This is the accepted manuscript made available via CHORUS. The article has been published as:

Inhomogeneous Relaxation of a Molecular Layer on an Insulator due to Compressive Stress

F. Bocquet, L. Nony, S. C. B. Mannsfeld, V. Oison, R. Pawlak, L. Porte, and Ch. Loppacher

Phys. Rev. Lett. **108**, 206103 — Published 16 May 2012

DOI: 10.1103/PhysRevLett.108.206103

Inhomogeneous relaxation of a molecular layer on an insulator due to compressive stress

F. Bocquet[1], L. Nony[1]*, S.C.B. Mannsfeld[2], V. Oison[1], R. Pawlak[1], L. Porte[1], and Ch. Loppacher[1]
[(1)] *Aix-Marseille Université, IM2NP, Av. Normandie-Niemen, Case 151, F-13397 Marseille CEDEX 20, France; CNRS, IM2NP (UMR 7334), Marseille-Toulon, France;* [(2)] *Stanford Synchrotron Radiation Laboratory (SSRL), Stanford, CA 94305, United States.*
(Dated: March 23, 2012)

We discuss the inhomogeneous stress relaxation of a monolayer of hexahydroxytriphenylene (HHTP) which adopts the rare line-on-line (lol) coincidence on KCl(001) and forms Moiré patterns. The fact that the hexagonal HHTP layer is uniaxially compressed along the lol makes this system an ideal candidate to discuss the influence of inhomogeneous stress relaxation. Our work is a combination of noncontact atomic force microscopy experiments, of density functional theory and potential energy calculations, and of a thorough interpretation by means of the Frenkel-Kontorova model. We show that the assumption of a homogeneous molecular layer is not valid for this organic-inorganic heteroepitaxial system since the best calculated energy configuration correlates with the experimental data only if inhomogeneous relaxations of the layer are taken into account.

PACS numbers: 68.37.Ps; 68.43.Hn; 68.60.Bs; 33.15.Fm

Highly-ordered π-conjugated organic thin films grown on surfaces by molecular beam epitaxy represent a rewarding approach to new classes of heterojunctions for organic semiconductors with promising applications in optoelectronics [1]. In this regard, physisorption of organic molecules on *insulating substrates* is of inherent interest. Nevertheless, little is known about organic-inorganic heteroepitaxy (OIHE) on single-crystal insulators since only a few systems were investigated so far, most of them on alkali halide substrates [2–5]. OIHE is governed by a delicate balance between molecule/molecule interactions involving supramolecular interactions, and molecule/substrate interactions. For polar molecules on alkali halides the latter ones are a combination of van der Waals (vdW) interactions acting like an unstructured background, and electrostatic (ES) interactions governed by the periodicity of the ionic surface. If the physisorption of large molecular layers on insulators is mainly driven by vdW interactions, then the layer can be regarded as a floating, i.e. with no particular anchoring sites, 2D periodic overlayer with a structure close to a crystallographic plane of the organic bulk crystal. Now, if the ES interaction becomes significant, site-specific adsorption can determine the growth mode [2, 3]. In this case, the organic overlayer adsorbs on the substrate in a way to find the best lattice match. This process does not necessarily generate a commensurate molecular supercell, but may also promote more complex epitaxies such as the line-on-line (lol) epitaxy [6–8]. Although coincidence between the organic layer and the substrate maximizes its adsorption energy (stabilizing effect), it may as well introduce stress within the layer (destabilizing effect). For systems with large misfit between adsorbate bulk and substrate lattices, the stress may become so huge that the layer becomes unstable and dewetting takes place [3]. Nevertheless, for intermediate misfits, efficient stress release may be achieved by local relaxations within the layer (nearest neighbor scale) which would provoke the layer to deviate from its periodic, and hence homogeneous, 2D bulk structure. This is known to occur in metallic heterostructures by means of misfit dislocations. Surprisingly, for OIHE where conformational adaptation and relative flexibility of the supramolecular network might favor local inhomogeneity, to our knowledge, only few attempts were done to account for this effect so far [9, 10].

In this Letter, we study the model system 2,3,6,7,10,11 hexahydroxytriphenylene (HHTP)/KCl(001) in which the molecular monolayer is not dewetting despite an uniaxial compression of 4%. We demonstrate that such a large compressive stress can be energetically favorable if local, inhomogeneous, relaxations are taken into account. These are treated within the framework of the 1D Frenkel-Kontorova model [11].

