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Valence Band Structure of Highly Efficient p-type Thermoelectric PbTe-PbS Alloys

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Abstract

Experimental evidence is given relevant to the temperature-dependence of valence band structure of PbTe and PbTe_{1-x}S_x alloys (0.04 \leq x \leq 0.12), and its effect on the thermoelectric figure of merit *zT*. The *x* = 0.08 sample has *zT* ~ 1.55 at 773K. The magnetic field dependence of the hightemperature Hall resistivity of heavily *p*-type (> 10¹⁹ cm⁻³) Na-doped PbTe_{1-x}S_x reveals the presence of high-mobility electrons. This casts doubts on prior analyses of the Hall coefficient suggesting that temperature induces a rapid rise in energy of the "heavy" hole relative to the "light" hole bands. The electron-like behavior is likely induced by the topology of the Fermi surface when the L and Σ -bands merge. Negative values for the low-temperature thermopower are also observed. The data show that PbTe continues to be a direct gap semiconductor at temperatures where the *zT* and *S*² σ of *p*-type PbTe are optimal e.g. 700-900K.

Introduction

PbTe^{1,2} and its alloys with PbS^{3,4} are the materials with some of the highest thermoelectric figures of merit in the temperature range of 550-750 K suitable for waste heat recovery applications⁵. The figure of merit of a material is $zT=TS^2\sigma/\kappa$, with S the material's thermopower or Seebeck coefficient, κ its thermal conductivity, $\sigma = l/\rho$ its electrical conductivity (ρ = resistivity), and T the absolute temperature. Binary PbTe, both p-type⁶ and n-type,⁷ can be doped to yield thermoelectric figures of merit zT at 750K exceeding $zT \sim 1.4$. The main reasons for this are the favorable power factor $S^2 \sigma$ and low lattice thermal conductivity. In *n*-type material, the power factor benefits from the good mobility that results from the high dielectric constant that screens much of the impurity scattering. The power factor can be high in heavilydoped ($\geq -4 \times 10^{19}$ cm⁻³) *p*-type material because of the complex valence band structure: the Fermi surface of heavily p-type doped material can include numerous pockets. Screening of impurity scattering also is effective here. The low lattice thermal conductivity of even binary PbTe is due to the high anharmonicity of the bonds, which results in intense phonon-phonon scattering at high temperature. The recently identified^{8,9} soft behavior of the PbTe the lattice is related to this anharmonicity, which in turn can be ascribed to the lone-pair s-electrons of divalent Pb.¹⁰ This behavior was also identified in PbSe and PbS.⁸ An incipient phase transition of possibly a similar nature is also present¹¹ in the $PbTe_{1-x}S_x$ alloys studied here, and bodes well for their thermoelectric performance.

Some of the established literature reports a crossing of the light and heavy hole bands of PbTe below the operating temperature range of thermoelectric devices. It has thus been stated,¹ including by authors of the present article,³ that PbTe is an indirect gap semiconductor at temperatures where the zT and $S^2\sigma$ of p-type PbTe are optimal by virtue of the rapid rise in

energy of the so called "heavy" lower valence band (LVB) relative to the "light" upper valence band (UVB). In this article we present experimental evidence to the contrary. We start with a review of the literature that highlights weaknesses in the historical evidence that PbTe, a directgap semiconductor at low T, becomes an indirect gap one with two sections of the valence band crossing at $T_0 \sim 415$ K. We then report transport data that shed light on the T-dependent Hall coefficient $(R_H(T))$ measurements. The results are obtained on heavily-doped p-type state-of-theart samples that exhibit $zT \sim 1.55$ at 670 K < T < 750 K, and show that at $600 \text{ K} R_H(T)$ is not a good measurement of the hole concentration, but contains a substantial contribution from highmobility minority electrons. A possible origin is suggested for this electron-like behavior: it is likely a topological feature of the Fermi surface at hole concentrations where the carriers at the L-point and those around the Σ -points merge.¹² This is a feature similar to that in the Fermi surface of copper, which is responsible for the positive polarity of its thermopower. It is also possible that an impurity band forms due to the high Na concentrations. Revisiting the analysis of the Hall data in the light of these features suggests that high zT values are obtained in PbTe at temperatures where it continues to be a direct-gap semiconductor. This conclusion does not distract from the fact that the holes in the heavy LVB pockets play an important role in the final zT, whether they result from a very high chemical doping level, or are thermally induced in lessdoped samples.

Literature

A review of the experimental evidence for the crossing of two parts of the valence band of PbTe with increasing temperature is necessary to put the present work in context. The band structure of PbTe is long established experimentally at cryogenic temperatures with Fermi surface mapping techniques such as de Haas van Alphen, Shubnikov-de Haas and magnetooptics, and calculations¹² are consistent with the experiments. The valence band of PbTe consists¹² of a "light" UVB at the four L-points of the Brillouin zone, and a "heavy" LVB broadly distributed over the zone near the Σ -points and between the L and Σ -points; at a p-type doping level in the mid-10¹⁹ cm⁻³ range, the Fermi surfaces of these two bands merge together. At low temperature, PbTe is unambiguously a direct-gap semiconductor, with a gap located at the L-point of the Brillouin zone. As long as the Fermi level (ε_F) does not reach the near- Σ -point bands, the conduction (CB) and valence (the UVB) bands are nearly symmetrical and have a small effective mass (at 4K $m^*_{//[111]} \sim 0.24\pm0.05 m_e$; $m^*_{\perp(111]} \sim 0.023\pm0.001 m_e$; $m^*_{DOS} \sim 0.13 m_e$ at the band edge)¹³ and high mobility. The absolute values of the thermopower of *n*-type^{14,15} and UVB-dominated *p*-type¹⁶ materials, when plotted versus carrier concentration at 300 K, are almost identical. The mobility of UVB-holes is less than that of electrons. The direct energy gap is given by:¹⁷

