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Phys. Rev. B 95, 184110 - Published 30 May 2017
DOI: 10.1103/PhysRevB.95.184110

# Global search for low-lying crystal structures using the artificial force induced reaction method: A case study on carbon 

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#### Abstract

We propose a novel approach to perform the global search for low-lying crystal structures from first principles, by combining the artificial force induced reaction (AFIR) method and the periodic boundary conditions (PBCs). The AFIR method has


been applied extensively to molecular systems to elucidate the mechanism of chemical reactions such as homogeneous catalysis. The present PBC/AFIR approach found 274 local minima for carbon crystals in the $\mathrm{C}_{8} /$ unit-cell described by the GGA/PBE functional. Among many newly predicted structures, three low-lying structures, which exhibit somewhat higher energy compared with those previously predicted, such as $\mathrm{Cco}^{-\mathrm{C}_{8}}$ ( $Z$-carbon) and $M$-carbon, are further discussed with calculations of phonon and band dispersion curves. Furthermore, approaches to systematically explore two- or one-dimensional periodic structures are proposed and applied to the $\mathrm{C}_{8} /$ unit-cell with the slab model. These results suggest that the present approach is highly promising for predicting crystal structures.

## I. INTRODUCTION

Properties of materials depend not only on their composition but also on their crystal structure. Different crystal structures are formed depending on the generation conditions. For example, carbon can take various forms, ${ }^{[1]}$ such as diamond, graphite, lonsdaleite (hexagonal diamond), fullerene, carbon nanotube (CNT), graphene, graphene nanoribbon (GNR), and so on. Furthermore, many stable structures have been predicted theoretically. $M$-carbon, ${ }^{[2,3]} \mathrm{Cco}-\mathrm{C}_{8}$ or $Z$-carbon, ${ }^{[4,5]}$ and so forth have been predicted recently from first principles. As many as 315 different structures have already been registered to the Samara carbon allotrope database (SACADA), ${ }^{[6]}$ and there has been a debate regarding how to handle newly predicted structures. ${ }^{[7,8]}$ Two- or
one-dimensional periodic structures, such as graphene and CNT, have also attracted attention as novel materials and catalysts.

In this context, the crystal structure prediction from first principles has become one of the active fields in materials science. To date, various methods have been developed. ${ }^{[9-12]}$ Methods for the searching of two-dimensional periodic structures have also been proposed. ${ }^{[13]}$ In one sense, the crystal structure prediction is to find the most stable structure of a given atomic composition. Finding only the most stable structure would be easy in a small system like $\mathrm{C}_{8}$ in which the number of atoms in the unit-cell is only 8 . Development of methods that can find the lowest energy structure in more complex systems therefore is an important research direction. On the other hand, there would be another direction; that is to exhaustively enumerate possible low-lying structures of a given atomic composition. This still is not a trivial task even in a small system like $\mathrm{C}_{8} /$ unit-cell which is the focus of the present study.

In this paper, we combined the artificial force induced reaction (AFIR) ${ }^{[14-16]}$ method with the periodic boundary conditions (PBCs). The AFIR method has previously been employed extensively in automated exploration of structures and reaction pathways in molecular systems such as homogeneous catalysis. ${ }^{[17]}$ The present extension enables the exhaustive search for low-lying crystal structures by the AFIR method. As a case study, crystal structures of carbon described by the $\mathrm{C}_{8} /$ unit-cell were explored by the PBC version of the AFIR method. The search generated 274 local minimum (MIN) crystal structures starting from a single initial structure. The obtained extensive structural database includes not only known structures but also many
unreported structures. Additional implementations for the exploration of two- or one-dimensional periodic structures with the slab model were also introduced and applied to carbon in the $\mathrm{C}_{8} /$ unit-cell. The search yielded 122 and 49 local minimum structures for the two- and one-dimensional periodic systems, respectively. These results demonstrate the usefulness of the present approach in a systematic exploration of unknown crystal structures from first principles.

## II. METHOD

## A. AFIR method

The AFIR method pushes fragments A and B together or pulls them apart to induce a structural rearrangement. In Fig. 1, a schematic of the AFIR method is shown. The product can be reached easily simply by minimizing a model function, termed the AFIR function (dotted curve), which is composed of the potential energy surface (PES) and the force term (for the form of the force term, see ref. 14). As demonstrated previously, ${ }^{[14-16]}$ the minimization path of this function, termed the AFIR path, is a good approximation of the reaction path of the corresponding structural rearrangement. Although the first-order saddle point along the path can also be determined by geometry optimization starting from the highest energy point along the AFIR path, the saddle point optimization was not performed in this study because the present purpose was to explore MIN structures. In this study, the model collision energy parameter $\gamma$, which defines the strength of the artificial force, ${ }^{[14]}$ was set to a sufficiently large value, i.e., $1000.0 \mathrm{~kJ} / \mathrm{mol}$, so as not to limit the search area.


FIG. 1. A schematic of the AFIR method. Starting from the reactant region, the product region can be easily found by minimizing the AFIR function (dotted curve), which is composed of the PES and the force term (for the form of the force term, see ref. 14). By repeating the minimization of the AFIR function with systematically generated different force terms, it is possible to obtain many different products starting from a given initial structure.

