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Fundamental Limits of Exciton-Exciton Annihilation for Light Emission in Transition Metal Dichalcogenide Monolayers

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Abstract

We quantitatively illustrate the fundamental limit that exciton-exciton annihilation (EEA) may impose to the light emission of monolayer transition metal dichalcogenide (TMDC) materials. The EEA in TMDC monolayers shows dependence on the interaction with substrates as its rate increases from 0.1 cm²/s (0.05 cm²/s) to 0.3 cm²/s (0.1 cm²/s) with the substrates removed for WS₂ (MoS₂) monolayers. It turns to be the major pathway of exciton decay and dominates the luminescence efficiency when the exciton density is beyond 10¹⁰ cm⁻² in suspended monolayers or 10¹¹ cm⁻² in supported monolayers. This sets an upper limit on the density of injected charges in light emission devices for the realization of optimal luminescence efficiency. The strong EEA rate also dictates the pumping threshold for population inversion in the monolayers to be 12-18 MW/cm² (optically) or 2.5-4×10⁵ A/cm² (electrically).

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Two-dimensional (2D) transition metal dichalcogenide (TMDC) materials such as monolayer MoS₂ and WS₂ promise to enable the development of atomic-scale light emission devices owing to their semiconducting nature, perfect surface passivation, and strong exciton binding energy [1]. A key issue for the device development is to understand the exciton dynamics of these materials, which has been known bearing substantial difference from what observed at conventional materials. In particular, the extraordinary exciton binding energy in the TMDC monolayers [2-5] is expected to enable strong many-body interactions like exciton-exciton annihilation (EEA). Recent studies have demonstrated that the EEA rate in monolayer TMDC materials is indeed two orders of magnitude higher than that in conventional semiconductor materials [6-9]. However, much fundamental of the EEA has remained to be elusive. For instance, substantial discrepancy can be found in the previous studies as some reported negligible EEA in the monolayers[10-12] shown to have strong EEA by others[7,8]. It is also not clear how the EEA could depend on the nature of the materials and the environment at the proximity like substrates. Most importantly, although it is generally known that EEA may affect luminescence efficiency, there is no quantitatively understanding about how the strong EEA could affect the light emission efficiency of the monolayers in unusual ways. This understanding would provide useful guidance for the rational design of high-performance light-emission devices.

Here we quantitatively elucidate the fundamental limit that the strong EEA may impose to the luminescence efficiency and lasing threshold in monolayer TMDC materials. We evaluate the EEA and its effect on luminescence for both suspended monolayers and monolayers supported by substrates. The EEA is found subject to influence of substrates as the substrate may decrease the EEA rate and facilitates defect-assisted recombination that can compete with the EEA as the

pathway for excitons to decay. The EEA may turn to be the major decay pathway and dominate the luminescence efficiency when the density of excitons is in scale of 10^{10} cm^{-2} at suspended monolayers or 10^{11} cm^{-2} at supported monolayers. This sets an upper limit on the density of injected charges in light emission devices in order to achieve optimal luminescence efficiency. The strong EEA also dictates the pumping threshold for population inversion in the monolayers to be 12-18 MW/cm^2 at optically pumping or $2.5\text{-}4\times 10^5$ A/cm^2 at electrically pumping.

Fig. 1a-b shows the PL efficiencies (the number of emitted photons vs. the number of adsorbed photons) of suspended monolayer MoS_2 and WS_2 as a function of incident laser power. The samples were prepared by manually transferring chemical vapor deposition-grown monolayers from the growth substrate (sapphire) onto SiO_2/Si substrates pre-patterned with holes (see Supplemental Material [13] for the optical images)[14]. The efficiency is evaluated from PL measurements at room temperature with Rhodamine 6G used as a reference. While the efficiencies vary among these materials, all exhibit an exponential decrease with the incident power increasing, even at an incident power as low as 10 W/cm^2 . In stark contrast, the PL efficiencies of the as-grown monolayers show much milder dependence on the incident power (Fig. 1c-d). We can exclude out any substantial heating effects and the formation of bi-excitons in the measurement as the lineshape and position of the PL show negligible change through the measurement (see Supplemental Material [13] for Methods). It has been known that the PL would redshift or broaden at elevated temperatures and show new peaks at lower energy with the formation of bi-excitons [15-20]. With the exclusion of heating effects and biexcitons, the observed power-dependent PL efficiency may be correlated to another non-linear process: exciton-exciton annihilation (EEA).