$$\varepsilon_{g}(T) = 171 + \sqrt{164 + 0.19(T + 20)^{2}} \text{ meV} (1),$$

and, at T > 100K, increases nearly linearly with T with d $_g/dT \sim 0.41$ to 0.45 meV K^{-1.13} Bands move little with T for T < 80K because the main origin of the temperature dependence of the band structure is the lattice expansion, and thermal expansion decreases rapidly below the Debye temperature. Because this small energy gap is determined by spin-orbit interactions, the UVB and CB are strongly non-parabolic and follow a Dirac-like dispersion, with effective masses that increase as function of energy into the bands following approximately:¹⁸

$$m^{*}() = m^{*}(=0)(1+2 / g)$$
 (2)

At low temperature (T < 80K), the energy separation between the maxima of the UVB and the LVB is $\Delta_v (0 \text{ K}) = 160-170 \text{ meV}.^{19,20}$ The Fermi surfaces of degenerately-doped samples with enough carriers to reach the LVB looks like a series of tubes arranged along the edges of a cube

inscribed inside the Brillouin zone.¹² The density of states in the LVB is much higher than in the UVB, with a DOS mass of the order of $m_{DOS-LVB}^* \sim 0.6$ to 2.5 m_e^{13} . This gives the LVB holes a higher thermopower than the UVB holes, and thus it is favorable to *zT* to have ε_F in the LVB. At operating temperatures of thermoelectric devices near 600 K, the LVB of heavily-doped $(p>5x10^{19}\text{cm}^{-3})$ PbTe is populated, and contributes to the high *zT* of the material at these temperatures, as has been described previously.^{1,3,12}

PbS has a band structure almost identical to PbTe, with a slightly larger direct L-point energy gap, but a very similar temperature-dependence (as does PbSe):¹⁷

$$_{g} = 263 + [400 + 0.265 T^{2}]^{1/2} \text{ meV}$$
 (3).

The position of the LVB, and its temperature dependence, is less well established than for PbTe, and estimated at Δ_{v} >300 meV at 0 K.²⁰

The transport properties of the valence band of heavily *p*-type doped Pb_{1-y}Na_yTe have been extensively investigated,^{16,21,22,23,24} and are summarized by Ravich et al.¹⁸ The temperaturedependence of $\Delta_{v}(T)$ is such that at *T* above some temperature T_{0} , defined by $\Delta_{v}(T_{0}) = 0$, PbTe turns into an indirect gap semiconductor with the CB minimum at L-point and the maximum of the valence band is the LVB. Two very different temperature dependences for $\Delta_{v}(T)$ above about 80K are reported in the literature (Fig. 1a).

1. Allgaier and Houston²⁵ suggest and Rogers¹⁹ reformulates (T > 80K):

$$\Delta_{\rm v}(T) \sim 160 \text{ meV} - 2.2 \text{ k}_{\rm B}T = 160 - 0.19 T \text{ meV}$$
 (4)

which is shown as the orange line in Fig. 1a.. Independently of this work, Airapetyants¹⁶ proposes a similar formula:

$$\Delta_{\rm v}(T) \sim 180 - 0.2 \ T \ {\rm meV},$$
 (4a)

This is not shown in Fig. 1a for clarity due to its similarity with Eq. 4. If Eq. 4 holds, T_0 ~800K and the band crossing does not play a role in the *zT* of *p*-type PbTe at operating temperatures, although the concentration of thermally-induced or chemically-placed holes in the LVB is still important at *T*>600 K.

2. In contrast, Refs. [1,18,21,26,27,28] report, for *T* > 80K:

$$\Delta_{\rm v}(T) \sim 170 - 0.41 \ T \ {\rm meV}$$
 (5),

which is shown as the blue line in Fig. 1a.. If Eq. 5 holds $T_0 \sim 415$ K, and PbTe is an indirect-gap semiconductor at the relevant temperatures. Coincidentally, in this case d $_g/dT|_{T<420}$ K $\sim d\Delta_v/dT$, which would pin the LVB to the CB.²⁶ There is no physical reason this should be so: $_g$ is dominated by spin-orbit interactions, and is of very different physical origin than the LVB structure. In contrast to that, Eq. 4 gives $d\Delta_v/dT \sim \frac{1}{2} d_g/dT|_{T<420}$ K, which pins the *T*-dependence of the LVB to the mid-gap at L-point. Under those circumstances, the physical origin of the *T*-dependence of the direct gap $_g$ is clearly dissociated from that of Δ_v . We now critique the existing experimental literature cited in support of either Eq. 4 or Eq. 5.

Compilations in Landolt-Börnstein¹³ and Ravich's monograph¹⁸ are consistent with Eq. 5 and $T_0 \sim 420$ K or at least $T_0 \leq 480$ K. The experimental evidence appears strong, because it consists of two independent measurements: optical measurements by Tauber et al.²⁶ and a strongly *T*-dependent Hall coefficient R_H which was confirmed experimentally by numerous authors, including co-authors of the present paper.³ We address the two types of measurements.

