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Bulk electronic structure of non-centrosymmetric $EuTGe_3$ (T=Co, Ni, Rh, Ir) studied by hard x-ray photoelectron spectroscopy

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Non-centrosymmetric $\operatorname{Eu}T\operatorname{Ge}_3$ ($T=\operatorname{Co}$, Ni, Rh, and Ir) possesses magnetic Eu^{2+} ions and antiferromagnetic ordering appears at low temperatures. Transition metal substitution leads to changes of the unit cell volume and of the magnetic ordering. However, the magnetic ordering temperature does not scale with the volume change and the Eu valence is expected to remain divalent. Here we study the bulk electronic structure of non-centrosymmetric $\operatorname{Eu}T\operatorname{Ge}_3$ ($T=\operatorname{Co}$, Ni, Rh, and Ir) by hard x-ray photoelectron spectroscopy. The Eu 3d core level spectrum confirms the robust Eu^{2+} valence state against the transition metal substitution with a small contribution from Eu^{3+} . The estimated Eu mean-valence is around 2.1 in these compounds as confirmed by multiplet calculations. In contrast, the Ge 2p spectrum shifts to higher binding energy upon changing the transition metal from 3d to 4d to 5d elements, hinting of a change in the Ge-T bonding strength. The valence bands of the different compounds are found to be well reproduced by ab initio band structure calculations.

I. INTRODUCTION

Strongly correlated 4f-electron systems have been a platform for studying various anomalous phenomena, such as valence fluctuations, unconventional superconductivity, heavy fermion behavior and spin/charge ordering^{1,2}. The ground state property of these compounds are characterized by competing Kondo effects or Rudermann-Kittel-Kasuya-Yoshida (RKKY) interactions. Both interactions originate from the interplay of localized f electrons and itinerant conduction electrons, though the former quenches the magnetic moments, while the latter leads to magnetic ordering in the ground state. The competition between Kondo effect and RKKY interactions in Ce- and Yb-compounds are often discussed within the Doniach phase diagram³. In the vicinity of the quantum critical point (QCP), where the non-thermal parameter controlled phase transition happens at absolute zero temperature, particularly, quantum fluctuations accommodate exotic phenomena⁴.

Eu-compounds exhibit very different phase diagrams from Ce- and Yb-compounds, and an absence of a QCP. Most of the reported Eu-compounds favor a $\mathrm{Eu^{2+}}$ ($4f^7$, J=7/2) valence state with an antiferromagnetic ground state. However, the energy difference between $\mathrm{Eu^{2+}}$ and the non-magnetic $\mathrm{Eu^{3+}}$ ($4f^6$, J=0) valence state is not so large⁵ and is reachable by applying external pressure or chemical substitution. Indeed, amongst the most extensively studied Eu-compounds series with the $\mathrm{ThCr_2Si_2}$ -type crystal structure, pressure or chemical substitution controlled first-order valence transitions

and valence fluctuations are frequently reported⁶. In the $\text{Eu}(\text{Pd}_{1-x}\text{Au}_x)_2\text{Si}_2$ system, EuAu_2Si_2 possesses a Eu^{2+} valence state and exhibits antiferromagnetic ordering below the Néel temperature (T_N) of $\sim 15.5 \text{ K}^7$. Substitution of smaller Pd ions decreases the lattice parameter and, by contrast, increases $T_{\rm N}$. Above $x \sim 0.25$, the magnetic transition is suddenly taken over by a first-order valence transition to ${\rm Eu}^{3+}.^{8}$ The Eu valence deviates from integer values, to so called intermediate valence states, and is ~ 2.8 in EuPd₂Si₂ below 150 K⁹. A similar tendency is also reported for $Eu(Pt_{1-x}Ni_x)_2Si_2^{10}$ and $\text{EuNi}_2(\text{Si}_{1-x}\text{Ge}_x)_2^{11}$ systems in such a way that the substitution by elements with small ionic radii works in the same way as pressure and leads to a non-magnetic ground state. Application of external pressure shows a consistent behavior compared with chemical substitutions¹². Due to the different ionic size between Eu²⁺ and Eu³⁺, the change of the Eu valence state is often assigned to Kondo volume collapse effects^{13,14}. In contrast, changes of the Eu valence state and ground state property are found to be independent of the volume effect in $Eu(Rh_{1-x}Ir_x)_2Si_2$ system. The conversion from divalent $EuRh_2Si_2$ to valence-fluctuating $EuIr_2Si_2$ involves only a $\sim 1.5\%$ volume change, indicating its origins in electronic structure changes¹⁵. Very recently, exotic behavior has been discovered in EuRhSi₃ and Eu₂Ni₃Ge₅¹⁶⁻¹⁸ that cannot be explained by the conventional phase diagram of Eucompounds. Both EuRhSi₃ and Eu₂Ni₃Ge₅ have magnetic Eu²⁺ ions and exhibit antiferromagnetism below $T_{\rm N}$ =49 K and 19 K, respectively, at ambient pressure. Electrical resistivity measurements under pressure have

reported the suppression of a magnetic ordering temperature and a successive phase transition to a non-magnetic heavy fermion state without a hint of a valence transition^{17,18}. These behaviors are similar to Ce- and Yb-compounds and indicate the possible existence of a QCP. These new aspects of Eu-compounds urge a systematic study of the electronic structure and its relation to physical properties.

