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Intrinsic Berry curvature driven anomalous Nernst thermopower in the semimetallic Heusler alloy CoFeVSb

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1	Intrinsic Berry curvature driven anomalous Nernst thermopower
2	in the semi-metallic Heusler alloy CoFeVSb
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9	Abstract
10	Understanding of spin-heat coupling mechanisms and magneto-thermoelectric phenomena,
11	including the anomalous Nernst effect (ANE), in emergent quaternary Heusler alloys is of
12	practical importance for applications in thermal management and energy harvesting. Here, we
13	demonstrate an intrinsic Berry curvature mediated anomalous Nernst thermopower in
14	CoFeVSb, which orders magnetically at high temperature ($T_c \approx 850$ K) with a large saturation
15	magnetization of $\approx 2.2~\mu_B/f.~u.$ at room temperature. We show that the electron-electron elastic
16	and electron-magnon inelastic scattering dominate longitudinal electrical transport at low
17	temperatures ($T \le 50$ K), whereas the electron-phonon and electron-magnon scatterings
18	govern it at higher T . The longitudinal thermopower is resulted mainly from the diffusive
19	contribution, with a very large longitudinal Seebeck coefficient ($\approx 42 \ \mu V. \ K^{-1}$ at 395 K). The
20	value of the anomalous Nernst coefficient (S_{ANE}) for CoFeVSb at room temperature is \approx
21	0.039 μ V. K ⁻¹ which is higher than the compressively strained SrRuO ₃ film (0.03 μ V. K ⁻¹) as
22	well as the spin gapless semiconductor CoFeCrGa (0.018 μ V.K ⁻¹). On lowering <i>T</i> , both the
23	ordinary Nernst coefficient and carrier mobility increase but an opposite trend is found for
24	S_{ANE} . Our ab-initio simulations reveal the topological semimetallic nature of CoFeVSb with a

1	pair of Weyl points. These Weyl crossings result in a significant contribution to the Berry
2	curvature, leading to an intrinsic anomalous Hall conductivity (σ_{xy}^{AHE}) of ≈ 85 S/cm, which
3	matches well with experiment (77 S/cm at 2 K). Our experimental findings and ab-initio
4	calculations support the dominance of the intrinsic Berry curvature in the observed ANE. The
5	ratio of σ_{xy}^{AHE} to the transverse anomalous thermoelectric conductivity (α_{xy}^{ANE}) shows an
6	increasing trend with T attaining a sizable fraction of $\frac{k_B}{e}$ ($\approx 0.35 \frac{k_B}{e}$) at room temperature.
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8	Keywords: Heusler alloys, transverse thermoelectricity, anomalous Nernst effect,
9	thermoelectric conductivity, Berry curvature
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1 I. INTRODUCTION

2 Current advancement in the field of spin-heat coupling [1–4] and transport of thermally 3 generated charge carriers has revived the interest in transverse magneto-thermoelectric 4 phenomena, *e.g.*, the anomalous Nernst effect (ANE) [5], the anomalous Ettingshausen effect 5 (AEE), [6] etc. The ANE is the thermal analog of the anomalous Hall effect (AHE), and 6 described as a magneto-thermoelectric phenomenon wherein an electric field is generated by 7 applying a thermal gradient and a transverse magnetic field to a magnetic 8 conductor/semiconductor [5,7]. Because of the large Berry curvature near the Fermi level, 9 topological ferromagnets Co₂MnGa [8,9], Co₃Sn₂S₂ [10,11], Fe₃Ga [12], Fe₃Sn [13] etc., exhibit very large anomalous Nernst conductivity within the range $0.5 - 5 \text{ A} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$. Very 10 11 recently, the non-centrosymmetric Kagome ferromagnet UCo_{0.8}Ru_{0.2}Al [14] and the 12 noncollinear topological antiferromagnet YbMnBi₂ [15] have been reported to show colossal anomalous Nernst conductivities of 15 A \cdot m⁻¹ \cdot K⁻¹ and 10 A \cdot m⁻¹ \cdot K⁻¹, respectively at low 13 14 temperatures. While the intrinsic Berry curvature driven large ANEs have been reported in 15 some Heusler alloys based topological ferromagnets [8,9,16], other Heusler ferromagnets exhibit an extrinsic skew scattering dominated ANE [17–19]. 16

