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Phys. Rev. B **92**, 014423 — Published 28 July 2015

DOI: [10.1103/PhysRevB.92.014423](https://doi.org/10.1103/PhysRevB.92.014423)

Band-filling effect on magnetic anisotropy using a Green's function method

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(Dated: July 6, 2015)

Abstract

We use an analytical model to describe the magnetocrystalline anisotropy energy (MAE) in solids as a function of band-filling. The MAE is evaluated in second-order perturbation theory, which makes it possible to decompose the MAE into a sum of transitions between occupied and unoccupied pairs. The model enables us to characterize the MAE as a sum of contributions from different, often competing terms. The nitridometalates $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ with $T=\text{Mn, Fe, Co, Ni}$ provide a system where the model is very effective, because atomic-like orbital characters are preserved and the decomposition is fairly clean. Model results are also compared against MAE evaluated directly from first principles calculations for this system. Good qualitative agreement is found.

I. INTRODUCTION

Magnetocrystalline anisotropy is a particularly important intrinsic magnetic property[1]. Materials with perpendicular magnetic anisotropy are used in an enormous variety of applications, including permanent magnets, magnetic random access memory, magnetic storage devices and other spintronics applications.[2–5]

Modern band theory methods have been widely used to investigate the MAE in many systems[6, 7]. The MAE in a uniaxial system can be obtained by calculating the total energy difference between different spin orientations (out-of-plane and in-plane). However, MAE is usually a small quantity and a reliable *ab initio* calculation requires very precise, extensive calculations. Moreover, MAE is in general harder to interpret from the electronic structure than other properties, such as the magnetization. MAE often depends on very delicate details of the electronic structure[8]. Using perturbation theory, the MAE can be decomposed into virtual transitions between different orbital pairs. In practice the d bandwidth is large enough that it is non-trivial to meaningfully resolve the MAE into orbital components, and predict its dependence on band filling.

The magnetocrystalline anisotropy originates from spin-orbit coupling (SOC)[9], more precisely the change in SOC as the spin quantization axis rotates. Including the relativistic corrections to the Hamiltonian lowers the system energy and breaks the rotational invariance with respect to the spin quantization axis. Here we refer to the additional energy due to the relativistic correction as SOC energy or relativistic energy E^r . MAE is a result of the interplay between SOC and the crystal field[10]. The MAE and change in orbital moment on rotation of the spin quantization axis are closely related. We describe this below and denote them as K and K_L , respectively. Without the SOC, the orbital moment is totally quenched by the crystal field in solids. Except for very heavy elements such as the actinides, SOC usually alleviates only a small part of the quenching and induces a small orbital moment relative to the spin moment. For

$3d$ transition metals, SOC is often much smaller than the bandwidth and crystal field splitting, thus can be neglected in a first approximation. While the E^r is generally small, its anisotropy with respect to spin rotation is even orders of magnitude smaller than SOC energy itself.

Recently, it had been found that a very high magnetic anisotropy can be obtained in $3d$ systems such as Lithium nitridoferrate $\text{Li}_2[(\text{Li}_{1-x}\text{Fe}_x)\text{N}]$ [11–14], which can be viewed as an $\alpha\text{-Li}_3\text{N}$ crystal with Fe impurities. As found both in experiments[15] and calculations[12, 13] using density functional theory (DFT), the $\text{Li}_2(\text{Li}_{1-x}\text{Fe}_x)\text{N}$ system possesses an extraordinary uniaxial anisotropy that originates from Fe impurities. The linear geometry of Fe impurity sites results in an atomic-like orbital and then a large MAE. As found in both x-ray absorption spectroscopy[11] and DFT calculations[11–13], $3d$ ions T have a unusually low oxidation state (+1) in $\text{Li}_2(\text{Li}_{1-x}\text{T}_x)\text{N}$ for $T=\text{Mn, Fe, Co, and Ni}$. Recently, Jesche et al.[16] developed a single crystal growth technique for these system and directly observed that the MAE oscillates when progressing from $T=\text{Mn}\rightarrow\text{Fe}\rightarrow\text{Co}\rightarrow\text{Ni}$. [16] Electronic structure calculations also show that the atomic-like orbital features are preserved for different T elements. Considering the rather large MAE and well separated DOS peaks in this system, it provides us an unique platform to investigate the MAE as a function of bandfilling.

Li and N are very light elements with s and p electrons respectively. They barely contribute to the MAE in $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$, rather MAE is dominated by single-ion anisotropy from impurity T atoms, especially for lower T concentration, where T - T atoms become well separated. In this work, we investigate the magnetic anisotropy with different T elements based on second-order perturbation theory by using a Green's function method. Lorentzians are used to represent local impurity densities of states and calculate the MAE as a continuous function of band-filling. First principles calculations of MAE are also performed to compare with our analytical modeling.

The present paper is organized in the following way.

In section II we overview the general formalism of the single-ion anisotropy[17, 18] with Green's functions and second order perturbation approach[19–24]. Analytical modeling and calculational details are discussed. In section III, we discuss the scalar-relativistic electronic structure of these systems. The bandfilling effect on MAE in $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ with $T=\text{Mn, Fe, Co and Ni}$ are examined within our analytical model and results are compared with first principles DFT calculations. The results are summarized in Section IV.

II. THEORY AND COMPUTATIONAL DETAILS

A. Perturbation theory of the magnetocrystalline anisotropy and orbital moment

Perturbation theory allows us to calculate magnetic anisotropy directly from the unperturbed bandstructure. Orbital moment, SOC energy and their anisotropies can be written in terms of the susceptibility.[7, 17, 21, 23] The relativistic energy E^r due to the spin-orbit interaction $\Delta V_{so}=\xi\mathbf{L}\cdot\mathbf{S}$ can be written as

$$E^r = -\frac{1}{2}\int_{-\infty}^{E_F} \frac{dE}{\pi} \Im(\text{Tr}[\mathbf{G}(E)\Delta V_{so}]) \quad (1)$$

where $\mathbf{G}(E)$ is the full Green's function which includes SOC and can be constructed from the non-perturbed Green's function \mathbf{G}_0 . Using second order perturbation theory (here we consider only systems with a uniaxial geometry), the relativistic energy can be written as

$$\begin{aligned} E^r &= -\frac{1}{2}\Im \sum_{ij} \int_{-\infty}^{E_F} \frac{dE}{\pi} \text{Tr}\{G_0^{ij}(E)\Delta V_{so}^j G_0^{ji}(E)\Delta V_{so}^i\} \\ &= -\frac{1}{2} \sum_i \xi_i^2 \sum_{\sigma=\pm 1} \sum_{m,m'} |\langle m\sigma | \vec{l} \cdot \vec{s} | m'\sigma' \rangle|^2 \chi_{mm'}^{\sigma\sigma'(i)} \\ &\quad + \text{intersite terms} \end{aligned} \quad (2)$$

Green's functions are represented in basis of orthonormalized atomic functions $|i, m, \sigma\rangle$ and i labels atomic sites, m subbands (in cubic harmonics), and σ the spin. The local susceptibility $\chi_{mm'}^{\sigma\sigma'}$, characterizing the transition between two subbands $|m, \sigma\rangle$ and $|m', \sigma'\rangle$, is defined as

$$\chi_{mm'}^{\sigma\sigma'}(E_F) = \chi_{m'm}^{\sigma'\sigma}(E_F) = \int_{-\infty}^{E_F} \frac{dE}{\pi} \Im\{g_m^\sigma g_{m'}^{\sigma'}\}, \quad (3)$$

where g_m^σ is the unperturbed on-site Green's function. Because we only consider the on-site contribution of MAE, only the on-site Green's function or local susceptibility are needed to investigate MAE. We further assume that on-site Green's functions diagonalize in real harmonic space. The angular dependence and bandstructure dependence of relativistic energy E^r are decoupled. In the following, we assume that MAE is dominated by a particular site i , and consider only its contribution.

