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Strain-stiffening in random packings of entangled granular chains

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Random packings of granular chains are presented as a model system to investigate the contribution of entanglements to strain-stiffening. The chain packings are sheared in uniaxial compression experiments. For short chain lengths, these packings yield when the shear stress exceeds the scale of the confining pressure, similar to granular packings of unconnected particles. In contrast, packings of chains which are long enough to form loops exhibit strain-stiffening, in which the effective stiffness of the material increases with strain, similar to many polymer materials. The latter packings can sustain stresses orders-of-magnitude greater than the confining pressure, and do not yield until the chain links break. X-ray tomography measurements reveal that the strain-stiffening packings contain system-spanning clusters of entangled chains.

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Most materials become weaker the further they are strained. In packings of granular materials, in particular, the maximum sustainable stress is limited by interparticle attraction forces or an external confining pressure which hold the structure together along force chains that span the system. Here we report experiments on packings of granular chains consisting of beads connected by flexible links and show that they exhibit the radically different behavior of strain stiffening, whereby the effective stiffness increases as the material is further strained, and they can sustain stresses that far exceed the contributions from attractions and external confinement.

Strain-stiffening is known to occur in many polymeric materials [1]. Theories suggest strain-stiffening could depend on many factors including chain stiffness, density, temperature, strain rate, and in particular on structures such as entanglements between different chains [2– 5]. However, it has not been possible to directly measure entanglements in polymer experiments because these structures are very small. In this letter, we investigate the role of entanglements in granular strain-stiffening. Macroscopic chains have several advantages over molecular polymer systems for investigating entanglements. First, the macroscopic size allows for imaging the structure to measure the precise positions of each particle and link. Second, we can isolate entanglement effects from temperature and strain rate dependent effects because the macroscopic chains have no inherent time scales due to Brownian motion or relaxation.

Previous work on granular chains focused on their packing structure near the jamming transition, but did not address their response to stress [6–8]. We go beyond those works by first demonstrating that packings of granular chains of sufficient length exhibit strain-stiffening and can sustain stresses orders-of-magnitude greater than those of unlinked granular materials. We then use x-ray tomography to measure the precise packing structure and identify entanglements to demonstrate quantitative connections between entanglements and strain-stiffening.

As a model material, we used macroscopic granular chains consisting of hollow spherical brass beads shown

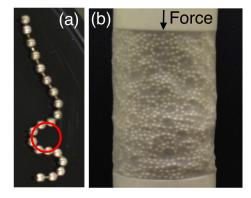


FIG. 1: (a) An individual chain with length N = 20 beads. The minimum loop size of $\xi = 8$ beads is highlighted by the red circle. (b) Cylindrical packing of chains with $N \gg \xi = 8$ inside a thin latex membrane used for uniaxial compression measurements. Many tight loops near the minimum loop size can be seen in the packing through the membrane.

in Fig. 1a. The beads are flexibly connected into chains by enclosing dog-bone-shaped brass links. These connections have essentially zero stiffness for small bend angles, but they have a maximum bend angle beyond which plastic deformation occurs. This defines a minimum loop circumference ξ the chains can bend into, as shown in Fig. 1a. We used two sets of chains with beads of slightly different diameters of 1.9 mm and 2.1 mm which causes them to have different minimum loop circumferences of $\xi = 8$ and $\xi = 11$ beads, respectively [9].

We measured the stress response of granular chain packings under uniaxial compression [10]. The grains were poured into a flexible elastic membrane in a cylindrical shape with solid end caps on the top and bottom, as shown in Fig. 1b. The chains can be seen to form tight loops within the packing, as small as the minimum loop circumference [7]. The membrane allows for free radial expansion with just enough confining stress to hold up the sample. An Instron materials tester was used to quasi-statically push the top end cap downward while

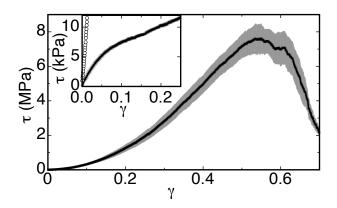


FIG. 2: Stress τ vs. strain γ for packings of chains with length $N \approx 10^4$. The shaded bands represent 1 standard deviation based on 5 repeated measurements. Strong strain-stiffening is seen as the positive curvature in $\tau(\gamma)$. Inset: $\tau(\gamma)$ for N = 1 (unlinked beads). Open symbols: reproduction of $N \approx 10^4$ data for comparison on the smaller scale.

measuring its height and the required force for samples with diameter and initial height of 50 mm. This average compressive stress τ is calculated as the force over the cross-sectional area of an end cap and the compressive strain γ is calculated as the height change (positive downward) over the initial height.

