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Unified understanding of the first-order nature of the transition in $TbCo_2$

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Determination of the nature of phase transitions, especially those that involve two symmetry-breaking order parameters, is a fundamental issue in condensed matter physics. For the Laves-phase rare-earth - transition-metal intermetallic compounds, their phase transitions involve both magnetic ordering and structural ordering. As a typical material of the Laves-phase intermetallics, $TbCo_2$ has been studied extensively for its transition around 230 K. However, the understanding on the nature of this transition remains controversial (first/second-order) for decades. Here in this work, based on the criteria that determine first-order and second-order transitions for magnetic materials: (1) latent heat, (2) thermal hysteresis, (3) coexistence of phases and (4) Banerjee criterion, we show direct evidence to reveal the first-order nature of the transition in $TbCo_2$, which is further interpreted by a Landau theory based phenomenological approach. Our work reconciles the lasting arguments on the transition of $TbCo_2$ and may pave the way for deepening the understanding on the transitions of magnetic materials that involve both magnetic and structural transitions.

I. INTRODUCTION

Phase transitions have been lying the foundation of magnetic functional materials, in particular the ones that involve the symmetry breaking of more than one order parameters[1]. The relationship between such transitions and functionalities is well exemplified by Laves-phase rare-earth - transition-metal compounds $\mathbf{RT}_2$ (**R** refers to rare-earth elements and **T** refers to transition-metal elements), which undergo a magnetic transition associated with a structural change[2].

Owing to the competition between the rare-earth - transition-metal exchange interaction and crystalline electric field effect[3,4], RT_2 alloys show diverse interesting effects, e.g., magnetostriction[5,6], magnetocaloricity[7,8], tunable thermal expansion[9], magnetoresistance[10,11], etc. Therefore, given the direct relationship between the transition and physical properties, it is crucial to investigate the nature (first/second-order) of these phase transitions in view of both fundamental theory and application.

As one typical compound in RT_2 family, $TbCo_2$ has been studied extensively on both magnetic transition and structural transition around 230 K. Early studies indicate that the transition of $TbCo_2$ is a second-order magnetic transition associated with a structural transition[12–17], which violates the common knowledge on the determination of the nature of phase transition[18,19]. A recent study reported the first-order nature of this transition, but pointed out that the magnetic transition and structural transition decouple[20], contradicting with a wealth of experimental evidence[9,21–23]. In one word, the nature of the transition in $TbCo_2$ remains an open question of interest.

Experimentally, for magnetic materials, the criteria for determining second-order phase transition (**SOPT**) or first-order phase transition (**FOPT**) are: (1) latent heat[14,24], (2) thermal hysteresis[14,24], (3) phase-coexistence[25], and (4) Banerjee criterion[26–29]. In this work, based on the evidence from the above-mentioned experimental results, we show that the paramagnetic-ferrimagnetic transition in $TbCo_2$ is of first-order.

II. EXPERIMENTS

The $TbCo_2$ alloy was prepared by arc melting method with the raw materials of Tb (99.9%) and Co (99.9%) in an argon atmosphere. To ensure compositional homogeneity, the sample (about 8 g) was melted four times. The as-cast ingot was cut into slices with the thickness of 1 mm and sealed into a quartz tube, filled with argon gas; then the sealed sample was annealed at 1273 K for 72 hours, and naturally furnace-cooled to room temperature. The synchrotron X-ray diffraction (XRD) was carried out at the BL15XU NIMS beam line of Spring-8 with the X-ray wavelength λ=0.6538 Å. The samples for synchrotron XRD were well-ground powders and sealed into Kapton capillaries. The capillary was rotated during the measurement to reduce the preferred orientation effect and to average the intensity. The crystal structures were refined using the Rietveld algorithm[30,31]. The samples used for the physical property measurements are polycrystalline bulk. The heat flow on heating and cooling processes across Curie temperature (T_C) is monitored using differential scanning calorimeter (DSC, Q2000, TA Instruments). The magnetic measurements were performed on the superconducting quantum interference device (MPMS-SQUID, Quantum Design). The magneti-

zation (M) versus temperature (T) curve was measured on cooling at the rate of 2 K/min from 260 K to 100 K, under the field of 500 Oe. The magnetic susceptibility (χ) versus temperature (T) curve was measured on cooling at the rate of 2 K/min from 260 K to 100 K, under the field 2 Oe with the frequency of 133 Hz. Before the measurement of isotherm M(H) curves, magnet reset process was performed to make sure there would be no frozen field to influence the magnetization behavior at low field.

