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Anharmonic thermodynamic properties and phase boundary across the post-perovskite transition in MgSiO₃

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Abstract

To address the effects of lattice anharmonicity across the perovskite to post-perovskite transition in MgSiO₃, we conduct calculations using the phonon quasiparticle (PHQ) approach. The PHQ is based on *ab initio* molecular dynamics and, in principle, captures full anharmonicity. Free energies in the thermodynamic limit ($N \rightarrow \infty$) are computed using temperature-dependent quasiparticle dispersions within the phonon gas model. Systematic results on anharmonic thermodynamic properties and phase boundary are reported. Both the local density approximation (LDA) and the generalized gradient approximation (GGA) calculations are performed to provide confident constraints on these properties. Anharmonic effects are demonstrated by comparing results with those obtained using the quasiharmonic approximation (QHA). The inadequacy of the QHA is indicated by its overestimation of thermal expansivity and thermodynamic Grüneisen parameter and its converged isochoric heat capacity in the high-temperature limit. The PHQ phase boundary has a Clapeyron slope (dP/dT) that increases with temperature. This result contrasts with the nearly zero curvature of the QHA phase boundary. Anharmonicity bends the phase boundary to lower temperatures at high pressures. Implications for the double-crossing of the phase boundary by the mantle geotherm are discussed.

31 I. INTRODUCTION

32 MgSiO₃ perovskite (Pv) with the *Pbnm* space group, also known as bridgmanite, is the most
33 abundant mineral in the Earth's lower mantle (LM), composing about 75 vol% of this region [1].
34 It undergoes a structural phase transition to MgSiO₃ post-perovskite (PPv) with the *Cmcm* space
35 group under the lowermost mantle conditions, i.e., above ~125 GPa and ~2500 K [2-4]. The
36 lowermost mantle with several hundred kilometers of depth, i.e., the D'' layer, is one of the least
37 understood regions in the Earth's interior [5]. It is the region where the D'' seismic discontinuity
38 is observed [6,7], and the PPv phase transition may be a reasonable interpretation of such seismic
39 discontinuity. Previous experimental measurements for the phase boundary in pure MgSiO₃ were
40 conducted up to ~3000 K, yielding significant discrepancies in the Clapeyron slopes, dP/dT
41 (~4.7–11.5 MPa/K), and the transition pressures (~113–131 GPa at 2500 K) [2,3,5,8,9]. Numerous
42 theoretical predictions for the phase boundary in MgSiO₃ were also carried out, giving various
43 Clapeyron slopes (~7.5–9.9 MPa/K) and also a significant discrepancy in the transition pressures
44 (~108–128 GPa at 2500 K) [3,4,9,10]. Previous theoretical studies of this phase boundary [3,4,9,10]
45 used the quasiharmonic approximation (QHA), which disregards intrinsic lattice anharmonic
46 effects. Therefore, the role of anharmonic effects on the phase boundary should still be investigated.

47 To capture and understand the role of anharmonic effects on the PPv phase transition, we use
48 the phonon quasiparticle (PHQ) approach [11,12] to compute the anharmonic phonon dispersions
49 and vibrational free energies. The PHQ method relies on the atomic trajectories obtained from *ab*
50 *initio* molecular dynamics (AIMD) simulations to compute the mode-projected velocity
51 autocorrelation function (VAF). This method, in principle, treats anharmonicity exactly to all
52 orders in perturbation theory. It deals with lattice anharmonicity by extracting phonon quasiparticle
53 properties, i.e., renormalized phonon frequencies and phonon lifetimes, from the mode-projected
54 VAFs. Unlike the QHA, in which phonon frequencies depend on volume only, PHQ accounts for
55 full anharmonicity and produces phonon frequencies explicitly temperature-dependent. Then the
56 vibrational free energy can be computed using these renormalized temperature-dependent
57 frequencies. The direct free energy method, such as thermodynamic integration (TI) [13], can also
58 address lattice anharmonicity. However, to approach the thermodynamic limit ($N \rightarrow \infty$),
59 performing TI using AIMD with a sufficiently large supercell is computationally unaffordable. In
60 the present approach, the renormalized frequencies are Fourier interpolated on a sufficiently dense
61 \mathbf{q} point mesh to overcome finite-size effects on the vibrational free energy. Then the vibrational

62 entropy and free energy are computed in the thermodynamic limit from the interpolated
 63 renormalized frequencies, i.e., the anharmonic phonon dispersions, within the phonon gas model
 64 (PGM) [14,15].

65 The PHQ approach was proposed and verified in Pv [11], of which irregular temperature-
 66 induced frequency shifts [16-18] observed in the Raman spectrum were successfully reproduced.
 67 In this study, we compute the anharmonic phonon dispersions of Pv and PPv using both the local
 68 density approximation (LDA) [19] and the Perdew-Burke-Ernzerhof generalized gradient
 69 approximation (PBE-GGA) [20]. Isochoric AIMD simulations at a series of V, T conditions are
 70 conducted to cover LM conditions, i.e., $23 < P < 135$ GPa, $2000 < T < 4000$ K. Systematic
 71 results on anharmonic thermodynamic properties and phase boundary across the PPv phase
 72 transition in MgSiO_3 are reported. Anharmonic effects on the thermodynamic properties and phase
 73 boundary are demonstrated by comparing to results obtained with the QHA.