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18 Among the Heusler systems of current interest, CoFeVSb possesses a very high ordering temperature ($T_C \approx 850$ K) and a large saturation magnetization of $\approx 2.2 \,\mu_B/f$. u.at 19 20 room temperature [20]. This quaternary Heusler alloy shows asymmetric magnetoresistance 21 (MR) at room temperature, indicative of the spin-valve like nature of the material in its bulk 22 form [20]. Ab initio calculations have shown that this system is composed of energetically 23 competing ferromagnetic(FM)/antiferromagnetic(AFM) interface structures embedded in an 24 FM matrix, which gives rise to the spin valve-like MR feature [20]. The AHE has also been observed in this system over a broad temperature range ($4K \le T \le 395K$) and attributed to 25

the dominating intrinsic Berry phase contribution. Furthermore, this sample possesses a very high value of thermoelectric power factor $\approx 0.62 \text{mW} \cdot \text{m}^{-1} \cdot \text{K}^{-2}$ at room temperature, making it a promising thermoelectric material. The coexistence of these striking physical properties in a single phase of CoFeVSb prompted us to investigate its spin-caloritronic properties. Here, we report on a comprehensive investigation of electrical, thermal, and magnetothermoelectric properties with an emphasis on the ANE in CoFeVSb.

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8 II. EXPERIMENTAL DETAILS

9 Polycrystalline sample of CoFeVSb were synthesized by arc melting technique. The synthesis 10 method and the crystal-phase characterization of the sample have been reported elsewhere [20]. 11 The static magnetic properties of the sample were characterized using a vibrating sample 12 magnetometer (VSM) attachment with a physical property measurement system (PPMS), Quantum Design. The longitudinal electrical resistivity (ρ_{xx}), longitudinal thermopower (S_{xx}), 13 and thermal conductivity (κ_{xx}) were simultaneously measured with the Thermal Transport 14 Option (TTO) of the PPMS. A sample with dimensions $5 \times 4 \times 1 \text{ mm}^3$ was used for 15 16 Hall/Nernst measurements. A standard four-probe technique was employed for the Hall measurements. A DC current, I_{DC} =8mA was sourced through the current leads using a Keithley 17 2400 source-meter and the average Hall voltage was recorded as, $V_H = \frac{[V(+I)-V(-I)]}{2}$ using a 18 19 Keithley 2182A nanovoltmeter. The background-corrected Hall voltage was estimated as, $V_{xy}^{Hall} = \frac{[V_H(+H_{max}) - V_H(-H_{max})]}{2}$, where, H_{max} is the maximum value of applied magnetic field. 20 The Hall resistivity was estimated as $\rho_{xy} = \frac{V_{xy}^{Hall}L_Z}{I_{DC}}$, where, L_Z = sample thickness. The Nernst 21 22 measurements on the CoFeVSb sample were performed by employing a home-built spin-23 caloritronic measurement set up integrated with the PPMS. A detailed description of our 24 experimental set up is reported elsewhere [17] and in the Supplemental Material (SM) [21].

1 III. RESULTS AND DISCUSSION

The main panel of Fig. 1(a) demonstrates the temperature (T)-dependence of ρ_{xx} for 2 CoFeVSb in the range: 4K $\leq T \leq$ 395K. Evidently, $\rho_{xx}(T)$ exhibits metallic-like resistivity 3 $\left(\frac{\partial \rho_{xx}}{\partial T} > 0\right)$ throughout the *T*-range along with a slope-change below 50K (see inset of Fig. 4 1(a)). The $\rho_{xx}(T)$ data for CoFeVSb was found to fit well with the expression: $\rho_{xx}(T) = \rho_0 + \rho_0$ 5 $A.\left(\frac{T}{\theta_{D}}\right)^{5} \int_{0}^{\frac{\theta_{D}}{T}} \frac{e^{x} x^{5} dx}{(e^{x}-1)^{2}} + C.T^{2} e^{-(T/\Delta)}$ in the high-*T* region (above 50K), where, ρ_{0} represents the 6 7 T-independent residual resistivity, the second-term represents the electron-phonon scattering contribution according to the Bloch-Grüneisen model ($A = \text{constant pre-factor and } \theta_D = \text{Debye}$ 8 9 temperature) [22–24] and, the third-term signifies the contribution from electron-magnon 10 scattering (C = constant pre-factor, $k_B\Delta$ is the spin wave energy gap which represents the 11 difference between the Fermi level and the nearest band edge of unoccupied minority spins [24], and k_B is the Boltzmann constant). In the case of half-metals, the T^2 -dependent electron-12 13 magnon scattering contribution is exponentially suppressed due to the gapped spin-flip scattering [24–26]. We fitted the $\rho_{xx}(T)$ curve with the expression: $\rho_{xx}(T) = \rho_0 + \rho_0$ 14 $\rho_{elastic}T^{1/2} + C.T^2e^{-(T/\Delta)}$ in the low-T region (below 50K) region [26], where the second-15 term accounts for the electron-electron elastic scattering contribution with a $T^{1/2}$ -dependence 16 at low-T [27]. From the high-T fit, we obtained $\rho_0 = (1.3 \pm 0.01) \times 10^{-4} \Omega$ cm , A =17 $(2.04 \pm 0.06) \times 10^{-4} \ \Omega. \text{ cm. } \text{K}^{-5}, \quad C = (5.07 \pm 0.2) \times 10^{-11} \ \Omega. \text{ cm. } \text{K}^{-2}, \quad \theta_D = (313 \pm 0.2) \times 10^{-11} \ \Omega. \text{ cm. } \text{K}^{-2}$ 18 2) K and $\Delta = (40 \pm 5)$ K. The fitting parameters extracted from the low-T fit are $\rho_0 =$ 19 $(1.2 \pm 0.02) \times 10^{-4} \,\Omega.\,\mathrm{cm}$, $\rho_{elastic} = (-3.34 \pm 0.1) \times 10^{-7} \,\Omega.\,\mathrm{cm}$. $\mathrm{K}^{-1/2}$, $C = (2.7 \pm 0.02) \times 10^{-7} \,\Omega.\,\mathrm{cm}$ 20 $0.1) \times 10^{-9} \Omega. \text{ cm. K}^{-2} \text{ and } \Delta = (40 \pm 3) \text{ K}.$ 21



