

CHCRUS

This is the accepted manuscript made available via CHORUS. The article has been published as:

Critical behavior of ferromagnetic perovskite ruthenates

J.-G. Cheng, J.-S. Zhou, J. B. Goodenough, and C.-Q. Jin Phys. Rev. B **85**, 184430 — Published 29 May 2012 DOI: 10.1103/PhysRevB.85.184430

Critical behavior of ferromagnetic perovskite ruthenates

J.-G. Cheng¹, J.-S. Zhou^{1*}, J. B. Goodenough¹, and C.-Q. Jin²

¹ The Materials Science Program/Mechanical Engineering, University of Texas at Austin, Austin, Texas 78712, USA

² Institute of Physics, Chinese Academy of Science, Beijing 100080, China

Stoner-Wohlfarth itinerant-electron ferromagnetism has been believed to be applicable to the metallic, ferromagnetic perovskite SrRuO₃. A key evidence is the linearity of the Arrott plot M^2 versus H/M obtained in a previous study. However, a typical Heisenberg behavior observed in isostructural BaRuO₃ raises a question how the small structural change between them can place these compounds into two extremes of magnetic interactions. For clarification of this issue, we carried out magnetization measurements in the vicinity of T_c for the whole series of ferromagnetic ruthenates from Ca_{0.5}Sr_{0.5}RuO₃ to SrRuO₃ to BaRuO₃. Moreover, we have studied the effects of the A-site size variance and hydrostatic versus non-hydrostatic pressure on the Arrott plot of SrRuO₃. Our results reveal that the magnetization behavior as presented by the Arrott plot is extremely sensitive to structural distortions. The linearity of the Arrott plot for $SrRuO_3$ is actually part of a continuous curvature evolution as lattice distortions and spin-orbit coupling change; it may have little to do with Stoner-Wohlfarth itinerant-electron ferromagnetism. The data for these ruthenates points to a distinct phase appearing between localizedelectron and broad-band behavior that exhibits a strong ferromagnetism stabilized by strong correlations among electrons of a narrow band.

PACS numbers: 75.30.-m, 75.40.-s, 75.10.Lp, 75.30.Et

I. Introduction.

Ferromagnetism has been described for two limiting cases, by Stoner-Wohlfarth itinerant-electron magnetism in metals and Heisenberg magnetism in either metals or insulators. The mean-field theory works well to describe magnetic properties in paramagnetic and magnetically ordered phases except in the vicinity of a magnetic transition where critical fluctuations become severe, especially for Heisenberg magnets. Exponents in the power law to describe critical fluctuations can be extracted from the Arrott plot¹ of the magnetization M(H) near T_c. Very weak itinerant-electron ferromagnetism² always shows parallel isothermal lines in the plot of M² versus H/M for both paramagnetic and ferromagnetically ordered phases whereas the Heisenberg ferromagnetism³ exhibits curved isotherms with a uniform concave curvature as is illustrated in Fig.1 for temperatures on crossing T_c. In the case of Heisenberg ferromagnetism, the linearity can be restored in a more general plot of $M^{1/\beta}$ versus $(H/M)^{1/\gamma}$ with $\beta < 0.5$ and $\gamma > 1$. Exponents can be further refined through iterations between the power-law fitting in $M_s(T)$, $\chi_0(T)$ obtained in Fig.1 and the modified Arrott plot with exponents from the power-law fitting. The self-consistent renormalization (SCR) model⁴⁵ has been put forward to account for ferromagnetism in metallic systems where the electron-electron correlations are too strong for Stoner-Wohlfarth itinerantelectron ferromagnetism. The latest development of this model by Takahashi⁶ can account explicitly for a deviation from lines to curves with a slightly concave curvature in an Arrott plot depending on the ratio of $\eta = T_c/T_0$, where T_c is the Curie temperature and T_0 is the characteristic temperature of spin fluctuations, which can be derived directly from neutron inelastic scattering or an NMR measurement. Linearity in the Arrott plot M²