HHTP (see inset in Fig.1A) is a planar molecule with three-fold symmetry and 6 peripheral hydroxyl groups. Crystallographic data are only reported for the hydrated crystal which forms a monoclinic unit cell (u.c.) [12]. HHTP is known as a prototypical discotic liquid crystal [13]. Its discotic nature combined with its hydroxyl groups make it well-suited to promote 2D self-organization. Indeed, on metals, HHTP forms extended and highly organized layers which are linked by supramolecular [14], or covalent interactions [15].

Sub-monolayers of HHTP were evaporated on KCl(001) single crystals in ultra-high vacuum according to standard procedure (see e.g. ref. [5] and supporting information (SI) for details) and imaged at room temperature by noncontact atomic force microscopy (nc-AFM). HHTP forms stable and extended 2D molecular monolayers which exhibit 3 orientations (see arrows 1,

*Corresponding author: laurent.nony@im2np.fr.

2, and 3 in Fig.1A). The apparent height of the layers is 370 pm, which indicates that molecules are lying flat on the substrate. Henceforth, all crystallographic axes related to the substrate and the molecular overlayer are subscripted s and m, respectively; the KCl(001) surface is described by its square primitive u.c. High-resolution images, i.e. properly drift-corrected and rescaled with respect to (w.r.t) the KCl(001) u.c., acquired on different domains reveal the same quasi-hexagonal molecular arrangement (see Fig.1B). Soft image processing (FFT and autocorrelation) permits to accurately measure the molecular u.c.: $\mathbf{a_{1m}}$ (along $\langle 10\rangle_m$) measures $\|\mathbf{a_{1m}}\| = (11.0 \pm 0.3)$ Å, and $\mathbf{a_{2m}}$ (along $\langle 01\rangle_m$) measures $\|\mathbf{a_{2m}}\| = (11.5 \pm 0.3)$ Å. The angle between these vectors is $(\widehat{\mathbf{a_{1m}}, \mathbf{a_{2m}}}) = (120 \pm 3)°$. Combined analysis in the reciprocal and direct space (see Fig.1B to 1D) shows that the $\mathbf{a_{1m}}$ vector is $-(27 \pm 3)°$ tilted w.r.t the $\langle 10\rangle_s$ axis. This direction matches the $\langle 2\bar{1}\rangle_s$ axis, the exact orientation of which is given by $-26.6°$. Therefore, the dense $\langle 10\rangle_m$ molecular rows exactly lie on non primitive $\langle 2\bar{1}\rangle_s$ substrate rows. Moreover, the distance between two adjacent $\langle 10\rangle_m$ rows, $d_m = \|\mathbf{a_{2m}}\| \cos(30°) = 9.96$ Å, matches within 0.1% the distance between five substrate $\langle 2\bar{1}\rangle_s$ rows (see Fig.1D, the distance between adjacent $\langle 2\bar{1}\rangle_s$ rows is $d_s = 4.45/\sqrt{2^2+1^2} = 1.99$ Å, where 4.45 Å is the u.c. parameter of KCl(001), hence $5\times d_s = 9.95$ Å). The epitaxy of HHTP on KCl(001) is therefore an exact lol epitaxy [7, 8] with $\langle 10\rangle_m \| \langle 2\bar{1}\rangle_s$.

A striking feature of the experimental images are the periodic fringes which appear in the domains and form well-defined Moiré patterns with a periodicity of $a_M = (45.5 \pm 1)$ Å and a tilt of $(100 \pm 3)°$ w.r.t. the $\langle 10\rangle_m$ axis (see Fig.1A and Fig.2A). These domains are all equivalent as their relative orientations can be deduced by applying the symmetry element of the punctual group of the substrate. A recurrent observation is that domain borders (strong adsorption sites) are not oriented along dense molecular rows, but rather along Moiré fringes (see yellow arrow in Fig.1A), which suggests that the fringes result from coincidences (or quasi-coincidences) between the molecular and the substrate lattice points. To address that issue, simulations of the Moiré pattern were performed by superposing the HHTP and KCl lattices [7] (see Fig.2B). The contrast of the resulting pattern spreads from black (100% coincident) to white (0% coincident). To properly reproduce the experimental Moiré pattern (periodicity and orientation of the fringes), the molecular u.c. has to be set to $\|\mathbf{a^{sim}_{1m}}\| = 11.15$ Å, $\|\mathbf{a^{sim}_{2m}}\| = 11.28$ Å, and $(\widehat{\mathbf{a^{sim}_{1m}}, \mathbf{a^{sim}_{2m}}}) = 118.1°$. This structure depicts a slightly distorted hexagonal arrangement; in agreement with the experimental data and the lol epitaxy. Any tiny change of these values drastically changes the simulated pattern which then becomes incompatible with the experimental fringes. Both, the Moiré fringes and the bright spots visible in the autocorrelation images (circles in Fig.2A) are reproduced in the simulated pattern, which therefore testifies that quasi-coincidences occur between the molecular overlayer and the substrate lattice points.