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FIG. 1(a) Main panel: T-dependence of ρ_{xx} of CoFeVSb, fitted with $\rho_{xx}(T) = \rho_0 + \rho_0$ 2 $A.\left(\frac{T}{\theta_{D}}\right)^{5}\int_{0}^{\frac{\theta_{D}}{T}}\frac{e^{x}x^{5}dx}{(e^{x}-1)^{2}}+C.T^{2}e^{-(T/\Delta)}$ in the high-*T* region and inset: fitting with $\rho_{xx}(T)=\rho_{0}+C$ 3 $\rho_{elastic}T^{1/2} + C.T^2 e^{-(T/\Delta)}$ in the low-T region. (b) T-dependence of S_{xx} . (c) Main panel: T-4 dependence of κ_{xx}^{Tot} and κ_{xx}^{el} , inset: *T*-dependence of thermoelectric figure of merit, *ZT*. (d) *T*-5 dependence of κ_{xx}^L fitted with Callaway's model. (e) Schematic illustration of Nernst 6 measurement. (f) Left y-scale: H-dependence of the Nernst voltage, $V_{xy}(H)$ for $\Delta T = +15$ K 7 8 and right y-scale: M(H) at 295K. (g) Main panel: $V_{xy}(H)$ for different values of ΔT at 295K, inset: background corrected anomalous Nernst voltage, $S_{ANE} \times \Delta T_{eff}$ as a function of ΔT_{eff} . 9 10

11 The values of Δ obtained from the fits for CoFeVSb are close to that reported for the 12 well-known half-metallic CrO₂ ($\Delta \approx 80$ K), [25] Fe₂Si ($\Delta \approx 85$ K), [26] and Co₂FeSi 13 ($\Delta \approx 100$ K), [24] which strongly indicates the half-metallic nature of this sample. Fig. 1(b) 14 represents the *T*-dependence of S_{xx} for CoFeVSb. The sign of $S_{xx}(T)$ is negative throughout 15 the measured *T*-range, indicating electrons as the dominant carrier for the thermally-driven 1 charge-transport. Moreover, $|S_{xx}|$ decreases linearly from $\approx 42 \mu V.K^{-1}$ at 395K and approaches 2 zero at low-*T* indicating the dominant contribution of the diffusive component of 3 thermopower [28].