versus H/M is obtained for a ferromagnet with small η as in ZrZn₂ having $\eta = 0.053$, for instance. The SCR model ⁶ has been applied successfully to explain the anomalous ferromagnetism in MnSi ($\eta \approx 0.13$), which shows curves with a slightly concave curvature in an Arrott plot. However, it is a challenge to apply the SCR model to explain the nearly linear parallel lines of the Arrott plot found in the M(H) of the perovskite SrRuO₃⁷ in which η is at least as large ⁸ as that of MnSi. The linearity of the Arrott plot and a large η appear contradictory to each other on the basis of current models. On the other hand, the close relationship between metallic conductivity and ferromagnetism found in the ultrathin film samples of SrRuO₃ seems to support the model of itinerantelectron ferromagnetism.⁹ As reported recently,¹⁰ the cubic BaRuO₃, which has essentially the same physical properties as that of SrRuO₃, exhibits in the Arrott plot isotherms with a uniform concave curvature typical of a Heisenberg ferromagnet. A much recuded T_c and the complete suppression of T_c under P = 8 GPa in the cubic BaRuO₃ indicate that the cubic BaRuO₃ has a much broader bandwidth than that in SrRuO₃. The bandwidth consideration is clearly not compatible with the argument of Stoner-Wohlfarth itinerant-electron ferromagnetism in SrRuO₃ and Heisenberg ferromagnetism in BaRuO₃. In order to clarify this issue, we have carried out a systematic study of the magnetization in the vicinity of T_c for all ferromagnetic members in the perovskite family of ARuO₃ with $A = Ca_{1-x}Sr_x$, $Sr_{1-y}Ba_y$. We have also fund that the critical behavior is extremely sensitive to lattice strains and local disorder due to either the size variance effect and non-hydrostatic component if high pressure is applied. We have argued that the strong ferromagnetism of the ARuO₃ perovskites is a distinguishable phase appearing at a crossover from Pauli paramagnetism to localized-electron behavior

in a 4d band where spin-orbit coupling is progressively enhanced as the bandwidth narrows. The interplay between spin-spin interactions and the strong spin-orbit coupling makes invalid the classic criterion for itinerant-electron ferromagnetism, viz. the linearity of the Arrott plot.

II. Experimental Methods

All samples in this study were made with starting materials ACO_3 (A = Ca, Sr, Ba) and Ru or RuO₂. The well-mixed starting materials were sintered at temperatures 800-1100°C in air with several intermediate grindings. This procedure produced a single phase with the orthorhombic *Pbnm* perovskite structure for all compositions of $Sr_{1-v}Ca_vRuO_3$ ($0 \le y$ \leq 1), but produced hexagonal polytypes, i.e. 4H, 6H, 9R, or mixed phases depending on x for the compositions of $Sr_{1-x}Ba_xRuO_3$ ($0 \le x \le 1$). While keeping the average A-cation size $\langle r_A \rangle$ constant, a size variance $\sigma^2 = \langle r_A^2 \rangle - \langle r_A \rangle^2$ was introduced into these samples by substituting (Ba_{.49}Ca_{.51}) for Sr. For example, $\langle r_A \rangle$ remains the same while σ^2 increases gradually with increasing z in the series $Sr_{1-z}(Ba_{49}Ca_{51})_z RuO_3$. Because the effective ionic radii of Ca^{2+} , Sr^{2+} , and Ba^{2+} depend sensitively on their coordination number, which decreases from XII for cubic BaRuO₃ to VIII for highly distorted orthorhombic CaRuO₃, we have chosen the effective ionic radius of $Ca^{2+}(VIII)$: 1.12 Å, $Sr^{2+}(X)$: 1.36 Å, and Ba²⁺ (XII): 1.61 Å for calculation of $\langle r_A \rangle$ and σ^2 in the present study. For those compositions containing Ba, high-pressure and high-temperature synthesis under 3-18 GPa and 1000°C is required in order to obtain a single phase with perovskite structure; it was carried out in a Walker multianvil module (Rockland Research). The X-ray powder diffraction was performed with a Philips X'pert diffractometer and the Rietveld

refinement on these diffraction spectra was made with FULLPROF. Magnetic properties of the ruthenates were measured in a SQUID magnetometer (Quantum Design). The pressure effect on the magnetization was carried out in a Be-Cu self-clamped cell fitting into a commercial SQUID magnetometer. We have used two kinds of fluids, 3M Fluorinert FC77 & FC72 and silicon oil, as the pressure medium. The sample's demagnetization factor is determined by the magnetization at low field. However, subtraction of the demagnetization field from the external field H_0 does not influence the output of the Arrott plot, so this correction has not been made in our plots.