There are two modes in the AFIR method, i.e., the multicomponent mode (MC-AFIR) and the single-component mode (SC-AFIR). The MC-AFIR was not described in this paper because it has been employed in the exploration of pathways of bimolecular or multicomponent reactions. ${ }^{[14,16]}$ In the SC-AFIR, fragments A and B were systematically defined in a given structure. The AFIR path was then computed for all the automatically defined fragment pairs. The systematic procedure to generate fragment pairs has been described in our previous reports. ${ }^{[15,16]}$ The force term with a given $\gamma$ is uniquely determined at any structure when the fragment pair is defined. By
computing AFIR paths for different fragment pairs, various pathways that depart from the initial structure and lead to new MINs can be obtained. Further applications of the same procedure to newly obtained MINs produce many more structures.

By applying the SC-AFIR to all obtained MINs, a full network of reaction pathways in the PES area accessible by a given $\gamma$ can be generated. Though such an exhaustive search can be highly demanding, some limiting search options are also available. Two such options have mainly been employed previously: a stochastic search targeted to low-energy structures and a restricted search targeted to structures having a specified chemical bond connectivity. The stochastic search option was adopted in this study to efficiently search for low-lying crystal structures, where the model temperature parameter $T_{\mathrm{R}},{ }^{[16]}$ which determines how frequently high-energy MINs are chosen, was set to a very large value of 10000.0 K to explore a wide variety of structures. In the stochastic search, the calculation was terminated if the last $P$ AFIR paths did not update the set of lowest $M$ MINs. In this study, $M$ and $P$ were set to $f$ and $3 f$, respectively, where $f$ is the number of internal degrees of freedom defined below.

## B. Internal degrees of freedom

The PBCs are used to describe periodic systems. In the search of three-dimensional crystal structures, Cartesian coordinates of three translation vectors (TVs) and all $N$ atoms in the unit-cell, i.e., $3 N+9$ Cartesian coordinates, were considered as variables. In each optimization step, a set of $3 N+3$ orthonormal vectors was defined, by eliminating 3 modes for the translational motions of all atoms along the
$x$-, $y$-, and $z$-axes and 3 modes for the rotational motions of the whole system around the $x$-, $y$-, and $z$-axes from the $3 N+9$ Cartesian coordinates. Then, geometrical displacements were invoked in the $3 N+3$ dimensional hyperspace, since neither the three translational motions of all atoms nor the three rotational motions of the whole system changed the total energy of a system without any external field.

In the case where two-dimensional periodic structures were searched, one TV was fixed on the $z$-axis and the other two TVs were restricted on the $x y$-plane. In other words, the $x$ and $y$ coordinates of the two TVs and all $N$ atoms in the unit-cell, i.e., $3 N+4$ Cartesian coordinates, were considered as variables. In this study, the length of the TV fixed on the $z$-axis was set to $20.0 \AA$. In each optimization step, a set of 3 N orthonormal vectors was defined, by eliminating 3 modes for the translational motions of all atoms along the $x$-, $y$-, and $z$-axes and 1 mode for a rotational motion of the whole system around the $z$-axis from the $3 N+4$ Cartesian coordinates. Then, geometrical displacements were invoked in the $3 N$ dimensional hyperspace.

In the case where one-dimensional periodic structures were searched, two TVs were fixed on the $z$ - and $y$-axes, respectively, and the remaining TV was restricted on the $x$-axis. Hence, the $x$ coordinate of one TV and all $N$ atoms in the unit-cell, i.e., $3 N+$ 1 Cartesian coordinates, were considered as variables. In this study, the lengths of the two TVs fixed on the $y$ - and $z$-axes were both set to $20.0 \AA$. In each optimization step, a set of $3 N-3$ orthonormal vectors was defined, by eliminating 3 modes for the translational motions of all atoms along the $x$-, $y$-, and $z$-axes and 1 mode for a rotational motion of the whole system around the $x$-axis from the $3 N+1$ Cartesian
coordinates. Then, geometrical displacements were invoked in the $3 N-3$ dimensional hyperspace.

Many electronic structure calculation programs provide the force acting on atoms in the unit-cell and the stress tensor acting on the unit-cell. To perform the search in the abovementioned coordinate system, the first-derivative of the total energy for TVs, i.e., $\partial E / \partial h$, is required. In this study, $\partial E / \partial h$ was obtained with the following equation

$$
\begin{equation*}
\left.\frac{\partial E}{\partial \mathbf{h}}\right|_{\mathrm{h}_{0}}=-V_{0} \sigma^{t} \mathbf{h}_{0}^{-1}, \tag{1}
\end{equation*}
$$

where $E$ is the potential energy, $\mathbf{h}$ is the unit-cell, $\mathbf{h}_{0}$ is the unit cell with which the stress is calculated., $V_{0}$ is the volume of the unit-cell, and $\boldsymbol{\sigma}$ is the stress tensor. The derivation of Eq. (1) is shown in Supplemental Material S2.