In Ref. [26] high-temperature optical absorption data versus wavelength were extrapolated to zero, and these intercepts were interpreted as giving a net optical energy gap. This gap then appears to be T dependent following Eq. 1 up to about 400 K, above which the T-

dependence appears to vanish:²⁹ so for $T_0 \sim 400$ K, (d $_g/dT$)_{*T*>*T0*} = 0, while (d $_g/dT$)_{*T*<*T0*} = 0.43±0.02 meV K⁻¹, supporting Eq. 5. This interpretation was influenced by the behavior of $R_{H}(T)$ near the same temperature, which is discussed below. As already discussed in Ref [12], because of (1) Fermi surface smearing, (2) the effect of indirect (L- Σ) optical transitions which require electron-phonon interactions, (3) the possible existence of Urbach tails, and (4) the strong and temperature-dependent non-parabolicity of the bands (Eqs. 1 & 2), the interpretation of the high-temperature energy-dependent optical absorption curves becomes overly sensitive to both the energy range selected for the extrapolations and the assumptions made in the model used. To illustrate the point further, the optical absorption spectra of PbTe, PbSe and PbS all three show a similar temperature-dependence,³⁰ yet only in PbTe are the results interpreted as above, and even then several different interpretations can be given to the same data, as can be seen by comparing Refs. [30] and [31] . Ekuma et al,³² and several older discussions of the optical data^{19,33} show them compatible with Eq. 4, even if the older discussion may be prone to larger uncertainties.

Next, we review the literature on the temperature-dependent Hall coefficient $R_H(T)$. The non-monotonic dependence of $R_H(T)$ in Na-doped PbTe has been the object of numerous investigations, some¹⁸ of which concluding that the maximum in $R_H(T)$ near $T \sim 415$ K identifies the LVB and UVB band crossing ($\Delta_v=0$) temperature (Eq. 5), although this was not generally accepted in the old literature.²² Here we first point out that the literature used for $R_H(T)$ does not specify what field the transverse Hall resistivities were measured at, only that it is fixed positive and negative fields. The field dependence is studied experimentally here. High-temperature measurements on heavily *p*-type doped material yield a negative slope to the field-dependent transverse (Hall) resistivity for B < 0.5T yet a positive slope for B > 1.5T. This provides evidence for the existence of an additional electron pocket that was ignored in the older analyzes.

The data^{21,27} used to conclude that $T_0 \le 480$ K show that $R_H(T > 400-550$ K) < 0, which is attributed to the presence of intrinsic electrons thermally excited across the direct gap g. The authors of Ref [21] do note that "the anomalously low temperature at which intrinsic conductivity appears is worthy of note", because, when such elemental calculations are repeated (see below), the influence of intrinsic electrons thermally excited across the gap only appears at temperatures much in excess of 500K.

Secondly, the Hall coefficient R_H is related to the carrier concentration p through the Hall prefactor r_H in the relation $R_H = r_H / p.q$. The Hall prefactor itself is the product of two factors, the first (≤ 1) due to the anisotropy of the Fermi surface, and the second (which can be >1) to scattering. The prefactor r_H was not taken into account in the discussion of the evidence for Eq. 5, even though it had been determined experimentally³⁴ for group IV-VI rocksalt semiconductors: r_H is shown to vary between 0.6 (in heavily p-type SnTe) and 1.7 (see below), and to have a very strong and non-monotonic dependence on doping level. A discussion of $r_H(p,T)$ is given by Allgaier.²⁴ Thirdly, we note that the maximum in $R_H(T)$ in a 2-band system occurs when the conductances of the two bands are equal, not when $\Delta_v = 0$; these conditions are not equivalent, because the LVB and UVB conductances are functions of both carrier densities and mobilities, which can be expected to be very different in the two bands.

Finally, Airapetyants & al.¹⁶ discusses the hole concentration dependence of the thermopower S(p,T) (called a Pisarenko relation) at 300K, alongside that of $R_H(p,T)$, and a summary of how he uses this to derive Eq.4a is given next. He assumes r_H =1 to calculate the hole concentration value from Hall data, whereas experimentally²³ 1.1 < r_H < 1.7 at these concentrations, so that his concentrations are underestimated by 10 to 70%. Below his value of 3-4x10¹⁹ cm⁻³ holes, the data S(p, T=300K) match a Pisarenko relation $S_{UVB}(p_{UVB})$ calculated