III. Results and discussions

Since all compositions containing Ba need to be synthesized under high pressure and the high-pressure synthesis has an obvious influence on the magnetization even for samples in which the high-pressure synthesis is not necessary, we show in Fig.2 the Arrott plots for high-pressure ferromagnetic products of all the ruthenate perovskites in order to demonstrate the A-cation size effect. We start from A=Sr_{.5}Ca_{.5}, which has the smallest average A-cation size $\langle r_A \rangle$ in this study. The feature inside the dashed circle around the origin in this Arrott plot will be discussed in the last paragraph of the section; we first focus on isotherms outside the circle.

As illustrated in Fig.2, isotherms for temperatures on crossing T_c have a uniform convex curvature for the sample $Sr_5Ca_5RuO_3$; this behavior of the Arrott plot cannot be accounted for with existing models. The degree of convex curvature decreases gradually as $\langle r_A \rangle$ increases. Almost perfectly linear isotherms are found in the Arrott plot for

SrRuO₃. As $\langle r_A \rangle$ increases further by substituting Ba for Sr, isotherms in the Arrott plot are no longer uniform and the curvature changes at different values of H/M. Since isotherms with uniform concave curvature are found in BaRuO₃, which has the largest $\langle r_A \rangle$, it is reasonable to attribute a concave curvature at higher H/M to the effect due to enlarging $\langle r_A \rangle.$ On the other hand, Ba substitution introduces the A-cation size variance σ^2 $= \langle r_A^2 \rangle - \langle r_A \rangle^2$ as well as increasing $\langle r_A \rangle$. A convex curvature at low H/M, identified by an arrow in the A= Sr_{.7}Ba_{.3} sample, develops as σ^2 increases; but it totally disappears in BaRuO₃ with $\sigma^2 = 0$. The size-variance effect is not easily distinguished in the Ca substituted samples where the structural distortion due to small $\langle r_A \rangle$ appears to contribute an overall convex curvature for isotherms in the Arrott plot. It is likely that the size variance generally causes a convex curvature at low H/M. In order to isolate the size variance effect on the curvature of isotherms, we have taken isothermal M(H) curves for a series of samples $Sr_{1-z}(Ba_{49}Ca_{51})_z RuO_3$ where σ^2 increases with increasing z while $\langle r_A \rangle$ \approx r_{Sr} remains a constant for this series of samples. Increasing σ^2 reduces T_c and the saturation moment $M_s(5K)$ at 5 T; but more importantly, as shown in Fig.3(a,b,c), it converts nearly linear lines of the Arrott plot in the SrRuO₃ into isotherms with slightly convex curvature at H/M \approx 1000 Oe/emu for temperatures that cross T_c.

We have also found that the curvature of isotherms in the Arrott plot depends sensitively on how the sample is made. Taking SrRuO₃, for example, an overall convex curvature of the isotherms in the Arrott plot is found for a polycrystalline SrRuO₃ sample synthesized by standard solid-state reaction at ambient pressure (AP). For a melt-growth singlecrystal sample,⁷ the convex curvature in the isothermal lines of the Arrott plot is clearly