Eu-based ternary germanides $EuTGe_3$ (T=Co, Ni, Rh,Ir) and EuRhSi₃ are isostructural, possess a BaNiSn₃type structure (I4mm) which is similar to the ThCr₂Si₂type structure, though without centrosymmetry¹⁹. Magnetic susceptibility measurements^{19–22} and Mössbauer spectroscopy^{23,24} report the presence of magnetic Eu²⁺ ions in all the compounds and localized Eu 4f moments order antiferromagnetically at similar temper-The magnetic moments order antiferromagnetically along the c-axis at $T_N=15.4$, 13.5 and 12.3 K for EuCoGe₃, EuNiGe₃ and EuIrGe₃, respectively. EuCoGe₃ and EuIrGe₃ exhibit additional magnetic transitions at 13.4 and 7.5 K, respectively, due to a change of the antiferromagnetic structure 19,22. Recently, three antiferromagnetic phases have been discovered in EuIrGe₃ and a helical magnetic structure based on the Dzyaloshinskii-Moriya interaction was proposed.²⁵ Conversely in EuRhGe3, the magnetic moments order perpendicular to the c-axis at \sim 12 K^{19,22}. For each compound, the effective magnetic moments are close to the $\mathrm{Eu^{2+}}$ ionic value of 7.90 μ_{B}^{19} . All the transition metals are nonmagnetic in EuTGe₃. Despite the variation of transition metal substitution and the change in the unit cell volume, the Eu ions seems to have a robust $\mathrm{Eu^{2+}}$ valence state with T_N being barely affected. In the $\text{EuNi}(\text{Si}_{1-x}\text{Ge}_x)_3$ system, transport measurements reported a monotonous decrease of $T_{\rm N}$ with an increase in Ge substitution indicating its strong connection with the volume change²⁶. However, the change of $T_{\rm N}$ in EuTGe₃ by transition metal substitution does not show a propotional change with the unit cell volume. This implies that variation of the physical properties of the EuTGe₃ by transition metal substitution is rather dominated by the change in electronic structure rather than the unit cell volume effect. In order to study the transition metal substitution effect on the electronic structure of $EuTGe_3$, we performed hard x-ray photoelectron spectroscopy (HAX-PES). By using the bulk sensitive HAXPES method, we can unambiguously determine the Eu valence from Eu 3dcore level spectra and suppress the surface contribution. The Eu 3d core level spectra confirmed that the Eu²⁺ valence state is robust against transition metal substitution. The estimated Eu valence is close to 2.1. In contrast, Ge 2p core level spectrum shifts to high binding energy by changing transition metal from 3d to 4dto 5d elements. A similar trend was observed in the Eu 4f spectrum in the valence band. We compare the measured valence band electronic structure with ab initio band structure calculations.

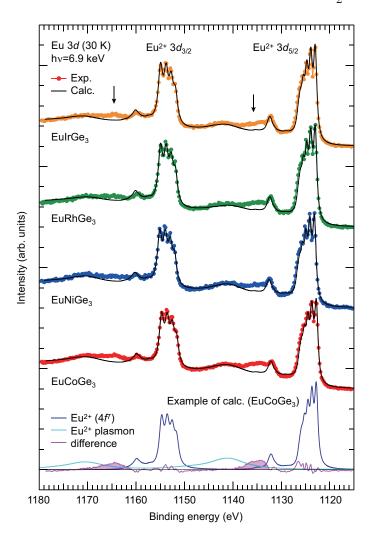


FIG. 1. (Color online). Eu 3d core level spectra of EuTGe₃ (T=Co, Ni, Rh and Ir). The experimental data are displayed using filled circles. The solid black lines represent the simulated spectra from atomic multiplet calculations with a $4f^7$ configuration including plasmon satellites and an integral background³⁴. The arrows indicate the position of Eu³⁺ components. The simulated Eu²⁺ spectrum (dark blue line) and its plasmon satellites (light blue line) for EuCoGe₃ are shown in the bottom of the figure as an example. The difference (purple line) spectrum obtained by subtracting the simulated spectrum from the experimental one reveals the Eu³⁺ components (purple area).

II. EXPERIMENTAL

HAXPES measurements were performed at the GALAXIES beamline^{27,28} of the SOLEIL synchrotron. The incident energy was selected by using the third order of the Si(111) monochromator ($h\nu$ =6.9 keV) yielding a photon bandwidth of ~ 200 meV. The photon beam was linearly polarized with the electrical field vector in the plane of the storage ring. Photoelectrons were collected by using a hemispherical analyzer EW4000 (VG Scienta). The binding energy of spectra was calibrated