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5 As shown in the main panel of Fig. 1(c), the total longitudinal thermal conductivity, $\kappa_{xx}^{Tot}(T)$ remains almost unchanged in the *T*-range: 200K $\leq T \leq$ 395K, but decreases slowly 6 below 200K and rapidly below 100K. The T-dependence of the electronic thermal conductivity, 7 $\kappa_{xx}^{el}(T)$ is usually estimated from the Wiedemann Franz law: $\kappa_{xx}^{el} = \frac{L_0 T}{\rho_{xx}}$, where, $L_0 = \frac{\pi^2 k_B^2}{3e^2} =$ 8 2.44×10^{-8} W. Ω . K⁻¹ is the Lorenz number for free electrons (degenerate-limit) [29]. 9 However, L_0 was shown to deviate from the degenerate-limit for semiconductors with large 10 value of S_{xx} . [30] To account for this discrepancy, L_0 for such semiconductors was proposed 11 to be accurately estimated using the expression, $L_0 = 1.5 + \exp\left(-\frac{|S_{xx}|}{116}\right)$. [30] Since $|S_{xx}|$ for 12 our sample is very high at high-*T*, we estimated $\kappa_{xx}^{el}(T)$ using the expression, [30] $\kappa_{xx}^{el}(T) =$ 13 $\frac{\left[1.5 + \exp\left(-\frac{|S_{XX}|}{116}\right)\right]T}{\rho_{vv}}$, as shown in the main panel of **Fig. 1**(c). The main panel of **Fig. 1**(d) shows 14 the T-dependence of the lattice thermal conductivity, $\kappa_{xx}^L(T)$ obtained from the expression: 15 $\kappa_{xx}^{L}(T) = [\kappa_{xx}^{Tot}(T) - \kappa_{xx}^{el}(T)]$. Evidently, $\kappa_{xx}^{L}(T)$ increases considerably with decreasing 16 17 temperature down to 100K at which it shows a broad maximum and then decreases rapidly 18 with further lowering of the temperature. As shown in the main panel of Fig. 1(d), we fitted $\kappa_{xx}^{L}(T)$ for CoFeVSb with the Debye-Callaway model, [31,32] 19

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$$\kappa_{xx}^{L}(T) = \frac{k_{B}}{2\pi^{2}v_{S}} \left(\frac{2\pi k_{B}T}{h}\right)^{3} \int_{0}^{\left(\frac{\theta}{D}\right/_{T}\right)} \frac{\tau_{L} z^{4} e^{z}}{(e^{z} - 1)^{2}} dz \tag{1}$$

Here, v_S is the average phonon velocity (approximately equals to the sound velocity), $z = \frac{\hbar\omega}{k_BT}$, $\omega =$ phonon frequency, and $\tau_L =$ total phonon relaxation-time. The values of the fitting parameters *A*, *B*, *C* and θ_D are (9.95 ± 2) x 10⁻⁴² s³, (7.45 ± 3) x 10⁻²³ s. K⁻¹, (4.5 ± 1) x 10⁻¹⁸ s 1 and (310 ± 5) K, respectively. The thermoelectric figure of merit, $ZT = \frac{S_{xx}^2}{\kappa_{xx}^{Tot}\rho_{xx}}T$ [33] for 2 CoFeVSb (inset of Fig. 1(c)) is \approx 0.07 at 395K which decreases gradually with decreasing 3 temperature.

4

5 The schematic illustration of our transverse thermoelectric transport measurement is 6 shown in Fig. 1(e). The DC magnetic field (H) is applied along the x-axis and the T-gradient is 7 along the sample thickness (z-axis). The Nernst voltage generated along the y-axis was 8 measured using a Keithley 2182A nanovoltmeter while sweeping H. The left y-scale of Fig. 1(f) exhibits the H-dependence of the Nernst voltage, $V_{xy}(H)$ for a fixed value of the T-9 difference between the top plate (T_{hot}) and the bottom plate $(T_{cold}; T_{hot} > T_{cold}), \Delta T =$ 10 $(T_{hot} - T_{cold}) = +15$ K for the CoFeVSb sample at $T = \frac{T_{hot} + T_{cold}}{2} = 295$ K. The right y-scale 11 of Fig. 1(f) shows the isothermal magnetization, M(H) at 295K. Clearly, the $V_{xy}(H)$ loop 12 13 shows negligible hysteresis and replicates the corresponding M(H) behavior. Since the 14 thermally generated electric field induced by ANE is proportional to the magnetization, *i.e.*, $\overrightarrow{E_{ANE}} \propto (\mu_0 \overrightarrow{M} \times \overrightarrow{\nabla T})$, [8,34] the observed feature of the $V_{xy}(H)$ loop originates from the ANE. 15

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17 Note that the bulk thermal resistances of the sample and the N-grease layers as well as 18 the interfacial thermal resistances cause drop in the actual *T*-gradient across the sample. [35] 19 Ignoring the interfacial thermal resistances between the N-grease and the hot/cold plates as 20 well as between the sample and N-grease layers, [17] the total *T*-difference between the hot 21 and cold plates (ΔT) = effective *T*-difference across the sample (ΔT_{eff}) + drop in ΔT in the N-22 grease layers on both sides of the sample. Therefore, considering the 3-slab model, ΔT_{eff} can 23 be expressed as, [17,36] $\Delta T_{eff} = \frac{\Delta T}{\left[1 + \left(\frac{2d_{N-Grease}}{\kappa_{N-Grease}}\right)\left(\frac{\kappa_{COFeVSb}}{d_{COFeVSb}}\right)\right]}$, where, $d_{N-Grease}(d_{COFeVSb})$ is