reduced, but it is still visible as seen in Fig.3(e). Linear lines can be seen only in the HP SrRuO₃ sample made under 4 GPa and at 1000°C. Moreover, the HP sample shows $T_c \approx$ 167 K and $M_s(5K) \approx 1.6 \,\mu_B/Ru$ at 5 T, the highest values ever reported. Different procedures for the sample's preparation may lead to a subtle difference in the crystal structure. In order to demonstrate the relationship between the magnetization near T_c and crystal structure, we have made a thorough structural study on these ruthenates. Ruthenates undergo Fig.4(a) structural transitions from the orthorhombic (*Pbnm*) phase to the orthorhombic (*Imma*) phase to the cubic (*Pm-3m*) phase as $\langle r_A \rangle$ increases. Fig.4(a) illustrates the phase diagram which includes the evolution of lattice parameters in these phases as $\langle r_A \rangle$. The difference of lattice parameters between AP and HP samples of SrRuO₃ is too small to be visualized in these phase diagram. Because of the intrinsic structural distortion in the *Pbnm* perovskite,¹¹ *a* and *b* cross as $\langle r_A \rangle$ increases; *a-b* peaks out inside the Pbnm phase near the Pbnm/Imma phase boundary. The subtle change of ab as a function of $\langle r_A \rangle$ in the *Pbnm* phase can be used to identify the difference between AP and HP samples. Fig.4(b) shows *a-b* versus $\langle r_A \rangle$ in a narrow range of $\langle r_A \rangle$ near A=Sr. The overall change of isotherms from a convex curvature to a concave curvature as a function of $\langle r_A \rangle$ from Fig.2 are color-coded in this plot. It is clear that the AP sample has smaller *a-b* than that in the HP sample, which makes this sample equivalent to a composition on the side of Ca substituted samples. As matter of fact, *a-b* of the single crystal sample of ref.7 is also slightly smaller than that of the HP sample. Therefore, linear isotherms of an Arrott plot exist for ruthenates only within an extremely narrow range of $\langle r_A \rangle$ in the *Pbnm* phase. The HP SrRuO₃ just happens to fall within this range.

After demonstrating the effects of chemical substitutions on the Arrott plot, it is interesting to compare them with the pressure effect since it provides a clean tuning on the bandwidth. Depending on the property of the pressure medium, however, some nonhydrostatic pressure component, typically sheer stress, is introduced at higher pressure. Hydrostatic pressure changes the cooperative octahedral-site rotations as well as shrinking the M-O bond length in an orthorhombic perovskite AMO₃. In contrast, the shear stress due to non-hydrostatic pressure results in some local bond-length disorder. We have studied these pressure effects in two samples, the HP SrRuO₃ and the HP Sr₇Ba₃RuO₃ up to 1 GPa. The top two panels (a) and (b) in Fig.5 show the results of the HP SrRuO₃ sample loaded in a high-pressure chamber filled with Fluorinert, which provides nearly hydrostatic pressure to about 1 GPa. Hydrostatic pressure does not cause any obvious changes in terms of the linear lines of the Arrott plot. In sharp contrast, nonhydrostatic pressure with silicone oil used as the pressure medium causes a clear convex curvature (indicated by an arrow in Fig.5(d)), which is similar to the effect due to Acation size variance in Fig.3. Isotherms at ambient pressure of the HP Sr₇Ba₃RuO₃ sample measured with the high-pressure cell are displayed in Fig.5(e); they are identical to those for the same sample measured with a straw in Fig.2. As mentioned in the analysis of Fig.2, complicated features of the Arrott plot in the HP Sr_{.7}Ba_{.3}RuO_{.3} sample are due to a combination of a convex curvature at low values of H/M due to the sizevariance effect and a concave curvature at high values of H/M due to a larger $\langle r_A \rangle$ in this Ba substituted sample. We have measured isotherms of the HP Sr 7Ba 3RuO3 sample near T_c under high pressure with silicone oil used as the pressure medium. As shown in Fig.5(f), isotherms of the sample under P = 1.1 GPa for the same temperature range do

not have interceptions at the vertical axis, which means that T_c reduces dramatically. Moreover, a convex curvature at about H/M = 1500 Oe g/emu is clearly enhanced. These data suggest that disturbing the easy magnetization axis due to local structural distortions is the source of the convex curvature.