by measuring the Fermi edge of a Au film. The overall energy resolution was estimated to be $\sim 250 \text{ meV}$ from Au Fermi edge fitting. EuTGe₃ (T=Co, Ni, Rh and Ir) single crystals were grown by the metal-flux method¹⁹. The grown crystals were characterized by xray diffraction, magnetic susceptibility and electrical resistivity measurements. The clean surfaces of the samples were obtained by fracturing *in-situ* under vacuum (better than 5×10^{-8} mbar) and immediately transferred into the analysis chamber with a base pressure of 5×10^{-9} mbar. The samples were aligned in a grazing incidence (normal emission) geometry. In order to avoid irradiation damage, all the measurements were performed at the lowest reachable temperature of 30 K. The EuCoGe₃ spectra have been collected within 6 hours (multi bunch operation mode: 450 mA with using attenuator to reduce the beam intensity down to 15%) and those for EuNiGe₃, EuRhGe₃ and EuIrGe₃ (single bunch operation mode:16 mA) have been collected within 12 hours after cleaving under the pressure 5×10^{-9} mbar. In order to check for oxidation or contamination, we have observed wide-range and Eu 3d spectra for each compound just after the cleaving and at the end of measurement which showed no remarkable changes. Furthermore, the accumulation of Eu 3d and valence band spectra was divided to several sets and reproducibility of the spectra has been checked.

III. RESULTS AND DISCUSSION

Figure 1 shows the Eu 3d core level spectra of EuTGe₃ (T=Co, Ni, Rh and Ir) measured at 30 K. The Eu 3d spectra are split into a $3d_{5/2}$ (1120-1145 eV) and $3d_{3/2}$ (1150-1175 eV) components due to spin-orbit interaction. Each spin-orbit partner further splits into a Eu²⁺ component at lower binding energy and the Eu³⁺ component at higher binding energy representing the Eu $4f^7 \rightarrow \underline{c} \ 4f^7 + e$ and Eu $4f^6 \rightarrow \underline{c} \ 4f^6 + e$ transitions, respectively. Here, \underline{c} denotes a 3d core hole and e the outgoing photoelectron. Broad structures around 1140 and 1170 eV are attributed to plasmon statellites related to the Eu²⁺ 3d photoemission process. Compared to the Eu 3d spectrum of other divalent Eu compounds, such as EuRh₂Si₂ (mean-Eu valence $v \sim 2.1$ at 300-20 K)²⁹ and EuNi₂(Si_{0.21}Ge_{0.79})₂ ($v \sim 2.2$ at 300 K)³⁰, the relative intensity of Eu^{3+} components to those of Eu^{2+} is small and buried in the tail of the Eu²⁺ components and its statellite structures. As expected from magnetic susceptibility and Mössbauer measurements, the Eu valence states in the $EuTGe_3$ series are very close to Eu^{2+} . In order to elucidate Eu³⁺ contributions, a simulation analysis was performed by carrying out atomic multiplet calculations to account for the lineshape of the Eu 3d core level spectra. The Eu 3d spectra were simulated by using the XTLS (version 9.01) $code^{31}$ with a $4f^7$ (Eu²⁺) groundstate configuration. The electrostatic and exchange parameters were obtained by Cowan's atomic Hartree-Fock program with relativistic corrections³². The exchange parameters were scaled down to 86% of their Hartree-Fock values. The calculated spectra are convoluted with a Lorentzian function for lifetime broadening and a Gaussian to account for the experimental resolution. The broadening parameters as well as the values used for the Coulomb and exchange multiplet interactions are listed in Ref. 33. An example of the simulation for EuCoGe₃ is shown in the bottom of Fig. 1. The plasmon satellites (light blue line) are reproduced by broadening the simulated Eu²⁺ atomic multiplet spectrum and shifting in order to be in agreement with the experimental energy. Their relative intensity and the energy position to the Eu²⁺ 3d components were calibrated using the Ge 2p peak and its plasmon position (see inset of Fig. 2 (b)). The solid black lines in Fig. 1 represent the simulated spectra including the atomic multiplet spectrum, plasmon satellites and integral background³⁴. The experimental spectra are fitted by adjusting the intensity of the calculated spectra such that the difference between the experimental and the calculated spectra are minimized. As seen in Fig. 1, the simulations can well reproduce the multiplet structures of the experimental spectra. Since the simulations only take into account the Eu²⁺ contribution, the deviations from the simulated spectrum at \sim 1135 and 1165 eV are assigned to the Eu³⁺ contributions. We extracted the Eu³⁺ component by subtracting the simulated spectrum from that of experiment. A contribution from the Eu^{3+} plasmon satellites to the Eu 3dspectrum is negligibly small and therefore not considered in this analysis. Some residual wiggling feature on the difference spectrum (purple line) originates mostly from tiny deviations in the peak positions and peak widths of the multiplet structures. The Eu valence was estimated by using the formula $v = 2 + I_{3+}/(I_{2+} + I_{3+})$. Here, I_{2+} and I_{3+} denote integrated spectral intensities of the simulated Eu²⁺ spectrum (dark blue line) and the extracted Eu³⁺ component (purple area in Fig. 1), respectively. The obtained Eu valences are v=2.11, 2.09, 2.08 and 2.09 (±0.01) for EuCoGe₃, EuNiGe₃, EuRhGe₃ and EuIrGe₃, respectively. We should note that the estimated inelastic mean-free path is $\sim 73 \text{ Å for } 5.7 \text{ keV photoelectrons}^{35}$, therefore, the Eu³⁺ signal is not likely to be coming from the surface states.