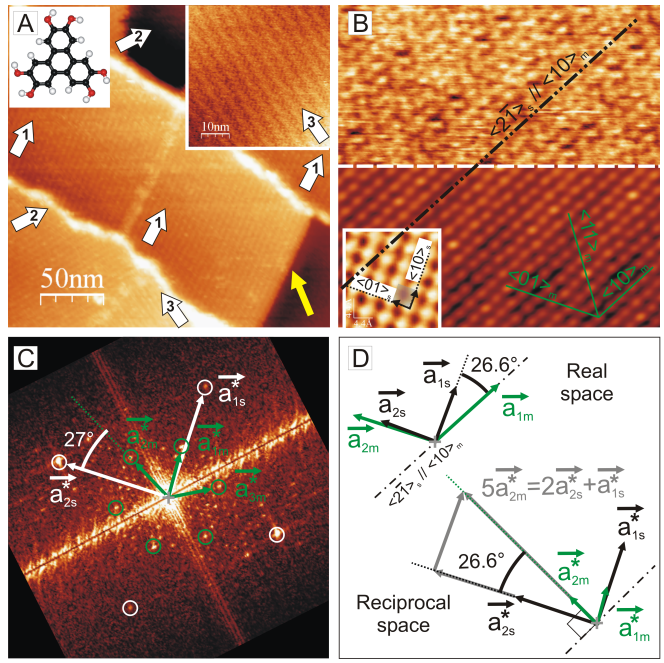

FIG. 1: (Color online) A- Molecular domains of HHTP on KCl(001) (300 × 300 nm^2, normalized Δf: $\gamma = -0.03$ nN.$\sqrt{\text{nm}}$). The domains consist of fringes forming a Moiré pattern with a periodicity of 45.5 Å. Three distinct orientations are visible (numbered arrows). Insets: HHTP molecule (left) and zoom on a type 3 domain showing both, the fringes orientation and a molecular axis (right, 40 × 40 nm^2, $\gamma = -0.12$ nN.$\sqrt{\text{nm}}$). B- Zoom on a molecular domain (25 × 25 nm^2, $\gamma = -0.27$ nN.$\sqrt{\text{nm}}$). Upper part: acquired topography signal. Lower part: topography signal autocorrelation showing the molecular hexagonal arrangement in a lol epitaxy ($\langle 10\rangle_m$ and $\langle 2\bar{1}\rangle_s$ are parallel). Inset: KCl imaged close to the molecular domain (2.2 × 2.2 nm^2, $\gamma = -0.33$ nN.$\sqrt{\text{nm}}$). C- FFT showing both the diffraction spots of HHTP and of KCl (yellow and white circles, respectively): raw data in Fig.S1 of SI. D- Sketch illustrating the correspondence between real and reciprocal space. The lol results from a coincidence of the ends of molecular and substrate reciprocal vectors: $5\mathbf{a_{2m}} = 2\mathbf{a_{2s}} + \mathbf{a_{1s}}$.

To investigate the energy configuration of the layer, Density Functional Theory (DFT) and Potential Energy (PE) calculations were carried out (see Fig.3 and SI). DFT calculations give the most stable molecular conformation and its corresponding freestanding layer configuration (see inset in Fig.1A and Fig.3A, respectively). The molecules assemble in a planar and regular hexagonal u.c. via three H-bonds. No molecular orbitals hybridization takes place but partial charges are present within the molecule (see SI). The cohesion energy of the structure is $E_{\text{coh},0} = -0.36$ eV/molecule and its uniaxial stiffness along the $\langle 10\rangle_m$ axis is estimated to be $k_{\text{DFT}} = 15.3$ N/m

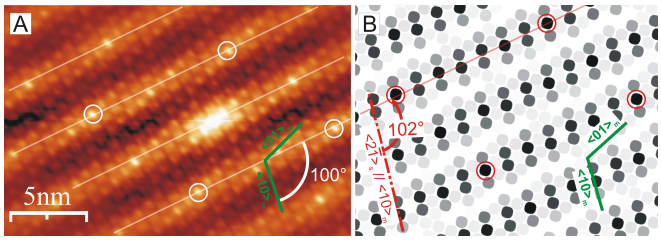

FIG. 2: (Color online) A- Self-correlation image of a molecular domain showing Moiré fringes and periodic bright spots. The orientation of the fringes w.r.t. $\langle 10\rangle_m$ is $(100 \pm 3)^\circ$. B- Simulated Moiré pattern which reproduces both the experimental fringes and the bright spots, respectively.

