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Phys. Rev. Fluids **4**, 091301 — Published 9 September 2019

DOI: [10.1103/PhysRevFluids.4.091301](https://doi.org/10.1103/PhysRevFluids.4.091301)

On the Extensional Rheology of a Dilute, Particle-Laden Viscoelastic Solution

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(Dated: August 21, 2019)

Recent discoveries have allowed a mathematical and computational foundation for understanding the rheology of particles suspended in viscoelastic fluids – we employ these new tools to understand the extensional rheology of such suspensions. We accomplish this by first calculating the renormalized particle contribution to the extensional viscosity in such a suspension in the dilute particle limit over a wide range of extensional Weissenberg number and Hencky strain. The models we use for the suspending fluids are the simplest dumbbell models – The Oldroyd-B, FENE-P and Giesekus models – such that our results are general for polymer solutions which exhibit strong strain hardening at values of the Weissenberg number above those which engender the coil-stretch transition, $Wi \geq 0.5$. We demonstrate that the effect of particles on the “extra elongational viscosity” relative to the fluid is non-monotonic in strain (increasing for small strain and then decreasing for large strain). Thus at a fixed strain, the particle “extra viscosity” relative to the fluid may increase or decrease with Wi . We demonstrate that this interesting behavior is due to the interplay between the two contributions of the “particle induced fluid stress” (PIFS) and the “stresslet” to the extra viscosity. The contribution of the “particle induced fluid stress” to the suspension viscosity increases at small strain but plateaus and then decreases at higher values of the strain. Thus, at small strain the local velocity gradients near a particle increase the polymer stretch, while for greater strain, polymers which have undergone the coil-stretch transition collapse in the neighborhood of a given particle. On the other hand, the “stresslet” contribution to the viscosity relative to the fluid decreases monotonically as the polymer stretch surrounding a given particle “shields” the particle and thus reduces the local surface tractions. Beyond Hencky strain of approximately 2 the decreasing value of the stresslet coupled to the plateauing of the PIFS, causes the overall reduction in the particle-induced extra viscosity relative to that of the fluid.

PACS numbers: May be entered using the `\pacs{#1}` command.

I. INTRODUCTION

The presence of strong strain-hardening in polymer solutions is one of their most important rheological characteristics [1–4]. Thus the extensional viscosity of a polymer solution, at sufficiently large strain rates and Hencky strain, can be orders of magnitude larger than the solvent alone. The control of processing flows of these materials is essentially dictated by this strain hardening [1–4], which has historically been associated with the famous coil-stretch transition of the individual polymer chains [5–7]. In the last 20 years, this connection has been made unambiguous by the simultaneous measurement of the extensional viscosity of DNA solutions and microscopy of the molecular configuration [8–13]. Strain hardening and the coil-stretch transition is still a very active area of research with a number of groups seeking to understand the effect of intermolecular interactions on the coil-stretch transition. The direct application of these rheological studies ranges from extrusion molding to fiber spinning. However a great many of the associated practical materials contain particle additives – they are hence *a suspension in a viscoelastic fluid*. These suspensions play key roles in many energy applications (e.g. fracking

fluids)[14], materials design applications (e.g. injected composite materials, pastes and paints) [15], and consumer product applications (liquid medicines and drugs).

Recently, there have been very significant advances in understanding the *shear* rheology of particle suspensions in highly elastic solutions [16–21]. Indeed the most salient experimental feature in this regard is the strong shear thickening of the suspensions (both in the viscosity and the primary normal stress coefficient), which is entirely absent in the fluids alone [18, 19, 22–24]. These advances include (a) methods for renormalization of the long range interactions in these suspensions, such that dilute suspension theory could be applied [18–20, 25] (b) the first calculation of the dilute limiting shear viscosity for a suspension of spheres for “weak flow” using these renormalizations, (c) accurate large scale computations of the renormalized shear flow rheological quantities beyond the weak flow limit [19–21] and (d) qualitative reproduction of the experiments [21]. It is the purpose of this manuscript to, for the first time, apply these advances to an understanding of the extensional rheology of particle suspensions in a viscoelastic polymer solution. **Previous studies [26–28] focus on 2D inertialess particle suspensions in Newtonian or viscoelastic fluids under planar elongational flows. In this study, we present 3D numerical results for a dilute viscoelastic suspension of spherical particles in uniaxial extension. We perform 3D simulations**

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to ensure elastic instabilities do not develop during evolution of polymeric flow with time.

