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## Topological Surface States Originated Spin-Orbit Torques in Bi<sub>2</sub>Se<sub>3</sub>

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Three dimensional topological insulator bismuth selenide (Bi<sub>2</sub>Se<sub>3</sub>) is expected to possess strong spin-orbit coupling and spin-textured topological surface states, and thus exhibit a high charge to spin current conversion efficiency. We evaluate spin-orbit torques in Bi<sub>2</sub>Se<sub>3</sub>/Co<sub>40</sub>Fe<sub>40</sub>B<sub>20</sub> devices at different temperatures by spin torque ferromagnetic resonance measurements. As temperature decreases, the spin-orbit torque ratio increases from ~ 0.047 at 300 K to ~ 0.42 below 50 K. Moreover, we observe a significant out-of-plane torque at low temperatures. Detailed analysis indicates that the origin of the observed spin-orbit torques is topological surface states in Bi<sub>2</sub>Se<sub>3</sub>. Our results suggest that topological insulators with strong spin-orbit coupling could be promising candidates as highly efficient spin current sources for exploring next generation of spintronic applications.

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The realization of functional devices such as the non-volatile memories and spin logic applications is of key importance in spintronic research [1]. The functions of these magnetic devices require highly efficient magnetization manipulation in a ferromagnet (FM), which can be achieved by an external magnetic field or a spin polarized current by spin transfer torque (STT). Recent advances have demonstrated that pure spin currents resulting from charge currents via spin-orbit coupling in heavy metals, such as Pt [2-7], Ta [8-10], and W [11], can produce strong spin-orbit torques on the adjacent magnetic layers. The reported amplitude of spin Hall angles (i.e. efficiency of spin-orbit torques) in Pt and Ta is in the range of ~ 0.012 to ~ 0.15, and in W is ~ 0.33. The exploration for new materials exhibiting new physics and possessing an even higher conversion efficiency between the charge current density ( $J_c$ ) and spin current density ( $J_s$ ) is crucial to exploit next generation spintronic devices.

The three dimensional (3D) topological insulators (TI) are a new class of quantum state of materials that have an insulating bulk and spin-momentum-locked metallic surface states [12-14]. They exhibit strong spin-orbit coupling and are expected to show a high charge to spin current conversion efficiency. So far, by extensively employing angle-resolved photoemission spectroscopy (ARPES) and spin-resolved ARPES, the Dirac cones and the helical spin polarized topological surface states (TSS) have been observed and the topological nature has been confirmed in TIs [15,16]. The surface state dominant conduction has also been confirmed by thickness dependent transport measurements in Bi<sub>2</sub>Se<sub>3</sub> [17].

The TSS in TI is immune to the nonmagnetic impurities due to the time reversal symmetry protection. Although a gap opening in the TSS dispersion was reported in Bi<sub>2</sub>Se<sub>3</sub>

doped with Fe in the bulk [18], most recently reports have confirmed that the TSS is intact in Bi<sub>2</sub>Se<sub>3</sub> covered with Fe [19,20] or Co [21] with in-plane magnetic anisotropy. The spin dependent transport is known to be significant near the Fermi level in the Bi<sub>2</sub>Se<sub>3</sub> surface states. However, limited spin dependent transport experiments have been focused on TI/FM heterostructures. Only recently, spin-orbit effects have been reported by spin pumping measurements [22-24] and magnetoresistance measurements [25,26]. Direct charge current induced spin-orbit torque on the FM layer has been demonstrated by spin torque ferromagnetic resonance (ST-FMR) measurement only at room temperature [27] and magnetization switching at cryogenic temperature [28]. It is known that for Bi<sub>2</sub>Se<sub>3</sub> the bulk channel provides an inevitable contribution to transport at room temperature and mag diminish the signals of spin-orbit torques arising from surface states. At low temperatures, however, the surface contribution should become significant [17], and spin-orbit torques in TI/FM heterostructures should be enhanced [28].

