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# Pressure-induced topological insulator-to-metal transition and superconductivity in Sn-doped $\text{Bi}_{1.1}\text{Sb}_{0.9}\text{Te}_2\text{S}$

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Tetradymite-type topological insulator (TI) Sn-doped  $\text{Bi}_{1.1}\text{Sb}_{0.9}\text{Te}_2\text{S}$  (Sn-BSTS), with a surface state Dirac point energy well isolated from the bulk valence and conduction bands, is an ideal platform for studying the topological transport phenomena. Here, we present high-pressure transport studies on single crystal Sn-BSTS, combined with Raman scattering and synchrotron x-ray diffraction measurements. Over the studied pressure range of 0.7-37.2 GPa, three critical pressure points can be observed: (i) at  $\sim 9$  GPa, a pressure-induced topological insulator-to metal transition is revealed due to closure of the bulk bandgap, which is accompanied by changes in slope of the Raman frequencies and a minimum in  $c/a$  within the pristine rhombohedral structure ( $R\bar{3}m$ ); (ii) at  $\sim 13$  GPa, superconductivity is observed to emerge, along with the  $R\bar{3}m$  to a  $C2/c$  (monoclinic) structural transition; (iii) at  $\sim 24$  GPa, the superconducting transition onset temperature  $T_C$  reaches a maximum of  $\sim 12$  K, accompanied by a second structural transition from the  $C2/c$  to a body centered cubic  $Im\bar{3}m$  phase.

## 32 I. INTRODUCTION

33 Pressure tuning of structural and electrical properties in tetradymite-type  
34 topological insulators (TIs) like  $\text{Bi}_2\text{Se}_3$ ,  $\text{Bi}_2\text{Te}_3$  and  $\text{Sb}_2\text{Te}_3$  have attracted much  
35 attention because of discoveries of a variety of interesting phenomena [1-22]. For  
36 instance, synchrotron x-ray diffraction and Raman scattering measurements in  $\text{Bi}_2\text{Se}_3$   
37 reveal a minimum in the lattice parameters ratio  $c/a$  and changes in pressure  
38 dependences of the Raman frequencies at the same low pressure far before a structural  
39 phase transition [1]; meanwhile, discontinuities in conductivity are observed around  
40 this critical pressure [2]. Similar situations are also encountered in other members of  
41 tetradymite-type TIs [3-10], despite that the critical pressures are somewhat different.  
42 Due to the absence of a structural transition, these anomalies were usually related to a  
43 pressure-induced electronic topological transition (ETT) or Lifshitz transition. As is  
44 known, the Lifshitz transition [23] links to the van Hove singularity associated with  
45 the band extrema passing through the Fermi level, and in the presence of such  
46 transition the distribution of carriers and Fermi surface topology changes. Considering  
47 the lack of such changes, Bera *et al.* [11] thus termed the transition as an isostructural  
48 transition, instead of the ETT. Obviously, the origins of these anomalies still remain  
49 elusive. Besides, superconductivity is commonly observed in these TIs at high  
50 pressures [12-17]. Interest is widely sparked by the expectation of realizing  
51 topological superconductivity [16], in spite of the absence of direct experimental  
52 evidences so far.

53 TI is characterized by the insulating bulk state and metallic topological surface  
54 states (TSSs) that are protected by time reversal symmetry. Ideally, a surface state  
55 Dirac point energy is isolated well from the bulk valence and conduction bands.  
56 Experimentally, for as-grown tetradymite-type nominally stoichiometric TIs, a  
57 metallic bulk conduction arising from inherent Se/Te deficiencies usually overwhelms  
58 the surface transport [24-26]. Therefore, it is hard to unveil the pressure evolutions of  
59 the intrinsic conduction of the bulk insulating state and TSSs via transport, which in

60 turn may result in misunderstanding of the structural responses to external pressure.