II. PROBLEM FORMULATION

A. Bulk Stress in a viscoelastic suspension

The general averaging procedure to obtain the macroscopic stress in a suspension of rigid particles has been discussed in [16, 19, 20, 29, 30]. Here, we give a very brief recapitulation of the method. The bulk stress in a suspension can be obtained by averaging over a sufficiently large volume with the volume averaged velocity field matching the ensemble averaged velocity field and this is equivalent to the ensemble average [29] for a statistically homogenous suspension. Thus the ‘‘bulk’’ stress defined in terms of the volume average is

$$\overline{\sigma_{ij}} = \frac{1}{V} \int_V \sigma_{ij} dV, \quad (1)$$

where the overbar denotes volume average and V is the total suspension (particles and fluid) volume. If the stress contributions from the particle phase σ_{ij}^p and the fluid phase σ_{ij}^f are separated and furthermore, if the stress contributions due to suspending fluid without any particles σ_{ij}^{f0} and the stress contributions due to particles are made distinct then, as shown in [18–20], Eq.(1) becomes

$$\overline{\sigma_{ij}} = \sigma_{ij}^{f0} + n \left(\int_{V_1} (\sigma_{ij}^f - \sigma_{ij}^{f0}) dV + \int_{A_1} x_j \sigma_{ik}^f n_k dA \right), \quad (2)$$

where diluteness has been assumed and the total particle contribution to the stress is obtained by multiplying the number density of particles n in suspension with the per particle contribution. In the above equation, V_1 is the domain volume that contains a single particle and is large enough such that any disturbance due to the particle decays far from the particle and at the boundaries of domain. The symbol A_1 represents a single particle surface and n_k is the unit normal vector pointing out of the particle surface. The number density n is related to particle volume fraction ϕ as $n = \frac{\phi}{V_p}$ where V_p is the volume of a single particle. The per particle contribution to the extension viscosity η_E^p is then the sum of contributions from the particle induced fluid stress (PIFS) Σ_{ij} and the stresslet S_{ij} , viz.

$$\Sigma_{ij} = \frac{1}{V_p} \int_{V_1} (\sigma_{ij}^f - \sigma_{ij}^{f0}) dV; \quad S_{ij} = \frac{1}{V_p} \int_{A_1} x_j \sigma_{ik}^f n_k dA \quad (3)$$

Assuming that the flow is homogeneous, steady uniaxial extensional flow, where the rate of strain tensor, $E_{ij} = \dot{\epsilon} [\delta_{i1}\delta_{j1} - \frac{1}{2}(\delta_{i2}\delta_{j2} + \delta_{i3}\delta_{j3})]$, we can define a per particle contribution to the extensional viscosity of the suspension η_E^ϕ through the relation :

$$\eta_E^p \dot{\epsilon} = \left(\Sigma_{11} - \frac{\Sigma_{22} + \Sigma_{33}}{2} \right) + \left(S_{11} - \frac{S_{22} + S_{33}}{2} \right). \quad (4)$$

The total suspension viscosity for any dilute volume fraction ϕ follows directly from Eq.(2) after the definition of particle viscosity is substituted and the particle number density is replaced by the volume fraction,

$$\eta_E^\phi = \eta_E^{\phi=0} + \phi \eta_E^p \quad (5)$$

where $\eta_E^{\phi=0} \dot{\epsilon} = \sigma_{11}^{f0} - \frac{1}{2}(\sigma_{22}^{f0} + \sigma_{33}^{f0})$ and therefore $\eta_E^{\phi=0}$ is the extensional viscosity of the suspending fluid. After rearranging, we obtain an equation for the relative difference between suspension and suspending fluid extensional viscosity,

$$\frac{\eta_E^\phi}{\eta_E^{\phi=0}} - 1 = \phi \frac{\eta_E^p}{\eta_E^{\phi=0}}, \quad (6)$$

which can be calculated in dilute simulations by computing the ratio of particle viscosity to fluid viscosity.