III. RESULTS AND DISCUSSION

Figure 1 shows the magnetization (M) versus temperature (T) and susceptibility (χ) versus temperature curves. From the $\partial M/\partial T$-T (inset of Fig.1(a)) curve and fitted $1/\chi$'-T curve (inset of Fig.1(b), deduction of the fitting can be referred to the Supplemental Material[32]), T_C is determined as 229.57±0.01 K, agreeing well with the reported values[9,20].

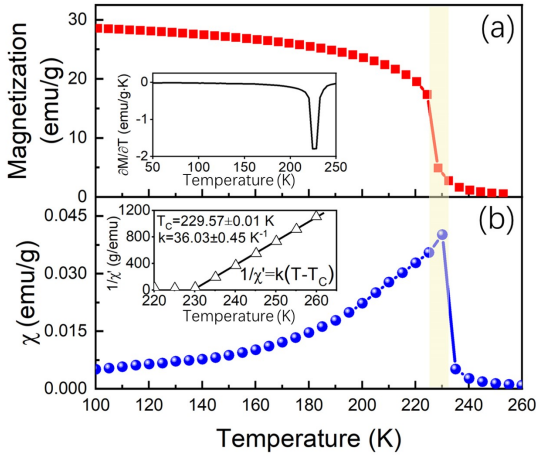

FIG. 1. (a) M-T curve and (b) χ-T curve of $TbCo_2$. The derivative of magnetization over temperature and fitted $1/\chi$'-T curve are shown in the insets of (a) and (b), respectively.

The XRD profile and refined pattern at 130 K (below T_C) are shown in Fig. 2(a), and the crystallographic information is listed in Table 1. The refined XRD pattern reveals that in ferrimagnetic state, $TbCo_2$ crystallizes in rhombohedral structure with the space group R$\bar{3}$m (No. 166).

The evolution of crystal structure from above T_C to below T_C, as reflected by the evolution of the characteristic reflections {222} and {800} from 260 K to 200 K observed from *in situ* synchrotron XRD measurements, is shown in Fig. 2(b1). At 260 K, no splitting in both {222} and {800} reflections is in accordance with the detected cubic crystal structure[21]. At 220 K and 200 K, the splitting in {222} reflection and no splitting in {800} reflection, indicate the rhombohedral crystal symmetry[33].

TABLE I. Crystal data and structure refinement for $TbCo_2$ at 130 K.

Wavelength		0.6538 Å	
Crystal system		Trigonal	
Space group		R$\bar{3}$m(No. 166)	
Unit cell dimensions		a = 5.0962(1) Å b = 5.0962(1) Å c = 12.5336(1) Å α = 90° β = 90° γ = 120°	
Volume		281.905(2) Å^3	
Goodness-of-fit on F^2		0.925	
R indices (all data)		Rp = 6.62%	Rwp = 9.34%
Atomic parameters			
Atom	Wyckoff	Position	Occ.
Tb	6c	0,0,0.1243	1
Co1	3b	0,0,0.5	1
Co2	9e	0.5,0,0	1

At 230 K, which is the detected magnetic transition temperature from M-T curve (Fig. 1(a)) and χ-T curve (Fig. 1(b)), the asymmetric peak shape of the reflections {222} and {800}, especially of {800}, is observed clearly, which indicates the phase-coexistence state. Besides, the asymmetric peaks are consistent with the superposition of cubic and rhombohedral profiles. Further on cooling, the emergent splitting of {222} reflection below T_C unambiguously suggests a structure phase transition, and the observed phase coexistence at 230 K further proves the first-order nature of the transition. The calculated lattice parameters and spontaneous lattice strain ε (the calculation of ε can be referred to Ref.[25]) are shown in Fig.2(b2).