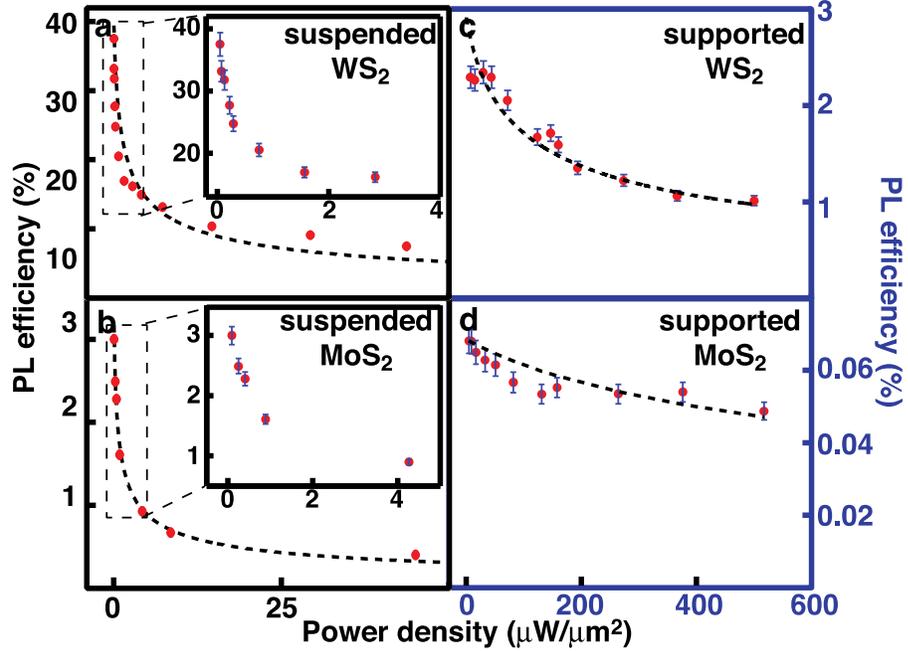


FIG 1. PL efficiencies of (a) suspended monolayer WS₂, (b) suspended monolayer MoS₂, (c) as-grown monolayer WS₂ on sapphire substrates, and (d) as-grown monolayer MoS₂ on sapphire substrates as a function of the incident power density. The dashed lines are simulation results using eq. (3) and the parameters given in Table 1. The insets in (a) and (b) are to better illustrate the results in the corresponding dashed box. All the given error bars are 10%. The error bars in the (a) and (b) are ignored for visual convenience.

To better understand the EEA, we examined the exciton dynamics in the suspended monolayers using pump-probe techniques (see Methods). What we measured is the differential reflection $\Delta R/R$ of a delayed probe beam from the monolayers photoexcited by a pump beam (590 nm). The wavelength of the probe beam is chosen to match the A exciton of the monolayer, and the pumping fluence is set to be small enough to ensure the absorption far below saturation. As a result, the differential reflection ($\Delta R/R$) can be linearly correlated to the density of photo-generated charge carriers at the band edges. Fig. 2 shows the transient differential reflection $\Delta R/R$ collected from suspended WS₂ monolayers (see Supplemental Material [13] for the result of suspended MoS₂ monolayers). We confirmed no substantial heating effect in the experiments by ensuring a reasonable linear dependence of the $\Delta R/R$ at the 0s delay $(\Delta R/R)_0$ on the

pumping fluence because $(\Delta R/R)_0$ is sensitive to the temperature (see Supplemental Material [13]). The decay rate can be found increasing with the pumping fluence (Fig. 2a), consistent with what expected from EEA. The increase of the decay rate also indicates negligible formation of bi-excitons, which would otherwise show the decay rate slowing with the pump fluence increasing [20].