61 In view of the above, here we present a systematic study of the electrical,  
62 structural and vibrational properties on a recently discovered TI Sn-doped  
63  $\text{Bi}_{1.1}\text{Sb}_{0.9}\text{Te}_2\text{S}$  (Sn-BSTS) [27] in diamond anvil cells with pressures up to 37.2 GPa,  
64 which has a well-isolated bulk state from the TSSs. Thanks to such unique band  
65 structure of Sn-BSTS, we can successfully trace the pressure evolutions of the bulk  
66 and TSSs conductivities separately, which are vital for unveiling the topological  
67 insulator-to-metal transition-related structural and vibrational anomalies, instead of an  
68 ETT. Interestingly, the  $T_C$ -maximum of  $\sim 12$  K obtained here is the highest value  
69 among all the pressurized tetradymite-type TIs.

70

## 71 **II. EXPERIMENTAL DETAILS**

72 Sn-BSTS single crystals were grown by modified Bridgman technique [28]. The  
73 high quality of the single crystals was checked by single crystal and powder x-ray  
74 diffractions as well as energy dispersive spectra (see Fig. S1 and notes in the  
75 Supplemental Material for more details) [29]. Standard four-probe method was  
76 employed to perform the high-pressure electrical transport measurements in a  
77 temperature range of 1.8-300 K in a Be-Cu diamond anvil cell (DAC), designed by  
78 the Honest Machinery Designer's office (HMD), Japan. A piece of single crystal  
79 cleaved from the bulk Sn-BSTS single crystal was loaded with sodium chloride (NaCl)  
80 powder as the pressure transmitting medium and the current was introduced in the *ab*  
81 plane.

82 Pressure was generated by Mao-Bell type symmetric DAC for the synchrotron  
83 x-ray diffraction (XRD) and Raman scattering measurements. Raman scattering  
84 measurements were performed at room temperature on freshly cleaved Sn-BSTS  
85 single crystals using 532-nm solid-state laser for excitation with the power below 1%  
86 to avoid sample damage and any heating effect, at the Center for High Pressure  
87 Science and Technology Advanced Research (HPSTAR) in Shanghai. Neon was used

88 as the pressure transmitting medium. We also repeated our high-pressure Raman  
 89 measurements at the China High Magnetic Field Laboratory (CHMFL) in Hefei using  
 90 Daphne 7373 as the pressure medium. High pressure synchrotron powder XRD ( $\lambda =$   
 91 0.4133 Å) was performed at room temperature at the beamline of 16 BM-D, HPCAT  
 92 [30] of Advanced Photon Source, Argonne National Laboratory. Daphne 7373 was  
 93 used as the pressure transmitting medium. The DIOPTAS [31] program was used for  
 94 image integrations and Le Bail method was employed to fit the XRD data with the  
 95 RIETICA [32] program. Ruby fluorescence method [33] was used to determine the  
 96 pressure for all of the above experiments.

97

### 98 **III. RESULTS AND DISCUSSION**

99 Figure 1(a) displays the temperature ( $T$ ) dependence of the electrical resistance  
 100 ( $R$ ) of the Sn-BSTS single crystal at pressures from 0.7 to 10.5 GPa. At 0.7 GPa, a  
 101 huge resistance hump at  $T_h \sim 170$  K is observed, in agreement with that measured at  
 102 ambient pressure [27]. At ambient pressure, the conductance of Sn-BSTS is  
 103 considered as a sum of two parallel channels: surface ( $\sigma_s$ ) and bulk ( $\sigma_b$ )  
 104 conductances [34]. For simplicity, a two-channel model with the formula of  
 105  $\sigma(T) = \sigma_s(T) + \sigma_b(T)$  is employed. The former contribution is written as  
 106  $\sigma_s(T) = 1/\rho_s(T)$ , where  $\rho_s(T) = \rho_0 + bT^2 + cT$ , while the latter one is expressed as  
 107  $\sigma_b(T) = 1/\rho_b(T)$ , where  $\rho_b(T) = a \exp(E_g/2k_B T)$  [35]. Figure 1(b) shows the fitting  
 108 results with the two-channel model at 0.7 GPa, in which the dashed and solid lines  
 109 represent  $\sigma_s$  and  $\sigma_b$ , respectively. It yields a bulk bandgap  $E_g = 294$  meV, which is  
 110 a little smaller than 350 meV revealed by ARPES for Sn-BSTS at ambient pressure  
 111 [27]. Upon compression to 3.8 GPa, the  $T_h$  shifts towards lower temperatures rapidly,  
 112 which indicates that the contribution of the bulk state to the total conductivity  
 113 becomes increasingly dominant. As displayed in Fig. 1(c), the conductivity of the bulk