Figure 2(a) shows spectra with the Ge $2p_{3/2}$ component for all the compounds. The inset shows the Ge 2p complete spectra. The Ge 2p spectra tends to shift towards higher binding energy upon going from 3d to 4d to 5d elements. In order to precisely obtain this energy shift, a fitting analysis was performed on the spectra by using Gaussian and Lorentzian functions³⁶. In addition, a Mahan function³⁷ with $\alpha = 0.16$ is used to account for the asymmetry of the line shapes. The result of the fitting analysis is plotted in Fig. 2 (b) with an example of such a fit in the inset. A large energy shift of ~ 250 meV is observed between EuCoGe₃ and EuNiGe₃. The energy shift of the Ge 2p peak for EuRhGe₃ and EuIrGe₃ relative to EuCoGe₃ are ~ 120 and 180 meV, respectively. We note that the Fermi level $(E_{\rm F})$ was carefully checked just

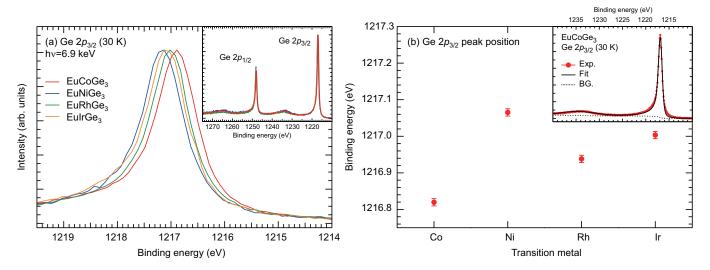


FIG. 2. (Color online) (a) Ge $2p_{3/2}$ core level spectra measured at 30 K. The inset shows the complete Ge 2p spectra. (b) Binding energy of the Ge $2p_{3/2}$ peaks of EuTGe₃ as a function of T (T=Co, Ni, Rh, and Ir). The inset shows an example of the fit. The dashed line represents the integral background³⁴ (BG).

before and after each measurement to correct for possible drift of the incident photon energy. The energy shift between $EuCoGe_3$ and $EuNiGe_3$ can be understood as a result of the energy shift of $E_{\rm F}$ towards higher-energy in the conduction bands with increasing 3d occupation.

In the related ternary compounds in the form of $RT_2\text{Ge}_2$ (R: Rare earth, T: transition metal), the strong T-Ge bonding based upon hybridization of T-d states with Ge-sp states is reported^{38,39}. In a similar vein, a strong bonding between T-Ge can also be expected in the $\text{Eu}T\text{Ge}_3$ family as well. The energy shift of the $\text{Ge}\ 2p$ binding energy amongst EuCoGe_3 , EuRhGe_3 and EuIrGe_3 indicates the change of bonding by transition metal substitution.

In order to study the atomic orbital character of the valence band electronic structure, we performed band structure calculations using the full-potential nonorthogonal local orbital code (FPLO)^{40,41}. The local density approximation (LDA) with the Perdew and Wang flavor⁴² of the exchange and correlation potential was chosen. Additionally, the strong Coulomb replusion between the Eu 4f electrons was included in a mean-field way by applying the LDA+U method. The calculations were performed for the experimentally obtained lattice parameters reported in Ref. 19 with $J_{\rm H}$ =0.7 eV and varying U from 5 to 7 eV. It should be noted that varying Uonly changes the energy separation between the filled and unfilled Eu 4f states and does not change the results qualitatively. The calculated total and partial density of states (PDOS) are presented in Fig. 3. The europium ions in the unit cell are configured such that they have a ferromagnetic arrangement in the ab-plane and antiferromagnetically aligned along the c-axis. Therefore, the DOSs are symmetric for both the spin channels except for Eu 4f. The DOSs in the minority-spin channel are inverted and added together with the majority-spin DOSs for comparison to experiments. The majority-spin states of Eu 4f (solid red line) are fully occupied and appear as a localized sharp peak around 1 eV while the Eu 4f minority-spin states (dashed red line) remain unoccupied. These results reflect the magnetic Eu $4f^7$ state in all compounds. The Co and Ni 3d PDOSs appear centered at -1.5 eV and -2 eV, respectively. An increase of the 3d electron number in EuNiGe₃ shifts the center of 3d PDOS away from $E_{\rm F}$ that decreases the hybridization to Eu 4f. The Rh 4d and Ir 5d PODSs show a more extended nature than 3d PDOS, appear from $E_{\rm F}$ to -6 eV and then below -7 eV. The occupied Ge 4p PDOS are mainly distributed from $E_{\rm F}$ to -6 eV, hybridizing with transition metal d and Eu 4f PDOSs. The Ge 4sPDOS appears at -7 to -10 eV. Commonly in all four compounds, a quasi-gap-like low DOS region appears ~ 1 eV above $E_{\rm F}$.