When the spin quantization axis is along the 001 direction, the spin-parallel (longitudinal) components of SO interaction l_z couple orbitals with the same $|m|$ quantum number ($m=-m'$), while the spin-flip (transverse) ones l_\pm couple orbitals with different $|m|$ numbers ($|m|=|m|\pm 1$). Hereafter we refer those two types of coupling as intra- $|m|$ and inter- $|m|$ types respectively. According to Eq. (2) and absorbing the site index i , the relativistic energy can be written as

$$E_{001}^r = -\frac{\xi^2}{8} \sum_{\sigma=\pm 1} \sum_{m,m'} (A_{mm'} \chi_{mm'}^{\sigma\sigma} + 2B_{mm'} \chi_{mm'}^{-\sigma\sigma}) \quad (4)$$

Positive-definite coefficients A and B are just the spin-parallel and spin-flip parts of the $|\mathbf{L}\cdot\mathbf{S}|^2$ matrix elements. They can be written as

$$A_{mm'} = m^2 \delta_{m,-m'} \quad (5)$$

$$B_{mm'} = \frac{1}{4}(l(l+1) - m(m\pm 1))\delta_{|m|,|m'|\pm 1}. \quad (6)$$

A and B correspond to intra- $|m|$ and inter- $|m|$ transitions respectively. An interesting property of the coefficient matrices is

$$\sum_{mm'} B_{mm'} = \sum_{mm'} A_{mm'} \quad (7)$$

For an arbitrary spin orientation other than the 001 direction, one can either obtain the relativistic energy E^r by rotating G_0 [7] or V_{so} [25, 26] in spin subspace. Here we use the latter approach and the relativistic energy with spin being along 110 direction can be written as

$$E_{110}^r = -\frac{\xi^2}{8} \sum_{\sigma=\pm 1} \sum_{m,m'} (B_{mm'} \chi_{mm'}^{\sigma\sigma} + (A_{mm'} + B_{mm'}) \chi_{mm'}^{-\sigma\sigma}) \quad (8)$$

Notice that spin-parallel coefficients in Eq. (8) are exactly half of the spin-flip coefficients in Eq. (4). If the susceptibility matrix χ is relatively homogeneous with respect to spin, then according to Eqns. (4,7,8) we should expect the spin-flip components of the relativistic energy E^r to be about twice as large as the spin-parallel components[27]. This is true for the weakly magnetic atoms in different compounds.

Let us define the orbital moment anisotropy (OMA) and MAE respectively as $K_L = \langle \mathbf{L}_z \rangle_{001} - \langle \mathbf{L}_z \rangle_{110}$ and $K = E_{110}^r - E_{001}^r$. In this definition, a positive K indicates that the system has a uniaxial anisotropy. If K_L is also positive, the system has a larger orbital magnetic moment along the easy axis. Using Eq. (4) and Eq. (8), the MAE K can be written as

$$K = \frac{\xi^2}{8} \sum_{m,m'} (A_{mm'} - B_{mm'}) (\chi_{mm'}^{\uparrow\uparrow} + \chi_{mm'}^{\downarrow\downarrow} - \chi_{mm'}^{\uparrow\downarrow} - \chi_{mm'}^{\downarrow\uparrow}). \quad (9)$$

MAE is resolved into allowed transitions between all pairs of orbitals $|m, \sigma\rangle \leftrightarrow |m', \sigma'\rangle$, corresponding to the $\chi_{mm'}^{\sigma\sigma'}$ terms. Since A and B are positive-definite, the coefficient of $\chi_{mm'}^{\sigma\sigma'}$ is positive when $(m=-m'$ and $\sigma=\sigma')$ or $(|m|=|m'| \pm 1$ and $\sigma=-\sigma')$, and is negative when $(m=-m'$ and $\sigma=-\sigma')$ or $(|m|=|m'| \pm 1$ and $\sigma=\sigma')$. In general the local susceptibility $\chi_{mm'}^{\sigma\sigma'}$ is also positive-definite, hence we have the following simple selection rule for MAE: For intra- $|m|$ orbital pairs, transitions between same (different) spin channels promote easy-axis (easy-plane) anisotropy; For inter- $|m|$ pairs, the sign is the other way around, transitions between same (different) spin channels promote easy-plane (easy-axis) anisotropy. This simple rule is illustrated in Fig. 1.

Similarly, the OMA K_L can be written as

$$K_L = \frac{\xi}{2} \sum_{m,m'} (A_{mm'} - B_{mm'}) (\chi_{mm'}^{\downarrow\downarrow} - \chi_{mm'}^{\uparrow\uparrow}) \quad (10)$$

Hence, OMA originates from the difference between $\uparrow\uparrow$ and $\downarrow\downarrow$ components of each pair susceptibility while MAE originates from the difference between the spin-parallel and spin-flip components. If we sum over contributions from all the spin components from each pair of orbitals (m, m') and define

$$\chi_{mm'}^\epsilon = \chi_{mm'}^{\uparrow\uparrow} + \chi_{mm'}^{\downarrow\downarrow} - \chi_{mm'}^{\uparrow\downarrow} - \chi_{mm'}^{\downarrow\uparrow} \quad (11)$$

$$\chi_{mm'}^l = \chi_{mm'}^{\downarrow\downarrow} - \chi_{mm'}^{\uparrow\uparrow}, \quad (12)$$

then Eqns. (9,10) can be written as

$$\frac{4}{\xi^2} K = \frac{1}{2} \sum_{m,m'} (A_{mm'} - B_{mm'}) \chi_{mm'}^\epsilon \quad (13)$$

$$\frac{1}{\xi} K_L = \frac{1}{2} \sum_{m,m'} (A_{mm'} - B_{mm'}) \chi_{mm'}^l \quad (14)$$

Obviously, the correlation between OMA and MAE[28] only happens when the susceptibility is dominated only by one of the spin-parallel components. If it is dominated by $\chi^{\uparrow\uparrow}$, then the system has a smaller orbital moment along the easy axis[27]. If it is dominated by $\chi^{\downarrow\downarrow}$, then the system has a larger orbital moment along the easy axis and we have $K = \frac{\xi}{4} K_L$.

