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Superconductivity in a new intermetallic structure type based on endohedral Ta@Ir₇Ge₄ clusters

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Abstract

We report the observation of superconductivity at a temperature near 3.5 K for the previously unreported compound TaIr₂Ge₂. In addition to being a superconductor, this material displays a new crystal structure type that contains endohedral clusters, as determined by single crystal Xray diffraction structure refinement; the structure is more complex than those of the commonly observed tetragonal 122 intermetallic phases. Despite the strong metal-metal interactions, electronic structure calculations show the presence of a relatively simple set of states at the Fermi level, consisting mostly of Ta atom orbital contributions. The superconducting transition is characterized by temperature-dependent resistivity, magnetic susceptibility and specific heat measurements, and is of the weak coupling BCS type with $\Delta C/\gamma T_c = 1.55$. The upper critical field at 0 K is estimated to be 2 Tesla. Basic characterization of the superconductivity suggests that this material is similar to PbTaSe₂, for which the effects of spin-orbit coupling may be important.

Introduction

Heavy elements such as iridium often form materials with interesting physical properties due to strong spin orbit coupling. Ir-based materials have been of particular recent interest for oxides,1 and a variety of intermetallic materials based on Ir have been the subjects of recent study.² Our interest is in superconductivity in Ir-containing materials where the presence of strong spin orbit coupling may affect the superconducting properties. Some of the most favored structures for Ir-containing superconductors are the LaIrSi-type and the tetragonal symmetry "1:2:2" CaBe₂Ge₂-type. Tetragonal "1:2:2" type materials are particularly ubiquitous among intermetallic phases.³ In these layered 1:2:2 AB_2X_2 structures, B and X closely bond and form the electron-acceptor layers while the electropositive A element (usually rare-earth, alkaline, or alkali-earth metals) acts as the electron-donor layer.⁴ Remarkably, despite the extreme structural similarities and known structural phase transitions between the CaBe₂Ge₂ structure type and the ThCr₂Si₂-type structures, which are related through the inversion of the order of one of the constituent layers, almost all of the superconducting 1:2:2 phases based on Ir exist in the CaBe₂Ge₂ structure type only - for materials that occur in both of these common 1:2:2 structures, it has been observed that superconductivity is only present in the CaBe₂Ge₂ structure, disappearing when the same elements form a ThCr₂Si₂-type structure.^{5,6}

Here we report a new, superconducting 1:2:2 Ir-based material with a previously unreported crystal structure type, TaIr₂Ge₂. Instead of hosting layered features like the CaBe₂Ge₂ and ThCr₂Si₂-type structures, the use of the less electropositive element Ta results in cluster-type structural fragments like are found in many metallic alloys.⁷ The previously unreported structure type displayed by this new ternary compound is understood as consisting of corner and face - sharing Ta@Ir₇Ge₄ clusters. We show the crystal structure, calculated electronic structure, and superconducting property measurements of TaIr₂Ge₂. This material has strong spin-orbit coupling and displays superconducting behavior below a critical temperature of 3.5 K. Our specific heat data confirm bulk superconductivity and indicate that TaIr₂Ge₂ is a weak-coupling BCS superconductor. The superconducting transition is further characterized by upper and lower critical field measurements.

Experimental

Polycrystalline samples of TaIr₂Ge₂ were synthesized by arc melting on a water-cooled copper hearth under an argon atmosphere. Tantalum (foil, 99.95%, Alfa Aesar), iridium metal

(powder, 99.95%, Alfa Aesar), and germanium (crystalline pieces, 99.9999%, Alfa Aesar) were measured out in the TaIr₂Ge₂ stoichiometric ratio, and the iridium powder was pressed into a pellet. The Ir powder pellet and Ge pieces were first arc melted together and then combined with the Ta foil. Samples were turned over and arc melted repeatedly to increase homogeneity. The total mass of each sample was approximately 100 mg, and weight losses from arc melting did not exceed 2.5%. Annealing experiments were performed in sealed evacuated quartz glass tubes for temperatures below 1150 °C, and in an argon-backfilled induction furnace for temperatures between 1150 °C and 1450 °C.