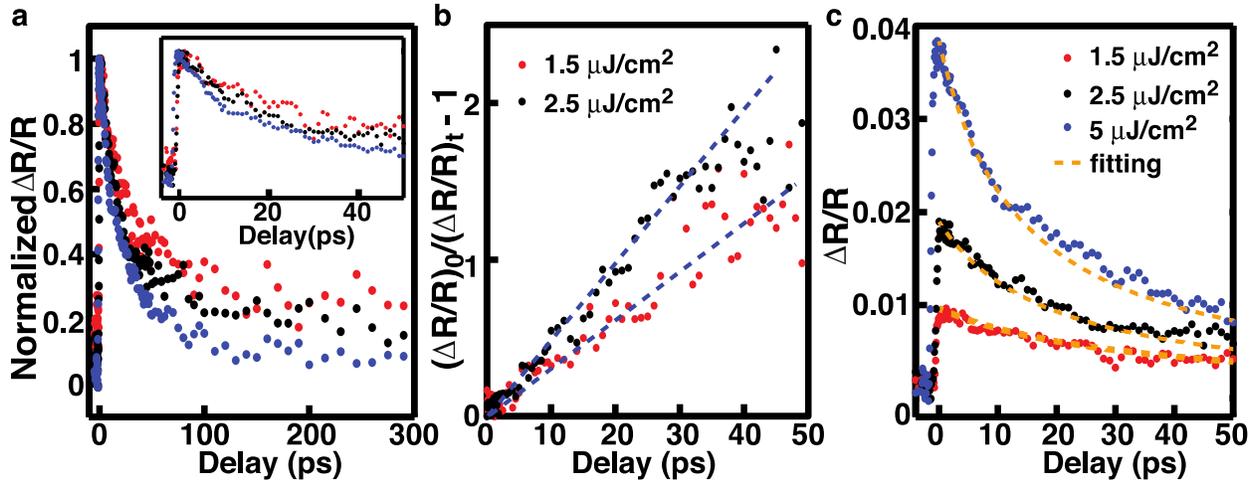


FIG 2. (a) Normalized differential reflection of suspended WS_2 with different pumping fluences, $1.5 \mu\text{J}/\text{cm}^2$ (red), $2.5 \mu\text{J}/\text{cm}^2$ (blue), and $5.0 \mu\text{J}/\text{cm}^2$ (black). Inset: the results for the early stage of the decay. (b) The result of $(\Delta R/R)_0/(\Delta R/R)_t - 1$ derived from the data in (a). The dashed line serves to illustrate the slope of the result. (c) Fitting for the measured differential reflection of suspended WS_2 with different pumping fluences as labeled, The fitted results are plotted in dashed lines and the experimental results are dots.

We can evaluate the rate constant of the EEA based on the pump-probe measurement. Should the exciton decay be dominated by EEA, the rate equation of exciton density would be written as a function of the EEA rate k_{ee} , $dN/dt = -k_{ee}N^2$. And the exciton density $N(t)$ would be correlated to the total photo-generated excitons N_0 as