Eq. (9) is useful to explain the MAE in two extreme cases. (i) Non-magnetic limit; Since the orbitals are spin-independent, we have $\chi_{mm'}^{\uparrow\uparrow} = \chi_{mm'}^{\uparrow\downarrow} = \chi_{mm'}^{\downarrow\uparrow} = \chi_{mm'}^{\downarrow\downarrow}$. $\chi_{mm'}^\epsilon$ vanishes for every pair of subbands mm' because the spin-parallel components cancel out the spin-flip

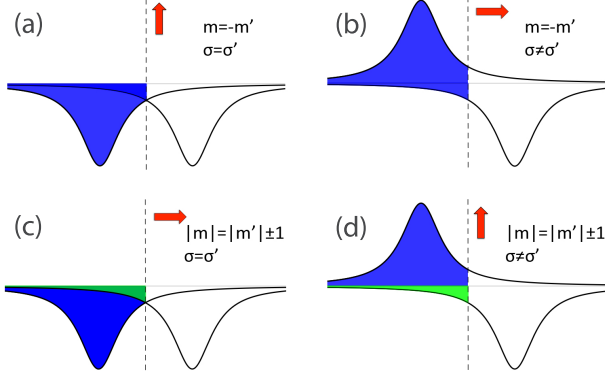


FIG. 1: (Color online) Illustration of the dependence of the easy axis direction on the orbital quantum numbers (m, m') and the spin quantum numbers (σ, σ') of two subbands. Configurations (a) and (d) favor uniaxial anisotropy, while (b) and (c) favor easy-plane anisotropy. The vertical dotted line corresponds to the Fermi energy, E_F . The horizontal line separates the majority (up) and minority (down) spin channels. Occupied states with different $|m|$ numbers are filled with different color.

ones. (ii) Zero crystal field limit; Since orbitals are degenerate, $\sum_{mm'}(A_{mm'} - B_{mm'})\chi_{mm'}^{\sigma\sigma'}$ in Eq. (9) vanishes for each of the four spin components $\sigma\sigma'$. Thus the total

anisotropy vanishes as in a free atom.

Using the expressions of coefficients in Eqns. (5,6), for a d -orbital system, Eq. (9) can be written as

$$\frac{4}{\xi^2}K = 4\chi_{-2,2}^\epsilon + \chi_{-1,1}^\epsilon - \frac{3}{2}(\chi_{-1,0}^\epsilon + \chi_{0,1}^\epsilon) - \frac{1}{2}(\chi_{-2,-1}^\epsilon + \chi_{-2,1}^\epsilon + \chi_{-1,2}^\epsilon + \chi_{1,2}^\epsilon) \quad (15)$$

where the ordering of the states is $|-2\rangle=d_{xy}$, $|-1\rangle=d_{yz}$, $|0\rangle=d_{z^2}$, $|1\rangle=d_{xz}$ and $|2\rangle=d_{x^2-y^2}$. Different point group symmetry results in different orbital degeneracy on site i . By summing up the coefficients of equivalent orbital pairs, Eq. (15) can be simplified.

For tetragonal, square planar or square pyramidal geometries, one pair of orbitals (d_{xz}, d_{yz}) is degenerate. Eq. (15) can be written as

$$\frac{4}{\xi^2}K = 4\chi_{-22}^\epsilon + \chi_{11}^\epsilon - \chi_{12}^\epsilon - 3\chi_{01}^\epsilon - \chi_{-2,1}^\epsilon \quad (16)$$

For linear, trigonal, pentagonal bipyramidal and square antiprismatic geometries, besides (d_{xz}, d_{yz}) orbitals, ($d_{x^2-y^2}, d_{xy}$) orbitals are also degenerate. Eq. (16) can be further simplified as

$$\frac{4}{\xi^2}K = 4\chi_{22}^\epsilon + \chi_{11}^\epsilon - 3\chi_{01}^\epsilon - 2\chi_{12}^\epsilon \quad (17)$$

We recover Eq.(13) in Ref.[21]. On the other hand, for tetrahedral and octahedral geometries, five d orbitals split into two groups E_g and T_{2g} , namely ($d_{z^2}, d_{x^2-y^2}$) and (d_{xy}, d_{yz}, d_{xz}). One can easily show that the right side of the Eq. (15) vanishes as expected for cubic geometry.

Similarly, with the coefficient matrices and orbital degeneracy, one easily recovers the formulas for the orbital moment in the tetragonal system as in Ref.[17] or A1 and A2 in Ref.[7].

B. Bandfilling effect on MAE in a two-level model

As shown in Eq. (9), the MAE and OMA can be resolved into contributions from allowed transitions between all pairs of orbitals. The sign and weight of the contribution are determined by coefficients $A_{m,m'}$ and $B_{m,m'}$, which only depend on the orbital characters of the corresponding orbital pairs. On the other hand, $\chi_{mm'}^\epsilon$ or its four components $\chi_{mm'}^{\sigma\sigma'}$, are determined by the electronic structure, namely the Fermi level (electron occupancy or band-filling), band width, crystal field splitting and spin splitting. Here we investigate the bandfilling effect on the MAE contribution from a single pair of orbitals. For each orbital pair mm' , there are four spin components, two spin-parallel ($\uparrow\uparrow$ and $\downarrow\downarrow$) terms and two

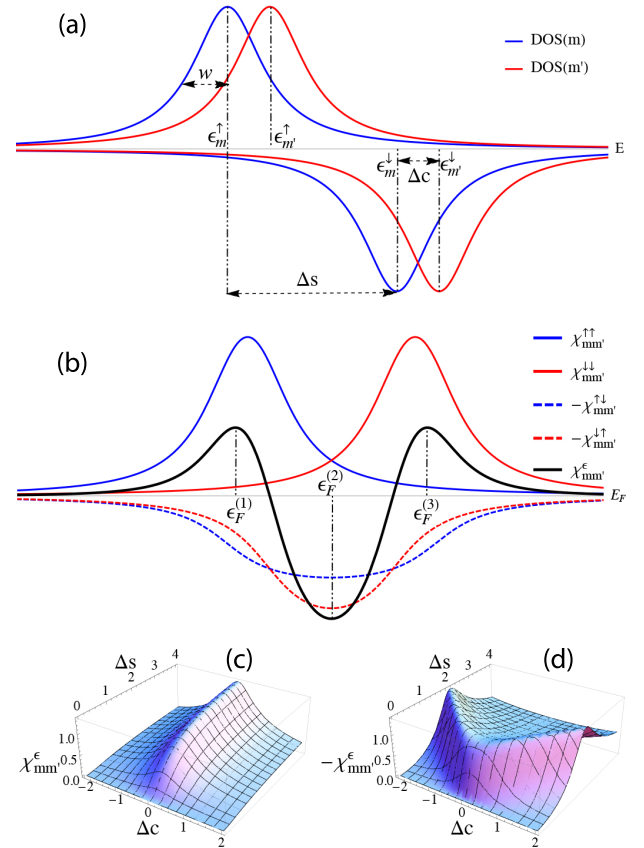


FIG. 2: (Color online) (a) Schematic Lorentzian shape Densities of states for subbands m and m' . (b) $\chi_{mm'}^\epsilon$ and its four spin components as functions of Fermi energy. The amplitudes of $\chi_{mm'}^\epsilon$ maximum at $\epsilon_F^{(1,3)}$ (c) and minimum at $\epsilon_F^{(2)}$ (d) as functions of spin splitting Δs and crystal field splitting Δc .

spin-flip terms ($\uparrow\downarrow$ and $\downarrow\uparrow$). As assumed in the Anderson model, Lorentzians are used to represent the local densities of state (LDOS) in our analytical model to illustrate the electronic structure dependence of $\chi_{mm'}^{\sigma\sigma'}$ and MAE. Similarly, Ebert et al.[17] used Lorentzians DOS to analytically investigate the orbital magnetic moment and relate it to the impurity density of states at the Fermi level. For simplicity, we use the same width for every Lorentzians orbital and the on-site Green's function for

subband $|m\rangle$ in one spin channel σ is given by

$$g_m^\sigma(E) = \frac{1}{E - \varepsilon_m^\sigma + iw} \quad (18)$$

where ε_m^σ is the band center and w is the half-width. The corresponding LDOS for subbands $|m\rangle$ and $|m'\rangle$ in two spin channels are shown in Fig. 2(a). For simplicity, we

further assume that the two subbands have the same spin splitting $\varepsilon_m^\sigma - \varepsilon_m^{\sigma'} = \varepsilon_{m'}^\sigma - \varepsilon_{m'}^{\sigma'} \equiv \Delta s$, or equivalently, have the same crystal field splitting $\varepsilon_m^\sigma - \varepsilon_{m'}^\sigma = \varepsilon_m^{\sigma'} - \varepsilon_{m'}^{\sigma'} \equiv \Delta c$ in the two spin channels.