B. Governing equations

We consider a neutrally buoyant sphere placed at the centre of a rectangular computation domain $(\pm L_1/2, \pm L_2/2, \pm L_3/2)$ in a mean flow field $\bar{u}_i = (\dot{\epsilon}x, -\dot{\epsilon}y/2, -\dot{\epsilon}z/2)$ where $\dot{\epsilon}$ is the extension rate. The mean flow is imposed on the walls of the domain far from the particle. There is no net force or torque on the sphere due to the symmetric nature of imposed flow. The dimensionless governing equations are

$$\frac{\partial u'_i}{\partial x'_i} = 0; \quad Re \left(\frac{\partial u'_i}{\partial t'} + u'_j \frac{\partial u'_i}{\partial x'_j} \right) = -\frac{\partial p'}{\partial x'_i} + \frac{\partial \sigma'^f_{ij}}{\partial x'_j} \quad (7)$$

where prime denotes a nondimensional variable. Length is made dimensionless by particle radius a , time by inverse extension rate $\dot{\epsilon}$, velocity by $\dot{\epsilon}a$ and stress by $\eta_0 \dot{\epsilon}$ where η_0 is the zero shear viscosity of the suspending fluid. The fluid stress is the sum of Newtonian and polymer stress:

$$\sigma'^f_{ij} = \beta \left(\frac{\partial u'_i}{\partial x'_j} + \frac{\partial u'_j}{\partial x'_i} \right) + \tau'^p_{ij} \quad (8)$$

where β is the ratio of Newtonian solvent viscosity to the zero shear viscosity and τ'^p_{ij} is the nondimensional polymer stress. The FENE-P equation describing the evolution of the polymer conformation tensor C_{ij} is

$$\frac{\partial C_{ij}}{\partial t'} + u'_k \frac{\partial C_{ij}}{\partial x'_k} - C_{ik} \frac{\partial u'_j}{\partial x'_k} - C_{jk} \frac{\partial u'_i}{\partial x'_k} = -\frac{1}{Wi} \left(\frac{C_{ij}}{1 - \frac{C_{kk}}{L^2}} - \delta_{ij} \right) \quad (9)$$

where the conformation tensor components C_{ij} and the finite polymer extensibility L are non-dimensionalized by

the polymer radius of gyration. There are two dimensionless groups: the particle Reynolds number defined as $Re = \rho \epsilon a^2 / \eta_0$ and the Weissenberg number $Wi = \lambda \dot{\epsilon}$ where λ is the longest polymer relaxation time. The polymer stress is related to conformation tensor through,

$$\tau_{ij}^{p'} = \frac{1 - \beta}{Wi} \left(\frac{C_{ij}}{1 - \frac{C_{kk}}{L^2}} - \delta_{ij} \right). \quad (10)$$

When $L \rightarrow \infty$, the FENE-P equation is equivalent to the Oldroyd-B model. The Giesekus equation, in the same notation, is:

$$\begin{aligned} \frac{\partial C_{ij}}{\partial t'} + u'_k \frac{\partial C_{ij}}{\partial x'_k} - C_{ik} \frac{\partial u'_j}{\partial x'_k} - C_{jk} \frac{\partial u'_i}{\partial x'_k} = \\ - \frac{1}{Wi} (C_{ij} - \delta_{ij}) - \frac{\alpha}{Wi} (C_{ik} - \delta_{ik})(C_{kj} - \delta_{kj}) \end{aligned} \quad (11)$$

where α is the mobility parameter. When $\alpha = 0$, the equation is again equivalent to the Oldroyd-B model. To complete a calculation of the full time dependent problem, for any of these constitutive equations, we require initial conditions for the velocity and conformation tensor. For the initial velocity we choose the Newtonian creeping flow solution for flow past a sphere in uniaxial extensional flow and for the initial conformation tensor we choose the identity tensor (the dimensionless, equilibrium configuration thus producing no polymer stress). We also impose the boundary condition on conformation tensor such that the polymer conformation tensor at the walls of domain takes values corresponding to suspending fluid at different strains and Wi . This is allowed because the simulation domain is taken to be big enough such that there is no disturbance due to particle at the walls of domain. We drop primes in rest of the paper for convenience and all the variables are assumed to be non-dimensional.