## C. SC-AFIR method with PBCs

In the conventional SC-AFIR, fragments A and B that are composed of several atoms in a given molecular system are pushed together or pulled apart by minimizing the AFIR function, as illustrated in Fig. 2 (a) and (b). In the present implementation, a similar procedure was applied to the atoms in the unit-cell, where atoms in fragments A and B were pushed together or pulled apart as depicted in Fig. 2 (c) and (d), respectively. The algorithm to define fragments automatically was exactly the same as that proposed previously for molecular systems. ${ }^{[15,16]}$ In addition, assuming that there were dummy atoms at the origin and also at the positions of TVs, a pair of dummy atoms were also
pushed together or pulled apart as illustrated in Fig. 2 (e) and (f), to consider the deformation of the unit-cell.


FIG. 2. Schematic of the SC-AFIR method: (a) a positive force applied to a molecular system, (b) a negative force applied to a molecular system, (c) a positive force applied to atoms in a periodic system, (d) a negative force applied to atoms in a periodic system, (e) a positive force applied to a pair of TVs, and (f) a positive force applied to the origin
and one of the TVs. A green circle indicates a dummy atom.

## D. Structural clustering

A crystal structure can be expressed by a number of different ways regarding the shape of the unit-cell and atomic arrangements in the unit-cell. For example, there were a number of ways to describe the crystal structure of graphite. With the $\mathrm{C}_{8} /$ unit-cell, graphite can be written with a nearly cubic unit-cell, a largely elongated cuboid unit-cell, a highly distorted rhombohedral unit-cell, and many others. These graphite structures have similar but slightly different energies owing to numerical errors in the first principles calculations. Without any care, the SC-AFIR search, which was targeted to low-energy structures, tended to be applied only to graphite (or lowest-lying structures in a given system). This is because the probability that the lowest-lying structures are selected is typically large in the stochastic algorithm, ${ }^{[16]}$ and in most cases, the SC-AFIR is applied to one of the lowest-lying structures expressed with different unit-cells.

To overcome this problem, structural clustering was performed during the automated search. A similarity between two structures was checked by comparing them with a super-cell, and those judged to be similar were regarded to belong to the same group. From each group, the following structures were selected as representatives.
i. The most stable structure.
ii. The structure which has the largest $\chi$, where $\chi$ is a quantity defined as the length of shortest TV divided by the longest TV.
iii. The structure which has the smallest $\varphi$, where $\varphi$ is the sum of absolute values
of inner products between all pairs of normalized TVs.
iv. The structure which has the smallest $\lambda$, where $\lambda=(1-\chi+\varphi) / 2$.

The other structures were then excluded from the search. Structures that had a highly distorted unit-cell, in which an angle between a pair of TVs was smaller than $\pi / 4$ or larger than $3 \pi / 4$, were also excluded from the search .

Some structures that can be described only with highly elongated or largely distorted unit-cells may be missed, because the above conditions avoid searches starting from structures having such unit-cells. Actually, one previously reported structure, which is written with a highly elongated unit-cell, was missed in the present automated search, as discussed in Section IV. This is a serious problem when only a small unit-cell is applicable. However, even those with an elongated and/or distorted unit-cell in a given cell-size can be obtained with nearly cubic unit-cells with a larger cell-size. Any periodic structure can be described by a nearly cubic unit-cell, when a sufficiently large unit-cell is applied.

## E. Flow of the SC-AFIR search

The search generates a set of MINs. The list of MINs obtained by the search is termed as the MIN-list. The search proceeds as follows.

1. Start from initial structure(s).
2. Choose a pair of fragments in one of the MINs in the MIN-list by the procedure described in Section II.A combined with the structural clustering introduced in Section II.D.
3. Minimize the AFIR function for the chosen fragment pair to obtain the AFIR path, which is the approximate reaction path.
4. Optimize the MINs starting from local minima along the AFIR path, and add newly obtained MINs to the MIN-list.
5. Exit from the automated search if the latest $P$ AFIR paths did not update the set of lowest $M$ MINs.
6. Return to the step 2, and continue the search.

The initial structure can either be one or more of the optimized MIN(s), or one or more random structure(s). In the case where the optimized MINs are given, these MINs are added to the MIN-list directly. Otherwise, randomly generated initial structures are optimized, and then these optimized MINs are added to the MIN-list.

In step 4, the MINs are optimized on the PES. Therefore, MINs in the MIN-list are all local minima on the PES. Minimization of both the PES in step 4 and the AFIR function in step 3 are done by the rational function optimization (RFO) method, ${ }^{[18]}$ where the RFO step is determined for the $3 N+3,3 N$, or $3 N-3$ internal degrees of freedom in three-, two-, or one-dimensional periodic systems, respectively, as discussed in Section II.B. In addition, steps 2 to 6 can be done in parallel.

After completion of the SC-AFIR calculation, not only the most stable structure but also a huge database of low-lying structures is generated. Furthermore, their network via the AFIR path is also obtained without any additional calculation. First-order saddle points can be located by optimizing the AFIR path by any path-optimization method. It is also possible to obtain the reaction path network via the
steepest descent path (SDP) starting from the first-order saddle points, by the additional SDP calculations. Systematic study on the path network in the crystal system generated by the SC-AFIR method will be the future subject.