In order to determine the identity of the new phase in the Ta-Ir-Ge system, single crystal x-ray diffraction was performed at room temperature. Small crystals (~0.01×0.01×0.01 mm) were selected from partially crushed polycrystalline samples and mounted on the tip of a Kapton loop. Ten different pieces were tested, and the same lattice parameters were obtained for each. A Bruker Apex II diffractometer was used to gather intensity data, with Mo radiation ($\lambda_{K\alpha}$ = 0.71073 Å) and an exposure time of 10 seconds per frame. Data were collected over a full sphere of reciprocal space, with 0.5° scans in ω and a 2 θ range from 5° to 70°. The data acquisition, extraction of intensity, and face-indexed numerical correction for Lorentz and polarization effects were performed using the Bruker SMART software.⁸ With the SHELXTL package,⁹ the crystal structures were solved using direct methods and refined by full-matrix least-squares on F². All crystal structure drawings were produced using the program *Vesta*.¹⁰

Powder x-ray diffraction patterns were obtained on a Bruker D8 Advance Eco diffractometer using Cu-K α radiation and a LynxEye-XE detector. A LeBail fit was performed using the FullProf Suite¹¹ with Thompson-Cox-Hastings pseudo-Voigt peak shapes starting with the cell parameters determined from single crystal diffraction. Lattice parameters from powder diffraction were found to be within <0.3% of the single crystal diffraction results. We therefore report the crystallographic parameters obtained from single crystal structure refinements.

Electronic structure calculations of the density of states (DOS) and band structure were completed using the Vienna ab-initio Simulation Package (VASP).^{12–15} The computation employed projector augmented-wave (PAW) pseudopotentials^{16,17} and the Perdew-Burke-Ernzerhof parameterization of the generalized gradient approximation (PBE-GGA),¹⁸ which includes scalar relativistic effects (spin-orbit coupling) on all atoms. All the calculations were performed on the experimental crystal structure data. The energy cutoff was 400 eV. Reciprocal

space integrations were completed over a $6 \times 4 \times 16$ Monkhorst-Pack k-points mesh, using the linear tetrahedron method. With these settings, the calculated total energy converged to less than 0.1 meV per atom.

Temperature-dependent resistivity and heat capacity measurements were performed on a Quantum Design Physical Property Measurement System (PPMS). Resistivity data were collected under a maximum applied current of 5 mA over a temperature range of 1.9-300K. A standard four-probe approach was used, and the 20µm-diameter platinum wire leads were attached to the sample by spot-welding. Magnetization measurements were also taken on the PPMS equipped with a Vibrating Sample Magnetometer (VSM). Zero-field cooled data were collected under an applied field of 20 Oe. The magnetization was also measured as a function of applied field (0 to 2T) at temperatures from 1.7 to 3.2K.

Results and Discussion

TaIr₂Ge₂ crystallizes in a previously unreported structure type. The crystal structure, which was determined by single crystal X-ray diffraction, is depicted in Figure 1. The compound is centrosymmetric, non-symmorphic, and orthorhombic (Pnnm, No. 58) with a unit cell volume of V = 312.6(2) Å³ and lattice parameters a = 8.515(4) Å, b = 12.553(4) Å, and c = 2.9247(10)Å. A summary of the crystallographic data from the structure refinement can be found in Table 1, and the atomic coordinates determined from the single crystal structure refinement are listed in Table 2. To determine whether the samples employed for property determination were the correct material, in high purity, powder X-ray diffraction was performed – the powder X-ray diffraction pattern for a sample employed for property characterization is shown in Figure 2, along with the positions of the expected peaks from the new TaIr₂Ge₂ material, calculated from the single crystal diffraction data. A high degree of sample purity is observed. Arc melting is the best method for synthesis of a pure material. In all the annealing experiments performed, which ranged in temperature between 1000 °C and 1450 °C for times between 30 minutes and 2 days, the compound was found to decompose with the accompanying disappearance of superconductivity – the common decomposition products were TaIrGe and TaIr₃. This indicates that TaIr₂Ge₂ is a high temperature phase.