from the known band parameters of the UVB, but above that critical concentration, the experimental values for S(p, T=300K) saturate at about 50-55 μ V/K. An analysis of the value of the thermopower $S_{UVB}(p_{UVB})$ at 300 K, using the band structure and equations in Ref. [18] and assuming acoustic phonon scattering, yields $S_{UVB}(p_{UVB} = 2 \times 10^{19} \text{ cm}^{-3}, 300 \text{ K}) \sim 120 \mu \text{V/K}$ and $S_{UVB}(p_{UVB} = 5 \times 10^{19} \text{ cm}^{-3}, 300 \text{ K}) \sim 50-55 \ \mu\text{V/K}$. Above the doping level where the LVB starts getting filled, the thermopower does not continue on the Pisarenko relation $S_{UVB}(p_{UVB})$ of the UVB, but exceeds it, because S(p) progressively moves from the Pisarenko relation of the UVB to that $S_{LVB}(p_{LVB})$ of the LVB, which lies higher $(S_{LVB}(p) > S_{UVB}(p))$ because the DOS of the LVB is larger than that of the UVB. This makes the thermopower S(p) appear to saturate,¹⁶ and the saturation value of the thermopower gives the hole concentration in the UVB when the Fermi energy is at the crossing point between UVB and LVB. The observed^{16,18} saturation thermopower at 300 K around 50-55 μ V/K therefore corresponds to a UVB that holds $p=5 \times 10^{19} \text{ cm}^{-3}$ holes when the Fermi energy reaches the LVB. Assuming the temperaturedependence of Eq. 4, and using the values of Ref [17] for the UVB, it is calculated to hold 5.2×10^{19} cm⁻³ holes at 300K when the Fermi level reaches the LVB: Eq. 4 agrees with the data of Airapetyants & al..¹⁶ In contrast, with the temperature-dependence of Eq. 5, the UVB can only hold 1.9×10^{19} cm⁻³ holes at 300K before the E_F reaches the top of the LVB (Fig. 1b): Eq. 5 does not agree with the data of Airapetyants & al.¹⁶ This calculation is also similar to that of Ref. [12], but the values for S in that reference are twice higher because they are calculated without taking the effect of electron scattering into account.

Experimental

Transport data are taken on $Pb_{0.99}Na_{0.01}Te_{1-x}S_x$ alloys with x=0, 0.04, 0.08, 0.12 and $Pb_{1-y}Na_yTe_{0.92}S_{0.08}$ with y=0.01, 0.005, 0.0025. A molar concentration of 0.01 in PbTe corresponds

to $\sim 1.5 \times 10^{20}$ cm⁻³ atoms. Stoichiometric amounts of starting elements were loaded into carbon coated quartz ampoules, which were then sealed under high vacuum. After heating the ampoules to 1373K, they were annealed at 1100K, thus generating single phase material.³⁵ Powder X-ray diffraction confirms the single phase nature and the lattice constant decreases linearly with increasing sulfur content in accordance with Vegard's law.

In an orthonormal set of axes [x,y,z], we apply a magnetic field B_z , and current of heat fluxes along x. Measurements are made of voltages along either x (longitudinal effects) or y (transverse effects), and temperature gradients along x. We measure electrical resistivity $\rho = \rho_{xx}$ and transverse Hall resistivity $\rho_{xy}(B_z)$ with a relative accuracy determined by the electrical noise in the signal of about $1 \times 10^{-8} \Omega m$ at a given temperature. The absolute accuracy of 3% on $\rho_{xy}(B_z)$ and 8% on $\rho_{xx}(B_z)$, are determined by the geometry of the sample. $\rho_{xy}(B_z)$ data yield the Hall coefficient $R_H = \lim_{B\to 0} (\frac{d\rho_{xy}(B_z)}{dB_z})$. We measure the thermopower or Seebeck coefficient $S = S_{xx}$ with an accuracy near 3%, and the transverse isothermal Nernst-Ettingshausen thermopower $S_{xy}(B_z)$ which yields the isothermal Nernst-Ettingshausen coefficient

$$N = \lim_{B \to 0} \left(\frac{dS_{xy}(B_z)}{dB_z} \right)$$
 with about 10⁻⁸ V K⁻¹T⁻¹ accuracy. This is done over 77K $\leq T \leq$

620K in a conventional flow cryostat in stepped temperature and magnetic field increments (-1.4T $\leq B \leq \pm 1.4$ T, and in some instances ± 1.8 T). The low-field limit for the data is defined by the condition that $\mu B < 1$, here in practice |B| < 0.5T. The measured coefficient is the adiabatic Nernst coefficient, the isothermal coefficient *N* is calculated applying the standard correction.³⁶ The error analysis in these measurements is similar to that in Ref. [36]. Measurements are extended to 2K in a Quantum Design Physical Properties Measurement System (PPMS) in the AC Transport and Thermal Transport options (TTO). In this temperature range, the error in *S* is increased to over 5%, due to the size of the thermometry of the TTO and unquantifiable uncertainties in the software analysis. The high temperature thermal conductivity $\kappa(T>300 \text{ K})$ is deduced from a measurements of the thermal diffusivity, α , in an Anter Corporation Flashline 3000, using the equation $\kappa = C_p \alpha \rho_d$ where ρ_d is the mass density and C_p is the material specific heat. ρ_d is measured at 300K using Archimedes's method and is adjusted over temperature for the literature values of the thermal expansion coefficient of PbTe. We measure C_p in the Heat Capacity Option in the PPMS (an absolute measurement) from 2K to 300 K, and then from 300 K to 800 K in a Netzsch DSC 404-C at 20 °C min⁻¹ under flowing Ar (<3% error) at Oak Ridge National Laboratory (ORNL). We renormalize the Netzsch C_p -values over their entire temperature range by the ratio of the values obtained on both instruments at 300K; this correction is less than 1%. High temperature *S* (static DC method, <5% error) and ρ (4-wire method, 6% error) are also measured at ORNL in a ULVAC ZEM-2 under a static 0.09 MPa He exchange gas pressure, and using two R-type thermocouples and Ni electrodes.

