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The evolution of the local structure of VO$_2$ was investigated across the pressure induced insulator to metal transition (IMT) by means of pair distribution function measurements. The pressure behavior of the V-V and V-O bond lengths have been determined. The data demonstrated that the pressure driven IMT is not activated by the suppression of the Peierls-type distortion. A clear octahedra symmetrization is observed in the metallic phase suggesting a link between structural degree of freedom and the metallization process.

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The microscopic mechanisms at the origin of the spectacular increase of conductivity in VO$_2$ is still elusive in spite of being the topic of many theoretical and experimental investigations for more than 50 years. The ultra-fast nature of the IMT creates several possibilities for logic and memory devices, some of which could potentially be transformative [1]. It is therefore crucial to clarify the physics governing such transition from both theoretical and applicative point of view.

The IMT is characterized by a change in the resistance of four orders of magnitude near 340 K with a simultaneous structural transition from a monoclinic insulating phase (M1) to a tetragonal (rutile) metallic phase (R) [1]. The monoclinic structure ($P2_1/c$ space group) consists of paired V-atoms displaced out of the octahedral planes, forming V-V dimers tilted with respect to the $a$ axis (see right panel of Fig.1). The R phase shows linear and equally spaced chains of V-atoms whereas. The electronic transition coincides with the structural changes suggesting a key role of the electron-lattice interactions. The transition is therefore ascribed to the removal of the V-V dimerization along the V-chain [2, 3]. On the other hand, theoretical and experimental evidences support the view of VO$_2$ as a Mott insulator [4–6]. Recent literature suggests that VO$_2$ should be considered a Peierls-Mott insulator where electron-electron correlations and dimerization of V ions both contribute to the opening of an insulating gap [7, 8].

VO$_2$ can assume several structural phases depending on doping, temperature and pressure. Two additional insulating phases appears on both applying uniaxial stress [9, 10] or doping [11, 12]: the monoclinic M2 phase (space group $C2/m$) [13] and the triclinic (T) phase [10, 12].

An even more complex scenario has recently emerged from high pressure (HP) experiments. Indeed, an IMT is also observed applying pressure [16–18]. Optical measurements identifies the onset of this transition at around 10 GPa [16]. Transport data confirm an electronic phase transition [17] and recent results suggest that VO$_2$ is fully metallic only above 34 GPa [18]. The nature of the P-induced IMT looks remarkably different from the T-induced IMT. The relationship between the structural degrees of freedom and the electronic transition is far to be fully assessed within an exhaustive framework.

In contrast with the ambient pressure case where the metallic phase and the rutile structure are closely coupled, HP metallic VO$_2$ displays different structural properties. A new HP structural phase has been first reported at around 13 GPa by Mitrano et al. [19] and recently confirmed [18]. This new structure M1$, has which has not be fully determined yet, retains the $P2_1/c$ space group and appears to be a distorted M1 structure [16, 18–20]. Above 34 GPa, an additional structural phase, named X, has been recently detected [18].

To get further insight into the HP phases of VO$_2$ and to clarify the relationship between the structural modification and the P-induced IMT, we investigate the short-range order with a challenging experiment performing X-ray total scattering measurements and atomic pair distribution function (PDF) using a diamond anvil cell [21].

An high performance experimental setup combined with a careful data analysis allowed us to determine the pressure dependence of the V-V and V-O bond distance thus providing the first experimental evidence that the removal of V-V dimers is not at the origin of the electronic transition observed at 13 GPa. Nevertheless, our PDF
data suggest that a correlation between structural and
the electronic degree of freedom still exists in the high
pressure regime.

In order to perform high pressure PDF experiment
we fully exploited the technical capabilities recently
developed at beamline 11-ID-B at the Advance Photon
Source at Argonne National laboratory. In particular,
we took advantage of the new experimental setup that
was successfully used to collect high pressure PDF
data up to $Q \geq \sqrt{20 A^{-1}}$ [21]. VO$\text{2}$ powder was
prepared by mixing $V_2O_3$ and $V_2O_5$ at high temperature
with argon gas flow, as described in Ref. [16]. A cross
DAC with 300 $\mu$m diamond cuets and a cell seat
with a 70 $\mu$m scattering aperture was employed [22].
The sample was dry-loaded in a 100 $\mu$m hole. A
dry loading (no pressure medium) avoids further
problems in data extraction because of the pressure
medium diffraction peaks. The diameter of the
X-ray beam was 80 $\mu$m. The sample pressure
was estimated using the ruby fluorescence method.
Total scattering data collected from an empty cell
were used to perform the background subtraction.
At each pressure, measurements were carried out in multiple
exposures ($\lambda = 0.137020 \AA$). The 2D data sets
were combined and integrated [23] and the Bragg and
diffuse scattering from diamonds was identified and masked
following the procedure described in ref.[21].