## F. Electronic structure calculation

SIESTA3.2 ${ }^{[19-21]}$ was used to compute the energy, force acting on atoms, and stress tensor acting on a unit-cell. These calculations were performed using density functional theory (DFT) with the PBE functional and the DZP basis set. Grimme's dispersion correction, ${ }^{[22]}$ with the parameters $R_{0}=2.904 \AA$ and $C 6=4.0 \mathrm{~kJ} \cdot \AA^{6} \cdot \mathrm{~mol}^{-1}$, was added, where the value of $C 6$ was adjusted so that the present computation reproduced the inter-layer distance of graphite of $3.4 \AA$. The pseudopotential of carbon was prepared using the parameters in the GGA Pseudopotential Database, ${ }^{[23]}$ where the core correction was not considered. A Monkhorst-Pack grid was, respectively, set to $4 \times 4 \times 4,4 \times 4 \times 1$, or $4 \times 1 \times 1$, including the $\Gamma$ point in the searches for three-, two-, or one-dimensional periodic structures for the $k$-point sampling. Collinear spin alignment was taken into account. The electronic temperature was set to 5.0 K . The mesh cutoff value was set to 50.0 Ry in the automated search, and finally increased to 200.0 Ry ; all structures discussed below are optimized MINs on the PES with the higher mesh cut off value.

## G. Computational procedure

The concrete procedure in this study of searching for low-lying crystal structures
of the carbon systems proceeds along the following five steps.
(a) Generate an initial structure in the $\mathrm{C}_{4} /$ unit-cell by optimizing a random structure.
(b) Search for MINs with the $\mathrm{C}_{4} /$ unit-cell by the SC-AFIR starting from the single initial structure generated in step (a).
(c) Generate initial structures in the $\mathrm{C}_{8} /$ unit-cell by extending the unit-cell in MINs obtained for the $\mathrm{C}_{4} /$ unit-cell in step (b) along the shortest TV of each MIN, and then re-optimizing them in the $\mathrm{C}_{8} /$ unit-cell.
(d) Search for MINs with the $\mathrm{C}_{8} /$ unit-cell by the SC-AFIR starting from the initial structures obtained in step (c).
(e) Re-optimize all structures generated in step (d) in the $\mathrm{C}_{8} /$ unit-cell with the higher mesh cutoff value (200.0 Ry).

Finally, further analyses were conducted for the important crystal structures obtained. Spglib-1.8.3 was used to determine the space group of the obtained crystal structures. ${ }^{[24]}$ Phonopy-1.10.8 was used to calculate the phonon dispersion. ${ }^{[25,26]}$ Prior to the band and phonon calculations, the corresponding structure was reoptimized with a Monkhorst-Pack grid of $12 \times 12 \times 12$ and was shown in Supplemental Material. Gnubans in SIESTA3.2 was used to calculate the band dispersion. The k paths were determined according to the method previously described in the literature. ${ }^{[27]}$

## III. RESULTS

## A. Three-dimensional crystal structures of carbon with the $\mathrm{C}_{8} /$ unit-cell

The search was initiated from a single MIN obtained by optimizing a randomly generated structure in the $\mathrm{C}_{4} /$ unit-cell, where the initial random structure is shown in Supplemental Material S3. In total, 274 MINs were obtained after the five steps (a)-(e) described in Section II.G. Table I shows degrees of freedom in each system, the numbers of AFIR paths and MINs obtained after each step and the number of gradient calculations required to finish each step. Cartesian coordinates of all the 274 MINs are shown in Supplemental Material S4.

TABLE I. Degrees of freedom $f$, the number of obtained AFIR paths $N_{\text {path }}$, obtained MIN structures $N_{\text {MIN }}$ and gradient calculations required in each step $N_{\text {gradient }}$, in the search of three-dimensional structures of carbon. ${ }^{a}$

| Step $^{b}$ | $f$ | $N_{\text {path }}$ | $N_{\text {MIN }}$ | $N_{\text {gradient }}$ |
| :---: | :---: | :---: | :---: | :---: |
| (a) | 15 | - | 1 | 34 |
| (b) | 15 | 158 | 35 | 46,838 |
| (c) | 27 | - | 24 | 2,010 |
| (d) | 27 | 697 | 377 | 323,321 |
| (e) | 27 | - | 274 | 19,653 |

[^0]The 30 most stable MINs among the 274 MINs are shown in Fig. 3, where the
$N$ th lowest MIN is labeled as 3D-min $N$. The names of those known previously, either experimentally or theoretically, are also shown in the structural labels. The space group indicated in the structural labels were determined by spglib-1.8.3. The energy values are relative to the most stable structure, hexagonal graphite (3D-min0). Both of the known graphite structures, hexagonal (alpha) and rhombohedral (beta), were obtained as 3D-min0 and 3D-min2, respectively. The former, with $A B$-stacking, showed a lower energy than the latter having $A B C$ stacking, consistent with the previously acquired knowledge. Hexagonal graphite (3D-min0) showed a lower energy than cubic diamond (3D-min1), and this was also consistent with the known trend. The structure with $A A$-stacking was not a local minimum with the present setup for electronic structure calculation, where the geometry optimization starting from such an initial structure collapsed into hexagonal graphite. Note that cubic diamond was energetically more stable than hexagonal graphite when the original PBE-D2 parameters were used together with the present setup for electronic structure calculation. This would arise from the parameters such as the DZP basis set, or the parameters such as the pseudopotential, charge density cutoff, and $k$-mesh and so forth that were adopted in this study. In this study, by adjusting the $C 6$ value of the PBE-D2 parameters, the energetics in Fig. 3, which was consistent with the previously acquired knowledge, was obtained. We also notice that the energy difference between hexagonal graphite and cubic diamond in Fig. 3 is still smaller than that in the more elaborate calculation ( $\sim 3$ $\mathrm{kJ} / \mathrm{mol} \cdot$ atoms). ${ }^{[28]}$