C. Numerical Simulations

The simulations are run using a massively parallel finite volume solver developed at Stanford and the details of the solver can be found in [19] and the references therein. The domain grid conforms to the body of a single sphere and is meshed using tetrahedral elements of size $a/60$ near the particle surface and bigger elements of size $a/40$ far from the particle. The simulations are run in the limit $Re \ll 1$. The domain convergence is tested for all simulations and the results are found to vary by less than 5% between two domain sizes ($40a, 40a, 40a$) and ($80a, 40a, 40a$) where the longest dimension of the box is in direction of principal extension of the flow i.e. the “1” direction.

III. RESULTS AND DISCUSSION

We numerically compute the ratio of the per particle extensional viscosity to the fluid extensional viscosity as

a function of strain for a range of Wi to study the effect of particles on the suspension viscosity. The ratio can be thought as the “Einstein coefficient” [31, 32] for the startup of uniaxial extensional flow of spheres in a viscoelastic fluid.

A. Small Strain Results

To gain a physical understanding and test the accuracy of our numerical solutions, we first use perturbation theory at small strains to calculate the PIFS and stresslet analytically for the Oldroyd-B model. The polymer conformation tensor, fluid velocity and pressure are expanded in the small strain ϵ in the usual manner:

$$C_{ij} = \delta_{ij} + \epsilon C_{ij}^{(1)} + \epsilon^2 C_{ij}^{(2)} + \dots \quad (12)$$

$$u_i = u_i^{(0)} + \epsilon u_i^{(1)} + \epsilon^2 u_i^{(2)} + \dots \quad (13)$$

$$p = p^{(0)} + \epsilon p^{(1)} + \epsilon^2 p^{(2)} + \dots \quad (14)$$

The leading order $O(1)$ flow field is the Newtonian field and the conformation tensor is equal to the identity tensor. Thus, there is no PIFS at $O(1)$ and the stresslet is given by $S_{ij}^{(0)} = 5E_{ij}$ where E_{ij} is the average rate of strain tensor in the fluid domain – this is in accord with the famous Einstein result [31, 32]. At $O(\epsilon)$ the conformation tensor is proportional to the rate-of-strain of the Newtonian field. Thus, again at this order the deviatoric stress is proportional to the deviatoric Newtonian stress at $O(1)$ and thus there is an associated pressure correction but no additional velocity correction, *viz.*

$$u_i^{(1)} = 0 \quad (15)$$

$$p^{(1)} = \frac{1 - \beta}{Wi} p^{(0)} \quad (16)$$

$$C_{ij}^{(1)} = 2e_{ij}^{(0)} \quad (17)$$

The fluid stress, at this order, is given by,

$$\sigma_{ij}^{(1)} = \frac{1 - \beta}{Wi} \left(-p^{(0)} \delta_{ij} + 2e_{ij}^{(0)} \right) \quad (18)$$

and the stresslet correction then simply reflects the change in the viscosity:

$$S_{ij}^{(1)} = \frac{1 - \beta}{Wi} 5E_{ij} \quad (19)$$

and there is no correction to the PIFS. Continuing, then the conformation tensor at $O(\epsilon^2)$ is given by

$$C_{ij}^{(2)} = -\frac{e_{ij}^{(0)}}{Wi} - u_m^{(0)} \frac{\partial e_{ij}^{(0)}}{\partial x_m} + \dot{\gamma}_{ik}^{(0)} e_{jk}^{(0)} + \dot{\gamma}_{jk}^{(0)} e_{ik}^{(0)} \quad (20)$$

where $e_{ij}^{(0)}$ is the local strain rate tensor and $\dot{\gamma}_{jk}^{(0)}$ is the local velocity gradient in the Newtonian flow. The flow field is now non-Newtonian at $O(\epsilon^2)$ and rather than determine the correct flow field, one can determine the stresslet correction using the reciprocal theorem [25]:

$$S_{ij}^{(2)} = \frac{1-\beta}{Wi} \frac{3}{14} E_{ik} E_{kj} - \frac{1-\beta}{Wi^2} 5E_{ij}. \quad (21)$$

The $O(\epsilon^2)$ correction to the PIFS is calculated using the renormalized averaging procedure of [25],

$$\Sigma_{ij}^{(2)} = \frac{1-\beta}{Wi} \frac{25}{7} E_{ik} E_{kj}. \quad (22)$$

We show the comparison of the small strain theory to our transient numerical simulation results for the stresslet and the PIFS in Figure (1). The theory and simulations agree well at small strains but the theoretical prediction for the stresslet departs from the numerical simulations at strains $\epsilon \approx 10^{-1}$ and the small strain theory no longer holds. Moreover, in the inset to each Figure, we see that the stresslet ratioed to the fluid stress decreases monotonically with strain, while the PIFS monotonically increases. This will be important in the physical description of our numerical results below.