The DSC measurement was carried out to check the heat flow during transition (Fig. 3). The appearance of exothermic peak on cooling process and the endothermic peak on heating process, proves FOPT around 230 K. The temperatures of exothermic peak and endothermic peak are 228 K and 233 K (at the rate of 6 K/min), respectively, agreeing well with the magnetic transition temperature (Fig.1). To eliminate the impact of temperature ramp rate, the rate-dependence (2, 4, 6, 8, 10, 12, 14 K/min) of measured peak temperatures is investigated and presented in the inset of Fig.3. Following the common treatment[34,35], linear fitting is adopted to fit the rate-dependence. The magnitude of the thermal hysteresis approximates to $\sim$ 1 K at an extrapolated zero ramp rate, comparable to the value reported recently (0.6 K)[20]. The non-zero latent heat (DSC peak) and the thermal hysteresis both reveal the first-order nature of the transition around 230 K. It should be noted that the thermal hysteresis between exothermic and endothermic peaks was previously detected but neglected, and thus

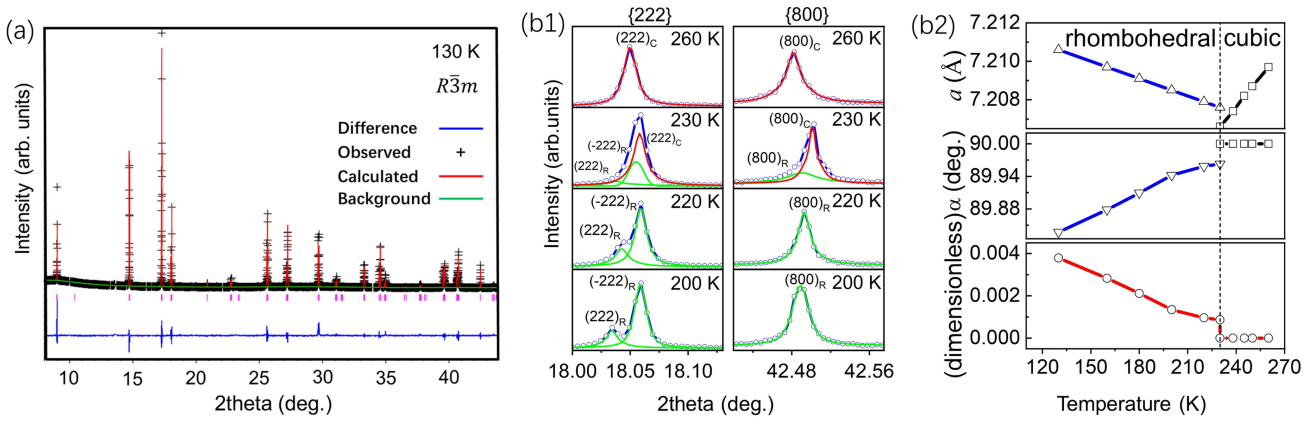

FIG. 2. (a) The refined XRD patterns for $TbCo_2$ at 130 K. The background and calculated Bragg peaks positions are shown below the observed (plus) and calculated (red line) intensities, and the difference is shown by blue line at the bottom; (b1) the characteristic XRD reflections {222} and {800} at 260 K, 230 K, 220 K and 200 K, respectively, with the red, green and blue lines denoting the cubic fit, rhombohedral fits and full sum, respectively; (b2) the lattice parameters and lattice strain within the temperature range 130 K∼260 K.

the transition at ∼230 K was incorrectly classified as second-order magnetic transition associated with a structural change[13].

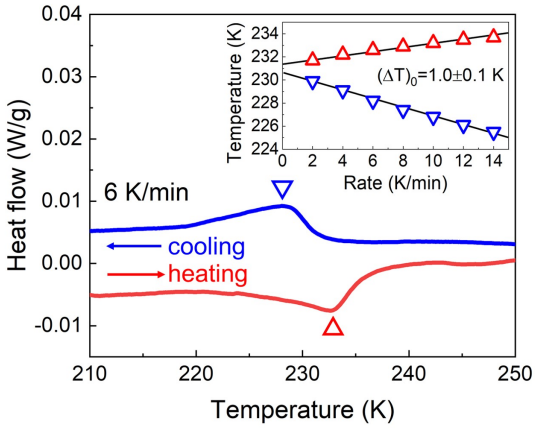

FIG. 3. Heat flow of $TbCo_2$ on heating and cooling processes at the rate of 6 K/min. The inset shows the ramp rate dependence (linear fitting) of exothermic and endothermic peak temperatures.

Since that the transition in $TbCo_2$ has been regarded second-order for long, the theoretical models for SOPT, i.e., scaling hypothesis and Heisenberg model[13], were employed for investigating the transition. Here in this work, the nature of transition of $TbCo_2$ is analyzed from the Arrott-plots method following the Banerjee criterion[26].