24 the thickness of the grease-layer (CoFeVSb), $\kappa_{N-Grease}(\kappa_{CoFeVSb})$ is the thermal conductivity

of the grease-layer (CoFeVSb), and A = cross-sectional area. Using the reported values of $\kappa_{\text{N-Grease}}(T)$ of the N-grease, [37], $d_{\text{N-Grease}} \approx 10 \mu\text{m}$, $d_{\text{CoFeVSb}} \approx 1 \text{mm}$, we estimated the *T*-dependence of ΔT_{eff} (see SM [21]).

4

5 In addition to the ANE contribution, the ordinary Nernst effect (ONE) also contributes to the total Nernst signal. The ONE contribution can be estimated from the slope of the slowly 6 7 varying segment $(H_{sat} \le H \le H_{max})$ of the *H*-dependent transverse Seebeck coefficient, $S_{xy}(H)$ defined as, $S_{xy}(H) = \frac{V_{xy}(H)}{\Delta T_{eff}} \times \left(\frac{L_z}{L_y}\right)$. [34,38] Here, L_y (=3mm) is the distance between 8 9 the voltage leads and L_z (= 1mm) is the sample thickness. The ordinary Nernst coefficient (S_{ONE}) at 295K for CoFeVSb was estimated as $S_{ONE} = (-0.176 \pm 0.05) \text{ nV.K}^{-1} \cdot \text{T}^{-1}$, which is 10 smaller compared to other materials e.g., Fe₃O₄ single crystal ($S_{ONE} = 10 \text{ nV}.\text{K}^{-1}.\text{T}^{-1}$ at room 11 12 temperature) [34]. The main panel of Fig. 1(g) demonstrates $V_{xy}(H)$ for different values of ΔT at 295K. Evidently, $V_{xy}(H)$ increases with ΔT . The inset of Fig. 1(g) shows the normalized 13 anomalous Nernst voltage, $S_{ANE} \times \Delta T_{eff} = V_{ANE} (\mu_0 H_{max}) \times \left(\frac{L_z}{L_v}\right)$, as a function of ΔT_{eff} at 14 295K, where, S_{ANE} is the background-corrected anomalous Nernst coefficient, $V_{ANE}(\mu_0 H_{max})$ 15 represents the background-corrected anomalous Nernst voltage defined as, $V_{ANE}(\mu_0 H_{max}) =$ 16 $\left[\frac{V_{xy}(+\mu_0H_{max})-V_{xy}(-\mu_0H_{max})}{2}\right] \text{ and } \mu_0H_{max} = \text{maximum applied magnetic field } (\mu_0H_{max} \gg 1)$ 17 $\mu_0 H_{sat}$). Clearly, $(S_{ANE} \times \Delta T_{eff})$ varies linearly with ΔT_{eff} . 18 19

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- 21



FIG. 2(a) $V_{xy}(H)$ for CoFeVSb at selected temperatures for $\Delta T = +15$ K. (b) $S_{xy}^{ANE}(H)$ and (c) $\rho_{xy}(H)$ for CoFeVSb at selected temperatures. (d) Left y-scale: *T*-dependence of S_{ANE} fitted with Eqn. (2), and right y-scale: *T*-dependence of ρ_{xy}^{AHE} . (e) Left y-scale: *T*-dependence of α_{xy}^{ANE} fitted with Eqn. (3), and right y-scale: *T*-dependence of $(\alpha_{xy}^{ANE} / \sigma_{xy}^{AHE})$. (f) Left y-scale: *T*dependence of $(|S_{ONE}|/T)$, and right y-scale: *T*-dependence of μ_H .