$$\frac{N_0}{N(t)} - 1 = k_{ee}N_0t \quad (1)$$

As $\Delta R/R$ can be linearly correlated to the density of photo-generated charge carriers, we may have $N_0/N(t) = (\Delta R/R)_0/(\Delta R/R)_t$. We can derive $(\Delta R/R)_0/(\Delta R/R)_t - 1$ from the result given in Fig. 2a, and plot it as a function of the delay time in Fig. 2b. The result shows that $(\Delta R/R)_0/(\Delta R/R)_t - 1$ linearly depends on the delay time at the early stage of the decay (up to 50-100 ps) and its slope linearly increases with the pumping fluence (Fig. 2b). This is consistent with what expected from eq. (1), indicating that the early-stage exciton decay in the suspended monolayer is dominated by EEA. We can also estimate the total photo-generated excitons N_0 from the incident fluence and the absorption efficiency of the monolayers. The absorption of suspended WS_2 and MoS_2 for the pump beam is estimated to be 0.058 and 0.022, respectively, using the refractive index we measured (See Ref. 3 and Supplemental Material [13]). The rate constant k_{ee} can thus be derived from the slope in Fig. 2b as $0.3 \text{ cm}^2/\text{s}$ and $0.1 \text{ cm}^2/\text{s}$ for suspended WS_2 and MoS_2 monolayers, respectively.

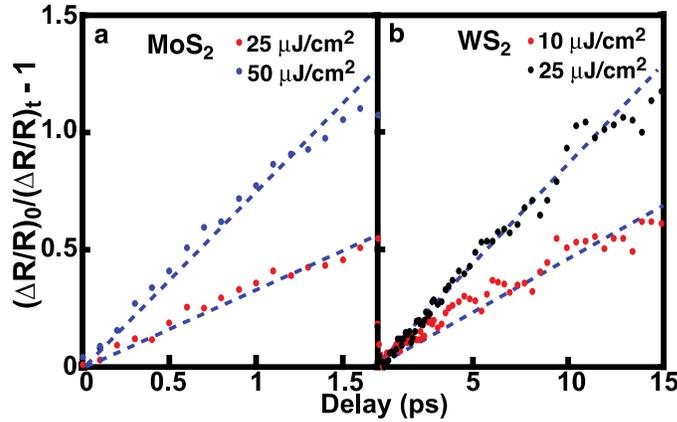


FIG 3. $(\Delta R/R)_0/(\Delta R/R)_t - 1$ of (a) as-grown MoS_2 and (b) as-grown WS_2 . The results are derived from the differential reflection measurement at these materials with different pumping fluences as labeled. The differential reflection measurement results are given in Fig. S4-S5.

To understand the different power dependence of PL efficiency in the supported monolayers, we performed similar pump-probe measurements and data analysis for the as-grown MoS_2 and WS_2 monolayers onto sapphire substrates (see Fig. 3 and Supplemental Material [13]). The EEA rate

is found to be 0.1 cm²/s and 0.05 cm²/s for the supported WS₂ and MoS₂, respectively. This smaller EEA rate indicates the effect of substrates, which may be understood from an intuitive perspective. Generally, the rate of EEA is related with the diffusion coefficient of excitons D and the annihilation radius R that represents the separation of two excitons when the annihilation may occur, $k_{ee} = 4\pi DR$ [21]. The presence of substrates may lower charge mobility and hence the diffusion coefficient [22]. The substrate may also lower the exciton binding energy [23,24], which could subsequently lead to a smaller R . Additionally, the substrate may facilitate defect-assisted recombination that can compete with the EEA as a pathway for excitons to decay [12,25]. While the presence of defect-assisted recombination may not change the EEA rate, it could make the experimental observation of the EEA more difficult, particularly when the defect-assisted decay rate is comparable to or even faster than the EEA rate. For instance, the EEA in the as-grown MoS₂ can be observed only in the first several ps (< 2 ps) and with relatively high pumping fluence ($> 25 \mu\text{J}/\text{cm}^2$) (see Fig.3a and Supplemental Material [13]). We found in experiments that generally it was generally more difficult to observe the EEA in the monolayers showing lower PL intensities. Given the significant effect of substrates on the EEA, we believe that the discrepancies in the previous studies, i.e., the demonstration of different EEA rates in the same materials [8,9,11-13], is likely due to difference in the effect of substrates.