Since the crystal structure type observed has not been previously reported, the structural determination was performed carefully and especially focused on the examinations of site preferences and atomic mixtures. The crystal structure of TaIr₂Ge₂ can be represented by Ta-

centered clusters, in which the Ir and Ge atoms surround Ta atoms to form 11-coordinate polyhedra. These endohedral Ta@Ir₇Ge₄ clusters extend into three dimensions through Ir vertex sharing or Ir₄ face sharing. Potential Ir/Ge mixtures were tested during the structural refinements to guarantee the veracity of the ordered atomic occupancies. Furthermore, multiple experimental attempts were made to synthesize compounds of similar compositions (for example, "TaIr₂- $_xGe_{2+x}$ " and "TaIr_{2+x}Ge_{2-x}"); only TaIr₂Ge₂ proved to be stable, supporting our determination of the clear site preference for Ir and Ge in the compound. The Ta-Ir bonds present have lengths of 2.76, 2.78, and 2.86 Å, while the Ta-Ge bonds have lengths of 2.88 and 2.90 Å. These bond lengths are similar to those in binary compounds containing the same elements. For example, the Ta-Ir bonds in TaIr₃ have a length of 2.75 Å,¹⁹ and the Ta-Ge bonds in TaGe₂ are either 2.66 or 2.84 Å in length.²⁰ These values are consistent with our observations for TaIr₂Ge₂. A brief analysis of the relative electronegativities of the three elements supports this understanding of the endohedral clusters in the new structure as follows: the electronegativity of Ir is greater than that of Ge, according to both the Pauling scale (χ_{Ir} = 2.20; χ_{Ge} = 2.01) and the Mulliken scale (χ_{Ir} = 2.14; χ_{Ge} = 1.88); on the other hand, Ta is significantly more electropositive than either Ir or Ge. Therefore, the two more electronegative atoms surround the electropositive Ta atoms. It is worth noting that Ge-Ge dimers are present in TaIr₂Ge₂, and since they have been observed in at least one other superconductor, they may be influential for the presence of superconductivity in the present material.²¹

In order to gain a better understanding of the cluster formation in TaIr₂Ge₂, the calculated band structure and the density of states based on the VASP results can be considered. The resulting diagrams can be found in Figure 3, with the band structure located in the right-most panel. Below values of 1.5 eV below the Fermi energy, primarily Ir and Ge bulk metallic states are observed. However, around the Fermi level, a few bands split off from the main set of states. These consist mostly of contributions from partially hybridized Ta 5*d* and 6*s* states, indicating a weak interaction between valence electrons from Ta in these states and those from Ir/Ge. Considering the total density of states near the Fermi level shown in the left panel of the figure, another significant feature is revealed - the deep pseudogap below the Fermi level. The pseudogap corresponds to 28 valence electrons per TaIr₂Ge₂ (14e- per Ta_{0.5}IrGe), which is associated with the stability of the compound.

The electronic structure calculation results shown in Figure 3 are consistent with our view of low T_c polar intermetallic compounds, as we observe that it is common for such superconductors to have a Fermi level just above or below the pseudogap in the density of states. In addition, one Van Hove singularities can be observed in the band structure of TaIr₂Ge₂ (Figure 3b.) These bands contribute to electronic instability often seen in superconducting systems. Moreover, the 14 valence electron count per Ta_{0.5}IrGe formula unit at the deep pseudogap inspired us to reinvestigate the electronic structure of superconductor IrGe, which has 13 valence electrons per formula unit.²² Not surprisingly, a broad pseudogap in the DOS of IrGe is located above the Fermi level, also corresponding to 14 valence electrons. A new material with two fewer electrons per formula unit than TaIr₂Ge₂ and the same crystal structure, if it could be synthesized, is expected to have its Fermi level quite close to a pseudogap in the density of states.

The main panel of Figure 4 shows the temperature dependence of electrical resistivity of TaIr₂Ge₂ between 1.9 and 300 K. Metallic behavior $(d\rho/dT > 0)$ is manifested in the whole temperature range. The temperature dependence of the resistivity is typical of what is seen in intermetallic compounds – a residual resistivity (ρ_0) is seen at low temperatures due to impurity scattering, and there is a saturation of the resistivity at high temperatures (ρ_{max}) where the mean-free path approaches the interatomic spacing, the so called Ioffe-Regal limit.^{23,24} This behavior is often fit by what is called a parallel resistor model (PRM)^{25–27} that includes Bloch-Grüneisen resistivity ρ_{BG} behavior combined with a parallel, temperature independent resistor ρ_{max} :

$$\rho(T)^{-1} = \rho_{max}^{-1} + (\rho_0 + \rho_{BG})^{-1},$$

where

$$\rho_{BG} = 4R\Theta(\frac{T}{\Theta})^5 \int \frac{x^5}{(\exp(x) - 1)(1 - \exp(-x))} dx.$$