Many previously reported structures, such as hexagonal graphite (3D-min0),
cubic diamond (3D-min1), rhombohedral graphite (3D-min2), lonsdaleite (or hexagonal diamond) (3D-min3), Cco-C8 or $Z$-carbon (3D-min4), ${ }^{[4,5]}$ crossed graphene (3D-min5), ${ }^{[29]} M$-carbon (3D-min6), ${ }^{[2,3]} 8$-tetra(2,2) tubulane or bct $\mathrm{C}_{4}(3 \mathrm{D}-\mathrm{min} 7),{ }^{[30-39]}$ LA5 or Y carbon (3D-min8), ${ }^{[38,40]} \mathrm{C} 2 / m-16$ (3D-min9), ${ }^{[41]} 3 \mathrm{D}-(5,0)$ (3D-min14), ${ }^{[39]}$ oI16-carbon (3D-min20), ${ }^{[42]} m C 16$-carbon (3D-min22), ${ }^{[42]} 3 \mathrm{D}-(4,4)$ carbon or squaroglitter (3D-min23), ${ }^{[39,43]}$ isoglitter (3D-min29), ${ }^{[44]} 8$-tetra(3,3) tubulane or E (3D-min33), ${ }^{[30,36,38,45]}$ LA7 (3D-min52), ${ }^{[38]}$ diam_cr43_bo (3D-min57), ${ }^{[46]}$ LA10 (3D-min69), ${ }^{[47]}$ a hypothetical metallic allotrope of carbon or bct-4 carbon (3D-min82), ${ }^{[2,48-51]}$ and F (3D-min196), ${ }^{[36]}$ were found in the automated search. The correspondence between the previously reported structures and those obtained by the present search was determined by comparing our structures with those with $2,4,8$, and 16 atoms per unit-cell in the SACADA, ${ }^{[6]}$ except for the case of 3D-min5 and 3D-min11. 3D-min 11 is a structure in which net $2 \mathrm{~b},{ }^{[52]}$ a two-dimensional sheet composed of fiveand seven-membered rings, is stacked.

In Fig. 3, 8-tetra(3,3) tubulane (3D-min33), LA7(3D-min52), diam_cr43_bo (3D-min57), LA10 (3D-min69), hypothetical metallic allotrope of carbon (3D-min82), and F (3D-min196), are not shown because these structures have higher energies of 35.2, 39.7, $41.1,46.1,50.1$, and $80.2 \mathrm{~kJ} / \mathrm{mol}$ atoms, respectively. To the best of our knowledge, the other structures have not previously been reported. This demonstrates that the present approach is useful for exploring unknown crystal structures.


FIG. 3. Thirty of the most stable crystal structures of carbon with the $\mathrm{C}_{8} /$ unit-cell. The names of structures (if available), space groups, and relative energies are shown in the labels. TVs are shown as red, green, and blue lines. Atoms in a unit-cell are highlighted
by the ball model.

## B. Two-dimensional periodic structures of carbon with the $\mathbf{C}_{8} /$ unit-cell

The search for the two-dimensional structures was also initiated from a single MIN obtained by optimizing a randomly generated structure in the $\mathrm{C}_{4} /$ unit-cell, where the initial random structure is shown in Supplemental Material S3. In total, 122 MINs were obtained after the five steps (a)-(e) described in Section II.G. Table II shows degrees of freedom in each system, the numbers of AFIR paths and MINs obtained after each step and the numbers of gradient calculations required to finish each step. Cartesian coordinates of all the 122 MINs are shown in Supplemental Material S4.

TABLE II. Degrees of freedom $f$, the number of obtained AFIR paths $N_{\text {path }}$, obtained MIN structures $N_{\text {MIN }}$ and gradient calculations required in each step $N_{\text {gradient }}$, in the search of two-dimensional structures of carbon. ${ }^{a}$

| Step $^{b}$ | $f$ | $N_{\text {path }}$ | $N_{\text {MIN }}$ | $N_{\text {gradient }}$ |
| :---: | :---: | :---: | :---: | :---: |
| (a) | 12 | - | 1 | 42 |
| (b) | 12 | 87 | 28 | 16,623 |
| (c) | 24 | - | 24 | 1,524 |
| (d) | 24 | 572 | 237 | 153,965 |
| (e) | 24 | - | 122 | 14,082 |

[^1]${ }^{b}$ the steps (a)-(e) are described in Section II.G.

The low-lying 15 structures are shown in Fig. 4. The Nth lowest structure is labeled as $2 \mathrm{D}-\min N$. The names of those known previously, either experimentally or theoretically, are shown in the structural labels. The energy values are relative to the most stable structure, hexagonal graphite (3D-min0).