In this work, we adopt extensively studied Bi<sub>2</sub>Se<sub>3</sub> as the TI layer and investigate the temperature dependence of charge-spin conversion efficiency, spin-orbit torque ratio ( $\theta_{\parallel} = J_s/J_c$ ), by the ST-FMR technique in Bi<sub>2</sub>Se<sub>3</sub>/Co<sub>40</sub>Fe<sub>40</sub>B<sub>20</sub> heterostructures. In this structure, the spin currents generated from charge currents flowing in Bi<sub>2</sub>Se<sub>3</sub> are injected into ferromagnetic Co<sub>40</sub>Fe<sub>40</sub>B<sub>20</sub> layer and exert torques on it. It must be pointed out that the spin-orbit torques could be attributed to either the spin Hall effect (SHE) in the Bi<sub>2</sub>Se<sub>3</sub> bulk, Rashba-split states at the interface [29-31], or Bi<sub>2</sub>Se<sub>3</sub> topological surface states [23,27,28,31]. We find that  $\theta_{\parallel}$  drastically increases when the temperature decreases to ~ 50 K. As the temperature decreases furthermore,  $\theta_{\parallel}$  reaches up to ~ 0.42, which is ~ 10 times larger than

that at 300 K. In addition, a significant out-of-plane torque is extracted at low temperatures. We argue that our observations could be correlated with the TSS in our  $Bi_2Se_3/Co_{40}Fe_{40}B_{20}$  heterostructures.

20 quintuple layer (QL, 1 QL  $\approx$  1 nm) of Bi<sub>2</sub>Se<sub>3</sub> films are grown on Al<sub>2</sub>O<sub>3</sub> (0001) substrates using a custom designed SVTA MOSV-2 molecular beam epitaxy (MBE) system with a base pressure  $< 3 \times 10^{-10}$  Torr. The detailed procedures for Bi<sub>2</sub>Se<sub>3</sub> thin film growth can be found in previous reports [17,32]. The temperature dependent resistivity of Bi<sub>2</sub>Se<sub>3</sub> film is measured by four probe method. Figure 1(a) shows a typical characteristic of Bi<sub>2</sub>Se<sub>3</sub> that the sheet resistivity decreases as temperature decreases and then saturates at temperature < 30 K [17,33]. High resistivity Co<sub>40</sub>Fe<sub>40</sub>B<sub>20</sub> (CFB) is chosen as the FM layer in order to minimize the current shutting effect thru the FM layer. We have prepared five Bi<sub>2</sub>Se<sub>3</sub>/CFB (*t*) samples (thickness *t* = 1.5, 2, 3, 4 and 5 nm) and measured their magnetization response as a function of external magnetic field as plotted in Fig. 1(b). From the inset of Fig. 1(b), the CFB dead layer in Bi<sub>2</sub>Se<sub>3</sub>/CFB samples is estimated to be 1.36 nm, similar to a recent report in which the Co dead layer at the interface of Bi<sub>2</sub>Se<sub>3</sub>/Co is ~ 1.2 nm [34].

The ST-FMR devices are fabricated by the following process. First, a 5 nm CFB layer is sputtered onto the Bi<sub>2</sub>Se<sub>3</sub> film at room temperature with a base pressure of  $3 \times 10^{-9}$  Torr followed by a MgO (1 nm)/SiO<sub>2</sub> (3 nm) capping layer to prevent CFB from oxidation. Then the film is patterned into rectangular shaped microstrips (dotted blue line) with dimensions of L (130 µm) × W (10 – 20 µm) by photolithography and Ar ion milling as shown in Fig. 2(a). In the next step, coplanar waveguides (CPWs) are fabricated. Different gaps (10 – 55 µm) between ground (G) and signal (S) electrodes are designed to tune the device impedance ~ 50

Ω. A radio frequency (RF) current ( $I_{rf}$ ) with frequencies from 7 to 10 GHz and a nominal power of 15 dBm from a signal generator (SG, Agilent E8257D) is applied to the Bi<sub>2</sub>Se<sub>3</sub>/CFB bilayer via a bias-tee, and the ST-FMR signal ( $V_{mix}$ ) is detected simultaneously by a lock-in amplifier. An in-plane external magnetic field ( $H_{ext}$ ) is applied at a fixed angle ( $\theta_{H}$ ) of 35° with respect to the microstrip length direction [6]. We present the data from three different devices, denoted as D1, D2 and D3.