114 state shows a rapid enhancement with increasing pressure, while that of the surface  
115 state increases marginally upon compression. At 6.4 GPa, the resistance exhibits a  
116 semiconducting-like behavior in the whole temperature range, suggesting that the  
117 contribution of topological surface states (TSSs) are overwhelmed by that of the bulk  
118 state. Further increasing the pressure to 10.5 GPa, a partial metallic behavior below  
119 190 K is observed, signaling the metallization of the bulk state.

120 As the contribution of the TSSs can also be identified through magnetotransport  
121 measurements, magnetoresistance ( $MR$ ) was measured at 5 K under various pressures  
122 up to 10.5 GPa with magnetic fields perpendicular to the  $ab$  plane, as shown in Fig.  
123 1(d). One can see that the  $MR$  decreases monotonically with increasing pressure. In  
124 the low pressure region of 0.7-3.8 GPa,  $MR$  exhibits a concave behavior, which is  
125 very similar to that observed in the tetradymite-type TIs and can be attributed to weak  
126 antilocalization originating from the TSSs [36, 37]. Further increasing the pressure,  
127 the  $MR$  changes to a convex behavior at 6.4 GPa, above which a classical power-law  
128 behavior ( $MR \propto H^2$ ) develops gradually. These observations indicate that the bulk state  
129 conductance dominates above 6.4 GPa, in excellent consistent with the results from  
130 the  $R$ - $T$  data shown in Fig. 1(a).

131 Upon further compression to 12.6 GPa, a pronounced resistance drop is clearly  
132 seen below  $\sim 4$  K as presented in Figs. 2(a) and 2(b). Zero resistance is observed at  
133 14.3 GPa, suggesting the appearance of superconductivity in the pressurized Sn-BSTS.  
134 The presence of superconductivity is further confirmed by the temperature  
135 dependence of resistance measurements under various magnetic fields perpendicular  
136 to the  $ab$  plane at 14.3 GPa. As shown in Fig. 2(c), the superconducting transition  
137 temperature  $T_C$  is monotonically decreased with increasing magnetic field and the  
138 resistance drop is almost smeared out at 1.5 T. The inset of Fig. 2(c) shows the upper  
139 critical field  $\mu_0 H_{C2}$  as a function of temperature. Here the  $\mu_0 H_{C2}$  value is defined from  
140 the resistance criterion of  $R_{\text{cri}} = 90\% R_n$  ( $R_n$  is the normal state resistance near  $T_C$ ).  
141 The upper critical field  $\mu_0 H_{C2}$  is estimated to be 2.31 T according to the

142 Werthamer-Helfand-Hohenberg (WHH) equation [38].

143 To check the relationship between superconductivity and structure, *in situ*  
144 high-pressure synchrotron XRD measurements were performed on powdered single  
145 crystal Sn-BSTS up to 31.6 GPa. The experimental patterns are presented in Fig. 3(a).  
146 Upon compression, the rhombohedral structure ( $R\bar{3}m$ ) of Sn-BSTS is found to be  
147 stable up to 11.7 GPa, above which several new peaks show up as denoted by arrows,  
148 suggesting occurrence of a structural transition. In the intermediate pressure range,  
149 this new structural phase is observed to coexist with the low-pressure pristine one.  
150 With increasing pressure above 19.1 GPa, the low-pressure  $R\bar{3}m$  phase disappears  
151 completely, while another new structural phase transition takes place, as indicated by  
152 appearance of new peaks. Both high pressure phases coexists up to the highest  
153 pressure of 31.6 GPa achieved in the present study. We also plot the phase  
154 quantification as a function of pressure in Fig. S2. In addition to two critical pressures  
155 relating to the structural phase transition, phase coexistences and their respective  
156 evolutions with pressure are also clearly revealed. After carefully analyzing and fitting  
157 the data, we found that the pressure-induced structural transition sequence in  
158 Sn-BSTS is  $R\bar{3}m \rightarrow C2/c \rightarrow Im\bar{3}m$ , in agreement with those observed in its brother  
159 compounds  $\text{Bi}_2\text{Te}_3$  and  $\text{Sb}_2\text{Te}_3$  [18-20]. Typical standard Rietveld refinements at 1.3,  
160 15.0 and 31.6 GPa are displayed in Fig. 3(b). The corresponding fit parameters are  
161 displayed in TABLE S1.