The experimentally measured valence band spectra after integral background correction³⁴ are displayed in Fig. 4. The HAXPES valence band spectra cannot be directly compared to the theoretical DOSs, since the photoionization cross-section of Eu 4f states is not the only one to contribute to the spectrum. Therefore the PDOSs are weighted with the corresponding photoionization cross-sections extracted or interpolated from Ref. 43–45. Then, the simulated theoretical spectra were convoluted with a Gaussian function (FWHM: 0.3 eV) and Fermi-Dirac function of 30 K. We consider the Eu 4f, T d, and Ge 4sp states as mainly contributing to the valence band. The simulated spectra show good accordance with experimental spectra. Note that we used the DOS with U=5 eV for EuCoGe₃ and U=6 eV for other compounds, based on the agreement of Eu 4f peak position. In all the compounds, a localized Eu^{2+} 4f peak is observed around 1 eV. A broad round shape of the Eu 4f peak is due to Eu²⁺ multiplet structures⁴⁶. A peak

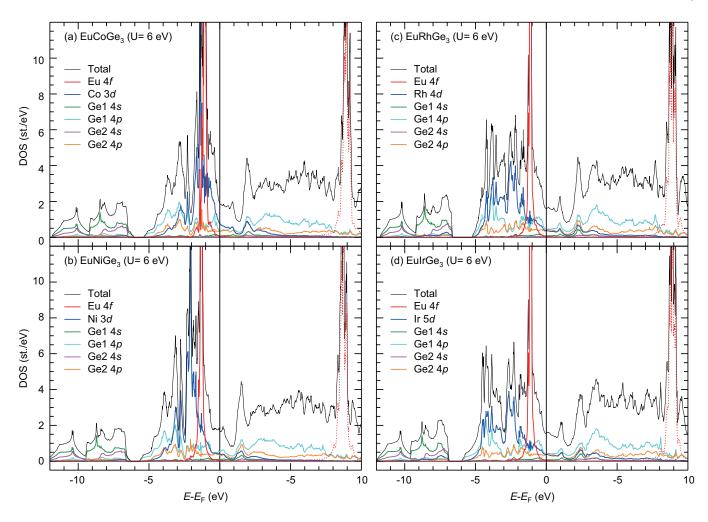


FIG. 3. (Color online) Total and partial density of states (DOS) of (a) $EuCoGe_3$, (b) $EuNiGe_3$, (c) $EuRhGe_3$ and (d) $EuIrGe_3$ simulated by LDA+U (U=6 eV) method. Since all the compounds show symmetric DOS for two spin channels, the DOSs in minority-spin channel are inverted and added together with the majority-spin DOSs, except Eu4f DOS. The majority-spin DOS of Eu4f (solid red line) are fully occupied and the minority-spin DOS (dashed red line) are unoccupied.

around 2 eV in EuNiGe₃, EuRhGe₃ and EuIrGe₃ corresponds to transition metal d states overlapping with the Ge 4sp states. Due to the larger photoionization cross-section of Ir 5d and Rh 4d compared to those of Ni 3d and Co 3d, the spectral weight of transition metal dstates is enhanced in EuIrGe₃ and EuRhGe₃. The broad structure between 6-12 eV is mainly attributed to Ge 4sstates. The Eu 4f peak shows a similar tendency as that of the Ge 2p spectrum, shifting to higher binding energy by changing transition metal atoms. However, unlike the Ge 2p spectrum, no significant energy shift was observed between EuRhGe₃ and EuIrGe₃. As seen from Fig. 3, the Eu 4f states are localized and do not hybridize with the other structural components in a significant way for all four compounds. The energy shift between the Eu 4f peaks between Co and Ni variants is due to the rigid band shift of the entire valence band spectrum due to the increase of one additional valence electron in Ni as compared to Co. To the contrary, Rh and Ir variants are isovalent with Co and therefore the Eu 4f peak remains pinned to a similar energy position. The experimental valence band spectra have low DOS at $E_{\rm F}$, especially for EuNiGe₃. The result is consistent with the reported transport measurements⁴⁷. The decrease in DOS at $E_{\rm F}$ from EuCoGe₃ to EuNiGe₃ can also be understood as a rigid-band shift due to an increase in 3d orbital occupation by substitution of Co by Ni which shifts the 3d states to higher binding energy.

Finally we comment on the pressure response of $EuTGe_3$. Recent temperature dependence of the electrical resistivity studies under pressure reported a successive increase of $T_{\rm N}$ for $EuTGe_3$ ($T={\rm Co, Ni, Rh, Ir}$) 25,47 . The antiferromagnetic ordering in $EuTGe_3$ stably exists up to 8 GPa and no sign of a Eu valence transition was observed. Our valence band spectra give an explanation to the robust Eu^{2+} magnetic states against pressure. The dominant part of the Eu 4f DOS of $EuTGe_3$ is localized at ~ 1 eV below $E_{\rm F}$ which is deeper than other Eucompounds possessing an intermediate Eu valence state or valence transition 29,48 . It hinders a charge transfer

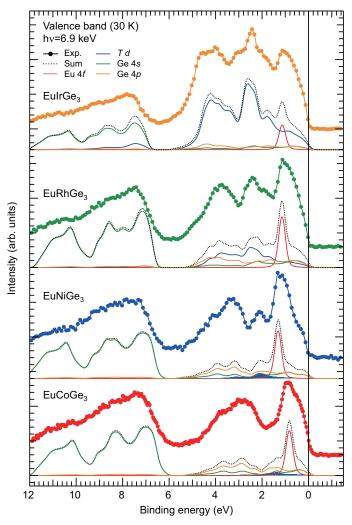


FIG. 4. (Color online) Valence band spectra (filled circles) of ${\rm Eu}T{\rm Ge}_3$ ($T{=}{\rm Co}$, Ni, Rh and Ir) after integral background correction and simulated spectra (lines). The majority- and minority-spin DOSs are added together to compare with experimental spectra. Each PDOSs are multiplied by the corresponding photoionization cross-section and convoluted by a Gaussian (0.3 eV: FWHM) and Fermi-Dirac functions at 30 K. The dashed lines denote the sum of the displayed PDOSs. Note that the intensities of simulated EuIrGe3 spectra are reduced to 50% relative to those of other compounds.