According to Eq. (3), the pairwise local susceptibility for orbitals $|m, \sigma\rangle$ and $|m', \sigma'\rangle$ can be written as

$$\chi_{mm'}^{\sigma\sigma'}(E_F) = \begin{cases} \frac{1}{\pi} \frac{1}{\varepsilon_{m'}^{\sigma'} - \varepsilon_m^\sigma} (\arctan[\frac{E_F - \varepsilon_m^\sigma}{w}] - \arctan[\frac{E_F - \varepsilon_{m'}^{\sigma'}}{w}]) & \text{if } \varepsilon_m^\sigma \neq \varepsilon_{m'}^{\sigma'} \\ D(E_F) = \frac{1}{\pi} \frac{w}{(E_F - \varepsilon_m^\sigma)^2 + w^2} & \text{if } \varepsilon_m^\sigma = \varepsilon_{m'}^{\sigma'} \end{cases} \quad (19)$$

$\chi_{mm'}^{\sigma\sigma'}(E_F)$ is a positive-definite function for any E_F and reaches the maximum at $E_F = (\varepsilon_m^\sigma + \varepsilon_{m'}^{\sigma'})/2$. The maximum value increases as the two band centers approach each other until becoming degenerate, because the energies required to transfer electrons from occupied states to the unoccupied states become smaller. Band narrowing increases $\chi_{mm'}^{\sigma\sigma'}$ quickly (nearly $1/w$) till it reaches the atomic limit. When the bandwidth becomes comparable to or smaller than the SOC constant, SOC can lift the orbital degeneracy and shift two states, one above and the other below the Fermi level E_F completely. On the other hand, if the Fermi level sits between two well-separated narrow subbands and bandwidth is small compared to the distance between the Fermi level and the two band centers, $w \ll E_F - \varepsilon_m^\sigma$ and $w \ll E_F - \varepsilon_{m'}^{\sigma'}$, according to Eq. (19) then $\chi_{mm'}^\varepsilon = 1/(\varepsilon_{m'}^{\sigma'} - \varepsilon_m^\sigma)$ does not depend on the Fermi energy.

Using Eqns. (11, 19), the dependencies of $\chi_{mm'}^\varepsilon$ and its four spin components on the Fermi energy E_F are shown in Fig. 2(b). There is one minimum at $\varepsilon_F^{(2)}$ and two maxima at $\varepsilon_F^{(1,3)}$, with

$$\varepsilon_F^{(i)} = \frac{\varepsilon_1 + \varepsilon_2 + \Delta s}{2} + \frac{i-2}{2} \sqrt{(\Delta c)^2 + (\Delta s)^2 + 4w^2} \quad (20)$$

The two maximum peaks originate from the two spin-parallel terms $\chi_{mm'}^{\uparrow\uparrow}$ and $\chi_{mm'}^{\downarrow\downarrow}$, while the minimum originates from the spin-flip terms $-(\chi_{mm'}^{\uparrow\downarrow} + \chi_{mm'}^{\downarrow\uparrow})$. In Eq. (20), each spin component $\chi_{mm'}^{\sigma\sigma'}$ has its maximum amplitude when the Fermi level is around the middle of corresponding two band centers. The two spin-flip components have their maximum values at the same Fermi level $\varepsilon_F^{(2)}$ because we assume that the two orbitals have the same spin splittings. Contributions from the two spin-flip components become identical when two states $|m\rangle$ and $|m'\rangle$ are degenerate.

As shown in Eqns. (9, 13), the MAE coefficients for intra- $|m|$ (A) and inter- $|m|$ terms ($-B$) have different signs. To have a large uniaxial anisotropy, the Fermi

level should be around the $\varepsilon_F^{(1)}$ or $\varepsilon_F^{(3)}$ for intra- $|m|$ orbital pairs, and $\varepsilon_F^{(2)}$ for inter- $|m|$ orbital pairs. Two orbitals can accommodate four electrons in two spin channels, and $\varepsilon_F^{(i)}$ roughly corresponds to bandfilling of one, two and three electrons with $i=1, 2$ and 3 respectively. Fig. 2(c, d) shows the maximum amplitude of $\chi_{ij}^\varepsilon(E_F = \varepsilon_F^{(i)})$ as functions of crystal splitting Δc and spin splitting Δs . For $E_F = \varepsilon_F^{(1,3)}$, it requires $\Delta c=0$ to align the two subbands in the same spin channel (two subbands becomes degenerate). For $E_F = \varepsilon_F^{(2)}$, it requires $\Delta s = \pm \Delta c$ to align the two subbands in different spin channels.

C. Crystal structures

$\text{Li}_2(\text{Li}_{1-x}\text{T}_x)\text{N}$ crystallizes in the $\alpha\text{-Li}_3\text{N}$ structure type, which is hexagonal and with space group $P6/mmm$ (no. 191). The unit cell of $\alpha\text{-Li}_3\text{N}$ contains one formula unit. There are two crystallographically inequivalent sets of Li atoms, Li_I (1b) and Li_{II} (2c) with $6/mmm$ and $-6m2$ point group symmetries, respectively. The Li_I atoms are sandwiched between two N atoms and form a linear $-\text{Li}_I\text{-N}-$ chain along the axial direction, while Li_{II} sites have twofold multiplicities and form coplanar hexagons which are centered at $-\text{Li}_I\text{-N}-$ chains and parallel to the basal plane. Li_{II} is more close-packed in lateral directions and $3d$ atoms randomly occupy Li_I sites. We carried out DFT calculations for small doping concentration with $x=0.166$ and found that all T elements with $T=\text{Mn, Fe, Co}$ and Ni indeed prefer to occupy Li_I sites. To calculate the electronic structure and MAE, we use a supercell which corresponds to a $\sqrt{3} \times \sqrt{3} \times 2$ superstructure of the original $\alpha\text{-Li}_3\text{N}$ unit cell. Details of the supercell construction can be found in Ref.[12]. For $x=0.5$, as shown in Fig. 3, there are three T atoms in the 24-atom supercell with one on the $1a$ site and the other two on the $2d$ sites. Both T_{1a} and T_{2d} sites are derived from the $1b$ site in the original $\alpha\text{-Li}_3\text{N}$. They have a linear geometry and a strong hybridization with neighboring N