2D-min5 net2 $b$ $26.5 \mathrm{kl} / \mathrm{mol} \cdot$ atoms

$20-m i n 1$ four-layer graphita $0.0 \mathrm{kl} / \mathrm{mol}$ 'atoms


2D-min6 pentahexocite $37.4 \mathrm{kl} / \mathrm{mol}$-atorms



2D-min4 graphene $4.1 \mathrm{kl} / \mathrm{mol}$-atoms

2D-min7 graphene + carbyne $38.7 \mathrm{kN} / \mathrm{mol}$ - 日toms


2D-ming


2D-min8 graphene + carbyne graphene + carbyne $38.7 \mathrm{~kJ} / \mathrm{mol} \cdot$ atoms $38.7 \mathrm{~kJ} / \mathrm{mol} \cdot$ atoms


2D-min 10 graphene + carbyne $38.8 \mathrm{~kJ} / \mathrm{mcl}$-atoms


2D-min11 graphene + carbyne $38.8 \mathrm{kl} / \mathrm{mol}$ - atorn


2D-min12 2D-min12
graphene + carbyne $39.1 \mathrm{~kJ} / \mathrm{mol}$-atoms


2D-min 13
$39.2 \mathrm{kl} / \mathrm{mol} \cdot \mathrm{atomg}$


2D-min 14 graphene + carbyne $39.3 \mathrm{~kJ} / \mathrm{mol}$-atoms

FIG. 4. Fifteen low-lying two-dimensional periodic structures of carbon with the $\mathrm{C}_{8}$ /unit-cell. The names of the structures (if available) and relative energies are shown in
the labels. Atoms in a unit-cell are highlighted by the ball model.

The lowest-lying structure is the four-layer graphite (2D-min0-2). A graphene sheet can be written with a unit-cell including two carbon atoms. With eight carbon atoms, not only graphene but also four-layer and bilayer graphite can be obtained. The stacking type of 2D-min0 is $A B C A$. In 2D-min1, the bottom three layers adopt an $A B C$ stacking. The top layer is similar to, but slightly deviated from $B$. In 2D-min2, the bottom three layers show an $A B B$ stacking, although the top layer is deviated from all the other three. The slight deviation from the $A B C$-layers seen in 2D-min1 and 2D-min2 arises from the present setup of the electronic structure calculation, which is not very accurate for describing weak interactions. Bilayer graphite (2D-min3) with an $A B$-stacking and graphene (2D-min4) are the next most stable structures. In addition, a lot of sheet structures that contain three-, four-, five-, six-, seven-, eight-, nine-, and/or ten-membered rings were obtained, where some of them correspond to those predicted from graph theory, such as net $2 b$ (2D-min5), ${ }^{[52-55]}$ pentahexoctite (2D-min6), ${ }^{[56]}$ and network7 (2D-min23). ${ }^{[53,57-61]}$ Carbynes are stabilized by interacting with a graphene (2D-min7-12, 2D-min14-18). Although they all are clusters between graphene and carbyne, they are different to each other in their interaction sites. In addition, two-dimensional sheets with a certain thickness, such as layered-diamond ${ }^{[62]}$ (2D-min19) and related structures (2D-min20-22), were also found.

## C. One-dimensional periodic structures of carbon with the $\mathbf{C}_{8} /$ unit-cell

The search for one-dimensional structures was initiated from a single MIN obtained by optimizing a randomly generated structure in the $\mathrm{C}_{4} /$ unit-cell, where the initial random structure is shown in Supplemental Material S3. In total, 49 MINs were obtained after the five steps (a)-(e) described in Section II.G. Table III shows degrees of freedom in each system, the numbers of AFIR paths and MINs obtained after each step and the numbers of gradient calculations required to finish each step. Cartesian coordinates of all the 49 MINs are shown in Supplemental Material S4.

TABLE III. Degrees of freedom $f$, the number of obtained AFIR paths $N_{\text {path, }}$, obtained MIN structures $N_{\text {MIN }}$ and gradient calculations required in each step $N_{\text {gradient }}$, in the search of one-dimensional structures of carbon. ${ }^{a}$

| Step $^{b}$ | $f$ | $N_{\text {path }}$ | $N_{\text {MIN }}$ | $N_{\text {gradient }}$ |
| :---: | :---: | :---: | :---: | :---: |
| (a) | 9 | - | 1 | 16 |
| (b) | 9 | 114 | 63 | 13,499 |
| (c) | 21 | - | 42 | 2,992 |
| (d) | 21 | 447 | 201 | 66,230 |
| (e) | 21 | - | 49 | 10,901 |

[^2]The low-lying 15 structures are shown in Fig. 5. The $N$ th lowest structure is labeled as 1D-min $N$. The names of the structures known previously, either experimentally or theoretically, are shown in the structural labels. The energy values are relative to the most stable structure, hexagonal graphite (3D-min0).