Fig.4 shows the magnetization isotherms and the corresponding Arrott plots (H/M-M^2) within the temperature range across T_C. From Fig.4(a), it is seen that the change of magnetization from neighboring temperatures reaches the maximum at 230 K. The negative slopes of H/M-M^2 curves are observed in Fig.4(b), demonstrating the FOPT according to Banerjees criterion[36,37]. Attention should be paid to that the negative slopes appear in the low magnetization region (inset of Fig.4(b)), where the data points might be missed if the measurement step of H is not small enough. And this may explain why the Arrott-plots method was used for $TbCo_2$ but the negative slopes were not observed[13,21,38].

An interesting fact is that, many of the previously reported "SOPT" of magnetic materials have been revised to FOPT in recent years, e.g. Fe, Ni, Co, $CoFe_2O_4$, $NdCo_2$ and $PrCo_2$[14,24,39]. This is not difficult to understand, as we will show in the following by using a Landau phenomenological approach.

Under an applied magnetic field, the change in Gibbs free energy per unit of volume (See Equ.(S1) in the Supplemental Material[32]) is modified as[40]:

$$\Delta G(M,H) = \frac{1}{2}a_0(T-T_C)M^2 + \frac{1}{4}bM^4 + \frac{1}{6}cM^6 - M\cdot H \tag{1}$$

where H is the applied magnetic field.

The equilibrium condition $\partial\Delta G(M,H)/\partial M = 0$ leads to:

$$H = a_0(T-T_C)M + bM^3 + cM^5 \tag{2}$$

Therefore, H and M follows the nonlinear relationship as Equ.(2) expresses. From the fitted curve (Fig.5(a)), the value of b is calculated to be 1.965×10^{-17} (J·m/A^4).

For the materials that the structural transition and the magnetic transition occur synchronously, strong coupling exists between the two order parameters of magnetization (primary) and lattice strain (secondary)[25], as demonstrated in the case of $TbCo_2$[41]. Then, the change of Gibbs free energy is rewritten as[24,25]:

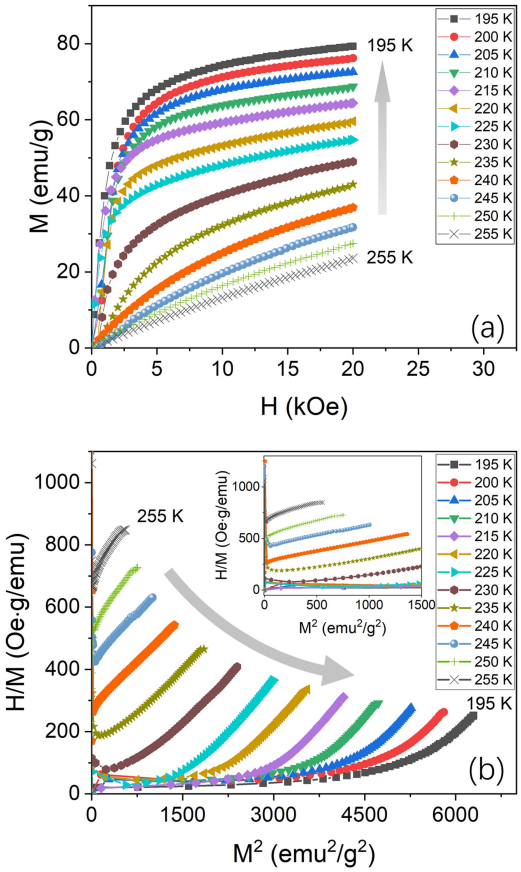

FIG. 4. (a) The magnetization isotherms measured at selected temperatures near T_C under the applied field of 2 Tesla; (b) the isotherms of H/M-M^2, the inset shows the curve of low magnetization region.