Results

Traces of $\rho_{xy}(B_z)$ on Pb_{0.99}Na_{0.01}Te_{1-x}S_x at a few selected temperatures are shown in Fig 2. At $T \le 400$ K, $\rho_{xy}(B_z)$ is linear with $d\rho_{xy}(B_z)/dB_z > 0$, but at $T \ge 600$ K, the low field $d\rho_{xy}(B_z)/dB_z$ is negative (x=0.08) or has a plateau, while the high-field $d\rho_{xy}(B_z)/dB_z$ is positive, and resumes a slope quite similar to that observed throughout the field range at $T \le 400$ K. This behavior will be shown in the next section to be evidence for the appearance of a high-mobility electron in these samples at $T \ge 600$ K. Taking the value of $\rho_{xy}(B_z)$ at two fixed values of B_z of opposite polarity, to deduce a pseudo-Hall coefficient R'_H defined as $R'_H \equiv \frac{\left[\rho_{xy}(B_z) - \rho_{xy}(-B_z)\right]}{2B_z}$, is deriving a

quantity that is not directly related to the carrier concentration because $p \neq \frac{1}{R'_{H}}e$. The

temperature dependence of R'_H is what was reported in the old literature and used as evidence for the bands crossing at $T_0 \sim 415$ K.

We show the galvanomagnetic properties from 80-420K in Fig. 3 on the samples with y=0.01. There is a plateau to R_H (Fig. 3a) for all samples at T< 100K: at 80K, taking r_H =0.65,^{22,23} $p = r_H/R_Hq = 1.4 \times 10^{20}$ cm⁻³ for x=0.08, which corresponds yields a doping efficiency of nearly unity (one Na atoms gives one hole). This varies slightly for the other samples with different x. Because $p > 1 \times 10^{20}$ cm⁻³, one can calculate from the values for the UVB¹⁷ that ε_F for all samples is in the LVB at all temperatures (Fig 1b). From 100K-460K, R_H increases strongly, and it can be seen from Fig 2 that R_H turns negative at some T > 460 K; this trend is a similar to earlier reports,^{21,37} and ascribed to the presence of additional electrons. This is in sharp contrast to the resonant PbTe-PbS:Tl^{2,4} alloys where R_H has almost no temperature dependence at the measured temperatures ≤ 420 K. There is no systematic trend in the magnitude of R_H with x: this most likely arises from unavoidable small variations in sample preparation due to the high reactivity of sodium and the necessity to anneal at 1100K to achieve single phase material; Crocker experienced similar issues when annealing at $1073K^{23}$. Because of this, we do not compare absolute values of ρ (Fig. 3b) or Hall mobility (μ_H) as function of x and p. The drop in ρ of the sample with x=0 & 0.04 at ~60K, which is absent in the others, is mentioned without explanation.

Moving to the thermomagnetic properties also shown in Fig. 3, we notice that the Nernst coefficient N (Fig. 3c) is large for a heavily doped semiconductor. Rogers³⁸ also reports on N for a heavily Na-doped sample with similar 77K R_H value: his temperature dependence is similar to that in Fig. 3, but the magnitude roughly double; because it is not reported if that measurement concerns the isothermal or the adiabatic coefficient, which would explain the difference, no

further analysis of the discrepancy is possible. The thermopower *S* (Fig. 3 d) is ~53 μ V/K at 300 K for all samples: this is the saturated value reported in Ref. [16] for all Na-doped samples where the ε_F presumably falls in the LVB. Below 140-120 K, *S* changes from positive to negative in sulfur-containing samples (*x*>0), even though these are degenerately *p*-type doped materials.

An investigation of the low-temperature S and its magnetic field dependence (longitudinal field B parallel to heat flux) is shown in Fig. 4. For pure PbTe (x=0, Fig 4a), S is positive at all temperature at 0 or 7T. A maximum of 18 µV/K is reached at 40K in zero field, most likely due to phonon drag. For x=0.04, Fig. 4b reports that S < 0 at T < 130K, with a small positive hump at 50K, but shows that with increasing magnetic field the temperature range over which S < 0 progressively shrinks, so that the 7T curve qualitatively looks like the curves for x=0. This can be viewed as a positive phonon-drag peak superimposed to a magnetic-field dependent diffusion thermopower that varies from negative at 0T for T < 130K to positive at 7T. For x=0.12, there is a region with S negative at all fields, and the phonon-drag peak appears strongly reduced or absent. A further study of low temperature S(B,T), N, and R_H of Pb_{1-v}Na_vTe_{0.92}S_{0.08} doped with y=0.01, 0.005, 0.0025 is carried out in Fig. 5. This places $\varepsilon_{\rm F}$ at different locations in the valence band, with [Na] and hole concentrations p. Table 1 shows the sodium concentration [Na], the saturated value of low-T Hall coefficient from Fig. 4a, and the calculated hole concentration p using r_H from Ref [23], illustrating that Na is a monovalent acceptor within the experimental error. Fig 5b shows the isothermal Nernst coefficient N, which has a minimum that increases in T with [Na]. Fig 5c shows that the T-range where S(H=0, T) < 0 is not a monotonic function of [Na]. As above, the effect of the magnetic field is to suppress the range where S < 0.