The characteristic temperature estimated from the fit is $\Theta = 183$ K; the temperature independent term, ρ_0 , is 0.204(3) m Ω cm; and $\rho_{max} = 2.28(7)$ m Ω cm. The inset shows $\rho(T)$ for TaIr₂Ge₂ in the vicinity of superconducting transition measured under zero field and applied magnetic field up to 1.0 T. With no applied magnetic field, we observe the resistivity drop to zero with a superconducting critical temperature of T_c = 3.9 K. The superconducting temperature was estimated as the midpoint of the resistivity transition (shown in the inset of Figure 4). T_c decreases with applied magnetic field as is expected for superconducting materials, and for μ_0 H = 1 T the critical temperature is T_c = 2.02 K.

The volume susceptibility $\chi_V(T)$ versus temperature of TaIr₂Ge₂ under an H = 20 Oe applied field is presented in Figure 5. The zero-field cooled (ZFC) susceptibility drops to a large negative value upon cooling. When corrected for the demagnetization factor N = 0.17 (obtained from the M(H) fit as discussed below), the diamagnetic signal at the lowest temperature T = 1.8 K is slightly below the expected value $\chi_V = -1/4\pi$, indicating that the sample volume is fully superconducting. A much smaller diamagnetic signal is observed in the field-cooled $\chi_V(T)$ transition (see the inset of Fig. 5) due to strong magnetic flux pinning in the polycrystalline TaIr₂Ge₂ sample.

The superconducting state is further characterized by magnetization versus magnetic field measurements M(H) as shown in Figure 6(a). Assuming that a linear response to a magnetic field indicates a perfect diamagnetic response, we obtain a demagnetization factor of N = 0.17. This value is consistent with the sample shape and was used for the $\chi(T)$ plot discussed above. Figure 6(b) presents the difference between the magnetization M(H) at temperatures from 1.7 K to 3.2 K and the fit to the data M_{fit}, which was determined in the low field range where the magnetization response is linear. The field at which M(H) starts to differ from a fitted M(H) is the lower critical field H_{c1}. Those values are plotted in Figure 6(c) as a function of temperature. These results were then fitted to the formula:

$$H_{c1}(T) = H_{c1}(0) \left[1 - \left(\frac{T}{T_c}\right)^2\right]$$

From this fit the lower critical field at 0 K for $TaIr_2Ge_2$ is estimated to be $H_{c1}(0) = 72(1)$ Oe.

Specific heat measurements were conducted in addition to magnetization and resistivity measurements to show that superconductivity is an intrinsic property of TaIr₂Ge₂. This data is presented in Figures 7 and 8. Figure 7(a) presents the overall temperature dependence of the specific heat. At room temperature C_p is close to the value calculated from the Dulong-Petit law $3nR \approx 125 \text{ J mol}^{-1} \text{ K}^{-1}$, where n is the number of atoms per formula unit (n=5) and R is the gas constant (R = 8.314 J mol^{-1} K^{-1}). The solid red line denotes a fit to a combined model: $C_p = C_{el} + k*C_{Debye}+(1-k)C_{Einstein}$. For this fit, k = 0.74 of the weight belongs to the Debye heat capacity C_{Debye} (dashed blue line), and 0.26 of the weight belongs to the Einstein heat capacity $C_{Eisntein}$ (solid green line). The electronic heat capacity C_{el} is equal to γT , where γ is the Sommerfeld coefficient.

$$C_{Debye}(T) = 9nR\left(\frac{T}{\Theta_D}\right)^3 \int \frac{x^4 \exp(x)}{\left[\exp(x) - 1\right]^2} dx ,$$

and

$$C_{Einstein}(T) = 3nR\left(\frac{\Theta_E}{T}\right)^2 \exp\left(\frac{\Theta_E}{T}\right) \left[\exp\left(\frac{\Theta_E}{T}\right) - 1\right]^{-2}.$$