162 The detailed lattice parameters and volume as a function of pressure are  
163 exhibited in Figs. 3(c) and 3(d). The equation of state (EoS) was fitted by using the  
164 third-order Birch-Murnaghan formula [39]:

$$165 \quad P = \frac{3}{2} B_0 [(V_0/V)^{\frac{7}{3}} - (V_0/V)^{\frac{5}{3}}] \left\{ 1 + \frac{3}{4} (B_0' - 4) [(V_0/V)^{\frac{2}{3}} - 1] \right\}, \quad (1)$$

166 where  $V_0$ ,  $B_0$  and  $B_0'$  are the volume, bulk modulus  $-V/(dV/dP)$ , and first  
167 order derivative of the bulk modulus at zero pressure, respectively. The fitting yields  
168  $V_0 = 153.7 \text{ \AA}^3$ ,  $B_0 = 51.4 \text{ GPa}$ , and  $B_0' = 7.0$  for  $R\bar{3}m$  phase;  $141.7 \text{ \AA}^3$ ,  $97.2 \text{ GPa}$

169 and 7.4 for  $C2/c$  phase; and 137.5 Å<sup>3</sup>, 122.9 GPa and 4.0 for  $Im-3m$  phase. In addition,  
170 the volume collapses at these two structural transition pressures are estimated to be  
171 about 1.8% and 3.0%, respectively, indicating both structural transitions are first  
172 order.

173 The structural transitions are also reflected in Raman measurements as shown in  
174 Fig. 4. At 1.3 GPa, two main peaks centered at 112.7 and 170.4 cm<sup>-1</sup> are observed,  
175 similar to those observed at ambient pressure [Fig. S1(d) and Ref. 27]. These two  
176 peaks can be assigned to the  $E_g$  and  $A_{2g}$  vibrational modes [27], respectively. With  
177 increasing pressure, both Raman peaks move towards higher frequencies  
178 monotonically. At 12.6 GPa, a new peak at 156.8 cm<sup>-1</sup> develops due to the structural  
179 transition from  $R-3m$  to  $C2/c$ , which can be assigned to the  $A_g$  mode [4, 6]. Upon  
180 further compression above 24.2 GPa, all Raman mode disappears relating to the  
181 structural transition from  $C2/c$  to  $Im-3m$ . **These conclusions are reproducible from a**  
182 **second high-pressure Raman experiment except for minor difference in the critical**  
183 **pressure values [Fig. S3].**

184 To obtain a comprehensive understanding of the pressure-induced metallization  
185 and superconductivity in TI Sn-BSTS, the pressure dependences of the frequency and  
186 full width at half maximum (FWHM) of Raman modes, lattice parameters ratio  $c/a$ ,  
187 bulk gap  $E_g$  and superconducting critical temperature  $T_C$  are plotted together in Fig. 5.  
188 It is found that the lattice ratio  $c/a$  shows a minimum around 9 GPa [Fig. 5(a)].  
189 Concurrently, clear changes in the slope of Raman frequencies and the FWHM of the  
190  $E_g$  and  $A_{2g}$  mode are observed [Figs. 5(b) and 5(c)]. Meanwhile, closure of the bulk  
191 gap is estimated at ~9 GPa by extrapolating the  $E_g$  vs.  $P$  curve [wine dashed line in  
192 Fig. 5(d)]. Consequently, all of these structural and vibrational anomalies at ~9 GPa  
193 should be related to the closure of the bulk gap, i.e. a pressure-induced topological  
194 insulator to metal transition, through electron-phonon coupling [40, 41].