The expected M1 structure was observed at ambient
pressure with lattice parameters $a = 5.74\AA$, $b = 4.49\AA$,
$c = 5.39\AA$ and angle $\beta = 122.6^\circ$. In the M1 structure,
the VO$_6$ octahedra are not regular (see Fig.1) since
there are two different apical V-O bonds and two short
and two long equatorial V-O bonds. Because of V-V
dimerization (Peiers distortion), V atoms form chains not parallel to
the a axis (yellow and orange arrows in Fig.1) and two dif-
ferent V-V$_1$ bonds are therefore present: $V$-$V_{1s} = 2.65\AA$
and $V$-$V_{1t} = 3.12\AA$.

In Fig.1, the extracted $G(r)/s$ obtained at each pressure
are displayed over the 1-10 $\AA$ range. The first peak below
2 $\AA$ contains the contribution from the V-O pairs, while
the two peaks observed between 2 and 3 $\AA$ correspond
to the V-V$_1$ bond lengths (dashed area in Fig.1). Finally
the peak around 3.5 $\AA$ contains contributions from
$V$-$V_2$ bond lengths. No significant changes are observed
for the V-O and V-$V_2$ peaks. The main and the most
important evidence is that the V-V$_1$ distances remain
distinction all up to 22 GPa (as also do the relative intensities of the two peaks) demonstrating that V-
V dimers still exist above the structural transition at 13
GPa (M1’ phase). At ambient pressure, the T-driven
IMT is characterized by the complete suppression of the
V-V dimerization which takes place with the monoclinic
to the rutile structural transition [1]. The P-driven met-
alization process does not show a strict correlation to the
disappearance of the V-V dimers. As a matter of fact,
most of the changes are observed above 4 $\AA$ and involve

FIG. 1: Extracted $G(r)$ as a function of pressure. The PDFs
are scaled up adding a constant to facilitate data visualization.
The V-V dimer bond distances are highlighted by
the dashed area. The VO$_2$ structure is reported too. Blue
atoms=Vanadium, red atoms=Oxygen. The yellow and orange
arrows correspond to the V-V dimer.

FIG. 2: PDF data at selected pressure: blue circle are used for
the extracted $G(r)$, red line indicates the calculated $G(r)$ and
the green line is the residual (shifted down to facilitate
data visualization). The black arrows in the two bottom plots
indicate the PDF features that cannot be modeled using the
$P2_1/c$ space group.
inter-polyhedra bond-lengths. Two shoulders appear respectively around 4.5 Å and 7.5 Å with the onset of the new M1’ phase and become more preeminent as pressure increases.

The PDFs of VO₂ as a function of pressure were modeled with the M1 structure (P2₁/c) over the 1 < R < 10 Å range using lattice parameters obtained from the Rietveld refinement of the average structure. Fitted parameters were: the scale factor, the correlation parameters, lattice parameters, atomic positions, and isotropic atomic displacement. In Fig. 2 the extracted G(r) (blue dots) are displayed together with the calculated G(r) (red line) for selected pressures. The M1 model describes very well the features of the PDF data up to 13 GPa. Around 12-13 GPa the structural transition to the HIP M1’ phase takes place together with the electronic transition [16–18]. The data collected above this pressure can still be well modeled using the M1 phase confirming that the M1’ phase is a distorted monoclinic structure which retains the P2₁/c space group [18, 19]. However, two shoulders fail to be properly reproduced (see Fig.2). These peaks contain contributions from all the possible pairs between V and O atoms so it is not possible to identify which bond length is more affected by this structural modification.