B. Numerical Results at Finite Strain and Weissenberg Number

We performed the full 3D simulations to calculate the per particle extra viscosity compared to the fluid at finite strains and Wi . We modeled the suspending fluid using the FENE-P equation with parameters $\beta = 0.68, L = 100$ and also using the Oldroyd-B equation with the same value of $\beta = 0.68$. The parameters of the Giesekus model used are $\alpha = 10^{-3}, \beta = 0.68$. **We used the value of viscosity ratio $\beta = 0.68$ in our simulations because the Giesekus model parameters $\beta = 0.68, \alpha = .0039$ fit the steady shear rheology of the Boger fluid used by Dai *et al.* [23] as shown in [19].** Each simulation is run for a total strain, $\dot{\epsilon}t$, of 7. Figure (2) shows the viscosity ratio with increasing Hencky strain for different models at a range of Wi . **We also explore the effect of smaller $\beta = 0.4$ value on the ratio of particle viscosity to suspending fluid viscosity for the Oldroyd-B model at different Wi . The inset shows the viscosity ratio for the Oldroyd-B model and $Wi = 1$ at two different β values. The dotted line is for $\beta = 0.68$ and the dashed line is for $\beta = 0.4$.** We see that the value of ratio (or Einstein's coefficient) is close to 2.5 for $Wi = 0.1$ at all values of strain for all models studied. Thus, we predict that the viscoelastic suspensions behave like Newtonian suspensions in the limit of small strain and Wi . However, the viscosity ratio is non-monotonic in strain for other values of Wi considered in the study. The non-monotonicity occurs for all model parameters studied i.e. different values

of L , α and viscosity ratio β and more importantly for all polymer constitutive models. **The inset in Figure (2) shows that the value of the viscosity ratio for $Wi = 1$ is unaffected by the value of β and we have also seen very little effect of varying β at other values of Wi .** We believe that this behavior is universal for spherical particles in dilute polymer solutions subject to (uniaxial) extensional flow. Furthermore, the viscosity ratio computed for both FENE-P and Oldroyd-B is identical until strains ≈ 6 after which the value of the finite extensibility parameter L makes a difference. The Giesekus model demonstrates lower values of the ratio of the particle viscosity to the fluid viscosity compared to the other two models starting from strains of about 2, but essentially demonstrates the same qualitative behavior.

The non-monotonicity in particle viscosity relative to fluid can be attributed to the PIFS and stresslet contributions to the particle viscosity. Figure (3) shows the ratio of the PIFS and stresslet to suspending fluid viscosity as a function of strain for different Wi . We see that the ratio of the stresslet to the fluid viscosity decreases monotonically with strain (as it did in the small strain theory) but the ratio of PIFS to fluid viscosity is non-monotonic with strain for all the $Wi > 0.1$ studied. The ratio of total particle viscosity to fluid viscosity increases at small strains (Figure (2)) because the increase in ratio of PIFS to fluid viscosity is stronger and offsets the decrease in the equivalent ratio of the stresslet. But as the Hencky strain increases, the ratio of the PIFS to the fluid viscosity plateaus and eventually starts to decrease with strain. This, combined with the monotonic decrease of the stresslet ratio, causes the total particle viscosity to decrease relative to the fluid at higher strains. Similar behavior was found for the particle first normal stress coefficient relative to the fluid (including the particle induced fluid stress and stresslet) in steady shear flow past an isolated sphere [19]. Thus, this nonmonotonicity in a material function is likely to be observed in other linear flows and it is important to understand the physical reasons for its origin.