We turn to the feature of isotherms inside the dashed circle of the Arrott plot in Fig.2, Fig.3, and Fig.5. Fig.6(a) is a zoom-in plot of the panel for A=Sr₅Ca₅ in Fig.2. At low fields, the local structural distortion and the A-cation size variance in this sample appear to curve isotherms so that no spontaneous M_s can be resolved at $T \leq T_c$ and also the inverse susceptibility χ^{-1} deduced from the intercept at M²=0 is much reduced from the intercept based on isotherms at higher H/M at $T > T_c$. These peculiar features of isotherms at low field are commonly seen in other ruthenate samples either with larger Acation size variance in Fig.3 or under non-hydrostatic pressure in Fig.5. Various types of site randomness alter long-range magnetic order in Heisenberg systems. Isotherms in the Arrott plot start to experience the influence of randomness at low fields. By taking into account a random uniaxial anisotropy, Aharony and Pytte¹² have predicted the behavior of isotherms at very low field for temperatures crossing T_c. Their prediction fits in principle what we have observed here. A characteristic of this type of randomness is to increase δ in H ~M^{δ} from δ < 3 (δ =3 is predicted by the mean-field theory) at T_c to δ > 3 at $T < T_c$ for a 3D Heisenberg system. An offset in the H/M axis for all isotherms in Fig. 6(a) is due to the demagnetization factor. Although isotherms at temperatures near T= 60 K cannot be fit with H $\sim M^{\delta}$ over a wide range on the H/M axis, the plot of logM versus logH in Fig.6(b) and their linear fitting indeed show a continuous increase of δ in

the vicinity of T_c with δ passing 3 eventually. These observations indicate that local lattice distortions and A-cation size variance result in a random uniaxial site anisotropy in the ruthenate perovskites.

In summary, the local disorders either from A-cation size variance or induced by nonhydrostatic pressure contribute complex features in an Arrott plot, which include nonparallel isotherms near the origin and a convex curvature near H/M ~ 1000 Oe g/emu as indicated by arrows in Fig.2, Fig.3, and Fig.5. After clarifying the effect of local structural disorder on the magnetization, it becomes more clear that the influence from the average A-cation size $\langle r_A \rangle$ on isotherms is to change a convex curvature to a linear line to a concave curvature at higher values of H/M as $\langle r_A \rangle$ increases.

Critical exponents for the ruthenates in Fig.2 have been obtained through iterations; an example for the HP SrRuO₃ sample is shown in Fig.7. We can find at least three classes of critical behavior, *i.e.* for a Heisenberg model ($\beta \sim 0.38$, $\gamma \sim 1.4$), for a mean-field model ($\beta \sim 0.5$, $\gamma \sim 1$), and for an unknown universality class ($\beta > 0.5$, γ which can't be defined well) in the plot of critical exponents versus $\langle r_A \rangle$ in Fig.8. It is also rare that critical exponents evolve continuously between these classes. Critical-behavior analysis normally serves as a window leading to the microscopic pictures of magnetic interactions from 3D Heisenberg, XY, and Ising models to itinerant-electron ferromagnetism. Does the ruthenate system offer a case in which the fundamental magnetic interaction can be continuously tuned by the bandwidth? In order to answer this question, we look into other ferromagnetic 4d and 5d oxides; their Arrott plots are shown in Fig.9. Although

isotherms in some Arrott plots look irregular (for example, those below the charge ordering temperature in 9R BaIrO₃ and above T_c in Ba₂NaOsO₃), critical-behavior analysis could be made within a very narrow temperature range near T_c. In general, straight lines and those with a convex curvature correspond to a $\beta \ge 0.5$, which corresponds to what is seen in the Ca substituted ruthenates. The mean-field model appears to be applicable to Ba₂NaOsO₆ and the post-perovskite CaIrO₃, ¹³ which are insulators. These results are certainly contradictory to Stoner-Wohlfarth itinerant-electron ferromagnetism. In addition to a generally higher β found in these 5d-metal-oxide ferromagnetic phase is approached at T > T_c are also highly unusual. As shown in Fig.10, the paramagnetic susceptibility of these 5d-metal oxides does not follow a Curie-Weiss law. A T_c less than a Weiss constant is always found in a regular ferromagnet. In contrast, all Weiss constants, if they can be obtained by extrapolating from the paramagnetic phase above T_c, are negative for the ferromagnets presented in Fig.9.