These samples match, within stated errors, the record values^{1,3} reported for zT in the ptype lead salt systems, as shown in Fig. 6. Here the measurements of *S*, ρ , κ , and zT on samples with x=0.08 & 0.12 are extended to higher temperatures. The two samples have similar *S*, which for 300K < T < 600K are linear with respect to temperature, then peak at ~280µV/K at 650K; as is usual for PbTe.¹⁶ ρ remains low, with a maximum of ~4.5 mΩ-cm for x=0.12 at 700K. Thermal conductivity shows a monatomic decrease with increasing sulfur content, presumably due to the reduction in lattice thermal conductivity. zT reaches a value of ~1.55 at 700K and remains above 1 at T>500K. The zT(T) from an Pb₉₇Tl₂Na₁Te₉₂S₈ alloy is also shown for comparison in Fig 6d, illustrating that its *T*-dependence is different from that of Tl-free PbTe alloys. This, the difference in $R_H(T)$, and the increase above the Pisarenko relation, demonstrate that the high zT in Tl-containing and Tl-free PbTe alloys have different physical origins.²

Analysis

The simultaneous presence in these degenerately *p*-type doped semiconductors of a negative thermopower at the low temperature end of the measurement range, and a negative Hall coefficient at the high-temperature end is, in our view, coincidental. From its magnetic field dependence, we conclude that the low-*T* thermopower is due to the energy dependence of the hole scattering mechanism. Indeed, S(B=0,T) is a function of the energy-dependence of both band structure and scattering mechanisms. The latter is parameterized by writing the energy dependence of the relaxation time $\tau(\varepsilon)$ as a function of a scattering exponent λ defined generally by:

$$\lambda \equiv \frac{\partial \ln(\tau(\varepsilon))}{\partial \ln(\varepsilon)}\Big|_{\varepsilon=\varepsilon_{F}}$$
(6).

The scattering exponent is indicative of the electron scattering mechanism ($\lambda = -1/2$ for acoustic phonon scattering, $\lambda = +1/2$ for polar optical phonon scattering above the Debye temperature, and $\lambda = 3/2$ for ionized impurity scattering). In contrast, the value of $\lim_{B\to\infty}(S(B,T))$ is scattering-independent and a function of the band structure only.¹⁸ Therefore in the regime where S(B=0,T) < 0, the fact that $S(B >> 1/\mu_{H_b}, T) > 0$ suggests that the dominant charge carrier are still holes. From this, and the fact that S(B=0,T) < 0, one concludes that the contribution of λ to S(B=0,T) must be important enough to change its sign. To illustrate how λ can induce a sign reversal of *S*, one can write the expressions for the diffusion thermopower (excluding phonondrag) for degenerately-doped semiconductors with parabolic dispersions (such as the LVB of PbTe) as:¹⁸

$$S = \frac{2\pi^{3/2}}{3^{5/2}} \frac{k_B}{e} k_B T \frac{m_{DOS}^*}{\hbar^2 p^{3/2}} (\lambda + \frac{3}{2})$$
(7a),

whereas for a non-parabolic bands (such as the UVB of PbTe):¹⁸

$$S = \frac{\pi^2}{3} \frac{k_B}{e} k_B T \frac{1}{\varepsilon_F} \left(\lambda + \frac{3}{2} - \frac{2^{\varepsilon_F} \varepsilon_G}{1 + 2^{\varepsilon_F} \varepsilon_G} + \frac{3}{2} \frac{\varepsilon_F}{1 + \varepsilon_F} \right)$$
(7b).

In both cases, while for $\lambda=0$ (the high-field limit) the sign of *S* will be that of the polarity of the charge carrier, when $\lambda < 0$ a sign reversal of the diffusion thermopower can be produced by the scattering mechanism. Because it is unlikely that acoustic phonon scattering ($\lambda = -1/2$) flips the sign of *S*, we propose that the negative thermopower is due to interband (L- Σ) scattering, since both valleys are occupied. In single-carrier systems, it is possible to deduce λ using the method of the four coefficients,³⁶ but here this approach applies only to the sample with *y*=0.0025 at low *T*.

The four-coefficients³⁶ fits to the *y*=0.0025 sample using the method of Ref [36] give $\varepsilon_{\rm F}$ (80K) = 83 meV and $m^*_{DOS} = 0.22$ m_e, both values consistent with the reported values for the UVB¹⁷ and the measured hole density. They indicate that the scattering parameter (λ) increases

monotonically with *T*. The values obtained at 80K for $\lambda \sim -0.6$ are too negative for acoustic phonon scattering, suggesting a contribution from interband scattering. λ becomes positive above 200K, indicating mostly acoustic phonon scattering with some optical phonon contribution. These concepts explain not only the observed magnetic field measurements, but also the quadratic dependence of *S* on temperature *T*, a unique feature in degenerately-doped semiconductors where usually $S \propto T'$. The usual T' dependence is to be multiplied by an additional *T*-dependence that originates from the increase of λ with *T*.

Besides the diffusion thermopower, phonon-drag acts on the majority carrier. Because we observe it to always have a positive contribution to the overall thermopower, this is another indication that the negative *S* does not arise from the presence of an unidentified electron pocket, reinforcing the conclusion that it must be related to a scattering effect. From Figs. 4 and 5, one can deduce that phonon-drag decreases with increasing sulfur content; this increases phonon-alloy scattering, which competes with phonon-electron interactions that give rise to phonon-drag. This explains why *S* never reaches negative values in heavily-doped binary PbTe: the negative diffusion thermopower is likely also present, but is masked by the large phonon-drag thermopower. Sulfur alloying may be needed to depress phonon-drag sufficiently to reveal the negative *S*.