We develop insight into the PIFS behavior by examining Eulerian contour plots of $C_{kk} - C_{kk}^{f0}$, the difference between polymer conformation trace, at each point in the $x_1 - x_2$ plane and the polymer conformation trace at a far field point (i.e. with negligible particle effect). Figure (4) shows the contour plots of $C_{kk} - C_{kk}^{f0}$ for the FENE-P model at two different values of strain $\epsilon = 2$, and $\epsilon = 4$ and $Wi = 0.8$. We see that there are regions of large polymer stretch relative to the fluid, along the principal axis of extension near the particle surface at both the strains considered. However at larger strains, we also observe the presence of blue regions around the particle where the polymer is significantly *less* stretched compared to the polymer stretch in the fluid alone. The stresslet contribution to the suspension viscosity decreases monotonically relative to the fluid as the polymer stretch surrounding a given particle "shields" the particle and thus reduces the local surface tractions. The origin of these regions can be

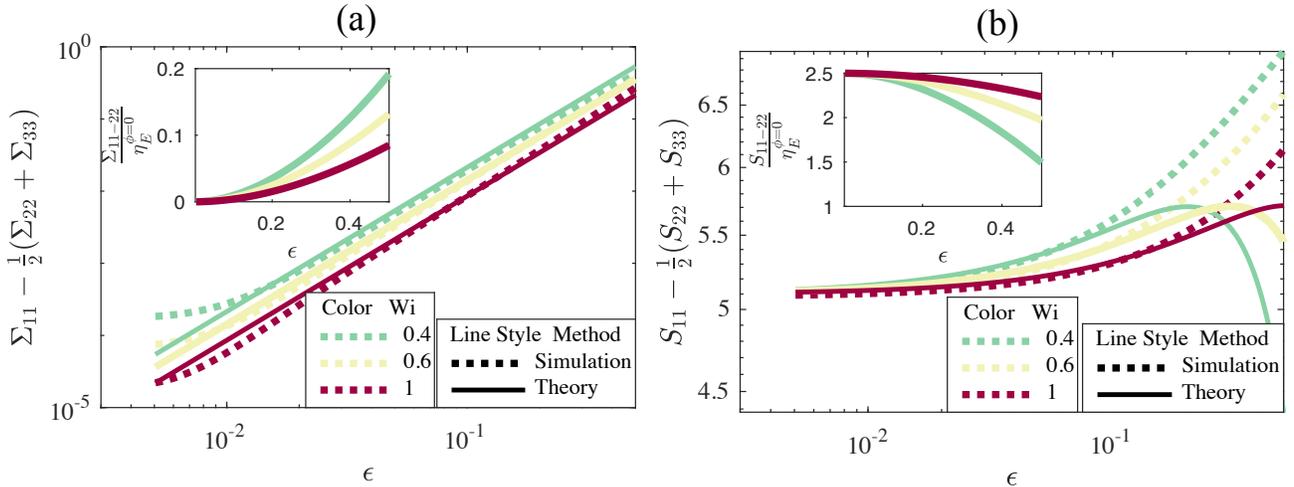


FIG. 1. Comparison of small strain theory with transient simulations as a function of strain for (a) stresslet and (b) particle induced fluid stress. The parameter $\beta = 0.68$ for Oldroyd-B model used in simulations.

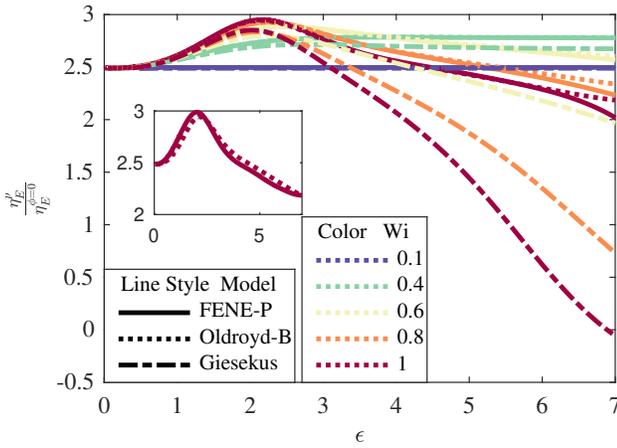


FIG. 2. Ratio of particle viscosity to suspending fluid viscosity with strain for a range of Weissenberg numbers for FENE-P, Oldroyd-B and Giesekus models. The parameters of FENE-P model are $\beta = 0.68$, $L = 100$, of Oldroyd-B model are $\beta = 0.68$ and of Giesekus model are $\alpha = 10^{-3}$, $\beta = 0.68$. The inset also shows ratio of particle viscosity to fluid viscosity for Oldroyd-B model and $Wi = 1$. The two curves correspond to two different β values. The β value for dotted curve is equal to 0.68 and the β value for dashed curve is equal to 0.4. We see that β does not have a significant effect on the ratio.

better understood by examining the $\beta \rightarrow 1$ limit because, in this limit, the flow is approximately Newtonian at all strains and we can thus analyze a known flow field as it effects the polymer evolution [18]. This simplification allows us to characterize the local velocity gradient using a Q-R decomposition [33] and determine the effect of extension or rotation dominated regions near the particle surface.