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Fig. 2(a) demonstrates the $V_{xy}(H)$ hysteresis loops for CoFeVSb at selected 8 temperatures for $\Delta T = +15$ K. Notably, $|V_{xy}(H)|$ decreases with lowering temperature. Most 9 importantly, $V_{xy}(H)$ shows prominent negative-slope in the high field region at low-T, and the 10 11 absolute value of the slope increases drastically below 200K, indicating the enhanced ONE 12 contribution at low-T. Fig. 2(b) shows the magnetic field dependence of the anomalous Nernst coefficient (ANC), $S_{xy}^{ANE}(H)$ at different temperatures, which was obtained by subtracting the 13 ONE contribution from the total transverse Seebeck coefficient, $S_{xy}(H)$ as, $S_{xy}^{ANE}(H) =$ 14 $[S_{xy}(H) - S_{xy}^{ONE}(H)] = [S_{xy}(H) - |S_{ONE}|.H]$. Clearly, $|S_{xy}^{ANE}(H)|$ gradually decreases with 15

1 decreasing-*T*. The left *y*-axis of **Fig. 2**(d) demonstrates the *T*-dependence of the background-2 corrected ANC, $S_{ANE}(\mu_0 H_{max}, T)$. At 295K, $|S_{ANE}| = 0.039 \ \mu\text{V.K}^{-1}$ for CoFeVSb which is 3 higher than the compressively-strained SrRuO₃ film (0.03 μ V.K⁻¹) [39] as well as the spin-4 gapless semiconductor CoFeCrGa (0.018 μ V.K⁻¹) [17].

5

Now, let us understand the origin of the non-linear T-dependence of $S_{ANE}(T)$ in 6 CoFeVSb. The transverse thermoelectric coefficient (S_{xy}) is defined as, $S_{xy} = \left[\frac{\alpha_{xy} - S_{xx}\sigma_{xy}}{\sigma_{xy}}\right]$, 7 where σ_{xy} and σ_{xx} are the transverse and longitudinal electrical conductivities which are 8 expressed as, [5,9,40] $\sigma_{xy} = \left[\frac{-\rho_{xy}}{(\rho_{xx})^2 + (\rho_{xy})^2}\right]$ and $\sigma_{xx} = \left[\frac{\rho_{xx}}{(\rho_{xx})^2 + (\rho_{xy})^2}\right]$, respectively, and α_{xy} 9 is the transverse thermoelectric conductivity [5,34]. According to the Mott's relations, S_{xx} = 10 $\frac{\pi^2 k_B^2 T}{3e\sigma_{xx}} \left(\frac{\partial \sigma_{xx}}{\partial E}\right)_{E=E_F}, \text{ and } \alpha_{xy} = \frac{\pi^2 k_B^2 T}{3e} \left(\frac{\partial \sigma_{xy}}{\partial E}\right)_{E=E_F}, \text{ where } e \text{ is the electronic charge and } E_F \text{ is the}$ 11 Fermi energy. [5,41] It is known that the ANE and AHE share the common physical origin [5] 12 Therefore, considering the power law for the AHE, $\rho_{xy}^{AHE} = \lambda M \rho_{xx}^n$, [5] where ρ_{xy}^{AHE} is the 13 anomalous Hall resistivity, ρ_{xx} is the longitudinal resistivity, λ is the spin-orbit coupling 14 15 constant and *n* is an exponent, the ANC can be expressed as, [5,34]

16
$$S_{xy}^{ANE} = \rho_{xx}^{n-1} \left[\frac{\pi^2 k_B^2 T}{3e} \left(\frac{\partial \lambda}{\partial E} \right)_{E=E_F} - (n-1) \lambda S_{xx} \right].$$
(2)

For n = 1, the extrinsic skew scattering is the leading mechanism for the anomalous Nernst/Hall transport, whereas for n = 2, the intrinsic Berry curvature becomes the dominating mechanism [42]. We fitted the $S_{ANE}(T)$ for CoFeVSb using Eqn. 2 and the best fit was obtained for $n = 2.24 \pm 0.2$ which indicates that the origin of ANE in CoFeVSb is dominated by the intrinsic Berry curvature or, the side jump mechanism [42]. To confirm the origin of the observed ANE, we need to understand the temperature dependence of the anomalous offdiagonal thermoelectric conductivity, $\alpha_{xy}^{ANE}(T)$, which can be expressed as, [10,11,43]

1
$$\alpha_{xy}^{ANE} = S_{xy}^{ANE} \sigma_{xx} + S_{xx} \sigma_{xy}^{AHE} = \left[\frac{S_{xy}^{ANE} \rho_{xx} - S_{xx} \rho_{xy}^{AHE}}{(\rho_{xx})^2 + (\rho_{xy}^{AHE})^2} \right]$$
. Using the Mott's relations, α_{xy}^{ANE} can be

2 written as, [5,34]

3
$$\alpha_{xy}^{ANE} = \rho_{xx}^{n-2} \left[\frac{\pi^2 k_B^2 T}{3e} \left(\frac{\partial \lambda}{\partial E} \right)_{E=E_F} - (n-2) \lambda S_{xx} \right]$$
(3)

