the samples, the RIXS spectra have no significant polarization dependence as shown in Supplementary information and thus we shows the calculated spectra for LaCoO₃/LSAT(110). There are two

kinds of strong peaks in the energy range between 0 and 2 eV in all three samples. The peaks observed around 0.3 eV can be assigned to the excitations from the HS ground states. On the other hand, the peaks around 1.3 eV observed in LaCoO₃/LSAT(111) and bulk correspond to the LS ground states. As the temperature increasing, the behavior that the peak intensities at 0.3 eV increase and those at 1.3 eV decrease is consistent with the fact that population of the HS state increases with increasing temperature. However, peaks which appear at 1.0 eV in LaCoO₃/LSAT(110) can be explained by neither the LS nor HS with O_h symmetry. As seen in the theoretical spectra, the peak for HS state is shifted to 1.0 eV by lowering the symmetry from O_h to D_{2h} . It indicates that the spin state of LaCoO₃/LSAT(110) consists of the HS states with different local symmetries, i.e. the mixture of O_h and D_{2h} symmetries. From the temperature dependence, the peak of $HS(D_{2h})$ still exists at 300 K. It is considered that the spin states are distributed randomly and/or fluctuate quickly after disappearing the spin-orbital ordering at 126 K [11].

To compare the present results with the previous study by resonant x-ray diffraction, we estimate the ratio of spin states by the theoretical spectra of various spin states. Since the theoretical spectra are normalized by the intensity of the theoretical XAS at A: $L_3 - 2.9$ eV, we can reproduce the mixed spin states by linear combinations of the spectra and estimate the ratio of the spin states by the peak intensity in each spin state. As shown in Fig. 4(a), $HS(O_h)$: $HS(D_{2h}) = 1$: 1 is in good agreement with the experimental spectrum of $LaCoO_3/LSAT(110)$, indicating that the CoO_6 octahedra in O_h and D_{2h} local symmetry coexist as the ratio of 1: 1. Since the spatial modulation of spin state is fourfold periodic from the x-ray diffraction in Ref. [11, 13], one plausible model is that the spin states are aligned in order of O_h - $D_{2h}(d_{yz})$ - O_h - $D_{2h}(d_{zx})$, where the HS states with D_{2h} have two different orbital states. In this study, we did not find the theoretical spectra of IS states which reproduce experimental results and thus could not confirm the spin state model of IS: HS = 3: 1. On the other hand, $HS(O_h)$: $LS(O_h) = 1$: 2 agrees with the experimental spectrum of LaCoO₃/LSAT(111) as shown in Fig. 4(b). The result is consistent with the x-ray diffraction study [12].

In conclusion, the spin states of LaCoO₃ change from LS to HS according to the magnitude of the epitaxial strain and the change of the symmetry, which could not be detected by conventional XAS but could be clarified by RIXS measurements. The LS ground state in bulk fully changes to HS states by tensile-strain as small as 1.0%. Especially in the thin film on LSAT(110), we observed the tensile-strain induced unique spin state, *i.e.* $HS(O_h): HS(D_{2h}) = 1: 1$. Since the spin state is not stabilized by temperature or hydrostatic pressure in bulk, applying the tensile-strain is an important factor for realizing novel spin states.

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