The unusual curvature and paramagnetic susceptibility are related to a strong spin-orbit coupling (SO) λ L·S that is on an equal footing with the spin-spin coupling in the 5d metal oxides. A strong SO interaction enhances both the single-ion anisotropy and the Dzyaloshinsky-Moriya interaction. The paramagnetic susceptibility of a system with a strong single-ion anisotropy energy and a Heisenberg spin-spin coupling normally does not follow a Curie-Weiss law.¹⁴ It is interesting to make a side-by-side comparison between the Ca substituted ruthenates and these 5d-oxide ferromagnets. Whereas the Ca

substituted samples, Sr_{.4}Ca_{.6}RuO₃, for example, share the same feature, *i.e.* T_c > Weiss constant, the transition region is much broader than that of these 5d oxide ferromagnets. Narrowing the bandwidth in the Ca substituted Sr_{1-x}Ca_xRuO₃ samples may lead to a phase segregation between strongly correlated and weakly correlated phases as suggested by Rozenberg *et al* ¹⁵ In the 4d and 5d metallic systems, an enhanced SO interaction alters the spin-spin coupling into clusters with strong correlations and clusters with J =0 would act to dilute a ferromagnetic spin-spin interaction in the matrix. As matter of fact, $\chi^{-1}(T)$ of Sr_{1-x}Ca_xRuO₃ closely resembles that typical of a Griffiths phase. The Griffiths phase is defined in the paramagnetic phase below a T_G where a $\chi^{-1}(T)$ deviates from the Curie-Weiss behavior and drops drastically as temperature lowers. The original concept was put forward by Griffith ¹⁶ to describe an Ising ferromagnetic system in which T_c is reduced due to the dilution of non-magnetic ions. The characteristic $\chi^{-1}(T)$ of a Griffiths phase has been found in other systems than an Ising magnet.¹⁷

The perovskite ruthenate system is unique. The SO coupling is negligible in the end member BaRuO₃ with the broadest bandwidth, and a β = 0.38 typical of Heisenberg magnet is found. In the Ca substituted Sr_{1-x}Ca_xRuO₃ samples, however, strong correlations result in a phase segregation into clusters with an enhanced SO coupling in a matrix of unaltered spin-spin interactions. As a result, these samples not only exhibit an unusual paramagnetic susceptibility, but also have a critical exponent $\beta \ge 0.5$ as found in other 5d-metal oxide ferromagnets. Therefore, a continuous evolution of the curvature in an Arrott plot from Ba substituted to Ca substituted samples implies strengthening a SO coupling as the bandwidth narrows. As in other 5d-oxide ferromagnets, such as the post

perovskite insulator CaIrO₃, observation of mean-field exponents $\beta \approx 0.5$ and $\gamma \approx 1$ in SrRuO₃ is part of a curvature evolution of the Arrott plot as a result of increasing spinorbit coupling in a narrow-band ferromagnet and does not signal Stoner-Wohlfarth ferromagnetism.

Moreover, the facts that the ferromagnetic phase in BaRuO₃ collapses abruptly to a Pauli paramagnetism at a $P_c < 8$ GPa ¹⁰ and that a localized low-spin Ru(IV) configuration could have a total quantum number J =0 in non-magnetic clusters of a Griffiths phase in the Ca substituted Sr_{1-x}Ca_xRuO₃ samples provide evidence that the strong ferromagnetism of ARuO₃ perovskites is a distinct phase appearing at the crossover from weakly correlated itinerant to localized Ru-4d electrons. The system provides a clear example of how the spin-orbit coupling of a narrow-band system increases with band narrowing until it induces a transition to localized-electron behavior.