1D-mine
1D-min3
1D-min4 $107.2 \mathrm{kl} / \mathrm{mol}$ 'atoms GNR + carbyne GNR + carbyne 2.1kimolatoms

1D-min7
$113.5 \mathrm{k} / \mathrm{mol}$ - atoms
1D-min6
1D-min9 carbyne $119.4 \mathrm{~kL} / \mathrm{mol}$-atoms



1D-min10
$121.3 \mathrm{~kJ} / \mathrm{mol}$-atoms

## 1D-minit <br> $126.5 \mathrm{~kJ} / \mathrm{mol}$-etoms

 $126.7 \mathrm{~kJ} / \mathrm{mol}$-atams1D-min13
carbyne
$127.8 \mathrm{~kJ} / \mathrm{mol}$-aboms
carbyne $127.9 \mathrm{kN} / \mathrm{mol}$-atoms

FIG. 5. Fifteen low-lying one-dimensional periodic structures of carbon with the $\mathrm{C}_{8} /$ unit-cell. Names of structures (if available) and relative energies are shown in labels.

Atoms in a unit-cell are highlighted by the ball model.

The most stable structure is GNR (1D-min0). Another stable structure is the (2,2)-CNT (1D-min1, 1D-min2), which is the smallest CNT. 1D-min2 is elongated
along the $x$-axis compared with 1D-min1. 1D-min3-5 are clusters of GNR and carbyne that resemble each other. These three are different in their interaction sites. Carbyne (1D-min9, 1D-min13, 1D-min14,) is less stable than the GNR-carbyne clusters (1D-min3-5). The polyacene-like thinnest GNR (1D-min25) was found as the 26th most stable structure. Some structures containing three-, four-, five-, six-, seven-, eight-, nine-, and/or ten-membered rings were also found as higher-energy structures.

## IV. DISCUSSION

Among the newly found crystal structures, 3D-min10 is the most stable. Its structure is shown in various ways in Fig. 6. Its primitive lattice contains eight carbon atoms, and its space group is $P 2 / \mathrm{m}$. This structure is composed of $s p^{2}$ and $s p^{3}$ carbons and corresponds to a structure in which graphene sheets in graphite are connected to each other by C-C bonds. We note that a similar structure has previously been reported as diam_cr43_bo (3D-min57). ${ }^{[46]}$ The difference is whether or not the structures contain four-membered rings, where 3D-min 10 does not have a four-membered ring and thus is more stable than 3D-min57. The phonon and band calculations were performed with the structure of 3D-min10, where the cell was redefined and re-optimized (see Supplemental Material S5 for the Cartesian coordinates). The phonon dispersion presented in Supplemental Material S6, shows that 3D-min10 is a local minimum structure. The band structure in Fig. 7 suggests that this structure is metallic. Furthermore, a Dirac cone was found between the A and X points in the band structure of 3D-min10.


FIG. 6. Views along the $a$-axis (a), $b$-axis (b), and $c$-axis (c), and the primitive-cell (d) of 3D-min10. TVs are shown as red, green, blue lines.


FIG. 7. Band structure of 3D-min10. The Fermi level is shown by a green line. The definitions of k points are described in ref. 27 and Supplemental Material S7.

The second most stable structure among those newly found in this study, i.e., 3D-min12, is depicted in Fig. 8. The space group of this structure is $P 1$. It includes eight carbon atoms in its primitive lattice. This structure is composed of $s p^{2}$ and $s p^{3}$ carbons
and is similar to 3D-(4,4) carbon (3D-min23). ${ }^{[39,42]}$ Both of these structure have a large hollow, and the size and shape of the hollow are different for 3D-min12 and 3D-min23. The phonon and band calculations were conducted for 3D-min 12 with the cell redefined and re-optimized (see Supplemental Material S5 for its Cartesian coordinates). Phonon dispersion indicates that 3D-min12 is a local minimum structure, as shown in Supplemental Material S6. The band structure in Fig. 9 shows that this structure is metallic.


FIG. 8. Views along the $a$-axis (a), $b$-axis (b), and $c$-axis (c), and the primitive-cell (d) of 3D-min12. TVs are shown as red, green, blue lines.


FIG. 9. Band structure of 3D-min12. The Fermi level is shown by a green line. The definitions of k points are described in ref. 27 and Supplemental Material S7.

The third most stable structure among those newly found in this study is 3D-min13. Its structure is depicted in Fig. 10 in various ways. The space group of this structure is $P 1$. It includes eight carbon atoms in its primitive lattice. This structure is composed of $s p^{2}$ and $s p^{3}$ carbons and is similar to $\mathrm{C}_{\mathrm{CAS}}{ }^{[63]}$ Although both of these structure have five-membered rings and a large hollow, they are different to each other in their size of the hollow. The phonon and band calculations were conducted for 3D-min13 with the cell redefined and re-optimized (see Supplemental Material S5 for its Cartesian coordinates). Phonon dispersion indicates that 3D-min13 is a local minimum structure, as shown in Supplemental Material S6. The band structure in Fig. 11 shows that this structure also is metallic. Furthermore, two Dirac cones were found around the Y point and the L point in the band structure of $3 \mathrm{D}-\min 13$.


FIG. 10. Views along the $a$-axis (a), $b$-axis (b), and $c$-axis (c), and the primitive-cell (d) of 3D-min13. TVs are shown as red, green, blue lines.


FIG. 11. Band structure of 3D-min13. The Fermi level is shown by a green line. The definitions of k points are described in ref. 27 and Supplemental Material S7.