The Debye temperature estimated from the fit is equal to $\Theta_D = 359(4)$ K, the Einstein temperature $\Theta_E = 150(2)$ K, and the weight k = 0.74(2). The Sommerfeld parameter value was held constant at $\gamma = 11 \text{ mJ mol}^{-1} \text{ K}^{-2}$ as obtained from the low temperature fit that is discussed below. Figure 7(b) shows temperature dependence of the specific heat C_p divided by T^3 . It can be shown that the temperature (T_{max}) at which the maximum of $C_{Einstein}/T^3$ occurs is equal to T_{max} = $\Theta_E/5$ Since T_{max} is about 29 K, Θ_E can be estimated to be approximately 145 K. This value of $\Theta_{\rm E}$ is in very good agreement with the Einstein temperature obtained from the fit over the full temperature range. Figure 7(c) presents C_p/T vs T in the vicinity of the transition temperature. The large anomaly displayed in the specific heat data supports the bulk superconductivity of TaIr₂Ge₂. A standard equal-area construction (yellow shading) was performed in order to determine the superconducting transition temperature and the magnitude of the specific heat jump. The critical temperature for TaIr₂Ge₂ was calculated to be $T_c = 3.5$ K, and the C_p/T jump was calculated to be about $\Delta C/T_c = 17.0 \text{ mJ mol}^{-1} \text{ K}^{-2}$. We note that the critical temperature determined from the specific heat calculations is somewhat lower than the transition temperature observed from the resistivity measurements. This again is the usual case for complex superconductors, where the zero resistivity path is frequently complete at temperatures higher than that determined from the equal area construction method applied to entropy loss data.

Figure 8(a) shows temperature dependence of the measured C_p/T in magnetic fields from 0T to 0.6 T. The solid vertical lines represent the midpoints of the superconducting transitions for each applied field. In part (b) of the same figure, the measurement of C_p/T versus T^2 is shown under a field of $\mu_0H = 1.5$ T. This field exceeds the upper critical field for TaIr₂Ge₂ over the temperature range measured. At sufficiently low temperatures (T < $\Theta_D/50$), the normal state specific heat can be described by the equation: $C_p = \gamma T + \beta T^3$, where γ and β are the electronic and phonon contributions to the specific heat, respectively. By fitting our data to this equation, we obtain the Sommerfeld coefficient $\gamma = 11.0(1)$ mJ mol⁻¹ K⁻² and $\beta = 0.173(4)$ mJ mol⁻¹K⁻⁴. In the simple Debye model β is related to the Debye temperature Θ_D through:

$$\Theta_D = \left(\frac{12\pi^4}{5\beta}nR\right)^{1/3}$$

where $R = 8.314 \text{ mJ mol}^{-1} \text{ K}^{-1}$ and n = 5 for TaIr₂Ge₂. The Debye temperature calculated using this formula $\Theta_D = 355 \text{ K}$ is very close to the value obtained from the fit to the combined model.

Knowing the value of Θ_D allows for the calculation of the electron-phonon coupling constant λ_{ep} , an important superconducting parameter. The inverted McMillan formula²⁸ relates these values:

$$\lambda_{ep} = \frac{1.04 + \mu * \ln\left(\frac{\Theta_D}{1.45T_c}\right)}{(1 - 0.62\mu *)\ln\left(\frac{\Theta_D}{1.45T_c}\right) - 1.04}$$

Taking $\mu^* = 0.13$, the value of $\lambda_{ep} = 0.56$ obtained suggests that TaIr₂Ge₂ is a weak coupling superconductor. Using the Sommerfeld parameter of $\gamma = 11.0(1)$ mJ mol⁻¹ K⁻² acquired above, the non-interacting density of states at the Fermi energy N(E_F) can be calculated from the equation:

$$N(E_F) = \frac{3\gamma}{\pi^2 k_B^2 (1 + \lambda_{ep})}$$

For TaIr₂Ge₂, N(E_F) = 3 states eV^{-1} per formula unit. For $\gamma = 11.0(1)$ mJ mol⁻¹ K⁻², the normalized specific heat jump value $\Delta C/\gamma T_c$ is found to be 1.55, which is slightly above the expected value of 1.43 for a weak-coupling BCS superconductor. This number is consistent with the values typically obtained for conventional BCS superconductors; therefore, bulk superconductivity in TaIr₂Ge₂ is confirmed.