Figure 2(b) shows the measured ST-FMR signals from D1 at different temperatures ranging from 20 to 300 K.  $V_{\text{mix}}$  can be fitted by a sum of symmetric and antisymmetric Lorentzian functions,  $V_{\text{mix}} = V_s F_{\text{sym}}(H_{\text{ext}}) + V_a F_{\text{asym}}(H_{\text{ext}})$  [3,6,27]. From fitting, the symmetric component  $V_s$  (corresponding to in-plane torque  $\tau_{\parallel}$  on CFB) and antisymmetric component  $V_a$ (corresponding to total out-of-plane torque  $\tau_{\perp}$ ) can be determined, simultaneously.

The spin-orbit torque ratio from ST-FMR measurements can be characterized by two methods. One is obtain  $\theta_{\parallel}$ from the analysis of  $V_{\rm s}/V_{\rm a}$ to via  $\theta_{\parallel} = (V_s/V_a)(e\mu_0 M_s t d/\hbar) [1 + (4\pi M_{eff}/H_{ext})]^{1/2}$  [3], where t and d represent the thickness of the CFB and  $Bi_2Se_3$  layer, respectively.  $M_s$  is the saturation magnetization of CFB and  $M_{eff}$  is the effective magnetization. This method (denoted as 'by  $V_s/V_a$ ' hereafter) is to date widely used in ST-FMR measurements of heavy metals Pt (or Ta)/FM bilayers [3,6,8]. However, one assumption of this method is that the  $V_a$  is only attributed to the Oersted field induced out-of-plane torque. However, in the case of a TI, the TSS in TI and/or Rashba-split states at the interface could also contribute to  $V_{a}$ , therefore, we cannot estimate the actual  $\theta_{\parallel}$  value by  $V_{\rm s}/V_{\rm a}$ . On the other hand, the second method can avoid such an issue by analyzing only the symmetric component  $V_s$  (denoted as 'by  $V_s$  only' hereafter) using the following equations:

$$V_{\rm s} = -\frac{I_{\rm rf}\gamma\cos\theta_{\rm H}}{4}\frac{dR}{d\theta_{\rm H}}\tau_{\parallel}\frac{1}{\Delta}F_{\rm sym}(H_{\rm ext}), \quad \sigma_{\rm s} = J_{\rm s}/E = \tau_{\parallel}M_{\rm s}t/E, \text{ and } \theta_{\parallel} = \sigma_{\rm s}/\sigma \quad [6,27], \text{ where } I_{\rm rf}$$

is the RF current flowing through the device,  $dR/d\theta_{\rm H}$  is the angular dependent magnetoresistance at  $\theta_{\rm H} = 35^{\circ}$ ,  $\Delta$  is the linewidth of ST-FMR signal,  $F_{\rm sym}$  ( $H_{\rm ext}$ ) is a symmetric Lorentzian,  $\tau_{\parallel}$  is the in-plane spin-orbit torque on unit CFB moment at  $\theta_{\rm H} = 0^{\circ}$ ,  $\sigma_{\rm s}$ is the Bi<sub>2</sub>Se<sub>3</sub> spin Hall conductivity,  $\sigma$  is the Bi<sub>2</sub>Se<sub>3</sub> conductivity, and E is the microwave field across the device. The second method avoids the possible contamination to  $\theta_{\parallel}$  arising from  $V_{\rm a}$ , therefore we can extract the  $\theta_{\parallel}$  values in Bi<sub>2</sub>Se<sub>3</sub> by analyzing only  $V_{\rm s}$ . At the same time. the total out-of-plane torque can be derived by using  $\tau_{\perp}$  $V_{\rm a} = -\frac{I_{\rm rf}\gamma\cos\theta_{\rm H}}{4}\frac{dR}{d\theta_{\rm H}}\tau_{\perp}\frac{\left[1+(\mu_0M_{\rm eff}/H_{\rm ext})\right]^{1/2}}{\Delta}F_{\rm asym}(H_{\rm ext})$  [27], where  $F_{\rm asym}(H_{\rm ext})$  is an

antisymmetric Lorentzian.