195 We note that similar anomalous behaviors have been reported in previous  
196 high-pressure studies of tetradymite-type TIs, which were attributed to an electronic

197 topological transition (ETT) [1, 4-9]. On the one hand, to check whether there is an  
198 ETT in Sn-BSTS or not, we have plotted the reduced pressure versus Eulerian strain  
199 [5]. In Fig. S4, one can see that only linear behaviors are observed, in stark contrast to  
200 previous cases, probably indicating the absence of ETT. On the other hand, as  
201 reported previously, the transport behavior in these parent TIs is dominated by the  
202 inherent impurity states [24, 26]. The impurity states contribute a metallic transport  
203 that overwhelms the contributions from the TSSs, resulting in the experimentally  
204 observed metallic feature in  $R-T$  [24]. Therefore, the interference from the impurity  
205 states may mask the pressure effect on inherent electrical, structural and vibrational  
206 responses of both TSSs and bulk state. In the present case, due to the unique electrical  
207 band structure of TI Sn-BSTS, we can trace the pressure dependencies of the TSSs  
208 and bulk state separately. Importantly, a pressure-induced topological  
209 insulator-to-metal transition of Sn-BSTS is successfully unveiled, which is  
210 simultaneously accompanied by structural and vibrational anomalies. In view of the  
211 above two points, we argue that the previously-observed anomalies in pressurized  
212 tetradymite-type parent TIs might be related to the closure of the bulk bandgap of the  
213 topological insulator, instead of the ETT [1, 4-7].

214       Upon further increasing pressure, another two critical pressures can be clearly  
215 discerned relating to the successive structural transitions. Accompanied by the first  
216 structural transition, superconductivity is observed at  $\sim 13$  GPa with  $T_C \sim 3.8$  K.  
217 Further increasing the pressure, the FWHM of the  $A_g$  mode is observed to enhance  
218 gradually, implying an increasing electron-phonon coupling [40, 41]. Meanwhile,  $T_C$   
219 also increases rapidly [Fig. 5(d)], probably hinting at a phonon-mediated  
220 superconductivity.  $T_C$  reaches a maximum value of  $\sim 12$  K at 21.8 GPa, around which  
221 the second structural transition happens.

222       In the tetradymite-type TIs family [12, 15-17, 42], the occurrence of  
223 superconductivity under high pressure is commonly accompanied by a structural  
224 transition or structural instability; the  $T_C$  increases rapidly and reaches a maximum at

225 a higher pressure, where a second structural transition appears. This is also true for the  
226 present case of Sn-BSTS. However, the  $T_C$  trend in the second high-pressure  
227 structural phase is somewhat different. In the Se-dominated compounds, for example  
228  $\text{Bi}_2\text{Se}_3$  [15] and  $\text{Sr}_{0.065}\text{Bi}_2\text{Se}_3$  [42],  $T_C$  keeps almost constant with pressure, which was  
229 considered as an indication of unconventional superconductivity; in Te-dominated  
230 compounds, for example  $\text{Bi}_2\text{Te}_3$  [12] and here Sn-BSTS,  $T_C$  decreases gradually with  
231 pressure. This might be related to difference in the high-pressure structural  
232 symmetries: generally a tetragonal  $I4/mmm$  or a BCC-like  $C2/m$  lattice develops for  
233 the former while a cubic  $Im-3m$  lattice for the latter. Interestingly, the  $T_C$ -maximum  
234 obtained here is the highest value in this TIs family studied so far.

235

#### 236 **IV. CONCLUSIONS**

237 In summary, we have investigated the high-pressure electrical, structural and  
238 vibrational properties of TI Sn-doped  $\text{Bi}_{1.1}\text{Sb}_{0.9}\text{Te}_2\text{S}$  with pressures up to 37.2 GPa.  
239 With increasing pressure, a pressure-induced metallization from closure of the bulk  
240 gap is revealed around 9 GPa, accompanied by a change in slope of the Raman modes  
241 and a minimum in the lattice ratio of  $c/a$ . Furthermore, superconductivity is observed  
242 at 12.6 GPa and  $T_C$  reaches the maximum of  $\sim 12$  K at 21.8 GPa, which is the highest  
243  $T_C$  ever reported in tetradymite-type TIs. Based on the synchrotron XRD and Raman  
244 measurements, the appearance of superconductivity and the decrease in  $T_C$  **could be**  
245 **related** to successive structural transitions.