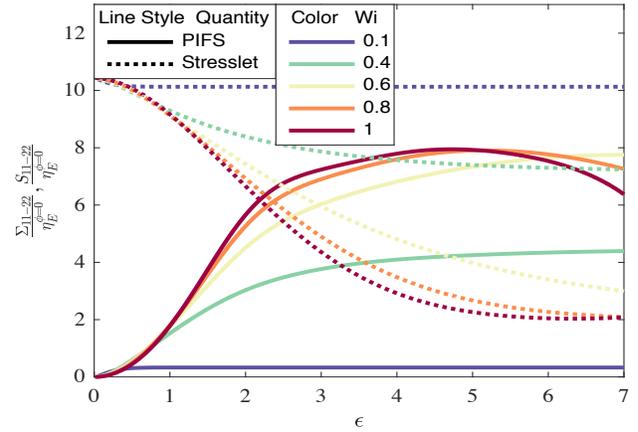


FIG. 3. Ratio of PIFS and stresslet to suspending fluid viscosity with strain for different Wi for FENE-P model. The parameters of FENE-P model used in simulations are $\beta = 0.68$, $L = 100$.

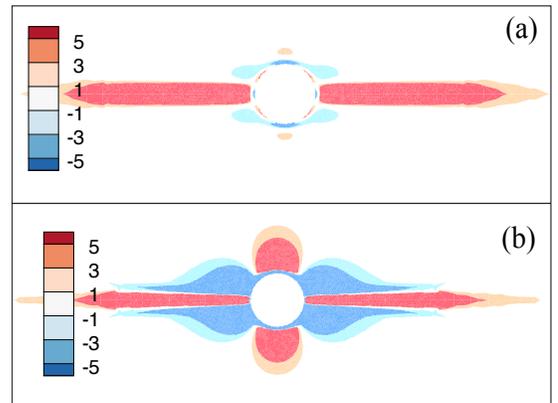


FIG. 4. Eulerian contours of $C_{kk} - C_{kk}^{f0}$ in $x_1 - x_2$ plane for FENE-P model parameters $L = 100$, $\beta = 0.68$, $Wi = 0.8$ and strain $\epsilon = 2$ in panel (a) and $\epsilon = 4$ in panel (b).

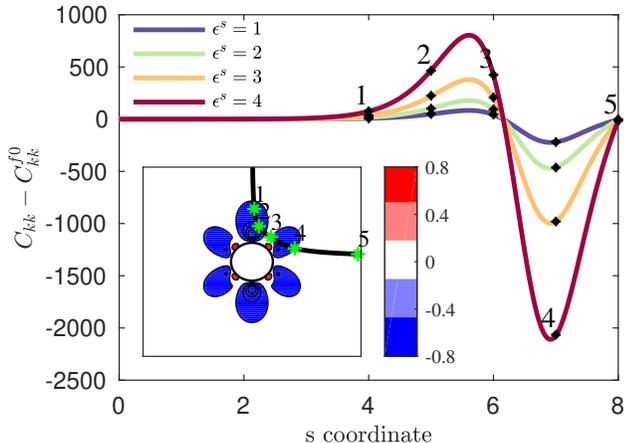


FIG. 5. Evolution of polymer conformation trace along the same streamline at different starting times for $Wi = 0.8$. Inset shows streamline plotted on top of discriminant D for uniaxial extension flow past a sphere with $D < 0$ indicating extension and $D > 0$ indicating rotation dominated regions.