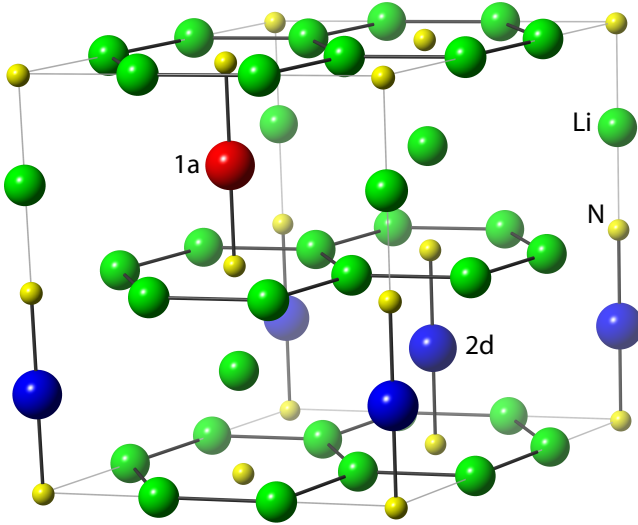


FIG. 3: (Color online) Schematic representation of the supercell used in DFT calculation for $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ with $x=0.5$. Both T_{1a} and T_{2d} sites are derived from the Li_I (1b) site in the original $\alpha\text{-Li}_3\text{N}$ structure while other Li atoms, which form coplanar hexagons, correspond to Li_{II} (2c) sites in the original $\alpha\text{-Li}_3\text{N}$ structure.

atoms along the axial direction. T_{1a} have six Li neighbors while T_{2d} have three T_{2d} and three Li neighbors in the T -Li plane. This structure (denotes as *hex2* in Ref.[12]) is of particular interest because two types of T sites, T_{1a} and T_{2d} , possess very different local surroundings and represent different local impurity concentrations. Along the in-plane direction, T - T distances are rather large especially for the $1a$ site. Since the T_{1a} site represents a relatively low impurity concentration and dominates the uniaxial MAE for $T=\text{Fe}$, most of the results in this work are focused on T_{1a} site in *hex2* supercell. We also consider other concentrations such as $x=0.16$ and $x=0.33$.

D. DFT calculational details

We carried out first principles DFT calculations using the Vienna ab initio simulation package (VASP)[29, 30] and a variant of the full-potential LMTO method[31]. We fully relaxed the atomic positions and lattice parameters while preserving the symmetry using VASP. The nuclei and core electrons were described by the projector augmented wave potential[32] and the wave functions of valence electrons were expanded in a plane-wave basis set with a cutoff energy of 520 eV. For relaxation, the generalized gradient approximation of Perdew, Burke, and Ernzerhof was used for the correlation and exchange potentials. The spin-orbit coupling is included using the second-variation procedure[33, 34]. We also calculated the MAE by carrying out all-electron calculations using the full-potential LMTO (FP-LMTO) method to check our calculational results. For the MAE calculation,

the k -point integration was performed using a modified tetrahedron method with Blöchl corrections with 16^3 k -points in the first Brillouin zone of the 24-atom unit cell. By evaluating the SOC matrix elements $\langle V_{SO} \rangle$ and its anisotropy[27], we resolve the anisotropy of orbital moment and MAE into sites, spins and orbital pairs. The correlation effects are also considered by using LDA+ U method. Here we choose the fully localized limit implementations of the double counting introduced by Liechtenstein et al.[35] considering it is more appropriate for materials with electrons localized on specific orbitals.

III. RESULTS AND DISCUSSIONS

A. Electronic structures

Without considering SOC, the axial crystal field on both T_{1a} and T_{2d} sites splits five $3d$ orbitals into three groups; degenerate ($d_{xy}, d_{x^2-y^2}$) states, degenerate (d_{yz}, d_{xz}) states and d_{z^2} state. Equivalently, they can be labeled as $m=\pm 2$, $m=\pm 1$ and $m=0$ using cubic harmonics.

The scalar-relativistic partial densities of states (PDOS) projected on the T_{1a} site are shown in Fig. 4. For $T=\text{Fe}$, the PDOS obtained is very similar to what was previously reported [12]. The Fe $3d$ shell has seven electrons and the majority spin channels of d orbitals are fully occupied with five electrons.

The Fe d_{z^2} states hybridize with p_z states of N atoms along the axial direction and mix with on-site $4s$ states, which causes the d_{z^2} orbital to be lower in energy than the other d orbitals.[12] The d_{z^2} states spread out and lie below the Fermi level and accommodates one electron in the minority spin channel. The last electron occupies half of the degenerate ($d_{xy}, d_{x^2-y^2}$) states in the minority spin channel. These states have a very narrow bandwidth and cross the Fermi level.

The linear geometry minimizes the in-plane hybridization between the T $3d$ orbitals and the neighboring atoms, making them atomic-like and resulting in narrower bands. The T_{2d} site shows a similar PDOS as T_{1a} site, however, the in-plane hybridization with other T_{2d} sites results in a much broader bandwidth than $1a$ sites.

For other T elements, the DOS peaks are well separated as in $T=\text{Fe}$. The minority spin channel clearly shows a different band-filling pattern with different T elements. The deviation from the rigid band model is also obvious. Spin splitting decreases from Mn to Ni while the crystal field splitting values (the energy difference between $m=\pm 1$ and $m=\pm 2$ states) are larger for $T=\text{Mn}$ and Fe than for $T=\text{Co}$ and Ni.

Fig. 5(a) shows the schematic Fe PDOS, and how the Fermi level changes with different T in a rigid band approximation (RBA). Different T elements correspond to different integer number of $3d$ electrons. Since each degenerate state pair can accommodate two electrons in one spin channel, the Fermi level either intersects the degenerate peaks or sits in the middle of two peaks.

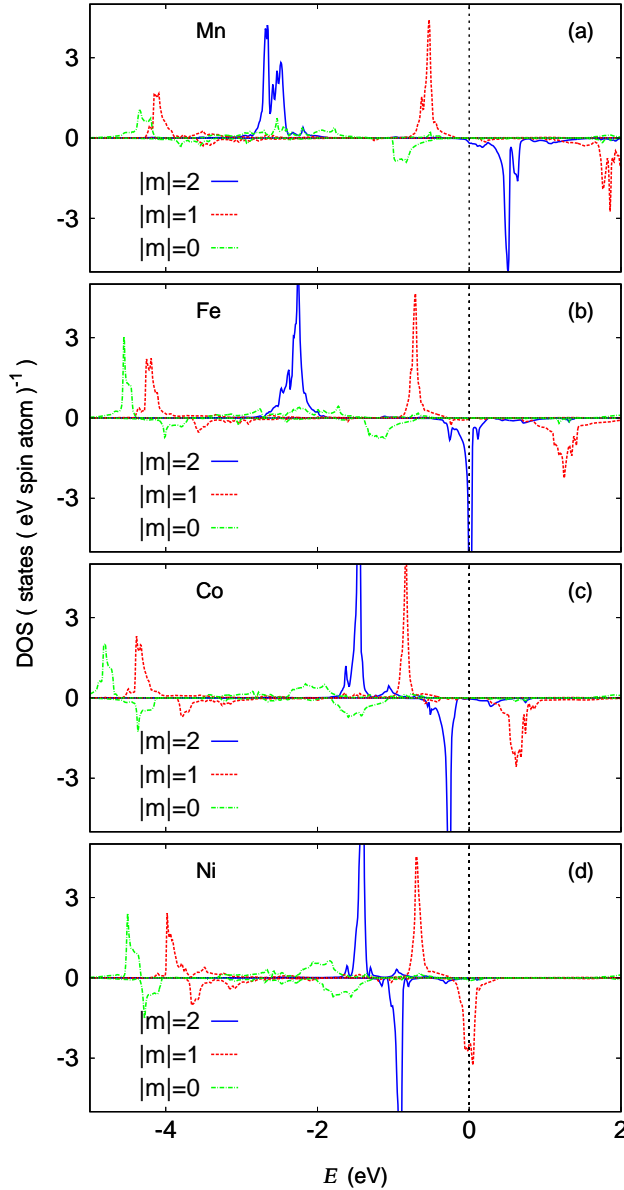


FIG. 4: (Color online) Partial densities of states projected on the 3d states of the T_{1a} site in $hex2$ structure in $Li_2[(Li_{1-x}T_x)N]$, where $x=0.5$ and T is Fe (a), Co (b) and Ni (d). The vertical dotted line corresponds to the Fermi energy, E_F . The horizontal dotted line separates the majority (up) and minority (down) spin channels. Calculation is within LDA, without spin-orbit coupling included.