from Eu 4f to the conduction band that make up the valence fluctuation or non-magnetic Eu³⁺ states. Moreover, the calculated DOS of EuTGe₃ (see Fig. 3) has a quasi-gap like region just above the $E_{\rm F}$ that can also prevent charge transfer. In divalent antiferromagnetic EuFe₂As₂, the Eu²⁺ 4f states are localized at 1-2 eV below $E_{\rm F}^{49}$. X-ray absorption spectroscopy under pressure reported a change of antiferromagnetic to ferromagnetic ordering above 8 GPa. Althoug the Eu valence gradually increases, the Eu²⁺ magnetic moments remain up to 20 GPa⁵⁰.

Notwithstanding the small energy difference between $\mathrm{Eu^{2+}}$ and $\mathrm{Eu^{3+}}$ valence state, not all Eu contain-

ing compounds are readily manifesting valence transitions/fluctuations upon chemical substitution, pressure, magnetic field, etc. For Eu-compounds containing non-magnetic transition metal ions, the main finger print to look for is the position of the Eu 4f valence band. When the occupied 4f states are pinned to the $E_{\rm F}$, they are more susceptible to possessing a non-integral amount of 4f electrons and therefore a valence transition upon chemical substitution/pressure is highly probable. For example, the calculated and measured valence band spectrum of valence fluctuating $\operatorname{EuNi}_2(\operatorname{P}_{1-x}\operatorname{Ge}_x)_2$ and $\operatorname{EuNi}_2(\operatorname{Si}_{1-x}\operatorname{Ge}_x)_2$ respectively corroborate this scenario.^{51,52} In this case, a strong coupling of the magnetic ordering temperature of the Eumoments with substitution/pressure can be anticipated. Alternatively, valence fluctuations remain rare in materials wherein the Eu 4f states are clearly half (integral) filled and around 1 to 2 eV below the $E_{\rm F}$ (i.e. the Fermi level is pinned by the d states of the transition metal ions or the p states of the ligands). The integral filling combined with the energy position enhances the localized character of the Eu 4f electrons and thereby the RKKY exchange and weakens the valence fluctuations. The EuTGe₃ systems presented here belong to this latter category and naturally explains the insensitivity of $T_{\rm N}$ to the different chemical substitutions.

IV. CONCLUSIONS

We have performed bulk sensitive HAXPES and successfully revealed the electronic structure of EuTGe₃ (T=Co, Ni, Rh and Ir). The Eu 3d core level spectrum revealed that the Eu valence states of all the compounds are almost Eu^{2+} with negligible contribution of Eu^{3+} . The estimated Eu valence is close to 2.1. The Ge 2pcore level spectrum shows the chemical shift to higher binding energy by changing transition metal from 3d to 4d and to 5d elements indicating the change of chemical bonding between T and Ge. The valence band electronic structure was systematically studied with the support of ab initio band structure calculations. The experimental valence band spectra shows good accordance with the theoretical simulation. The Eu^{2+} 4f states are localized at \sim 1 eV below $E_{\rm F}$ in all compounds. All the compounds have a quasi-gap like region just above $E_{\rm F}$. This favours the robust Eu²⁺ magnetic state against transition metal substitution and also gives an explanation to its stability against external pressure.