Figure 3(a-b) show the  $\tau_{\parallel}$  and  $\tau_{\perp}$  as functions of temperature, respectively, using the 2<sup>nd</sup> method. Here, the  $\tau_{\parallel}$  ( $\tau_{\perp}$ ) represents the mean value for different RF frequencies. At 300 K, the  $\tau_{\parallel}$  is ~ 0.43 Oe for D1 (~ 0.84 Oe for D2 and ~ 0.48 Oe for D3). As the temperature decreases from 300 to 100 K,  $\tau_{\parallel}$  for all three devices gradually increases. At ~ 50 K,  $\tau_{\parallel}$  shows a steep increase and finally reaches ~ 5.25 Oe for D1 (~ 4.11 Oe for D2 and ~ 2.26 Oe for D3), which is ~ 10 times larger than that at 300 K. It is noteworthy that the observed drastic temperature dependent behavior of  $\tau_{\parallel}$  is different from the recently reported results in heavy metals such as Ta [10,35] as well as Pt [6,36,37], where the damping-like torque (equivalent to  $\tau_{\parallel}$  here), often argued to arise mainly from the SHE, shows a weak temperature dependence. This difference indicates the SHE mechanism may not account for the observed  $\tau_{\parallel}$  in our Bi<sub>2</sub>Se<sub>3</sub>/CFB. Moreover, the  $\tau_{\perp}$  shows a similar temperature dependent behavior as  $\tau_{\parallel}$ 

shown in Fig. 3(b). It is worth noting that the difference in  $\tau_{\parallel}$  (and  $\tau_{\perp}$ ) among D1, D2 and D3 can be attributed to the slight variation of the Bi<sub>2</sub>Se<sub>3</sub>/CFB interface during the fabrication process considering recent challenges in TI film growth and device fabrication. However, a qualitatively similar temperature dependence of torques is observed in all devices.

The  $\theta_{\parallel}$  values as a function of temperature determined by above two methods have been shown in Fig. 3(c). From analysis by  $V_s$  only,  $\theta_{\parallel}$  is ~ 0.047 for D1 (~ 0.113 for D2 and ~ 0.072 for D3) at 300 K, and increases to ~ 0.158 for D1 (~ 0.225 for D2 and ~ 0.149 for D3) as temperature decreases to 100 K. In this temperature range (100 - 300 K),  $\theta_{\parallel}$  has similar amplitudes as the spin Hall angle in heavy metals such as Pt, Ta, and W [3,8,11,42-44]. However,  $\theta_{\parallel}$  increases sharply as temperature decreases to ~ 50 K and reaches maximum values of ~ 0.42 for D1 (~ 0.44 for D2 and ~ 0.30 for D3) at lower temperatures, respectively. Remarkably,  $\theta_{\parallel}$  increases ~ 10 times compared to that at 300 K for D1. Similarly, from the analysis by  $V_s/V_{as}$ ,  $\theta_{\parallel}$  also shows an abrupt increase as temperature decreases to ~ 50 K in Fig. 3(c). It is worth noting that we use the effective CFB thickness of t = 3.64 nm due to the dead layer for  $\theta_{\parallel}$  estimation by  $V_s/V_a$  at different temperatures. Interestingly, as shown in Fig. 3(d), the ratio of  $[\theta_{\parallel}$  (by  $V_s$  only)  $\theta_{\parallel}$  (by  $V_s/V_a$ )] $/\theta_{\parallel}$  (by  $V_s/V_a$ ) obtained by two different methods increases as temperature decreases and becomes more significant below ~ 50 K, as discussed later.

In the context of spin Hall mechanism, the spin Hall angle ( $\theta_{sh}$ ) is found to be almost independent of temperature from Pt [6,36], Ta [45], Cu<sub>99.5</sub>Bi<sub>0.5</sub>, and Ag<sub>99</sub>Bi<sub>1</sub> [46], which is attributed to the extrinsic mechanisms. In some cases,  $\theta_{sh}$  shows a gradual increase as the temperature decreases, which behaves as a typical intrinsic mechanism based on the degeneracy of *d*-orbits by spin-orbit coupling [47,48]. In contrast, in our Bi<sub>2</sub>Se<sub>3</sub>/CFB, the spin-orbit torque ratio ( $\theta_{\parallel}$ ) shows an abrupt and nonlinear increase as temperature decreases, especially below ~ 50 K. Therefore, the SHE from the Bi<sub>2</sub>Se<sub>3</sub> bulk is probably not the dominant mechanism for our observation of temperature dependent spin-orbit torque (ratio) in Bi<sub>2</sub>Se<sub>3</sub>/CFB. From the measured ST-FMR signals as shown in Fig. 2(b), we also find that the Rashba-split state at the Bi<sub>2</sub>Se<sub>3</sub>/CFB interface is not the main mechanism for our observations, since the Rashba-split states lead to opposite direction (and sign) of charge current-induced spin polarization (and  $\theta_{\parallel}$ ) on the basis of the spin structure [27,31]. Instead, we ascertain that the direction of in-plane spin polarization to the electron momentum in our Bi<sub>2</sub>Se<sub>3</sub>/CFB is consistent with expectations of the TSS of TIs (spin-momentum locking) [12-14,27,31,37]. From further analysis [37], we have found that in our devices a large portion of the charge current flows through the TSS in Bi<sub>2</sub>Se<sub>3</sub>. The effective  $\theta_{\parallel}$  attributed to only TSS is in the range from ~ 1.62 ± 0.18 to ~ 2.1 ± 0.39.