B. MAE in $Li_2[(Li_{1-x}T_x)N]$ with $T=Fe$

MAE in $Li_2[(Li_{1-x}T_x)N]$ with $T=Mn, Fe, Co$ and Ni and $x = 0.5$ are calculated in DFT and summarized in Table I. The system has uniaxial anisotropy with $T=Fe$ or Ni and easy-plane anisotropy with $T=Mn$ or Co . MAE is dominated by the contributions from the $1a$ site for $T=Fe$ or Ni . Results are in qualitative agreement with previous calculations.[11–13] The extraordinary MAE for

TABLE I: Lattice constants, total and site-resolved MAE $Li_2[(Li_{0.5}T_{0.5})N]$ with $T=Mn, Fe, Co$ and Ni . The MAE values for T_{2d} site are in unit of $meV/atom$, there are two T_{2d} atoms in the supercell.

T	Lattice parameters		$K(meV)$			
	$a(a.u.)$	c/a	cell	T_{1a}	T_{2d}	others
Mn	12.143	1.202	-1.14	-0.35	-0.38	-0.03
Fe	12.091	1.183	20.83	14.77	3.09	-0.12
Co	12.144	1.154	-3.69	-0.89	-1.32	-0.15
Ni	12.113	1.156	2.52	1.71	0.37	0.06

$T=Fe$ originates from the unique bandstructure in this system. Because the well-isolated Fe atoms, such as the Fe_{1a} site in the $hex2$ supercell, provide the major contribution to the uniaxial anisotropy, we focus on the Fe_{1a} site.

As shown in Fig. 1, the sign of the MAE contribution from transitions between a pair of subbands $|m, \sigma\rangle$ and $|m', \sigma'\rangle$, is determined by the spin and orbital character of the involved orbitals. Because the d_{z^2} orbital is spread out, relatively further below the Fermi level and contributes negligibly to the MAE, we only consider the transitions between subbands with $m = -2, -1, 1$ and 2 . Intra- $|m|$ transitions $|1\rangle \leftrightarrow |-1\rangle$ and $|2\rangle \leftrightarrow |-2\rangle$ promote easy-axis anisotropy when they are within the same spin channel, and easy-plane anisotropy when between different spin channels. For inter- $|m|$ transitions, it is the other way around. Transition $|\pm 1\rangle \leftrightarrow |\pm 2\rangle$ promotes easy-plane anisotropy when it is within the same spin channel and easy-axis anisotropy when between different spin channels. The signs and coefficients of the MAE contributions from different orbital pair transitions are indicated in Fig. 5(a). Transitions contribute to MAE only when they cross the Fermi level. The amplitude of MAE depends on the orbital characters and also the energy difference between the two band centers. When the Fermi level intersects the narrow degenerate states, the transition energy required to excite an electron across the Fermi level is very small (between 0 and bandwidth), making the MAE contribution from this pair of orbitals very large. On the other hand, when the Fermi level is between two well separated DOS peaks, the required transition energy is much larger so the amplitude is much smaller.

To elucidate the orbital contributions from the Fe_{1a} site to the MAE in $Li_2[(Li_{0.5}Fe_{0.5})N]$, we approximate the densities of states (DOS) of $|\pm 1\rangle$ (d_{xz}, d_{yz}) and $|\pm 2\rangle$ ($d_{xy}, d_{x^2-y^2}$) subbands with two Lorentzian functions. Crystal field splitting $\Delta c = \epsilon_{|m|=1} - \epsilon_{|m|=2} = 1.8eV$, spin splitting $\Delta s = 2.4 eV$ and half-width $w = 0.06eV$ are used to represent the DFT-calculated PDOS as shown in Fig. 4. The PDOS used in our model is shown in Fig. 5(a) and the MAE contribution from the $1a$ site and its decomposition into orbital pair transitions as functions of the Fermi energy are shown in Fig. 5(b). With $T=Fe$, Fermi level intersects the $|\pm 2, \downarrow\rangle$ states, which results in a large uniaxial anisotropy. Using Eq. (17), Fe_{1a} has

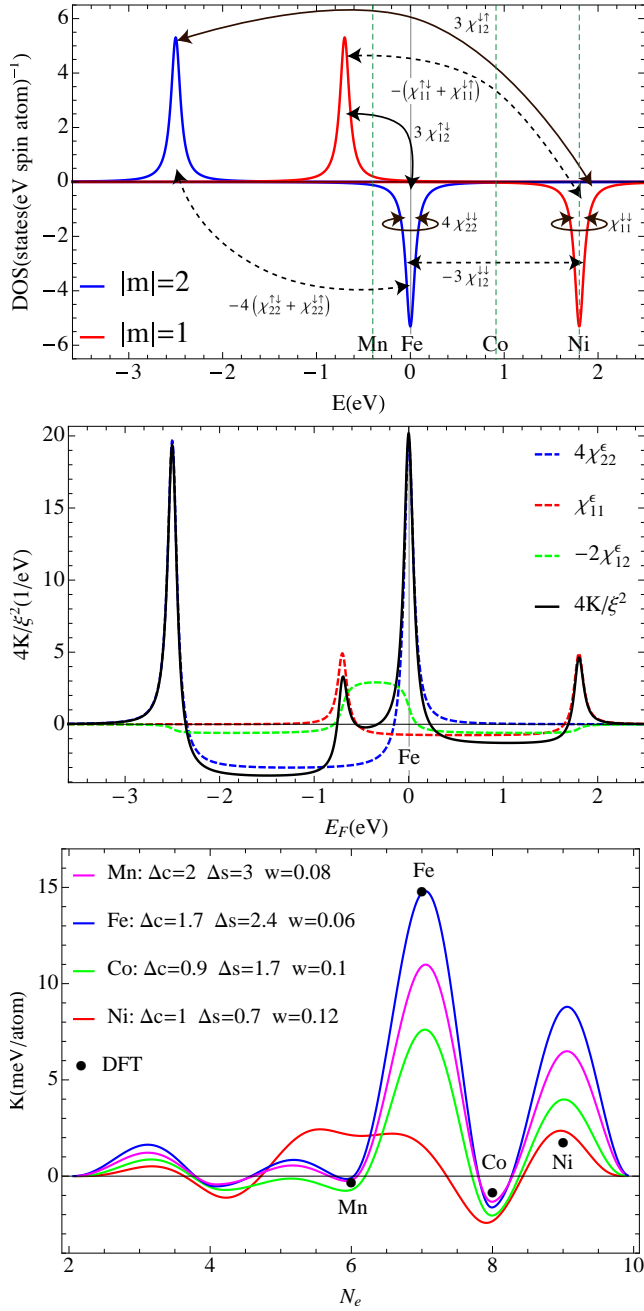


FIG. 5: (Color online) (a) Schematic partial densities of states projected on the 3d states of Fe_{1a} sites. Orbital transitions and the sign of their contributions to the MAE are also shown. Solid line indicates positive contribution (easy-axis) and the dashed line indicates negative contribution (easy-plane) to the easy-axis anisotropy. (b) Scaled MAE $4K/\xi^2$ from T_{1a} site and its decomposition into orbital susceptibilities as functions of band-filling (c) Magnetic anisotropy energy K from T_{1a} site as a function of T . Different sets of electronic structure parameters Δs , Δc and w are used to represent the DFT PDOS on T_{1a} sites in $\text{Li}_2[(\text{Li}_{0.5}\text{T}_{0.5})\text{N}]$ for different T elements.