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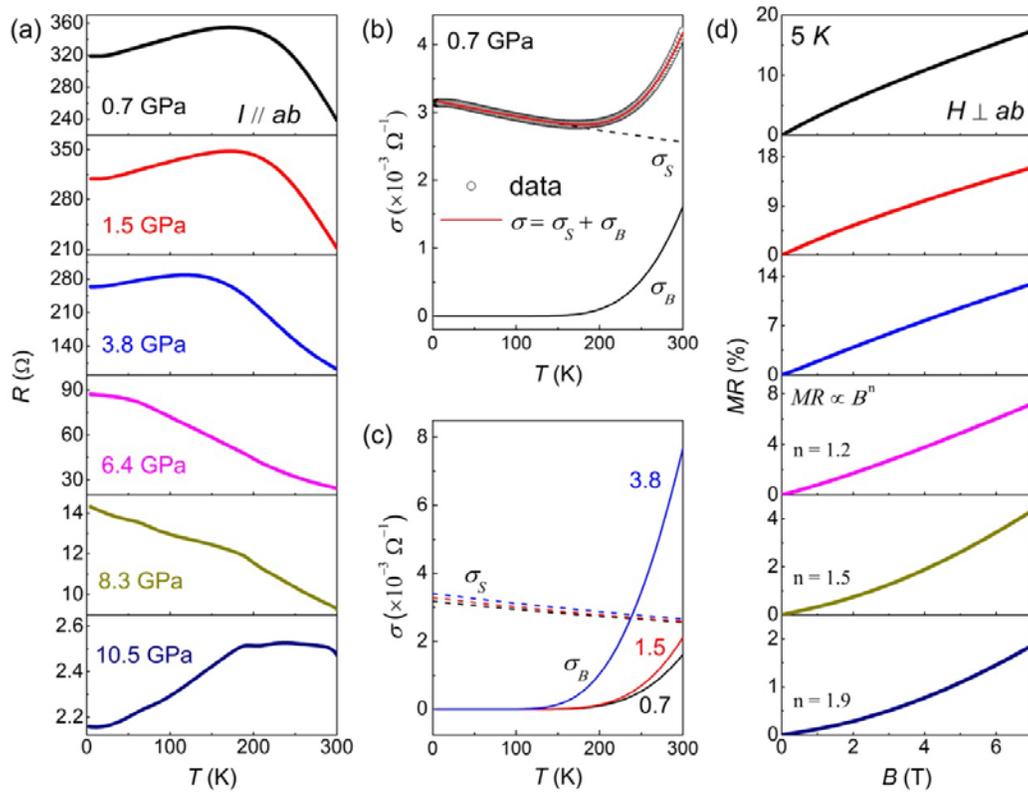
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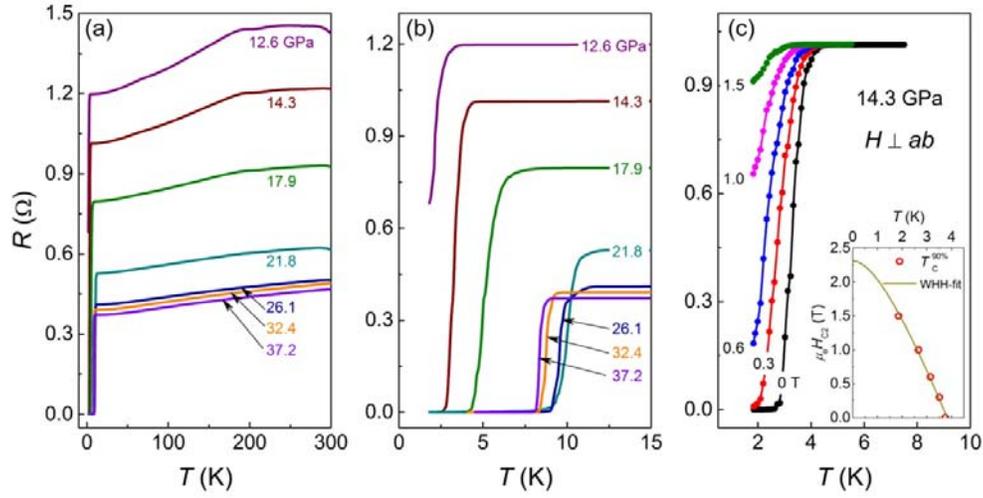
400 **FIGURES AND CAPTIONS**