(fit of the data shown in Fig.3B with a harmonic potential). The parameters of the u.c. optimized by DFT are: $\|\mathbf{a}^{\mathrm{DFT}}_{\mathbf{1m},\mathbf{2m}}\| = 11.62$ Å and $(\widehat{\mathbf{a}^{\mathrm{DFT}}_{\mathbf{1m}}, \mathbf{a}^{\mathrm{DFT}}_{\mathbf{2m}}}) = 120^\circ$. *A posteriori*, the experimentally observed lol epitaxy can now be justified since the distance between two dense molecular rows of the freestanding layer, $\|\mathbf{a}^{\mathrm{DFT}}_{\mathbf{2m}}\| \cos(30^\circ) = 10.06$ Å, fits within 1% to five distances between two adjacent $\langle 2\bar{1}\rangle_s$ rows. However, unlike other molecular films on KCl [5], the equilibrium configuration of the freestanding layer does not match the experimental one along the $\langle 2\bar{1}\rangle_s$ direction, but shows a significant discrepancy with a compressive stress of $(a^{\mathrm{sim}}_{1m} - a^{\mathrm{DFT}}_{1m})/a^{\mathrm{DFT}}_{1m} = -4.0\%$.

Such a 4% uniaxial compressive stress is unexpected since (i) along the lol direction, an almost perfect geometric coincidence could be achieved for 7 molecules (6×11.62 Å) lying over 8 substrate ions along the $\langle 2\bar{1}\rangle_s$ axis (7×9.95 Å) giving a nearly unstressed layer (0.1 % misfit), and (ii) the stress-induced elastic energy stored in the layer decreases by a large amount (35%) the cohesive energy w.r.t. the freestanding overlayer ($E_{\mathrm{el}} = \frac{1}{2} k_{\mathrm{DFT}} (a^{\mathrm{sim}}_{1m} - a^{\mathrm{DFT}}_{1m})^2 = 0.112$ eV). Only a sufficiently large and in-plane corrugated molecule/substrate ES interaction could explain the observed compression.

As both, large clusters and interactions with an ionic substrate are properly treated by PE calculations [7, 16], we proceeded as follows: first, the freestanding layer was calculated. The PE results are in excellent agreement with the DFT calculations. Second, the adhesion energy for a single molecule kept at its optimal orientation, $\alpha_{\mathrm{opt}} = 7^\circ$ w.r.t. $\langle 2\bar{1}\rangle_s$, is calculated for different lateral positions (see Fig.3C). The obtained energy landscape can be interpreted as a constant vdW background of 0.76 eV plus a locally modulated part of ± 0.24 eV due to ES interactions. Third, calculations involving a large cluster (3600 molecules) were done using, as input parameters, the values derived from the simulated Moiré pattern. This structure corresponds to a local energetic minimum and gives, when the whole 3600 molecules are translated in block over the surface, the energy landscape shown in Fig.3D. The low energetic corrugation (4 meV) perpendicularly to the $\langle 2\bar{1}\rangle_s$ axis is sufficiently large to