MAE contribution which is of order of 15 meV/Fe. As shown in Fig. 5(b), for $T=\text{Fe}$ nearly all MAE contributions are from the transitions $|2, \downarrow\rangle \leftrightarrow |-2, \downarrow\rangle$, in other words, between $d_{x^2-y^2}$ and d_{xy} orbitals in the minority spin channel.

To compare with the above analytical modeling, MAE calculations were carried out in both VASP and all-electron FP-LMTO. The difference of MAE values using two methods is less than 5% for $T=\text{Fe}$. To decompose the MAE, we evaluate the SOC matrix element $\langle V_{so} \rangle$ and its anisotropy $K(\langle V_{so} \rangle)$, which can be easily decomposed into sites, spins and orbital pairs[27]. We found that $K \approx K(\langle V_{so} \rangle)/2$ for all T compounds, which suggests that 2nd order perturbation theory is a good approximation. As shown in Table I, for $T=\text{Fe}$, the total MAE is 20.8 meV (per 24-atom cell) and MAE contributions from 1a and 2d sites are 14.77 meV/Fe and 3.09 meV/Fe, respectively. The contributions from Li and N atoms are nearly zero as expected. Thus, the impurity Fe (especially Fe_{1a}) atoms are essentially the only MAE providers. By further investigating the matrix element of SOC on the 1a site, we found that nearly all the MAE contributions came from intra- $|m|$ transitions of $|2, \downarrow\rangle \leftrightarrow |-2, \downarrow\rangle$. As shown in Table II, the $4\chi_{22}^\epsilon$ term (dominated by $\chi_{22}^{\uparrow\downarrow}$ for $T=\text{Fe}$) contributes 15.1 meV/Fe and the χ_{11}^ϵ term has a much smaller negative value of -0.42 meV/Fe, while other terms are negligible. Hence, DFT results agree with our model very well.

With magnetization along the c direction, the SOC can lift the orbital degeneracy and shift two narrow bands $m = \pm 2$, one below and the other above the Fermi level completely, with orbital quantum number $m^c = \pm 2$ respectively, where m^c is the orbital quantum number in the complex spherical harmonics. As a result, the density of states at the Fermi level becomes very small. Indeed, experiments[15] found that this system to be an insulator for $T=\text{Fe}$. It had been shown that [11–13, 36] the correlation effect further enhances the separation between occupied and unoccupied states. Using the LDA+ U method, we also found that correlation can enhance the orbital moment when the spin is along the axial direction.

Fe concentration and site disordering can significantly affect the MAE. As we have shown, the Fe_{2d} sites, which represent a high doping concentration, have much lower anisotropy than the Fe_{1a} sites, which represent a lower doping concentration. By replacing the Fe_{2d} sites back with Li atoms in the $\text{hex}2$ supercell, we calculated the MAE with a smaller concentration $x=0.166$ and found that MAE increase to 22 meV/Fe, which is in very good agreement with previous calculations.[12]. An interesting concentration is $x=0.33$. If only one of two 2d sites is occupied by Fe in the $\text{hex}2$ supercell as shown in Fig. 3, then this configuration would correspond to $x=0.33$ and the supercell has two well-isolated Fe atoms. The DFT calculation shows high MAE with a value of 20 meV/Fe. On the other hand, if the two Fe atoms occupy the 2d sites and then are not well separated, the resulted MAE is much smaller (2.8 meV/Fe). Even if we assume that

Fe atoms tend to separate, with a concentration beyond $x=0.33$, it is unavoidable to have Fe atoms neighboring each other and the hybridization between them cause the MAE (per Fe) to decrease. Furthermore, impurity sites are disordered as found in experiments. At least at a higher concentration, many Fe atoms would not have the symmetric lateral surroundings as the two Fe sites do in the *hex2* supercell we used in calculations. This site disordering may also have an effect on MAE by lowering the point group symmetry of Fe impurity sites. And the $m = \pm 2$ states on Fe sites are no longer degenerate, which may decrease MAE per Fe.

C. MAE in $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ with $T=\text{Mn, Co and Ni}$: The bandfilling effect

Fig. 5(a) shows how the Fermi level changes with different T elements in a simple rigid band picture. Only those transitions across the Fermi level contribute to MAE. With T elements other than Fe, the $|\pm 2, \downarrow\rangle$ states become either fully occupied or unoccupied. The large uniaxial anisotropy that originated from transition $|2, \downarrow\rangle \leftrightarrow |-2, \downarrow\rangle$ (term $4\chi_{22}^{\downarrow\downarrow}$) vanishes and other transitions become important depending on the position of the Fermi level. For $T=\text{Ni}$, the Fermi level intersects the degenerate $|\pm 1, \downarrow\rangle$ states. Hence anisotropy contributions are dominated by the transitions $|1, \downarrow\rangle \leftrightarrow |-1, \downarrow\rangle$ (term $\chi_{11}^{\downarrow\downarrow}$). This transition promotes the uniaxial anisotropy as $4\chi_{22}^{\downarrow\downarrow}$ does for $T=\text{Fe}$. For $T=\text{Co}$, the Fermi level is between $|\pm 2, \downarrow\rangle$ and $|\pm 1, \downarrow\rangle$ peaks. The transitions of $|\pm 2, \downarrow\rangle \leftrightarrow |\pm 1, \downarrow\rangle$ (term $-3\chi_{12}^{\downarrow\downarrow}$) and $|\pm 1, \uparrow\rangle \leftrightarrow |\mp 1, \downarrow\rangle$ (term $-(\chi_{11}^{\uparrow\downarrow} + \chi_{11}^{\downarrow\uparrow})$) support easy-plane anisotropy while the transition $|\pm 2, \uparrow\rangle \leftrightarrow |\pm 1, \downarrow\rangle$ (term $3\chi_{12}^{\uparrow\downarrow}$) promotes easy-axis anisotropy. However the two bands involved in the last transition are far away from each other and this contribution is relatively small. Hence, for $T=\text{Co}$ one should expect the system to have easy-plane anisotropy. For $T=\text{Mn}$, there are four transitions that contribute to the MAE, all of them are between the two spin channels; two inter- $|m|$ transitions $|\pm 1, \uparrow\rangle \leftrightarrow |\pm 2, \downarrow\rangle$ (term $3\chi_{12}^{\uparrow\downarrow}$) and $|\pm 2, \uparrow\rangle \leftrightarrow |\pm 1, \downarrow\rangle$ support easy-axis anisotropy, while two other intra- $|m|$ transitions $|\pm 1, \uparrow\rangle \leftrightarrow |\mp 1, \downarrow\rangle$ and $|\pm 2, \uparrow\rangle \leftrightarrow |\mp 2, \downarrow\rangle$ (term $-4(\chi_{22}^{\uparrow\downarrow} + \chi_{22}^{\downarrow\uparrow})$) support easy-plane anisotropy. The four transitions compete and the sign of the total MAE is not obvious and requires a more quantitative description.