We can better understand the effect of the EEA on luminescence efficiency (Fig.1) by correlating the power-dependent efficiency to the nonlinear and linear decay processes involved. The rate equation of exciton density for the time-averaged PL can be written as

$$\frac{dN}{dt} = -\left(\frac{1}{\tau_r} + \frac{1}{\tau_{nr}}\right)N - k_{ee}N^2 + \alpha I_0 \quad (2)$$

where τ_r and τ_{nr} represent the exciton lifetimes associated with radiative and linear non-radiative recombinations, α and I_0 are the absorption efficiency for the incident wavelength and the incident power density. From eq. (2) we can derive the efficiency of the time-averaged PL as

$$QY = \frac{N / \tau_r}{\alpha I_0} = \frac{\left[\sqrt{(1/\tau_r + 1/\tau_{nr})^2 + 4k_{ee}\alpha I_0} - (1/\tau_r + 1/\tau_{nr}) \right]}{2k_{ee}\alpha I_0 \tau_r} \quad (3)$$

The absorption efficiency α of suspended monolayer MoS₂ and WS₂ can be calculated using the refractive index we measured (See Ref. 3 and Supplemental Material [13]), which is 0.065 and 0.055 for the incidence of 532 nm, respectively. The EEA rate k_{ee} is known from the differential reflection measurement. Then we can evaluate τ_r and τ_{nr} by numerically fitting the measured power-dependent efficiency to eq. (3). The fitting results are plotted (dashed lines) along with the experimental results in Fig. 1 and the fitted value of τ_r and τ_{nr} are given in Table 1.

Table 1. EEA rate and lifetime

	k_{ee} (cm ² /s)	τ_r (ns)	τ_{nr} (ns)
<i>Sus</i> WS₂	0.3	1	0.76
<i>AG</i> WS₂	0.1	4.5	0.13
<i>Sus</i> MoS₂	0.1	28	1
<i>AG</i> MoS₂	0.05	80	0.05

This result may provide useful guidance for the rational design of light emission devices with optimal efficiency. It can guide the proper charge injection in light emission devices for the realization of optimal quantum yield. According to eq. (2), the EEA may turn to be the major pathway of exciton decay ($k_{ee}N^2 > (1/\tau_r + 1/\tau_{nr})N$) when the exciton density

$N > (1/\tau_r + 1/\tau_{nr})/k_{ee}$, which is in the scale of 10^{10} cm^{-2} and 10^{11} cm^{-2} for suspended and supported monolayers, respectively. The charge injection must be controlled to maintain the steady-state charge density well below those values. We can estimate the steady-state charge density as a function of the injected current density J using an equation modified from eq.(2) $N = [\sqrt{(1/\tau_r + 1/\tau_{nr}) + 4k_{ee}J} - (1/\tau_r + 1/\tau_{nr})]/2k_{ee}$. The calculation indicates that, in order to maintain the steady-state charge density well below (<10%) of the threshold values, the injected current density should be no more than 0.2-0.4 A/cm² and 15-30 A/cm² for suspended and supported monolayers, respectively (Fig. 4a). Additionally, the result may help predict the lasing threshold and optical gain coefficient. We use a simple three-level system to represent the pumping process in the monolayer (Fig.4b inset), in which the charges at the ground state 1 (valence band edge) are first pumped to the upper pump level 3 (a higher level in the conduction band) and then quickly decay to the level 2 (conduction band edge). Our analysis indicates the population inversion is completely dictated by the EEA with negligible influence from the linear recombination processes τ_r and τ_{nr} (see S1 in the Supplemental Materials) as

$$\Delta N = \left(\sqrt{W_p^2 + 4k_{ee}W_pN_t} - W_p \right) / k_{ee} - N_t \quad (4)$$