As mentioned before, the temperature dependent  $\theta_{\parallel}$  obtained from the above two methods shown in Fig. 3(c) should not show any difference, if  $V_a$  is attributed to only the charge current induced Oersted field. Therefore, the observed difference implies the existence of other contributions to  $V_a$  (i.e. to  $\tau_{\perp}$ ). For the Bi<sub>2</sub>Se<sub>3</sub>/CFB system, the difference can be attributed to the TSS in Bi<sub>2</sub>Se<sub>3</sub> [23,27,28,31] and/or Rashba-split states at the Bi<sub>2</sub>Se<sub>3</sub>/CFB interface [29-31]. We analyze  $\Delta \tau = \tau_{\perp} - \tau_{Oe}$  as the other contributions to the out-of-plane torque, where  $\tau_{\perp}$  is the total out-of-plane torque as shown in Fig. 3(b), and  $\tau_{Oe}$  is a partial out-of-plane torque from charge current (flowing in Bi<sub>2</sub>Se<sub>3</sub>) induced Oersted field. By using the measured  $\theta_{\parallel}$  by  $V_s$  only, we can deduce  $\tau_{Oe}$  and thus  $\Delta \tau$  by

$$\theta_{\parallel} = (V_{\rm s}/V_{\rm a}^{*})(e\mu_{0}M_{\rm s}td/\hbar)[1 + (4\pi M_{\rm eff}/H_{\rm ext})]^{1/2} , \qquad \text{and}$$

$$V_{\rm a}^* = -\frac{I_{\rm rf}\gamma\cos\theta_{\rm H}}{4}\frac{dR}{d\theta_{\rm H}}\tau_{\rm Oe}\frac{\left[1+(\mu_0M_{\rm eff}/H_{\rm ext})\right]^{1/2}}{\Delta}F_{\rm asym}(H_{\rm ext}) \quad [3,27], \quad \text{where} \quad V_{\rm a}^* \quad \text{is the}$$

equivalent antisymmetric component only due to the current induced Oersted field ( $\tau_{\text{Oe}}$ ). As shown in Fig. 4(a), the out-of-plane torque ( $\Delta \tau$ ) in all three devices becomes much larger at low temperatures < 50 K, compared to the  $\Delta \tau$  at high temperatures (100 – 300 K). Consequently, we can obtain the out-of-plane spin-orbit torque ratio ( $\theta_{\perp}$ ) as a function of temperature by using the same method by which we deduce  $\theta_{\parallel}$  from  $\tau_{\parallel}$  above. As shown in Fig. 4(b), we find that  $\theta_{\perp}$  in all three devices also becomes more significant at low temperatures (< 50 K). More interestingly, the  $\theta_{\perp}$  almost has the same order of magnitude compared to  $\theta_{\parallel}$ .

We now discuss the origin of the out-of-plane torque. As has been reported recently, a Rashba-split surface state in two dimensional electron gas (2DEG) coexists with TSS in the Bi<sub>2</sub>Se<sub>3</sub> surface due to the band bending and structural inversion asymmetry [29,30,49-52]. The Rashba effective magnetic field can be written as  $H_T = \alpha_R/\hbar(\hat{z} \times k)$  [49-51], where  $\hat{z}$ is a unit vector normal to film plane, *k* is the average electron Fermi wavevector, and  $\alpha_R$  is a characteristic parameter of the strength of Rashba splitting in 2DEG. Since the electron Fermi wavevector can be assumed to show a weak temperature dependence and the  $\alpha_R$ decrease as temperature decreases in a typical 2DEG [53,54],  $H_T$  is expected to decrease as temperature decreases in these semiconductor systems. In addition, the similar temperature dependent behavior of  $H_T$  has been recently reported in Ta/CoFeB heterostructures, where  $H_T$ decreases and eventually almost reaches to zero at low temperatures [10,35]. However, the observed  $\Delta \tau$  (equivalent to  $H_T$ ) in our Bi<sub>2</sub>Se<sub>3</sub>/CFB presents the opposite temperature dependent behavior which is not in line with the reports about Rashba induced torques. Therefore, we conclude that the Rashba-split surface state in 2DEG of  $Bi_2Se_3$  is not the main mechanism for the out-of-plane torque ( $\Delta \tau$ ).