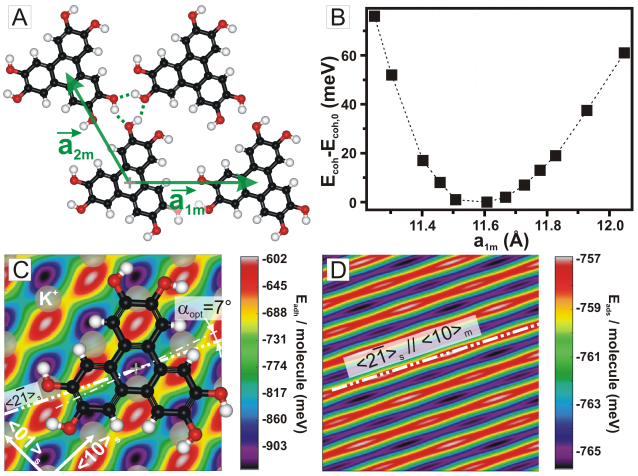

FIG. 3: (Color online) A- Unit cell of the HHTP freestanding layer as calculated by DFT. A H-bonded (dotted green lines) hexagonal structure with $\|\mathbf{a}^{\mathrm{DFT}}_{\mathbf{1m},\mathbf{2m}}\| = 11.62$ Å is found. B- Evolution of the cohesion energy per molecule as a function of a^{DFT}_{1m} (uniaxial stress), from which the uniaxial stiffness $k_{\mathrm{DFT}} = 15.3$ N/m of the layer is derived. C- Adhesion energy of a single HHTP adsorbed on KCl(001), as calculated by PE calculations. The constant background of -0.76 eV (mainly vdW interaction) is laterally corrugated due to ES molecule/substrate interaction. D- Adsorption energy per molecule for a cluster of 3600 molecules when translated with fixed angular orientation. The remaining 1D potential is 4 meV deep and carries the lol epitaxy direction.

promote the lol epitaxy [7]. However, when this structure is fully relaxed in order to find an absolute minimum, the calculations converge towards the quasi freestanding layer with the lol orientation maintained. Therefore the PE method, which strictly describes a homogeneous cluster, fails to predict the experimental stress along the $\langle 10\rangle_m$ axis. One possibility to relieve such stress is to consider a pseudo-periodic, inhomogeneous, 2D molecular structure. Describing large molecular domains with 2D local relaxations, including eventually variable orientations (α), is out of the scope of this letter. Nevertheless, as the experimentally observed stress is uniaxial, we can at least semi-quantitatively discuss the problem in the 1D Frenkel-Kontorova (FK) formalism [11], the details of which are given in SI.

The purpose of the FK model is to find the best equilibrium configuration for a set of harmonically interacting "point-masses" (here molecules initially separated by 11.62 Å) constrained in a 1D periodic potential (here the ES potential) by taking into account local variations of next neighbor distances. We assume the overlayer as a set of $\langle 10\rangle_m$ non-interacting parallel and uniaxially compressed 1D molecular chains. Along one chain, two next neighbors are linked by a spring with stiffness $k = k_{DFT}/3$ because a molecule possesses only 2 neigh-

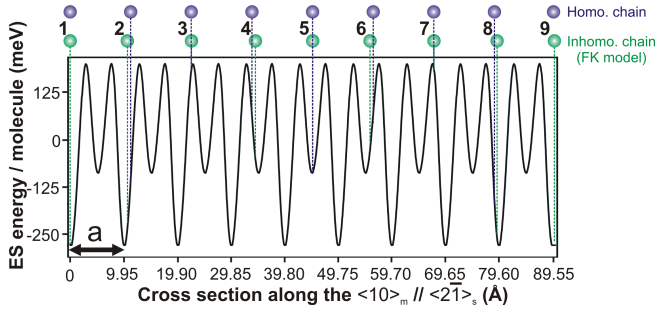

FIG. 4: (Color online) 9 molecules are forced within the distance of 10 minima of the ES energy profile along the $\langle 2\bar{1}\rangle_s$ for a single molecule (see Fig.3C). The adsorption energy of the molecules with even number is significantly larger when the molecules are spaced according to the Frenkel-Kontorova model (lower ones) compared to the equally spaced molecules (upper ones, homogeneous chain).

bors, despite 6 in the u.c. The molecules are exposed to an ES potential given by the one taken along the $\langle 2\bar{1}\rangle_s$ axis in Fig.3C and shown in Fig.4. Its periodicity a is obviously 9.95 Å. In order to simplify the approach, the potential is fitted by a sinusoidal curve with a peak-to-peak amplitude $W = 0.40$ eV (see SI). The key parameter of the FK model which compares the elastic and the adsorption energy is given by $l_0^{\text{theo}} = \sqrt{ka^2/2W} = 6.4$. Along the $\langle 2\bar{1}\rangle_s$ direction, each molecule in the chain is supposed to be identically oriented, i.e. with the optimal angle α_{opt}. Note that any local orientational variation (within a range of e.g. $\pm 5°$) between two adjacent molecules may additionally increase the adsorption energy [17], which should be considered for further studies.