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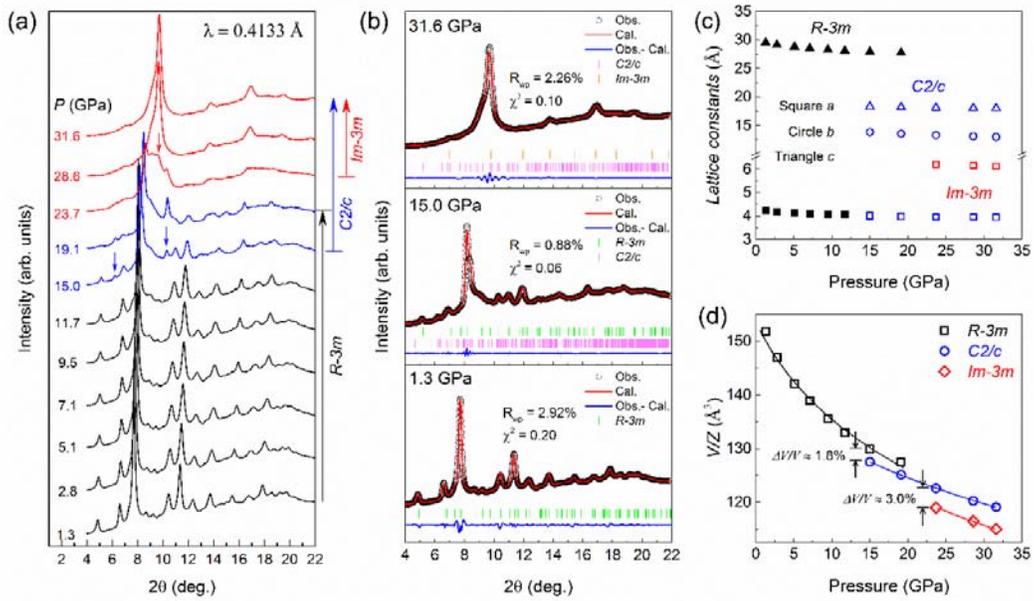
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403 FIG. 1. (Color online) (a) Resistance ( $R$ ) versus temperature ( $T$ ) curves in the pressure  
 404 range of 0.7-10.5 GPa. The current ( $I$ ) is introduced in the  $ab$  plane of the single  
 405 crystal. (b) Temperature dependence of the conductivity at 0.7 GPa fitted by the  
 406 formula of  $\sigma(T) = \sigma_S(T) + \sigma_B(T)$  (red line). More details can be found in the main  
 407 text. (c) Evolutions of the surface and bulk conductivities with pressures to 3.8 GPa.  
 408 (d) Magnetoresistance ( $MR$ ) measured at 5 K and 7 T under pressures to 10.5 GPa.