IV. Conclusion

The magnetizations as presented in an Arrott plot in the perovskite ruthenates has been thoroughly investigated. Results from this study include effects on an Arrott plot of the bandwidth, the size variance, and hydrostatic pressure versus non-hydrostatic pressure. At low H/M value, isotherms in an Arrott plot are very sensitive to local site disorder either from A-site size variance or induced by non-hydrostatic pressure. However, the curvature of isotherms at higher H/M depends only on the bandwidth. It has been found that the convex curvature is correlated with the unusual paramagnetic susceptibility in the Ca substituted samples and other 5d-metal oxide ferromagnets in which a spin-orbit

coupling is strong. The overall curvature change from the concave curvature in BaRuO₃ to linear lines in SrRuO₃ to convex curvature in Ca substituted $Sr_{1-x}Ca_xRuO_3$ reflects a progressive strengthening of a spin-orbit coupling as the Ru-4d bandwidth narrows in a distinct strong-ferromagnet phase appearing between a weakly correlated itinerant-electron phase and a localized-electron phase. Since the curvature in the Arrott plot of the magnetization in ruthenates is highly sensitive to structural distortions due to the chemical substitution on the A site, the critical analysis based on the Arrott plot becomes unreliable for the entire perovskite family of ARuO₃. Neither the linear Arrott plot in SrRuO₃ nor a Heisenberg behavior in a modified Arrott plot for BaRuO₃ truly reflects the fundamental magnetic coupling. Other thermodynamic measurements are needed in order to clarify this issue.

Acknowledgements

The work was supported by Robert A. Welch foundation and NSF (DMR 0904282, 1122603) in the US and MOST in China. JSZ is grateful to F. Rivadulla for informing us ref.11.



Fig.1 (Color online) Schematic drawing of the Arrott plot of the magnetization for ferromagnets in two extreme limits. Solid lines can be predicted by the equation of state $M^2 = M_0^2 [1-(T/T_c)^2 + 2\chi_0 H/M]$, which is applicable for a very weak, itinerant-electron ferromagnet. At T > T_c, isotherms intercept the H/M axis at the inverse susceptibility χ^{-1} , whereas they intercept the M² axis at the spontaneous magnetization M_s^2 at T < T_c. Curves of dashed lines with a concave curvature are typical for ferromagnets described by Heisenberg, XY, and Ising models.



Fig. 2 (Color online) Arrott plots M^2 versus H/M for perovskites $ARuO_3$ (I. The effect of A-cation size). These plots are organized according to their average A-cation size $< r_A >$

in the order shown by the diagram in the center of the figure. BaRuO₃ shows uniform curves with concave curvature on crossing T_c; the linearity is restored with a modified Arrott plot $M^{1/\beta}$ versus $(H/M)^{1/\gamma}$. An A-cation size variance distorts the plot so severely that the isothermal curves are no longer uniform on crossing T_c; the size variance creates a convex curvature in the curves below T_c as shown by the arrow in A= Sr_{.7}Ba_{.3}. The combination of the size variance and the structural distortion associated with a smaller $\langle r_A \rangle$ converts all curves to a convex curvature in the plot of A=Sr_{.5}Ca_{.5}.



Fig. 3 (Color online) Arrott plots M^2 versus H/M for perovskites ARuO₃ (II. The effects of size variance and high-pressure treatment). (a-c) These three samples have an identical $\langle r_A \rangle = r_{Sr}$ but different σ^2 as labeled inside each panel. Lines are a guide to the eyes. The linearity deteriorates as σ^2 increases. The size variance clearly causes the convex curvature at low fields as identified in (c). (d-f) These three samples are the polycrystalline SrRuO₃ synthesized under ambient pressure (AP), single crystal SrRuO₃ (Crystal), which plot is digitized from ref.7 (note: the unit of magnetization in this plot is emu/cm³), and polycrystalline SrRuO₃ synthesized under 4 GPa and 1000°C (HP).



Fig.4 (Color online) (a) Lattice parameters versus the averaged A-cation radius $\langle r_A \rangle$ in perovskite ruthenates; (b) Lattice parameters *a-b* versus $\langle r_A \rangle$ near A=Sr.