The SOC constant ξ changes with element. In Fig. 5(b) we plot the scaled MAE $\tilde{K}=K/4\xi^2$ and its orbital-resolved components as functions of the Fermi level by using parameters of Δs , Δc and w for $T=\text{Fe}$. In a rigid band picture, it clearly shows that Ni also has a uniaxial anisotropy with contributions coming from the $\chi_{11}^{\downarrow\downarrow}$ term. Since we are using a same half-width w of LDOS for $m=\pm 1$ and $m=\pm 2$ subbands, we have $\tilde{K}_{\text{Ni}} \approx \frac{1}{4}\tilde{K}_{\text{Fe}}$ because of the intra- $|m|$ transitions coefficients m^2 , as shown in Eqns. (5,9). Fig. 5(c) shows the MAE K as

a function of the number of occupied electrons by using different sets of Δs , Δc and w parameters to better present DFT-calculated PDOS for different T element, as shown in Fig. 4. The SOC constant ξ is interpolated by using DFT-calculated ξ values for $3d$ elements. Since ξ decreases with the atomic number within a given nl shell, K quickly decreases with smaller atomic numbers due to the factor ξ^2 . The DFT MAE values are also plotted to compare with the modeling MAE function. As shown in Fig. 5(c), with $T=\text{Fe}$ parameters, the modeling MAE (Fe rigid band approximation) can already describe correctly the MAE trend with different T elements.

Although, the RBA predicts the correct easy axis direction for $T=\text{Ni}$, the difference between RBA modeling and DFT is rather large. In RBA modeling, $K_{\text{Ni}}/K_{\text{Fe}}=(\xi_{\text{Ni}}/\xi_{\text{Fe}})^2/4 \approx 0.6$, while the DFT value (1.71meV/atom) for $T=\text{Ni}$ is about one-order of magnitude smaller than for $T=\text{Fe}$. This can be explained as follows. First, we use same bandwidth for all DOS peaks in our modeling. In fact, the $|\pm 1, \downarrow\rangle$ bands are much broader than $|\pm 2, \downarrow\rangle$ bands. The easy-axis anisotropy contribution from the transition between $|\pm 1, \downarrow\rangle$ states decreases with increasing bandwidth; Secondly, the Ni PDOS deviates from the Fe PDOS more than Mn or Co, so RBA is less appropriate for $T=\text{Ni}$. The spin splitting Δs and crystal field splitting Δc are much smaller in Ni than in Fe. This causes the amplitudes of the negative contributions from $|\pm 2, \downarrow\rangle \leftrightarrow |\pm 1, \downarrow\rangle$ and $|\pm 1, \uparrow\rangle \leftrightarrow |\mp 1, \downarrow\rangle$ to become larger and decrease the total uniaxial anisotropy. As shown in Fig. 5(c), if we use a smaller Δs , smaller Δc and larger w to better represent the Ni PDOS calculated from DFT calculations, then much better agreement between model and DFT values can be reached.

For $T=\text{Co}$, the model MAE is about twice the DFT value, probably because of the simplified model DOS. The orbital-resolved T_{1a} MAE calculated in DFT are summarized in Table II. Overall there is a qualitative agreement between DFT and the analytical model for the orbital-resolved MAE values for all T elements. It is interesting that with $T=\text{Co}$, the contribution of $4\chi_{22}^{\epsilon}$ term is comparable to that of $-2\chi_{12}^{\epsilon}$ and χ_{11}^{ϵ} in DFT, which is not expected in model. As shown in Fig. 4(c), there is a small portion of unoccupied $|\pm 2, \downarrow\rangle$ states right above the Fermi level in the minority spin channel, which make $4\chi_{22}^{\downarrow\downarrow}$ terms comparable to others. However, this electronic structure detail is not considered in the simplified DOS we use in modeling. If we neglect the $4\chi_{22}^{\epsilon}$ terms in DFT, then a better agreement between modeling and DFT can be achieved for $T=\text{Co}$.

Thus, the contributions from well-separated impurity sites with T can be well understood. For $T=\text{Mn}$ and Co , the easy-plane anisotropy is a result of competition between different transitions, instead of being dominated by the intra- $|m|$ transition, which strongly depends on the bandwidth of the degenerate $|\pm m\rangle$ states that are intersected by the Fermi level. As a result, the band narrowing effect on MAE is not as strong as for $T=\text{Fe}$ or Ni . As shown in Table I, the contributions from $2d$ sites

TABLE II: Orbital-resolved MAE from T_{1a} site in $\text{Li}_2[(\text{Li}_{0.5}\text{T}_{0.5})\text{N}]$ with $T=\text{Mn, Fe, Co}$ and Ni .

Term	Orbital Transition	K (meV)			
		Mn	Fe	Co	Ni
$4\chi_{22}^\epsilon$	$d_{xy} \Leftrightarrow d_{x^2-y^2}$	-0.86	15.10	0.71	-0.03
χ_{11}^ϵ	$d_{yz} \Leftrightarrow d_{xz}$	-0.22	-0.42	-0.78	3.68
$-2\chi_{12}^\epsilon$	$d_{yz}, d_{xz} \Leftrightarrow d_{xy}, d_{x^2-y^2}$	0.73	-0.18	-0.81	0.09
$-3\chi_{01}^\epsilon$	$d_{z^2} \Leftrightarrow d_{yz}, d_{xz}$	0.03	0.08	-0.01	-0.25

is comparable or even larger than $1a$ sites for $T=\text{Mn}$ and Co .

IV. SUMMARY AND CONCLUSION

Based on second-order perturbation theory, MAE is resolved into contributions from different pairs of orbital transitions, more precisely the difference between spin-parallel and spin-flip components of the orbital susceptibilities of the corresponding orbital pair. In the $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ systems, with $T=\text{Mn, Fe, Co}$ and Ni , the linear geometry of the T sites minimizes the in-plane hybridization and results in atomic-like orbitals around the Fermi level for all T elements. The MAE oscillates with the atomic number from $T=\text{Mn}$ to $T=\text{Ni}$, which is a result of the competition between contributions from all allowed orbital transitions. As the Fermi level evolves with T , different orbital pair transitions dominate the contribution to MAE. For $T=\text{Fe}$ and $T=\text{Ni}$, the intra- $|m|$ transitions within the minority spin channel dominate

the MAE contribution and result in a uniaxial anisotropy. For $T=\text{Mn}$ and Co , the easy-plane anisotropy is a result of the competition between contributions from several transitions with different signs. Using Lorentzian density of states, we investigate the bandfilling effect on MAE in an analytical model based on a Green's function technique. We show the MAE as a continuous function of atomic number. This analytical model can already describe the correct trend of the MAE obtained using DFT, by just using a simple rigid Fe band picture. If we take into account of the deviation from the rigid Fe band model and some details of DFT electronic structure, an even better agreement between Model and DFT can be found. To further validate our modeling analysis, we also calculate the orbital-resolved MAE by evaluating the SOC matrix element in DFT. Overall, $\text{Li}_2[(\text{Li}_{1-x}\text{T}_x)\text{N}]$ with $T=\text{Mn, Fe, Co}$ and Ni is a unique system which clearly shows the bandfilling effect on MAE and the nature of this effect can be understood in a very simple model.

V. ACKNOWLEDGMENTS

We would like to thank A. Jesche, P. Canfield, V. Antropov, A. Chantis, B. Harmon and D. Johnson for helpful discussions. The work was supported by the Department of Energy-Energy Efficiency and Renewable Energy, Vehicles Technology Office, PEEM program, under Contract No. DE-AC02-07CH11358 for the operation of Ames Laboratory (USDOE).

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