4 The main panel of Fig. 2(c) illustrates the magnetic field dependence of Hall resistivity $\rho_{xy}(H)$ for CoFeVSb at few selected temperatures in the *T*-range: $120K \le T \le 295K$. Similar to the 5 6 Nernst voltage, $\rho_{xy}(H)$ also shows the contributions from both ordinary Hall effect (OHE) and 7 AHE along-with negative slope in the high field region throughout the temperature range, 8 indicating electrons as the majority carriers. However, unlike the Nernst voltage, and the 9 absolute value of the slope in $\rho_{xy}(H)$ increases slowly with decreasing temperature. Especially, 10 at low-T (e.g., 120K), the negative slope in the Nernst voltage is more robust compared to that 11 in the Hall resistivity. Such striking difference in the Nernst and Hall signals has also been 12 observed in the half-metallic semimetal Co₃Sn₂S₂. [11] From the slope of the slowly varying 13 segment of $\rho_{xy}(H)$, we obtained the OHE contribution and estimated the *T*-dependence of the carrier-concentration, $n_H(T)$, as shown in the inset of Fig. 2(c). Clearly, $|n_H|$ decreases with 14 decreasing temperature. By subtracting the OHE contribution from $\rho_{xy}(H)$, we evaluated the 15 temperature dependence of anomalous Hall resistivity (ρ_{xy}^{AHE}), as shown on the right-y scale 16 of Fig. 2(d). [42] Similar to the ANC, ρ_{xy}^{AHE} also decreases with decreasing temperature. 17 Incorporating the *T*-dependences of S_{xx} , S_{ANE} , ρ_{xx} and ρ_{xy}^{AHE} , we evaluated the *T*-variation of 18 α_{xy}^{ANE} , as shown on the left-y scale of Fig. 2(e). We fitted $\alpha_{xy}^{ANE}(T)$ for CoFeVSb using Eqn. 19 20 3, and the best fit was obtained for $n = 2.02 \pm 0.1$, which is close to that obtained from the $S_{ANE}(T)$ fit. This further confirms that both the ANE and AHE in CoFeVSb is governed by the 21 intrinsic Berry curvature or, the side jump mechanism [42]. Notably, the ratio $\left[\left(\frac{\partial \lambda}{\partial E}\right)_{E=E_{T}}/\lambda\right]$ 22 obtained from the $S_{ANE}(T)$ and $\alpha_{xy}^{ANE}(T)$ fits are 1.33×10^{19} and 2.04×10^{19} , respectively. 23

1 Next, we examine the correlation between the anomalous transverse thermoelectric conductivity (α_{xy}^{ANE}) and anomalous Hall conductivity (σ_{xy}^{AHE}) . While α_{xy}^{ANE} represents the 2 transport of entropy, σ_{xy}^{ANE} determines the transport of charge-carriers and, $(\alpha_{xy}^{ANE}/\sigma_{xy}^{AHE})$ 3 should approach $\frac{k_B}{e}$ at high-*T* [14,44]. As shown on the left-y scale of Fig. 2(e), $\left| \alpha_{xy}^{ANE} / \sigma_{xy}^{AHE} \right|$ 4 increases with increasing temperature and attains $\approx 0.35 \frac{k_B}{e}$ at 295K, which is much lower than 5 UCo_{0.8}Ru_{0.2}Al $\left(\left| \alpha_{xy}^{ANE} / \sigma_{xy}^{AHE} \right| \approx 2 \frac{k_B}{e} \text{ at } 47 \text{K} \right)$ [14], but closer to that for Co₂MnGa, Mn₃Sn, 6 7 Co₃Sn₂S₂ and La_{0.7}Sr_{0.3}CoO₃ [44]. Since the ordering temperature of CoFeVSb is very high $(T_c \approx 850 \text{K})$ and, both $S_{ANE}(T)$ and $\left| \alpha_{xy}^{ANE}(T) / \sigma_{xy}^{AHE}(T) \right|$ for CoFeVSb are increasing with 8 9 increasing temperature even at room temperature, it would be worth investigating high 10 temperature ANE/AHE in this system to realize at what temperature(s) S_{ANE} shows a peak and $\left| \alpha_{xy}^{ANE}(T) / \sigma_{xy}^{AHE}(T) \right|$ approaches $\frac{k_B}{e}$. To confirm the origin of ANE in CoFeVSb, it is 11 imperative to quantify the carrier mobility (μ_H) . In semi-classical picture, $|S_{ONE}|$ scales 12 linearly with μ_H through the expression, [11,45,46] $|S_{ONE}| = \frac{\pi^2}{3} \left(\frac{k_B}{e}\right) \left(\frac{k_B T}{E_F}\right) \mu_H$. However, 13 $|S_{ANE}|$ is known to scale with $\left(\frac{1}{\mu_H}\right)$ if the origin of the ANE is dominated by the intrinsic Berry-14 curvature [11]. As shown in Fig. 2(f), both ($|S_{ONE}|/T$) and μ_H increase with decreasing 15 temperature. Furthermore, S_{ANE} decreases with decreasing temperature whereas μ_H shows the 16 17 opposite T-dependence, which is in agreement with the intrinsic Berry-curvature picture of 18 ANE [11].