and the optical gain coefficient as $\gamma = \sigma_{12}\Delta N$, where $W_p = \sigma_{13}I_p / h\nu_{13}$ representing the pumping rate. The total charge density N_t in monolayer WS₂ and MoS₂ can be estimated to be $4.17 \times 10^{12} \text{ cm}^{-2}$ and $6.27 \times 10^{12} \text{ cm}^{-2}$ by assuming parabolic band edges at K point and using the average effective mass reported in the literature ($0.4m_0$ and $0.6m_0$ for WS₂ and MoS₂)[26-29]. The stimulated emission (absorption) cross-section σ_{13} (σ_{12}) can be derived from the total charge density and the absorption efficiency α_{13} (α_{12}) as $\sigma_{13} = \alpha_{13} / N_t$ ($\sigma_{12} = \alpha_{12} / N_t$). Without losing generality, we use the pumping wavelength of 532 nm as an example to implement numerical

evaluation. Fig. 4b-c shows the calculated population inversion and optical gain coefficient as a function of the incident power at optically pumping (532 nm) and the injected current density at electrically pumping. The result indicates that the threshold is around 12-18 MW/cm² at optically pumping or 2.5-4 MA/cm² at electrically pumping. This calculation does not take into account any optical enhancement effects, heating effect during the pumping, and possible re-normalization of the bandgap [24,30]. It nevertheless provides useful guidance for the development of 2D TMDC lasers operated at room temperatures. This predicted threshold pumping power is reasonably consistent with one recent study, in which the threshold pumping power for lasing in supported WS₂ monolayer is estimated at 5-8 MW/cm² [31].

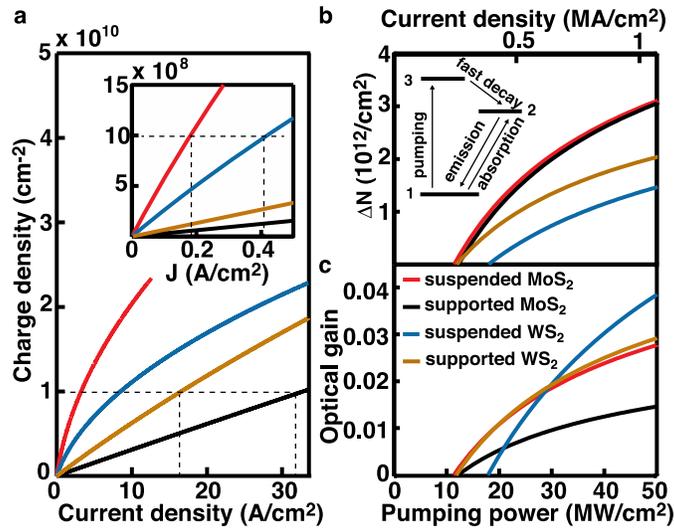


FIG. 4. (a) Steady-state charge density as a function of the injected current density in suspended monolayer MoS₂ (red), supported MoS₂ (black), suspended WS₂ (blue), and supported WS₂ (brown). The dash lines indicate the proper charge density and corresponding injection current density in order to have negligible effects from the EEA. (b) Calculated population inversion and (c) optical gain coefficients as a function of pumping power (optical) and injection current density (electrical) for different monolayers, including suspended monolayer MoS₂ (red), supported MoS₂ (black), suspended WS₂ (blue), and supported WS₂ (brown). Inset, a schematic illustration of the three-level model used for the calculation. The absorption efficiency is approximately set to be 5% for the conversion of the pumping power to the injection current density.

In conclusion, we have quantitatively evaluated the EEA and its effect on light emission for suspended and supported monolayer TMDC materials. The EEA is subject to strong influence of the substrate. It may turn to be the major pathway of exciton decay and dominates the luminescence efficiency when the exciton density is in scale of 10^{10} cm⁻² in suspended monolayers or 10^{11} cm⁻² in supported monolayers. This sets an upper limit for the density of injected charges in light emission devices in order to achieve optimal luminescence efficiency. The strong EEA also dictates the pumping threshold for population inversion in the monolayers to be 12-18 MW/cm² at optically pumping or 2.5-4 MA/cm² at electrically pumping. The result may provide useful guidance for the rational design of atomic-scale light emission devices, including LEDs and lasers.

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