Figure 8(c) shows the values of the upper critical field μ_0H_{c2} vs temperature obtained from specific heat (red squares) and resistivity (blue circles) measurements. The points plotted are the estimated midpoints of the C_p/T jumps (Fig. x.8(a)) and the resistivity drops (Inset of Fig. x.4). The data from resistivity measurements show an upward curvature of H_{c2}(T) near T_c, which may be due to an intrinsically anomalous pairing mechanism. The formula proposed by Micnas, et. al²⁹ $\mu_0H_{c2}(T) = \mu_0H_{c2}(0)(1 - (T/T_c)^{3/2})^{3/2}$, has been successfully used to fit the whole $\mu_0H_{c2}(T)$ data set for noncentrosymmetric Nb_{0.18}Re_{0.82}³⁰ and for the PbTaSe₂ superconductor.³¹ For TaIr₂Ge₂ the fit gives $\mu_0H_{c2}(0) = 1.95(1)$ T, which will be used for further calculations. Alternatively, for a single-band, BCS-type superconductor, the upper critical field can be calculated from the formula:

$$H_c(0) = -AT_c \frac{dH_c}{dT}\Big|_{T=T_c}$$

where A is 0.69 or 0.73 for the dirty or clean limit, respectively.³² The $d\mu_0H_c/dT$ slope for both series is almost identical and is about -0.6 T/K. Taking $T_c = 3.5$ the formula gives $\mu_0H_{c2}(0) = 1.45$ T and 1.53 T for the dirty and clean limit, respectively.

Using the upper critical field value of $\mu_0 H_{c2}(0) = 1.95(1)$ T and assuming that H_{c2} is purely orbital, the superconducting coherence length was estimated to be $\xi_{GL}= 13$ nm, from $H_{c2}(0) = \Phi_0/2\pi\xi_{GL}^2$ where $\Phi_0=$ h/2e. Combining this result with the value of $H_{c1}(0)$ determined above (72 Oe), the Ginzburg–Landau superconducting penetration depth $\lambda_{GL}=$ 262 nm was numerically estimated from a formula for the lower critical field:

$$H_{c1} = \frac{\phi_0}{4\pi\lambda_{GL}^2} \ln\frac{\lambda_{GL}}{\xi_{GL}}$$

The calculated Ginzburg–Landau parameter [$\kappa = \lambda_{GL}(0)/\xi_{GL}(0)$] is $\kappa = 20$, and this result confirms that TaIr₂Ge₂ is a type-II superconductor.

Finally, now that H_{c1} , H_{c2} and κ have been obtained, the thermodynamic critical field can be calculated from the relation:

$$H_{c1}H_{c2} = H_c^2 \ln \kappa$$

This yields an estimate for the thermodynamic critical field of $\mu_0 H_c = 70$ mT. All the estimated superconducting parameters for TaIr₂Ge₂ are gathered in Table 3, where they are compared to those of PbTaSe₂.³¹ The superconducting properties of these two materials are highly analogous. Spin orbit coupling was originally considered as inconsequential to the superconducting behavior of PbTaSe₂, but on further consideration, it has been proposed as a potential topological superconducting material.³³

Conclusions

We have described the crystal structure and properties of the previously unreported superconductor TaIr₂Ge₂. The compound crystallizes in a new orthorhombic structure type, which can be described by the vertex and face sharing of 11-coordinate Ta@Ir₇Ge₄ endohedral clusters. Resistivity, magnetic susceptibility and specific heat measurements show that the material becomes superconducting below a temperature of 3.5 K, and specific heat and critical field measurements indicate that it is a weak coupling type II BCS superconductor. There is no

obvious indication, in our simple characterization, of anomalous properties that might arise due to the strong spin orbit coupling of Ir, but analogies of the superconducting parameters of TaIr₂Ge₂ to those of PbTaSe₂ suggest that more detailed electronic characterization may be of future interest.

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TaIr ₂ Ge ₂			
F.W. (g/mol);	710.53		
Space group; Z	<i>Pnnm</i> (No.58); 4		
	<i>a</i> = 8.515 (4) Å		
Lattice Parameters	<i>b</i> = 12.553 (4) Å		
	<i>c</i> = 2.9247 (10) Å		
Volume ($Å^3$)	312.6 (2)		
Absorption Correction	Numerical		
$\mu(\text{mm}^{-1})$	0.0027(2)		
Θ range (deg)	2.89 to 33.20		
	$-12 \le h \le 12$		
<i>hkl</i> ranges	$-17 \le k \le 19$		
	$-4 \le l \le 4$		
No. reflections; R_{int}	3803; 0.0419		
No. independent reflections	696		
No. parameters	32		
R_1 ; wR_2 (all I)	0.0358; 0.0730		
Goodness of fit	1.134		
Diffraction peak and hole (e^{-/Λ^3})	5.516; -4.758		