Considering next the magnetic field dependence of the Hall conductance $\rho_{xy}(B_z)$, Fig. 2, the fact that the high-temperature curves have two distinct slopes reveals the simultaneous presence of two carriers of opposite polarities, electrons and holes. In two-carrier systems (the electron concentration and mobility are denoted *n* and μ_e ; the hole concentration and mobility *p* and μ_p) and in the presence of a transverse magnetic field B_z along the *z*-axis, the longitudinal electrical resistivity $\rho_{xx}(B_z)$ and transverse Hall resistivity $\rho_{yx}(B_z)$ are the diagonal and offdiagonal elements of the equivalent conductivity elements $\sigma_{xx}(B_z)$ and $\sigma_{xy}(B_z)$, given as function of the parameters above by:³⁹

$$\sigma_{xx} = \frac{ne\mu_e}{1 + \mu_e^2 B_z^2} + \frac{pe\mu_h}{1 + \mu_h^2 B_z^2}, \sigma_{xy} = \frac{ne\mu_e^2 B_z}{1 + \mu_e^2 B_z^2} + \frac{pe\mu_h^2 B_z}{1 + \mu_e^2 B_z^2}$$

$$\rho_{xx} = \frac{\sigma_{xx}}{\sigma_{xx}^2 + \sigma_{xy}^2}, \rho_{xy} = \frac{\sigma_{xy}}{\sigma_{xx}^2 + \sigma_{xy}^2}$$
(8)

At an arbitrary field, $\rho_{yx}(B_z)$ is no longer a linear function of B_z , and $d\rho_{yx}/dB_z$ can change sign with B_z . The low-field Hall coefficient R_H is the low-field slope of $\rho_{yx}(B_z)$:

$$R_{H} \equiv \lim_{B_{z} \to 0} \left(\frac{d\rho_{xy}}{dB_{z}} \right) = \frac{-n\mu_{e}^{2} + p\mu_{h}^{2}}{n\mu_{e} + p\mu_{h}}$$
(9),

while the high-field limit, Eq. (9) gives:

$$\lim_{B_z \to \infty} \left(\frac{d\rho_{xy}}{dB_z}\right) = \frac{1}{p-n}$$
(10).

From (10) and (11), we conclude that the low-field Hall resistivity $\rho_{yx}(B_z)$ is most sensitive to the presence of high-mobility minority carriers, and more so than the thermopower, which is given by $S = \frac{-S_e n\mu_e + S_h p\mu_h}{n\mu_e + p\mu_h}$, where S_e and S_p for electrons and holes respectively. In contrast, the

high-field $\rho_{yx}(B_z)$ measures the net carrier concentration, and is representative of the majority carrier concentration when $p \gg n$. Fig. 2 is therefore characteristic of a semiconductor that is dominantly p-type, but in which a low concentration of high-mobility electrons appears around 600K. One cannot deduce from Fig. 2 what electrons these are. Andreev²¹ makes similar observations and reaches the same conclusion, but suggests that the high-mobility minority electrons could be thermally activated in the L-point CB, although he expresses doubts about his own suggestion (see introduction). Fig. 7a reports the results for the calculation of the density of extrinsic holes (x-axis) and thermally activated electrons (y-axis) in PbTe using Fermi statistics⁴⁰ the temperature-dependent $m_{DOS}(p,T)$, $\varepsilon_g(T)$, and $\varepsilon_F(p)$ band structure parameters of PbTe,¹⁷ and the non-parabolicity of the dispersion relations.³⁶ The hatched region gives the doping range of our samples: at 600K, no more than 10^{16} cm⁻³ electrons can be present in the L-point CB, while the low-field negative slope in Fig. 2a corresponds to the mid- 10^{19} cm⁻³ electron concentration range. This discrepancy by over 3 orders of magnitude suggests that these electrons are not simply thermally activated across ε_g into the L-point CB, the discrepancy noted by Andreev.²¹

We suggest that the electron-like behavior is of topological origin, and arises from the regions in k-space where the pieces of the hole Fermi surfaces centered around L and Σ points touch. The hole Fermi surface of low-doped p-type PbTe is topologically multiply connected, consisting of 4 ellipsoid-like pockets centered at L-point, and 8 centered at Σ points. When the doping level is increased, each of these pockets increases in volume until they touch: now the entire Fermi surface is singly connected. This situation, illustrated in Fig. 7b, shows that "necks" (regions of inverted curvature) are induced by this change in topology of the band structure at the points where the different pockets touch. These regions will have a curvature of opposite polarity of the majority holes (L-W axis, Fig 1 of Ref. [12]); in response to a force induced by a potential or temperature gradient, they will respond as electrons. This latter effect is well known in the Fermi surface of copper, which is mostly a sphere. Because the Fermi sphere of copper is just a little too large to fit inside the first Brillouin zone, "necks" develop at the points where the spheres of adjacent Brillouin zones touch. Although each copper atom gives one electron to the solid, the necks have transport properties of holes. The positive thermopower of copper is explained by the presence of these topological holes. The samples in this study are sufficiently doped that ε_F falls below the maximum of the LVB (see Fig.7b), so that the topological electrons

do not exist at low temperature. When the temperature is increased sufficiently that the region of negative curvature comes within k_BT from ε_F , electron-like transport properties arise. As explained above, the fact that the Hall coefficient turns negative and the thermopower remains positive is an indication that the topological electrons have a high mobility.^{Error! Bookmark not defined.}

An alteration in properties due to the formation of an impurity band, possibly because of an inhomogeneous distribution throughout the samples of either S or Na atoms, is also possible. It is less likely, however, because such band is unlikely to have a mobility higher than that of the holes near the Σ points.