On the other hand, a possible out-of-plane spin polarization in the TSS has been theoretically predicted [55,56] and experimentally observed in Bi<sub>2</sub>Se<sub>3</sub> [57,58], which is attributed to the hexagonal warping effect in the Fermi surface [55,59]. This out-of-plane spin polarization in the TSS can account for the observed  $\Delta \tau$  especially in the low temperature range (< 50 K) and the  $\Delta \tau$  adds to the  $\tau_{0e}$  [27,31]. Moreover, as shown in Fig. 3(a) and 4(a), the out-of-plane torque ( $\Delta \tau$ ) has the same order of magnitude comparable to in-plane torque ( $\tau_{\parallel}$ ) below 50 K ( $\Delta \tau / \tau_{\parallel} \sim 60\%$ ) [37], which is in agreement with the behavior of hexagonal TSS in TI [55,56]. With the analysis from different aspects, our findings especially in the low temperature range (< 50 K) indicate a TSS origin of spin-orbit torques in Bi<sub>2</sub>Se<sub>3</sub>/CFB.

In summary, we have studied the temperature dependence of spin-orbit torques in  $Bi_2Se_3/CoFeB$  heterostructures. As temperature decreases, the spin-orbit torque ratio increases drastically and eventually reaches a maximum value of ~ 0.42, which is almost 10 times larger than that at 300 K. A significant out-of-plane torque ( $\Delta \tau$ ), in addition to charge current induced Oersted field torque ( $\tau_{Oe}$ ), can be observed below 50 K. The observed spin-orbit torques are attributed to the topological surface states in  $Bi_2Se_3$ . Our results suggest that topological insulators with strong spin-orbit coupling and spin-momentum locking are promising spin current sources for next generation of spintronic devices.

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### **Figure captions**

FIG. 1. (a) Temperature dependent sheet resistivity of  $Bi_2Se_3$  films (20 QL). (b) The magnetization versus field (*H*) for  $Bi_2Se_3$  (20 QL)/CFB (*t*) (*nominal* thickness *t* =1.5, 2, 3, 4 and 5 nm) at room temperature. The inset shows the magnetization per unit area versus CFB thickness.

FIG. 2. (a) The schematic diagram of the ST-FMR measurement, illustrating a bias-tee, lock-in amplifier, RF signal generator (SG), and ST-FMR device with a  $Bi_2Se_3/CFB$  (5 nm). Micro-strip is denoted by a dashed blue rectangle. (b) The measured ST-FMR signals from a  $Bi_2Se_3/CFB$  (5 nm) device (D1) at different temperatures.

FIG. 3. Temperature dependence of (a)  $\tau_{\parallel}$ , (b)  $\tau_{\perp}$ , (c)  $\theta_{\parallel}$ , and (d) [ $\theta_{\parallel}$  (by  $V_{\rm s}$  only)  $\theta_{\parallel}$  (by  $V_{\rm s}/V_{\rm a}$ )]/[ $\theta_{\parallel}$  (by  $V_{\rm s}/V_{\rm a}$ )] in Bi<sub>2</sub>Se<sub>3</sub>/CFB (5 nm) for D1, D2, and D3. The  $\theta_{\parallel}$  is analyzed by two different methods, by ' $V_{\rm s}$  only' and by ' $V_{\rm s}/V_{\rm a}$ '.

FIG. 4. (a) Temperature dependent out-of-plane torque ( $\Delta \tau = \tau_{\perp} - \tau_{Oe}$ ) and (b) out-of-plane torque ratio ( $\theta_{\perp}$ ) in Bi<sub>2</sub>Se<sub>3</sub>/CFB (5 nm) devices.



FIG. 1



FIG. 2



FIG. 3



FIG. 4