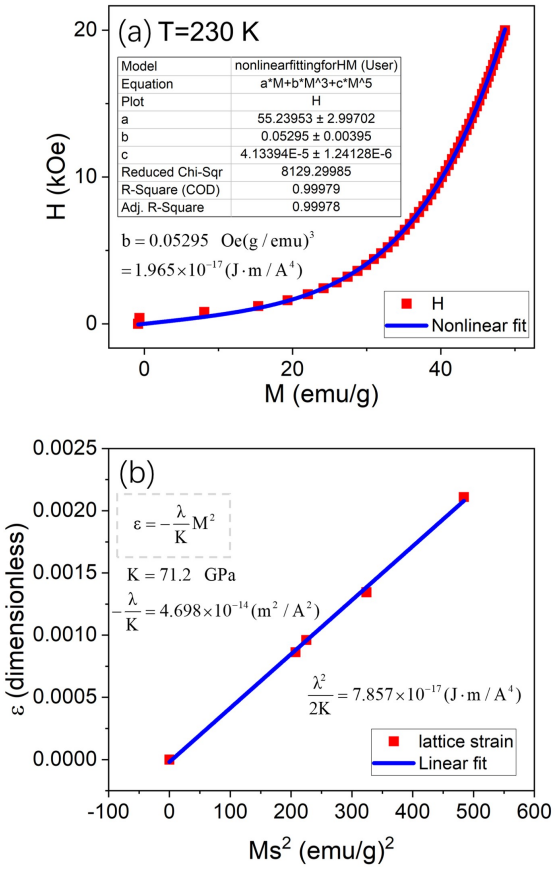

FIG. 5. (a) H versus M curve and nonlinear fitting, (b) lattice strain ε versus M^2 and linear fitting. In Fig.5(b), the values of M and ε are obtained from M-H loops and the synchrotron XRD patterns measured at 180 K, 200 K, 220 K and 230 K; the value of K (71.2 GPa) is taken from Ref.[20].

$$\Delta G(M,\varepsilon) = \frac{1}{2}a_0(T-T_C)M^2 + \frac{1}{4}bM^4 + \frac{1}{6}cM^6 + \frac{1}{2}K\varepsilon^2 + \lambda\varepsilon\cdot M^2 \quad (3)$$

where the elastic energy ($\frac{1}{2}K\varepsilon^2$) and the magnetoelastic coupling energy ($\lambda\varepsilon\cdot M^2$) are added, in which K is the elastic modulus, ε is the lattice strain and λ is the coupling coefficient.

Minimizing the energy with respect to the strain ($\partial\Delta G(M,\varepsilon)/\partial\varepsilon = 0$) yields a relation between ε and M^2:

$$\varepsilon = -\frac{\lambda}{K}M^2 \quad (4)$$

Substituting Equ.(4) into Equ.(3) leads to:

$$\Delta G = \frac{1}{2}a_0(T-T_C)M^2 + (\frac{1}{4}b - \frac{\lambda^2}{2K})M^4 + \frac{1}{6}cM^6 \quad (5)$$

Therefore, the order of the transition can be determined from the value of the coefficient of the fourth-order term ($\frac{b}{4} - \frac{\lambda^2}{2K}$)[28].

Based on the spontaneous magnetization (Ms, calculated using the law of approach to saturation (LAS)[42,43]) and lattice strain (ε, calculated from the XRD patterns (Fig.2(b2)), the linear relation $\varepsilon \sim Ms^2$ is fitted as shown in Fig.5(b). The value of $\lambda^2/2K$ is calculated to be 7.857×10^{-17} (J·m/A^4). As a result, the coefficient of the fourth-order term $\frac{b}{4} - \frac{\lambda^2}{2K} = -7.366\times10^{-17}$ (J·m/A^4), is small but indeed negative, suggesting the weak FOPT[44]. Such conclusion is consistent with that obtained from Figs.1-4.

Last but not least, it should be pointed out that the coupling of magnetization with the lattice strain, which results from the exchange interaction effect[45], would drive the magnetic ordering to become first-order[2,46,47]. If the exchange interaction is admitted being a function of lattice inter-atomic spacing and the lattice is deformable, then the transition would become FOPT, and yield latent heat as well as a spontaneous lattice strain at the transition point[48,49]. Again, we suggest the scarcity of purely second-order ferro(ferri)-magnetic transition, as long as the inevitable coupling between the magnetization and the crystal lattice exists[24].

IV. CONCLUSION

In conclusion, from the detected thermal hysteresis, latent heat, structural transition and Banerjee criterion, we show that the transition of $TbCo_2$ around 230 K is FOPT, which can be well understood based on Landau theory model. Moreover, our findings unifies the understanding on the transition of $TbCo_2$: FOPT with synchronous first-order structural transition and first-order ferrimagnetic transition. Our work may provide an insight into investigating the nature of other alleged second-order transitions for the magnetic materials that undergo both magnetic and structural transitions.

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