Summary and conclusions

The results presented here are on material with *zT* values reaching ~1.55 at 700K. We demonstrate that the temperature dependence of the Hall coefficient of heavily p-type doped PbTe, which was used as evidence in the literature for a crossing of two valence bands in PbTe at ~415K, is due to another cause, namely the appearance of an additional electron contribution. Complete magnetic-field dependent Hall-resistivity data must be taken to derive the proper carrier concentrations. The previously suggested temperature induced rapid rise in energy of the "heavy" hole LVB relative to the "light" hole UVB⁶ is not supported by the present experimental data. The temperature dependence posited by Allgaier and Houston²⁵ and Airapetyants¹⁶ is more plausible, and the energy overlap between LVB and UVB has a temperature dependence of d $\Delta_v/dT = -0.2$ (meV K⁻¹) which is about one half of the temperature dependence of the direct energy gap d $\Delta_v/dT \sim \frac{1}{2} d_g/dT|_{T<420 \text{ K}}$. Combining this with the low-temperature ($T \le 80$ K) value of $\Delta_v = 160-170$ meV from the literature compiled by Ravich et al.¹⁸ leads to the further conclusion that PbTe continues to be a direct gap semiconductor at temperatures where the *zT* and $S^2 \sigma$ of *p*-type PbTe are optimal e.g. 700-900K. This does not detract from the fact that the

heavy holes contribute dominantly to the zT of lead-salt-based semiconductors doped in the high-10¹⁹ cm⁻³ range.

The low temperature diffusion thermopower is negative below about 130 K, and magnetothermopower measurements show that that this is due to scattering, possibly intraband scattering between the LVB and UVB. The qualitative differences between the temperature dependences of the zT(T) of materials where the electronic properties are dominated by the heavy valence band, and the zT(T) of Tl-doped PbTe are also shown.

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Table 1					
Nominal Na	[Na]	R_H	$r_H(\text{Ref.23})$	$p=r_H/qR_H$	$R_{H}(300)/R_{H}(80)$
concentration	$(10^{19} \mathrm{cm}^{-3})$	$(10^7 \Omega m T^{-1})$		$(10^{19} \mathrm{cm}^{-3})$	
0.25%	3.72	1.96	1.3	4.1	1.7
0.50%	7.44	0.52	0.65	7.8	2.3
1.00%	14.88	0.29	0.65	14	2.1

Figure Captions

Figure 1. (color online) (a) Schematic temperature-dependences of the relative positions of the different bands of PbTe. The bands are the conduction (CB) and upper valence bands (UVB) at L-point, and a lower valence band (LVB) in a pocket near the Σ -points. Two temperature dependences are proposed in the literature for the relative positions of the UVB and LVB, which are separated by $\Delta \varepsilon_v$. The latter is given by either Eq. (4) or Eq. (5), see text. (b) Calculated density of holes in the UVB for the proposed temperature dependences of $\Delta \varepsilon_v(T)$ (insert, see text).

Figure 2. (color online) Hall resistivity $\rho_{xy}(B_z)$ as a function of magnetic field B_z for Pb_{0.99}Na_{0.01}Te_{1-x}S_x at selected temperatures.

Figure 3. (color online) Low temperature electrical properties of 1% Na doped samples from 2-420 K: (a) Hall coefficient, (b) isothermal Nernst coefficient, (c) electrical resistivity, and (d) Seebeck coefficient. The symbols are experimental points; the lines are added to guide the eye. **Figure 4.** (color online) Thermopower from 2-240K for Pb_{0.99}Na_{0.01}Te_{1-x}S_x samples with nominal concentration of [Na] = 1 at.%. (a) depicts data for sulfur concentrations x=0 & x=0.12 in 0 & 7T field; (b) depicts data for x=0.04 in 0, 3.5T & 7T field. The symbols are experimental points; the lines are added to guide the eye.

Figure 5. (color online) Low temperature electrical properties of $Pb_{1-[Na]}Na_{[Na]}Te_{1-x}S_{x}$, *x*=0.08 samples with different sodium concentrations [Na] indicated. (a) Hall coefficient (b) isothermal Nernst coefficient (c) thermopower in 0 and 7T longitudinal external magnetic field. The symbols are experimental points; the lines are added to guide the eye.

Figure 6. (color online) (a) Thermopower, (b) electrical resistivity, (c) thermal conductivity and (d) zT for Pb_{0.99}Na_{0.01}Te_{1-x}S_x samples with [Na] = 1%. The symbols are experimental points; the

lines are added to guide the eye. The added zT data is on a sample from Ref. [4] and contains the resonant impurity thallium.

Figure 7. (color online) (a) Calculated thermally excited electron concentration at various temperature as a function of hole carrier concentration. The grey band indicates the range of extrinsic hole densities in samples studied; (b) The valence band of PbTe alloys contains multiple pockets when the samples are doped lightly p-type: the hole Fermi surface is then multiply connected. When the Fermi level ε_F is lowered by doping to the level shown here, these multiple pockets merge into one singly-connected complex Fermi surface. The green band represents the width at half maximum of the energy derivative $df/d\varepsilon$ of the Fermi distribution function *f* at 600 K. When this band encompasses the regions in the dispersion relation $\varepsilon(\mathbf{k})$ where the upper valence bands, near the L-points of the Brillouin zone, touch the lower valence bands centered near the Σ -points, thermally excited carriers (orange) appear in parts of the Brillouin zone where the dispersion $\varepsilon(\mathbf{k})$ has curvature of polarity opposite to that of the holes. The thermally excited carriers in that orange region behave like electrons under the influence of an external electric field.

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