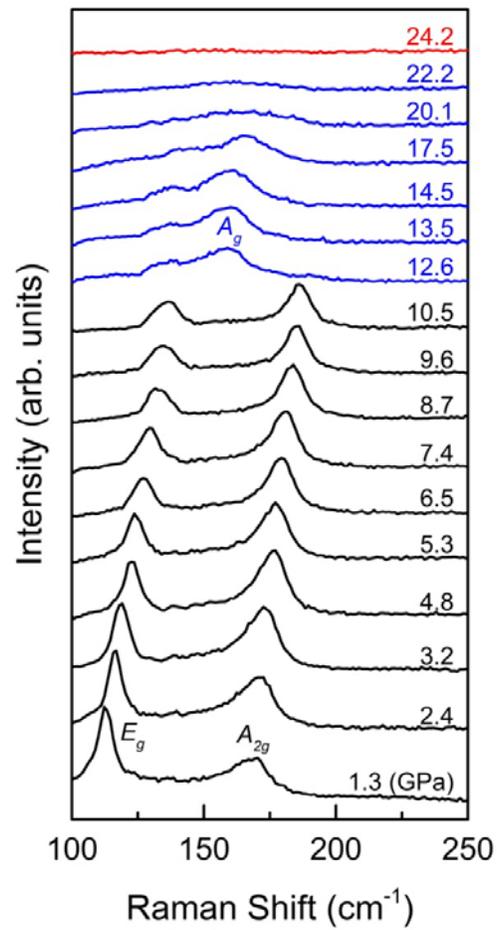


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410 FIG. 2. (Color online) (a) Temperature dependence of the resistance in the pressure  
 411 range of 12.6~37.2 GPa. (b) An enlarged view of the low temperature resistance,  
 412 highlighting the superconducting transition. (c) Temperature dependence of the  
 413 resistance under different magnetic fields up to 1.5 T. The applied pressure is 14.3  
 414 GPa. Inset of Figure (c) shows temperature dependence of the upper critical field  
 415  $\mu_0 H_{C2}(T)$ . The solid line is the WHH fit to the data and  $\mu_0 H_{C2}(0)$  is estimated to be  
 416 2.31 T.



417  
 418 FIG. 3. (Color online) (a) Synchrotron x-ray diffraction patterns of Sn-BSTS at room  
 419 temperature with pressures up to 31.6 GPa ( $\lambda=0.4133$  Å). (b) Typical Rietveld  
 420 refinement results at 1.3 GPa, 15.0 GPa and 31.6 GPa by using the Le Bail method. (c)  
 421 Lattice parameters as a function of pressure. (d) The unit-cell volume versus pressure.  
 422 The data was fitted by the third-order Birch-Murnaghan formula.  
 423

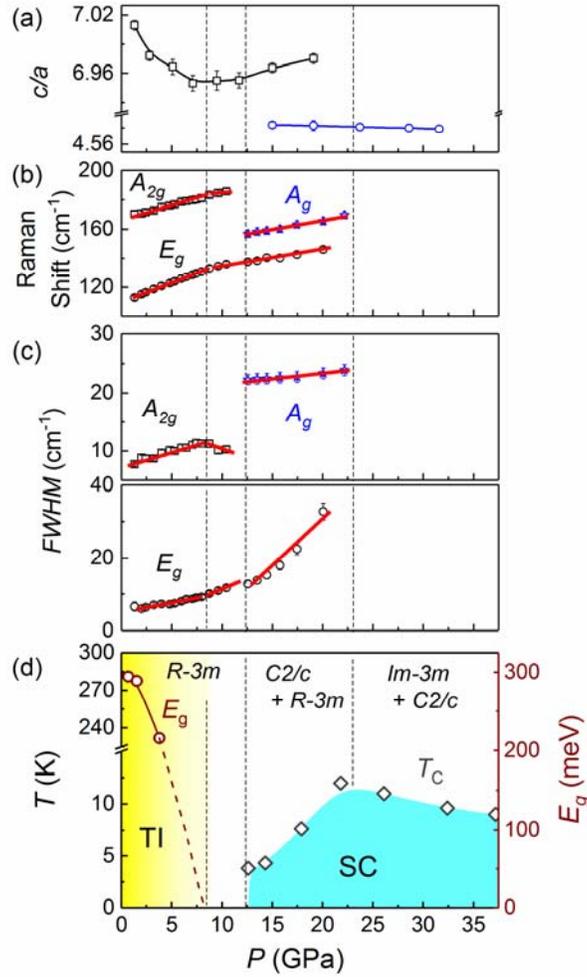


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425 FIG. 4. (Color online) Selected Raman spectra of Sn-BSTS at room temperature with

426 pressures up to 24.2 GPa.

427



428

429 FIG. 5. (Color online) (a) Lattice ratio of  $c/a$  as a function of pressure. (b) Pressure  
 430 dependence of Raman frequencies of Sn-BSTS. Solid lines are linear fits to data. (c)  
 431 Pressure dependence of FWHM of Sn-BSTS. Solid line denotes linear fits to data. (d)  
 432 Pressure-temperature diagram of Sn-BSTS. The left axis stands for temperature  $T$  and  
 433 the right axis corresponds to the bulk bandgap  $E_g$ . “TI” and “SC” denote topological  
 434 insulators and superconductivity, respectively.