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- ¹ C. M. Varma, Rev. Mod. Phys. **48**, 219 (1976).
- ² V. N. Antonov, L. V. Bekenov, and A. N. Yaresko, Adv. Condens. Matter Phys. **2011**, 107, (2011).
- ³ S. Doniach, Physica B+C **91**, 231-234 (1977).
- ⁴ Q. Si, and F. Steglich, Science **329**, 1161-1166 (2010).
- ⁵ E. R. Bauminger, D. Froindlich, I. Nowik, S. Ofer, I. Felner, and I. Mayer, Phys. Rev. Lett. 30, 1053 (1973).
- ⁶ Y. Onuki, A. Nakamura, F. Hondoa, D. Aoki, T. Tekeuchi, M. Nakashima, Y. Amako, H. Harima, K. Matsubayashi, Y. Uwatoko, S. Kayama, T. Kagayama, K. Shimizu, S. E. Muthu, D. Braithwaite, B. Salce, H. Shiba, T. Yara, Y. Ashitomi, H. Akamine, K. Tomori, M. Hedo, and T. Nakama, Phil. Mag. 97, 3399-3414 (2017).
- 7 I. Felner, J. Phys. Chem. Solids ${\bf 36},\,1063\text{-}1066$ (1975).
- ⁸ C. U. Segre, M. Croft, J. A. Hodges, V. Murgai, L. C. Gupta, and R. D. Parks, Phys. Rev. Lett. **49**, 1947 (1982).
- ⁹ E. V. Sampathkumaran, L. C. Gupta, R. Vijayaraghavan, K. V. Gopalakrishnan, R. G. Pillay, and H. G. Devare, J. Phys. C: Solid State Phys. 14 L237, (1981).
- A. Mitsuda, Y. Ikeda, N. Ietaka, S. Fukuda, and Y. Isikawa, J. Mag. Mag. Mat. **310** 319-321 (2007).
- ¹¹ H. Wada, M. F. Hundley, R. Movshovich, and J. D. Thompson, Phys. Rev. B **59**, 1141 (1999).
- ¹² H. J. Hesse, R. Lübbers, M. Winzenick, H. W. Neuling, and G. Wortmann, J. Alloys Compd. **246**, 220-231 (1997).
- ¹³ J. W. Allen, and R. M. Martin, Phys. Rev. Lett. **49**, 1106 (1982).
- ¹⁴ J. W. Allen, and L. Z. Liu, Phys. Rev. B **46**, 5047 (1992).
- ¹⁵ S. Seiro, and C. Geibel, J. Phys.: Condens. Matter 23, 375601 (2011).
- A. Nakamura, T. Okazak, M. Nakashima, Y. Amako, K. Matsubayashi, Y. Uwatoko, S. Kayama, T. Kagayama, K. Shimizu, T. Uejo, H. Akamine, M. Hedo, T. Nakama, Y. Ōnuki, and H. Shiba, J. Phys. Soc. Jpn 84, 053701 (2015).
- ¹⁷ S. E. Muthu, D. Braithwaite, B. Salce, A. Nakamura, M. Hedo, T. Nakama, Y. Ōnuki, J. Phys. Soc. Jpn. 85, 094603 (2016).
- M. Nakashima, Y. Amako, K. Matsuyabashi, Y. Uwatoko, M. Nada, K. Sugiyama, M. Hagiwara, Y. Haga, T. Takeuchi, A. Nakamura, H. Akamine, K. Tomori, T. Yara, Y. Ashitomi, M. Hedo, T. Nakama, and Y. Ōnuki, J. Phys. Soc. Jpn. 86, 034708 (2017).
- ¹⁹ O. Bednarchuk, A. Gagor and D. Kaczorowski, J. Alloys. Comp. **622**, 432-439 (2015).
- ²⁰ R. J. Goetsch, V. K. Anand, and D. C. Johnston, Phys. Rev. B 87, 064406 (2013).
- ²¹ O. Bednarchuk, and D. Kaczorowski, J. Alloys. Comp. 646, 291-297 (2015).
- O. Bednarchuk, and D. Kaczorowski, A. Phys. Pol. A 127, 418-420 (2015).
- ²³ A. Maurya, P. Bonville, A. Thamizhavel, and S. K. Dhar, J. Phys.: Condens. Matter 26, 216001 (2014).

- ²⁴ A. Maurya, P. Bonville, R. Kulkarni, A. Thamizhavel, and S. K. Dhar, J. Mag. Mag. Mat. **401**, 823-831 (2016).
- M. Kakihana, H. Akamine, K. Tomori, K. Nishimura, A. Teruya, A. Nakamura, F. Honda, D. Aoki, M. Nakashima, Y. Amako, K. Matsubayashi, Y. Uwatoko, T. Takeuchi, T. Kida, M. Hagiwara, Y. Haga, E. Yamamoto, H. Harima, M. Hedo, T. Nakama, and Y. Ōnuki, J. Alloys and Comp. 694, 439-451 (2017).
- ²⁶ K. Uchima, Y. Takaesu, H. Akamine, M. Kakihana, K. Tomori, T. Uejo, A. Teruya, A. Nakamura, M. Hedo, T. Nakama, K. Yagasaki, K. Matsubayashi, and Y. Uwatoko, J. Phys.: Conf. Ser. **568**, 042032 (2014).
- ²⁷ D. Céolin, J. M. Ablett, D. Prieur, T. Moreno, J. P. Rueff, T. Marchenko, L. Journel, R. Guillemin, B. Pilette, T. Marin, and M. Simon, J. Electron Spectrosc. Relat. Phenom. **190**, 188-192 (2013).
- ²⁸ J. P. Rueff, J. M. Ablett, D. Céolin, D. Prieur, T. Moreno, V. Balédent, B. Lassalle-Kaiser, J. E. Rault, M. Simon, and A. Shukla, J. Synchrotron Rad. 22, 175-179 (2015).
- ²⁹ K. Ichiki, T. Matsumoto, H. Anzai, R. Takeshita, K. Abe, S. Ishihara, T. Uozumi, H. Sato, A. Rousuli, S. Ueda, Y. Taguchi, K. Shimada, H. Namatame, M. Taniguchi, S. Hamano, A. Mitsuda, H. Wada, and K. Mimura, J. Electron Spectrosc. Relat. Phenom. 220, 28-32 (2017).
- ³⁰ K. Ichiki, K. Mimura, H. Anzai, T. Uozumi, H. Sato, Y. Utsumi, S. Ueda, A. Mitsuda, H. Wada, Y. Taguchi, K. Shimada, H. Namatame, and M. Taniguchi, Phys. Rev. B. 96, 045106 (2017).
- ³¹ A. Tanaka, and T. Jo, J. Phys. Soc. Jpn. **63**, 2788-2807 (1994).
- ³² Ř. D. Cowan, J. Opt. Soc. Am. **58**, 808-818 (1968); The Theory of Atomic Structure and Spectra (University of California Press, Berkeley, 1981).
- The value of the F and G Slater integrals and the spinorbit interaction (ζ) used for the atomic multiplet calculations were obtained by Cowan's atomic Hartree-Fock (HF) program³². The obtained HF values are reduced to 86% for the simulations of the Eu²⁺ spectrum. The parameters of the $3d^{10}4f^7$ and $3d^94f^7$ configurations in the simulations of the Eu²⁺ 3d spectra are: $(F_f^2, F_f^4, F_f^6, \zeta_f)$ = (11.23, 7.00, 5.03, 0.160) and $(F_f^2, F_f^4, F_f^6, \zeta_f)$ = (12.79,8.04, 5.75, 0.202), respectively, and $(F_{fd}^2, F_{fd}^4, G_{fd}^1, G_{fd}^3,$ $G_{fd}^5, \zeta_d)$ = (9.85, 3.76, 5.75, 3.37, 2.33, 11.24). All values are in units of eV. The simulated spectra are convoluted with Lorentzian functions with FWHM=0.7 and 1.0 eV for Eu²⁺ $3d_{5/2}$ and $3d_{3/2}$, respectively, and a Gaussina function with FWHM=0.25 eV.
- 34 D. A. Shirley, Phys. Rev. B 5, 4709 (1972).
- ³⁵ S. Tanuma, C. J. Powell, and D. R. Penn, Surf. Interface. Anal. **21**, 165 (1994).
- ³⁶ The Ge $2p_{3/2}$ peak and its plasmon satellite were fitted by a Gaussian function with FWHM=0.3 eV, and Lorentzian functions with FWHM=0.8 eV and 8 eV, respectively.
- ³⁷ G. D. Mahan, Phys. Rev. B **11**, 4814, (1975).
- ³⁸ Y. Jeon, B. Qi, F. Lu, and M. Croft, Phys. Rev. B 40, 1538 (1989).