Carbyne, shown in Fig. 12 (a), was obtained as the 266th lowest MIN (3D-min265) with a high relative energy of $112.3 \mathrm{~kJ} / \mathrm{mol} \cdot$ atoms. Carbyne is a linear, pure-carbon chain described as a resonance structure of polyyne $(-\mathrm{C} \equiv \mathrm{C}-)_{n}$ and polycumulene $(=\mathrm{C}=)_{n}$. Although short carbon chain molecules have been studied
extensively, ${ }^{[64-67]}$ the production of very long chains composed of more than 6000 atoms has been achieved only recently in a double-walled CNT. ${ }^{[68]}$ The present search found a structure, shown in Fig. 12 (b), in which carbyne is sandwiched by graphene sheets (3D-min36). Intriguingly, the relative energy of the sandwiched structure of 37.5 $\mathrm{kJ} / \mathrm{mol} \cdot$ atoms is lower than the average energy of graphite and carbyne, which is 56.2 $\mathrm{kJ} / \mathrm{mol} \cdot$ atoms. This suggests that carbyne is also stabilized by graphene sheets.


FIG. 12. The structures of (a) carbyne and (b) sandwiched carbyne. TVs are shown in red, green, blue lines.

The present search missed a previously reported structure called 4H-diamond ${ }^{[69]}$ even though it can be written with the $\mathrm{C}_{8} /$ unit-cell. In 4 H -diamond, structures of diamond and hexagonal diamond co-exist, and its unit-cell is highly elongated. The 4 H -diamond was missed because of the structural clustering, which was introduced in Section II.D. The algorithm restricts the search to areas in which the shape of the unit-cell is not far from cubic. As discussed in Section II.D, this structure is expected to be found with the present algorithm if a larger unit-cell is used. The other structures which were missed were $\mathrm{K}_{4}{ }^{[70-72]}$ and $D_{6 \mathrm{~h}}$ prism- $\mathrm{C}_{12} .{ }^{[73]}$ This is because these structures
have a much higher energy than 3D-min0. Hence, the present search, which is targeted to low-energy strictures ${ }^{[16]}$, was unable to find them. These structures can be found when a larger $P$ and $M$ values are adopted, where $P$ and $M$ are the parameters of the stochastic search algorithm as defined in Section II.A. We note that a $\mathrm{K}_{4}$-like structure was predicted in the present search as 3D-min223, where its energy $88.9 \mathrm{~kJ} / \mathrm{mol} \cdot$ atoms is slightly higher than $\mathrm{K}_{4} 83.3 \mathrm{~kJ} / \mathrm{mol}$ atoms at the present computational level.

## V. CONCLUSION

In this study, the AFIR method was combined with the PBCs. The AFIR method has been used extensively in the exploration of reaction pathways in molecular systems. The combined PBC/AFIR approach was applied to the carbon crystals described by the $\mathrm{C}_{8}$ /unit-cell. The search automatically generated the previously reported low-lying structures. Furthermore, a lot of unreported structures were found. This suggests that the AFIR method is effective not only in molecular systems but also for the global exploration of low-lying crystal structures of a given atomic composition.

Among the many newly found structures, three low-lying structures were further studied. These three structures have a slightly higher energy than those predicted previously, such as Cco- $\mathrm{C}_{8}$ (Z-carbon) and M-carbon. All the three structures did not exhibit a band-gap and were predicted to behave as a metal. Furthermore, Dirac cones were discovered in the band structures of two of these structures.

In addition, the present approach was extended for the exploration of two- or one-dimensional periodic structures. A search of two-dimensional periodic structures
with $\mathrm{C}_{8} /$ unit-cell gave many structures, such as two-dimensional sheets, composed of three-, four-, five-, six-, seven-, eight-, nine-, and/or ten-membered rings including graphene. Another search of the one-dimensional periodic structures with a $\mathrm{C}_{8} /$ unit-cell found GNR, the thinnest CNT, carbyne, and so forth.

The present results suggest that this PBC/AFIR approach is promising in exploring low-lying periodic structures systematically. Furthermore, this approach, in principle, is applicable to any periodic systems. Applications to systems including two or more elements and/or to molecular crystals will be interesting future avenues for research.

## ACKNOWLEDGMENTS

MT is supported by the Ministry of Education, Culture, Sports, Science and Technology through Program for Leading Graduate Schools (Hokkaido University "Ambitious Leader's Program"). This work was partly supported by a grant from Japan Science and Technology Agency with a Core Research for Evolutional Science and Technology (CREST) in the Area of "Establishment of Molecular Technology towards the Creation of New Functions" (Grant Number JPMJCR14L5) at Hokkaido University, Materials research by Information Integration Initiative $\left(\mathrm{MI}^{2} \mathrm{I}\right)$ project of the Support Program for Starting Up Innovation Hub from Japan Science and Technology Agency (JST), Japan.

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[^0]:    ${ }^{a}$ a single gradient calculation includes calculations of the energy, force, and stress tensor.
    ${ }^{b}$ the steps (a)-(e) are described in Section II.G.

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[^2]:    ${ }^{a}$ a single gradient calculation includes calculations of the energy, force, and stress tensor.
    ${ }^{b}$ the steps (a)-(e) are described in Section II.G.