Table 1. Crystallographic data for $TaIr_2Ge_2$ at 300(2) K

Atom	Wyck.	Occ.	x	у	Z	U_{eq}
Та	4 <i>c</i>	1	0.8914(1)	0.3691(1)	0	0.0029(1)
Ir1	4c	1	0.8557(1)	0.5559(1)	1/2	0.0023(1)
Ir2	4c	1	0.5826(1)	0.2783(1)	0	0.0025(1)
Gel	4c	1	0.6152(1)	0.4403(1)	1/2	0.0037(3)
Ge2	4 <i>c</i>	1	0.7937(1)	0.1823(1)	1/2	0.0036(3)

Table 2. Refined atomic coordinates and equivalent isotropic displacement parameters of TaIr₂Ge₂ (U_{eq} is defined as one-third of the trace of the orthogonalized U_{ij} tensor (Å²))

Parameter	Unit	TaIr ₂ Ge ₂	PbTaSe ₂
T_c	K	3.5	3.72
$\mu_0 H_{c1}(0)$	mT	7.2(1)	7.5
$\mu_0 H_{c2}(0)$	Т	1.95(1)	1.47
$\mu_0 H_c(0)$	mT	70	57
$\xi_{\text{GL}}(0)$	nm	13	15
$\lambda_{GL}(0)$	nm	262	248
κ_{GL}		20	17
γ	mJ mol ⁻¹ K ⁻²	11.0(1)	6.9
$\Delta C/\gamma T_c$		1.55	1.41
$\Theta_{\rm D}$	K	355(3)	112
λ_{ep}		0.56	0.74
$N(E_F)$	st. eV^{-1} per f.u.	3.0	1.7

Table 3. Superconducting parameters of TaIr₂Ge₂ compared to PbTaSe₂

Figures

Figure 1: Crystal structure of $TaIr_2Ge_2$, emphasizing the $Ta@Ir_7Ge_4$ clusters. The tantalum is shown in pale green, iridium in gold, and germanium in blue. Inset: A single 11-coordinate cluster.

Figure 2: Powder x-ray diffraction pattern of a polycrystalline sample of TaIr₂Ge₂ employed in the property determinations. The positions of the Bragg reflections, as determined from the single crystal structure refinement, are marked in green.

Figure 3: Calculated electronic structure for TaIr₂Ge₂. Left: density of states (DOS) in the vicinity of the Fermi level, right: band structure calculations. Total band structure and density of states curves were calculated using PAW pseudopotentials on VASP, with spin-orbit coupling included.

Figure 4: Dependence of superconducting transition on magnetic field measured through the resistivity. Main panel: Resistivity over a temperature range from 1.8 to 300 K, measured under zero external magnetic field (μ_0 H = 0T). Inset: resistivity at the superconducting transition measured in magnetic fields ranging from μ_0 H = 0T to 1T in steps of 0.1T.

Figure 5: The zero-field cooled (ZFC) and field cooled (FC) volume susceptibility through the superconducting transition, measured in a field of 20 Oe.

Figure 6: (a) Magnetization versus magnetic field for $TaIr_2Ge_2$ at various temperatures. (b) Difference between magnetization and the M_{fit} for various temperatures. (c) The estimation of H_{c1}

Figure 7: (a) The specific heat versus temperature (open circles) with fit with combined model (red solid line) with 74% of the weight is Einstein model (dashed blue line) and 26% is Debye model (sold green line). (b)Specific heat divided by T³ versus T. (c)Temperature dependence of

the specific heat of TaIr₂Ge₂ from 1.8 to 5.5 K. The sample was measured under zero applied field (μ_0 H = 0T). The solid lines were drawn from an equal-area entropy construction.

Figure 8: (a) C_p/T versus T with various applied magnetic fields. Solid lines represent equal-area constructions. (b) C_p/T versus T² in low temperature region measured under 1.5 T. Red solid line represents linear fit. (c) Plots of upper critical field obtained from specific heat (red squares) and resistivity (blue circles) measurements. Values were fitted with linear function and equation $H = H_c(0)[1 - (\frac{T}{T_c})^{1.5}]^{1.5}$, respectively.



Figure 1



Figure 2



Figure 3



Figure 4



Figure 5



Figure 6



Figure 7



Figure 8