Fig. 5 (Color online) Arrott plots M² versus H/M for perovskites ARuO₃ (III. The effects of hydrostatic pressure and non-hydrostatic pressure). (a, b) The HP SrRuO₃ sample in a pressure cell filled with Fluorinert (FC); (c,d) The HP SrRuO₃ sample in a pressure cell filled with silicone oil (oil); (e,f) The HP Sr_{.7}Ba_{.3}RuO₃ sample in a pressure cell filled with silicone oil.



Fig.6 (Color online) (a) A zoom-in plot of isotherms inside the dashed circle of the sample A=Sr_{.5}Ca_{.5} in Fig.2; (b) the logM vs logH plot of the same isotherms in (a). Inset is the temperature dependence of δ from the linear fitting to lines in (b)



Fig. 7 (Color online) (a) Attott plot of isotherms for HP SrRuO₃ and their linear fittings; (b) temperature dependences of M_s and ${\chi_0}^{-1}$ from (a) and fitting to a power law $M_s(T) \propto |t|^{\beta}$ for t < 0, and ${\chi_0}^{-1}(T) \propto |t|^{\gamma}$ for t > 0; $t \equiv (T-T_c)/T_c$



Fig.8 (Color online) Critical exponents as a function of averaged A-cation radius in perovskite ruthenates.



Fig.9 Arrott plots of isotherms for four 5d-oxide ferromagnets.



Fig.10 (Color online) Temperature dependence of inverse magnetic susceptibility of 4d and 5d oxide ferromagnets.

* jszhou@mail.utexas.edu

- ¹ A. Arrott and J. E. Noakes, Phys. Rev. Lett. **19**, 786 (1967).
- ² E. P. Wohlfarth, J. Appl. Phys. **39**, 1061 (1968).

³ H. E. Stanley, *Introduction to phase transition and critical phenomena* (Oxford

University Press, New York, 1971)

⁴ T. Moriya, *Spin fluctuations in itinerant electron magnetism* (Springer-Verlag, Berlin, 1985)

- ⁵ G. G. Lonzarich and L. Taillefer, J. Phys. C: Solid State Phys **18**, 4339 (1985).
- ⁶ Y. Takahashi, J. Phys. Soc. Japan **55**, 3553 (1986).
- ⁷ D. Kim, B. L. Zink, F. Hellman, S. McCall, G. Cao, and J. E. Crow, Phys. Rev. B **67**, R100406 (2003).

- ⁸ K. Yoshimura, T. Imai, T. Kiyama, K. R. Thurber, A. W. Hunt, and K. Kosuge, Phys. Rev. Lett. **83**, 4397 (1999).
- ⁹ Jing Xia, W. Siemons, G. Konster, M. R. Beasley, and A. Kapitulnik, Phys. Rev. B **79**, 140407 (2009).
- ¹⁰ J.-S. Zhou, K. Matsubayashi, Y. Uwatoko, C.-Q. Jin, J.-G. Cheng, J. B. Goodenough,
- Q. Q. Liu, T. Katsura, A. Shatskiy, and E. Ito, Phys. Rev. Lett. 101, 077206 (2008).
- ¹¹ J.-S. Zhou and J. B. Goodenough, Phys. Rev. Lett. **94**, 65501 (2005).
- ¹² A. Aharony and E. Pytte, Phys. Rev. Lett. **45**, 1583 (1980).
- ¹³ J.-G. Cheng, J.-S. Zhou, J. B. Goodneough, Y. Sui, Y. Ren, and M. R. Suchomel, Phys. Rev. B **83**, 64401 (2011).
- ¹⁴ J. Kanamori, Prog. Theor. Phys. **17**, 177, 197 (1957).
- ¹⁵ M. J. Rozenberg, G. Kotliar, and X. Y. Zhang, Phys. Rev. B **49**, 10181 (1994).
- ¹⁶ R. B. Griffiths, Phys. Rev. Lett. **23**, 17 (1969).
- ¹⁷ M. B. Salamon, P. Lin, and S. H. Chun, Phys. Rev. Lett. **88**, 197203 (2002).