The great advantage of PDF is the possibility of probing the local structure on different length scales by varying the range refined [26–28]. The medium-range order was investigated testing the M1 model over the 10 to 20 Å r-range. Fig.3 shows fits of data Fourier transformed from using the P2₁/c space group. At 5 GPa, the agreement factor RWF decreases moving from the 1-10 Å to the 10-20 Å range (two upper plots in Fig.3). This means that both medium and short-range order are characterized by the same structural features described by the M1 phase. On the contrary, no improvements of the RWF factor are observed extending the r-range above 13 GPa (two bottom plots in Fig.3). The greatest discrepancies between the calculated and the measured PDFs are actually found over the 10-20 Å range. This means that the structural modifications taking place at 13 GPa mostly affect the medium range order. Several VO₂ polymorph structures, in particular the M2 and the T structures were then tested on the data collected above 13 GPa. The best fit was still obtained using the M1 structure. Therefore we conclude that the P2₁/c model works well at the very local level ((1-10 Å)) over the entire pressure range. Pressure affects mostly the long range order structure in good agreement with previous X-ray diffraction results indicating the presence of a new distorted monoclinic phase (M1’) closely correlated to the M1 structure [18, 19].

The results of the quantitative analysis performed over the 1-10 Å range, are displayed in Fig.4 and in Fig.5. The pressure dependence of the V-V₁ (a), the V-V₂ (c) bond lengths and δ: V-V₁ twisting angle (b).
phase, HP transport data show an electronic transition be fully metallic only above 34 GPa in the X structural
is displayed in Fig.5(a). At ambient condition the VO
6 V-V dimerization is far to be removed. The distortion of the VO
6 was estimated using the equation reported in Ref.[14]. A more pronounced reduction of the octahedral distortion is observed between 0 and 8 GPa. The M1′ phase is thus characterized by more symmetric octahedra (see Fig.5(b)). This is also consistent with the pressure dependence of the V-V2 bond distances (see Fig.4(c)) which are mostly associated to structural modifications of the octahedra. This pressure behavior remarkably mimics the one observed in the V pre K-edge peaks of the system which correspond roughly to the e g and t2g contributions to the absorption [29]. This result sheds some light on the mechanism behind the IMT in VO2. Although VO2 appears to be fully metallic only above 34 GPa in the X structural phase, HP transport data show an electronic transition at 13 GPa with a reduction of the band gap and a resistance change of around 2 order of magnitude [17, 18] (M′ phase). The M1 phase can still be used to refine the high pressure PDF data. The structural modification affects mostly the medium range order structure confirming that the M1′ phase is a distorted monoclinic structure [18, 19]. Our PDF data show clearly that the VO6 octahedra are less distorted above 13 GPa and we provided the first experimental evidence that the V-V dimers are still present up to the highest pressure. The removal of the V-V dimerization is therefore not a crucial factor for inducing the electronic transition in VO2. It was recently discovered that orbital occupation can be controlled by strain in VO2 thin films [30]. In this case, the IMT temperature is reduced by 60 K in the maximally strained sample. Strain increases the difference between the apical and the equatorial V-O bond length values leading to a change of orbital occupation and to the decrease the IMT temperature [30]. An opposite mechanism may be invoked to explain the effects of pressure on VO2. The V-O bond distances display a closer distribution above 13 GPa, suggesting that a change in orbital occupation driven by octahedral symmetrization may be at the origin of electronic transition. The octahedral ligand-field splitting affects the hybridization strength between the V 3d and the O 2p orbitals leading to a different electronic structure arrangement.

In conclusion, we were carrying out a detailed investigation of the evolution of the local structure of VO2 up to 22 GPa using HP X-ray total scattering measurements and atomic pair distribution function. We provide the first experimental evidence that the V-V1 dimerization is not suppressed up to 22 GPa. The V-V1 bond distances remain well distinguished up to 22 GPa, demonstrating that the suppression of the Peierls distortion is not correlated in any way with the change of the electronic properties in the pressure range under investigation. However, at the conductive transformation around 13 GPa, structural modifications of both short and mainly medium range order do occur. The quantitative data analysis clearly shows an octahedral symmetrization which appears to be linked with the band gap reduction. We unambiguously demonstrated that the nature of the P-induced IMT is remarkably different from the T-induced IMT. This result underlines that orbital-lattice coupling plays a significant role in the pressure driven IMT and can represent a severe benchmark for any theoretical ab-initio calculation devoted to a deeper understanding of the physics of this system.

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