C. Method of characteristics for small polymer concentration $\beta = 1$

We use the method of characteristics [18] to study the polymer response to the local flow field along different streamlines. There is only “one way coupling” from flow to polymer in the $\beta = 1$ limit (meaning the given fixed flow affects the polymer configuration). We solve the set of ordinary differential equations shown below to obtain the polymer conformation along streamlines:

$$\frac{dx_i}{ds} = u_i^N; \quad \frac{dt}{ds} = 1 \quad (23)$$

$$\frac{dC_{ij}}{ds} = C_{ik} \frac{\partial u_j^N}{\partial x_k} + C_{jk} \frac{\partial u_i^N}{\partial x_k} - \frac{1}{Wi} (C_{ij} - \delta_{ij}) \quad (24)$$

where u_i^N is the Newtonian creeping flow solution for flow past a sphere in uniaxial extensional flow and s is the integration parameter along the streamline.

We show a streamline trajectory in inset of Figure (5) that starts far from the particle and the initial conformation tensor on the streamline is the same as the conformation in the fluid alone at the starting time. We study the evolution of polymer conformation along the same streamline at different dimensionless starting times denoted by ϵ^s for $Wi = 0.8$. The polymers in the far field get increasingly stretched as time increases and the initial condition for Equation(24) is different for different starting times. The polymers convected towards the particle get stretched by extension dominated regions (characterized by $D < 0$) around the particle as shown by points 1 and 2 in Figure (5). As the polymers enter the vorticity dominated regions around the particle (characterized by $D > 0$), they lose their alignment with the principal axis

of extension and collapse compared to far field flow. This is shown by points 3 and 4 in Figure (5). Note that the reduction in polymer stretch (stress) during the collapse (point 4) is far larger than the increase in stretch (stress) through the extension dominated region. At large distances from particle denoted by point 5 in the figure, the polymers attain the same configuration as the suspending fluid. The magnitude of polymer stretch and collapse increases as starting times increase because the polymers undergo the coil-stretch transition [5–7] in the far field as time increases. Nearly fully stretched polymers collapse dramatically in rotation regions around the particle [34] as shown in Figure (5). These regions where polymers have significantly smaller stretch compared to the far field contribute negatively to the PIFS and the size of these regions grows around the particle as dimensionless time/strain in the experiment increases. This eventually arrests the growth of the PIFS relative to the fluid, thus it plateaus and decreases at higher strains. **From numerical analysis of our flow fields, we claim that these microstructural origins of polymer stretch relative to the suspending fluid for $\beta = 1$ also hold for other β values where the flow field is not Newtonian and the polymer conformation has to be tracked along different time dependent particle paths for different Wi . The extension and rotation dominated regions (characterized by D) also change with time as the flow evolves for $\beta \neq 1$ and it is not easy or necessarily correct to correlate the polymer conformation with the local extension or rotation experienced by the polymer. Therefore, we present the results only for $\beta = 1$.**

IV. CONCLUSION

We performed 3D transient simulations of the bulk extensional rheology of dilute particle suspensions for a range of Wi and finite strains and calculated the renormalized per particle extra viscosity of the suspension. We also calculated the particle induced fluid stress and stresslet contributions analytically for small strain and these calculations agree well with numerical simulations at small strains. We found that the ratio of the per particle extra viscosity to suspending fluid viscosity is non-monotonic in strain for all constitutive models and for all values of $Wi > 0.5$ studied. Thus, at a fixed strain, the particle viscosity relative to the fluid may increase or decrease with Wi . The conformation contours show large polymer stretch near the particle in the principal extension direction at modest strains and regions around the particle where the polymers are significantly less stretched compared to the far field at high strains when polymers have undergone coil-stretch transition. These regions grow with strain and thus arrest the PIFS increase relative to the fluid. We believe the non-monotonicity in ratio of particle viscosity to fluid viscosity is universal for spherical particles in dilute polymer

solutions subject to (uniaxial) extensional flow. We look forward to experimental examination of our theoretical predictions in future work.

V. ACKNOWLEDGEMENTS

The authors of this paper would like to acknowledge support from NSF (CBET-1803765) and support

from the US Army High Performance Computation Research Center (AHPARC) with the grant number W911NF07200271. Computer simulations were also performed on the Stanford University Certainty computer cluster, which is funded by the American Recovery and Reinvestment Act of 2009. A.J. would like to acknowledge financial support from Stanford Graduate Fellowship for funding her Ph.D. work. Moreover, J.E. would like to thank the Knut & Alice Wallenberg Foundation for their generous postdoctoral fellowship, KAW 2015.0419.

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