- ³⁹ J. Chen, E. Kemly, M. Croft, Y. Jeon, X. Xu, S. A. Shaheen, and P. H. Ansari, Solid State Commun. 85, 291-296 (1993).
- ⁴⁰ K. Koepernik, B. Velický, R. Hayn, and H. Eschrig, Phys. Rev. B **55**, 5717 (1997).
- ⁴¹ K. Koepernik, and H. Eschrig, Phys. Rev. B **59**, 1743 (1999).
- ⁴² J. P. Perdew, and Y. Wang, Phys. Rev. B **45**, 13244 (1992).
- ⁴³ M. B. Trzhaskovskaya, V. I. Nefedov, and V. G. Yarzhemsky, Atomic Data Nuclear Data Tables 77, 97-159 (2001).
- ⁴⁴ M. B. Trzhaskovskaya, V. I. Nefedov, and V. G. Yarzhemsky, Atomic Data Nuclear Data Tables 82, 257-311 (2002).
- ⁴⁵ M. B. Trzhaskovskaya, V. K. Nikuliln, V. I. Nefedov, and V. G. Yarzhemsky, Atomic Data Nuclear Data Tables 92, 245-304 (2006).
- ⁴⁶ F. Gerken, J. Phys. F: Met. Phys. **13**, 703 (1983).
- ⁴⁷ K. Uchima, N. Arakaki, S. Hirakawa, Y. Hiranaka, T. Uejo, A. Teruya, A. Nakamura, M. Takeda, Y. Takaesu, M. Hedo, T. Nakama, K. Yagasaki and Y. Uwatoko, JPS

- Conf. Proc. 1, 012015 (2014).
- ⁴⁸ K. Mimura, Y. Taguchi, S. Fukuda, A. Mitsuda, J. Sakurai, K. Ichikawa, and O. Arita, Physica B 351, 292-294 (2004).
- ⁴⁹ G. Adhikary, N. Sahadev, D. Biswas, R. Bindu, N. Kumar, A. Thamizhavel, S. K. Dhar, and K. Maiti, J. Phys.: Condens. Matter 25, 225701 (2013).
- ⁵⁰ R. S. Kumar, Y. Zhang, A. Thamizhavel, A. Svane, G. Vaitheeswaran, V. Kanchana, Y. Xiao, P. Chow, C. Chen, and Y. Zhao, Appl. Phys. Lett. **104**, 042601 (2014).
- ⁵¹ U. B. Paramanik, A. Bar, D. Das, N. Caroca-Canales, R. Prasad, C. Geible, and Z. Hossain, J. Phys.: Condens. Matter 28, 166001 (2016).
- ⁵² T. Kinoshita, H. P. N. J. Gunasekara, Y. Takata, S. Kimura, M. Okuno, Y. Haruyama, N. Kosugi, K. G. Nath, H. Wada, A. Mitsuda, M. Shiga, T. Okuda, A. Harasawa, H. Oghasawara, and A. Kotani, J. Phys. Soc. Jpn. **71**, 148-155 (2002).