We show in Fig. 2 the stress-strain relation $\tau(\gamma)$ for a packing of chains with length $N \approx 10^4$ beads, much longer than their minimum loop circumference $\xi = 8$. Enormous strain-stiffening can be seen as the region of positive curvature of $\tau(\gamma)$. A striking feature is that the stress reaches the order of 10 MPa before the packing fails, which is indicated by the flattening of the stressstrain relation where the packing is able to shear without supporting additional stress. This ultimate strength is on the order of 1/10 the strength per density of the solid brass that makes up the chains (this packing is about 20% brass by volume) and comparable to polymer rubbers. In contrast, unconnected granular materials typically fail when the shear stress exceeds the confining pressure at the boundary in the absence of interparticle attractions [10]. This is shown for N = 1 (unlinked beads) in the inset of Fig. 2, where the total confining stress coming from gravity and the elastic membrane is only about 10 kPa, which is the maximum stress scale reached (the positive slope for $\gamma \gtrsim 0.05$ matches the contribution of the membrane stiffness as it is deformed). The chain packings exceed this confining stress by a factor of $\sim 10^3$. The strength of polymer materials is partly attributed to the extensional or bending strength of individual chains. Indeed, during the compression of the long chains in Fig. 2 we started to hear chains break at the point where the maximum stress was reached, and after each measurement was done we counted approximately 10% of the links in the chains to be broken. This breaking and the ability to support stresses much greater than the confining stress suggests that – similar to poly-

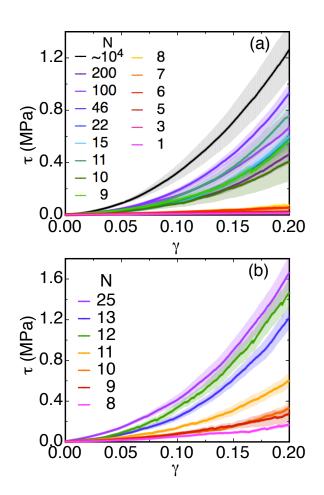


FIG. 3: Stress τ vs. strain γ for different chain lengths N shown in the key. (a) For chains of minimum loop circumference $\xi = 8$. (b) For $\xi = 11$. In both panels, the data for $N > \xi$ (cool colors) group into a band which exhibits strong strain-stiffening.

mer systems – the material strength of the chains themselves contributes and the packing can only shear and fail by breaking links in the chains, which would require the packing structure to produce some self-confinement.

To investigate the transition from granular behavior in the limit of unlinked beads with N = 1 and strainstiffening in the limit of large N, we show stress-strain curves for packings of chains with different lengths N in Fig. 3. Chains of minimum loop circumferences $\xi = 8$ and $\xi = 11$ are shown in panels a and b, respectively. It is seen in both panels that the data fall into two distinct bands: the upper band of data with chain lengths $N > \xi$ exhibits strong strain-stiffening, while the lower band of data with $N \leq \xi$ exhibits typical granular strain-softening or much weaker strain-stiffening. The fact that the transition length between these two bands scales with and is about equal to the minimum loop size ξ in both cases suggests a relationship between strain-stiffening and whether the chains form loops in the packings.

To investigate the role of entanglement in this relationship, we performed x-ray tomography at the Advanced

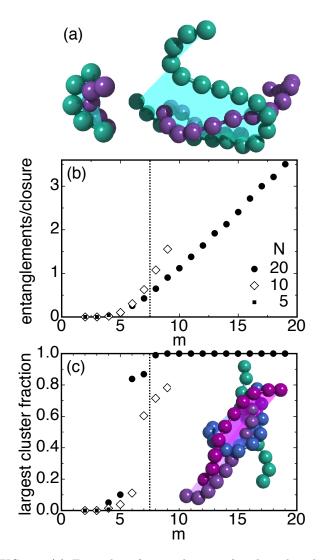


FIG. 4: (a) Examples of entanglements for chain lengths N = 5 (left) and N = 20 (right). The green chains are entangling the purple chains. The shaded region indicates the entanglement manifold for the closure connecting the ends of the chains (m = N - 1). (b) Average number of entanglements per closure as a function of contour length m for $\xi = 8$. The key indicates the chain lengths N. (c) Size of the largest entangled cluster as a fraction of the total number of chains in the sample. The dotted line indicates the onset of strong strain-stiffening, which is only found to occur when the contour length of a closure is large enough to that there are entangled clusters consisting of the majority of chains in the system. The inset image shows an example of a cluster of 4 chains with 5 entanglements between them for N = 10.

Photon Source, obtaining the precise positions of each bead and link in three dimensions [11]. For these experiments, we used aluminum chains with bead diameter 2.5 mm and $\xi = 7.5$ in cylindrical packings with diameter and height of 35 mm each. Here we present results for chain lengths N = 5, 10 and 20 each compressed to a strain of $\gamma = 0.2$.