19

We have further performed *ab-initio* band structure calculations to complement our experimental results. Computational details of these calculations are presented in reference [20] and SM [21]. Figure 3(a) shows the band structure revealing a pair of Weyl points above E_F (≈ 0.3 eV) arising from the intersections of semi-metallic bands 28 and 29 1 (highlighted by red circles). The contribution to the density of states at/near E_F primarily arises 2 from band numbers 27, 28 and 29 (see Fig. 3(b)). Figure 3(c) display the z-component of 3 simulated Berry curvature ($-\Omega_z(k)$), along with its 2D projection in the $k_z=0$ and $k_z=0.5$ 4 planes shown in Figs. 3(d)-(e). The spike in the Berry curvature along the L- Γ direction (see 5 Fig. 3(c)) is due to two spin-semi metallic bands, 27 and 28 (see Fig. 3(a)). This spike arises 6 from the small energy denominator in the Berry-curvature's definition along this *k*-path [42]. 7



8

FIG. 3 For CoFeVSb, (a) Electronic band structure showing the topological non-trivial feature involving a pair of Weyl points at ≈ 0.3 eV above E_F. (b) Widths of various semi metallic bands (27, 28, 29). (c) Calculated Berry curvature along the high symmetry path. (d)-(e) 2D projection of Berry curvature on k_z =0 and k_z =0.5 plane (f-h) Fermi surfaces originating from the semi metallic bands (27, 28, 29) indicating the location of electron/hole pockets and Weyl points.

14

15 A reasonably high intrinsic anomalous Hall-conductivity (AHC) value is obtained 16 which is solely attributed to the large spike in $\Omega_z(k)$. The following equation was used to 17 calculate AHC,

18
$$\sigma_{xy} = -\frac{e^2}{\hbar} \sum_p \int_{BZ} \frac{d^3k}{(2\pi)^3} f_p(k) \Omega_{p,Z}(k)$$
(4)

where $f_n(k)$ represents the Fermi-Dirac distribution function, $\Omega_{n,z}(k)$ is the z-component of 1 the Berry curvature for n^{th} band, and the integration is over the entire Brillouin zone (*BZ*). The 2 simulated value of intrinsic AHC is, $\sigma_{int}^A = 85$ S/cm, which matches well with the experimental 3 4 value (77 S/cm at 2K). This implies that both intrinsic AHC and ANE mainly stem from the 5 Berry curvature. We also simulated the Fermi surfaces related to the three semi-metallic bands 6 (27, 28, and 29), as shown in Figs. 3(f)-(h). Two hole-pockets emerged along L-Γ points due 7 to bands 27 and 28, while an electron-pocket appears from band 29. A pair of Weyl points is 8 clearly visible in the Fermi surface, originating from band 28 (see Fig. 3(g)), as observed in 9 the band structure along L- Γ , and U- Γ (see Fig. 3(a)), corroborating the topological non-trivial 10 features of this alloy.

11

12 IV. CONCLUSIONS

13 In summary, we present a comprehensive study of the longitudinal and transverse 14 thermoelectric properties with a special focus on the ANE in CoFeVSb. We have found that the anomalous Nernst coefficient (S_{ANE}) of CoFeVSb is $\approx 0.039 \,\mu\text{V}$. K⁻¹ at room temperature. 15 Both ordinary Nernst coefficient (S_{ONE}) and carrier mobility (μ_H) increase upon lowering 16 temperature, but an opposite trend is found for S_{ANE} . Our in-depth analysis confirmed that the 17 18 observed ANE in this material is dominated by intrinsic Berry-curvature which is also 19 supported by our *ab initio* calculations. Our band structure calculations confirm the topological non-trivial feature of CoFeVSb with a pair of Weyl points located slightly above E_F . 20

21

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