Accordingly to the data needed to simulate the Moiré pattern (i.e. with an intermolecular distance of 11.15 Å), we force 4%-compression along a molecular chain by distributing 9 molecules on 10 ES potential minima along the $\langle 2\bar{1}\rangle_s$ axis (8 × 11.62 Å on 9 × 9.95 Å). This yields $l_0 = 4.0$ and an averaged intermolecular distance of 11.19 Å instead of $l_0^{\text{theo}} = 6.4$ and 11.15 Å, respectively. The FK model gives the calculated positions of the molecules along the potential as depicted in Fig.4. Note that the local relaxation is reflected by a regular alternation of always two shortened and two prolonged intermolecular distances along the stressed line. The energy per molecule of this *relaxed* configuration is $E_{\text{rel}}^{(1)} = -34$ meV, while for an *homogeneously* 4%-compressed chain, it is $E_{\text{hom}} = 10$ meV. Therefore the inhomogeneous relaxation allows for an energy gain of $(E_{\text{hom}} - E_{\text{rel}}^{(1)}) \simeq 44$ meV per molecule. The ultimate test to validate our model of inhomogeneous relaxation is to see whether the structure displayed in Fig.4 remains energetically favorable compared to a nearly unstressed one (7 molecules over 8 substrate periods along the $\langle 2\bar{1}\rangle_s$ axis). For this structure, the FK model yields $l_0 = 8.7$ and inhomogeneous relaxation with an energy gain $E_{\text{rel}}^{(2)} = -6$ meV. Therefore, the energetically most favorable molecular structure is the 4%-compressed, but *inhomogeneously relaxed* layer, which is compliant with the experimental data.

In conclusion, we studied the adsorption of a monolayer of hexahydroxytriphenylene (HHTP) on KCl(001). To our knowledge, this is the first example where an organic layer adopts the rare line-on-line (lol) epitaxy on an ionic substrate. The fact that the hexagonal HHTP layer is uniaxially compressed along the lol direction made this system an ideal candidate to discuss the influence of stress-induced local relaxation, a point which was neglected in organic overlayers so far. Our results demonstrate that the observed compression of the organic layer becomes energetically favorable only and only if local inhomogeneous relaxations are considered. In more general, the adsorption energy of a large molecular cluster in a lol epitaxy does not vary upon changes of the molecular positions along the lol (see Fig. 3D). Thus, local relaxations are facilitated along this direction. The main criterion for the formation of a lol epitaxy is that along the other molecular directions, the distance between dense molecular rows of the freestanding layer and the one between substrate rows match closely.

F.B., L.N. and Ch.L. thank S.Clair and J.-M.Themlin for useful discussions and acknowledge support from the ANR and the PNano MolSiC (ANR-08-P058-36) and Nanokan (ANR-11-BS10-004) programs. This work was done in part at Stanford Synchrotron Radiation Lightsource (SSRL), operated by Department of Energy.

[1] R. Otero *et al.*, Advanced Materials **23**, 5148 (2011).
[2] T. Dienel *et al.*, Advanced Materials **20**, 959 (2008).
[3] S. Burke *et al.*, Phys. Rev. Lett. **100**, 186104 (2008).
[4] D.-M. Smilgies and J. E.J. Kintzel, Phys. Rev. B **79**, 235413 (2009).
[5] R. Pawlak *et al.*, J. Phys. Chem. C **114**, 9290 (2010).
[6] D. Hooks, T. Fritz, and M. Ward, Adv. Mat. **13**, 227 (2001).
[7] S. Mannsfeld, K. Leo, and T. Fritz, Phys. Rev. Lett. **94**, 056104 (2005).
[8] D. Kasemann *et al.*, Langmuir **25**, 12569 (2009).
[9] V. Oison, M. Koudia, M. Abel, and L. Porte, Phys. Rev. B **75**, 035428 (2007).
[10] P. Fenter *et al.*, Physica B **221**, 145 (1996).
[11] Y. I. Frenkel and T. Kontorova, Zh. Eksp. Teor. Fiz. **8**, 1340 (1938).
[12] T. Andresen, F. Krebs, N. Thorup, and K. Bechgaard, Chem. Mater. **12**, 2428 (2000).
[13] D. Adam *et al.*, Phys. Rev. Lett. **70**, 457 (1993).
[14] R. Pawlak *et al.*, ChemPhysChem **10**, 1032 1035 (2009).
[15] N. Zwaneveld *et al.*, Journal of American Chemical Society **130**, 6678 (2008).
[16] S. Mannsfeld and T. Fritz, Phys. Rev. B **71**, 235405 (2005).
[17] J.-Z. Wang *et al.*, Phys. Rev. B **83**, 235433 (2011).