We define an entanglement to occur if one chain wraps

	type 1	type 2
		11.0 ± 0.3
min. length for strong strain-stiffening	8.0 ± 0.5	11.0 ± 0.5
min. length for majority clusters	6.5 ± 1.0	-

TABLE I: Comparison of different length scales for the two chain types. The chain length required for strong strainstiffening is just above that required for most of the chains to be entangled in a single cluster, and consistent with the minimum loop circumference of the chains.

partially around another. Motivated by string knotting analysis [12], we first form a closure by drawing an imaginary line between any two beads of the entangling chain separated by a contour length m beads along the contour of the chain. We then define an entanglement manifold as the minimal two-dimensional manifold bounded by this closure. If another chain crosses through this manifold, it is counted as entangled by the entangling chain. Examples of entanglements along with the manifolds are shown in Fig. 4a. These particular examples would not be counted as entanglements based on typical polymer algorithms where entanglements are counted if chains catch on each other if they are contracted [13]. In contrast with elastic polymers, our granular chains are highly constrained from significant rearrangements due to their high packing density, so we expect these examples are still able to produce significant constraints against shearing from entanglement. Regardless, the majority of our entanglements would satisfy the typical requirements, and it is usually expected that different algorithms for measuring entanglement tend to produce qualitatively similar statistics [13].

We count all of the entanglements for each possible closure between beads and plot the average number of entanglements per closure as a function of the contour length m in Fig. 4b [14]. Regardless of chain length N, for contour lengths $m \leq 3$, no entanglements are observed because the size of the beads restricts other chains from fitting inside even the largest possible entanglement manifolds. For larger m, the average number of entanglements per closure increases monotonically with m. Since chains that are longer compared to their minimum loop size can be bent further, the area of the entanglement manifold tends to similarly increase with m and it is more likely in a random packing that other chains will cross this manifold to become entangled. Thus, the entanglement manifold method provides an intuitive way to understand why the probability of entanglements increases with separation or chain length.

The increase in the number of entanglements per closure with contour length seen in Fig. 4b could help explain why only the longer chains strain-stiffen. The longer chains with N = 10 and 20 which strain-stiffen reach more than 1 entanglement per closure for larger contour lengths, while the shorter chains with N = 5which do not strain-stiffen have no more than 0.04 entanglements per closure. Since the probability of entanglements as a function of contour length is similar for different N, the reason that the N = 5 chains do not entangle more is simply that they are too short compared to the minimum loop size $\xi = 8$ for them to bend enough to have large entanglement manifolds.

If entanglements are to sustain a stress, they must connect to form clusters that span the system [15]. If these are to prevent failure by shear so that the packing as a whole can sustain significantly more stress, these clusters must span the system densely so that they provide enough constraints to avoid weak points in the packing that could shear. We identify a cluster as subset of chains in entanglements that are transitively connected. An example of a cluster of four entangled chains is shown in the inset of Fig. 4c. The largest cluster size found for each N and m is plotted in Fig. 4c. For the strain-stiffening chains with N = 10 and 20, the largest cluster size jumps sharply to over 50% of the chains in the packing at m = 6and 7, respectively. This is just below the chain length N = 8 required for strong strain-stiffening (these length scales are summarized in Table I). These cluster sizes also large enough that they can span the system in many directions. In contrast, the N = 5 chains which do not strain-stiffen also do not have large enough clusters of entanglements to span the system.

To summarize, we demonstrated that random packings of granular chains can be used as a model system that exhibits strain-stiffening as in polymer materials, with the additional ability to precisely measure the structure of entanglements. These chains exhibit strainstiffening if they are longer than the minimum loop circumference they make in packings, which allows them to form system-spanning clusters of entanglements. Since

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the strength of granular materials is usually limited by the scale of the confining pressure at the boundaries, we propose that for the chain packings to reach the orders-of-magnitude higher strength, the confinement at the boundary must be functionally replaced by a selfconfinement due to the system-spanning clusters of entanglements. Strain-stiffening then results if these entanglements tighten up under strain, eventually locking up to prevent further shear unless the links break. This picture contrasts with recent theoretical arguments that strain-stiffening in polymers can be explained without entanglement. However, those arguments only explained much weaker strain-stiffening in which the stress increase is less than an order of magnitude [4, 5], so no additional confining stress mechanism was necessary in those cases. The questions of whether entanglement is the only way to get strong strain-stiffening, and how to quantitatively predict strain-stiffening from entangled structures, remain open for future work.

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tion images were taken at 720 different orientations of each sample. Using a Fourier-transform-based reconstruction algorithm these images were converted into three-dimensional data sets with a spatial resolution of 40 μ m. The bead and link positions were then obtained by fitting shape templates to the three-dimensional images (see Ref. [7] for details).

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