74

75 II. METHOD

76 We define a phonon quasiparticle numerically by the mode-projected VAF [11,12],

$$77 \quad \langle V_{\mathbf{q}s}(0) \cdot V_{\mathbf{q}s}(t) \rangle = \lim_{\tau \rightarrow \infty} \frac{1}{\tau} \int_0^\tau V_{\mathbf{q}s}^*(t') V_{\mathbf{q}s}(t' + t) dt', \quad (1)$$

78 where

$$79 \quad V_{\mathbf{q}s}(t) = \sum_{i=1}^N \sqrt{M_i} \mathbf{v}_i(t) e^{i\mathbf{q} \cdot \mathbf{R}_i} \cdot \hat{\mathbf{e}}_{\mathbf{q}s}^i \quad (2)$$

80 is the mass-weighted mode-projected velocity for normal mode (\mathbf{q}, s) . \mathbf{q} is the phonon wave vector,
 81 and s indexes the $3n$ phonon branches of an n -atom primitive cell. M_i , \mathbf{R}_i , and \mathbf{v}_i ($i = 1, \dots, N$) are
 82 the atomic mass, the atomic equilibrium coordinate, and the atomic velocity computed by AIMD
 83 simulations of an N -atom supercell, respectively. $\hat{\mathbf{e}}_{\mathbf{q}s}^i$ is the harmonic phonon polarization vector
 84 that is determined by the harmonic phonon calculations and repeats periodically within the
 85 supercell with the periodicity of the primitive cell. \mathbf{q} is commensurate with the supercell size. For
 86 a well-defined phonon quasiparticle, the VAF can be phenomenologically described as an
 87 exponentially decaying cosine function,

$$88 \quad \langle V_{\mathbf{q}s}(0) \cdot V_{\mathbf{q}s}(t) \rangle = A_{\mathbf{q}s} \cos(\tilde{\omega}_{\mathbf{q}s} t) e^{-\Gamma_{\mathbf{q}s} t}, \quad (3)$$

89 where $A_{\mathbf{q}s}$ is the oscillation amplitude, $\tilde{\omega}_{\mathbf{q}s}$ is the renormalized phonon frequency, and $\Gamma_{\mathbf{q}s}$ is the
 90 phonon linewidth inversely proportional to the lifetime, $\tau_{\mathbf{q}s} = 1/(2\Gamma_{\mathbf{q}s})$ [11,21]. The
 91 corresponding power spectrum,

92
$$G_{\mathbf{q}_s}(\omega) = \left| \int_0^\infty \langle V_{\mathbf{q}_s}(0) \cdot V_{\mathbf{q}_s}(t) \rangle e^{i\omega t} dt \right|^2, \quad (4)$$

93 should have a Lorentzian line shape with a single peak at $\tilde{\omega}_{\mathbf{q}_s}$ and a linewidth of $\Gamma_{\mathbf{q}_s}$ [11,21]. Both
 94 Eqs. (3) and (4) can be used to extract phonon quasiparticle properties from the mode-projected
 95 VAF. In this study, only Eq. (3) was used for all quasiparticles. In this study, the obtained $\tilde{\omega}_{\mathbf{q}_s}$ are
 96 used to compute the anharmonic thermodynamic properties and phase boundary, while $\tau_{\mathbf{q}_s}$ can be
 97 used to evaluate the lattice thermal conductivity [22]. To overcome the finite-size effects on the
 98 vibrational free energy and obtain the free energy in the thermodynamic limit, the anharmonic
 99 phonon spectrum is further calculated via Fourier interpolation for $\tilde{\omega}_{\mathbf{q}_s}$ [11,12]. Within the PGM,
 100 the vibrational entropy formula with temperature-dependent anharmonic phonon spectrum is
 101 applicable [11,12,21],

102
$$S_{\text{vib}}(T) = k_B \sum_{\mathbf{q}_s} [(n_{\mathbf{q}_s} + 1) \ln(n_{\mathbf{q}_s} + 1) - n_{\mathbf{q}_s} \ln n_{\mathbf{q}_s}], \quad (5)$$

103 where $n_{\mathbf{q}_s} = [\exp(\hbar\tilde{\omega}_{\mathbf{q}_s}(T)/k_B T) - 1]^{-1}$. $\tilde{\omega}_{\mathbf{q}_s}(T)$ at arbitrary temperatures were obtained by
 104 fitting $\tilde{\omega}_{\mathbf{q}_s}$ at several temperatures and constant volume to a second-order polynomial in T
 105 [11,23,24]. Then for insulators such as Pv and PPv, the Helmholtz free energy $F(T)$ at any T can
 106 be obtained by integrating the vibrational entropy [11,25,26],

107
$$F(T) = E_0 + \frac{1}{2} \sum_{\mathbf{q}_s} \hbar\omega_{\mathbf{q}_s} - \int_0^T S_{\text{vib}}(T') dT'. \quad (6)$$

108 E_0 is the static energy and $\frac{1}{2} \sum_{\mathbf{q}_s} \hbar\omega_{\mathbf{q}_s}$ is the zero-point energy, where $\omega_{\mathbf{q}_s}$ is the harmonic phonon
 109 frequency, i.e., the phonon frequency at zero temperature. The latter two terms on the right-hand
 110 side compose the vibrational free energy.

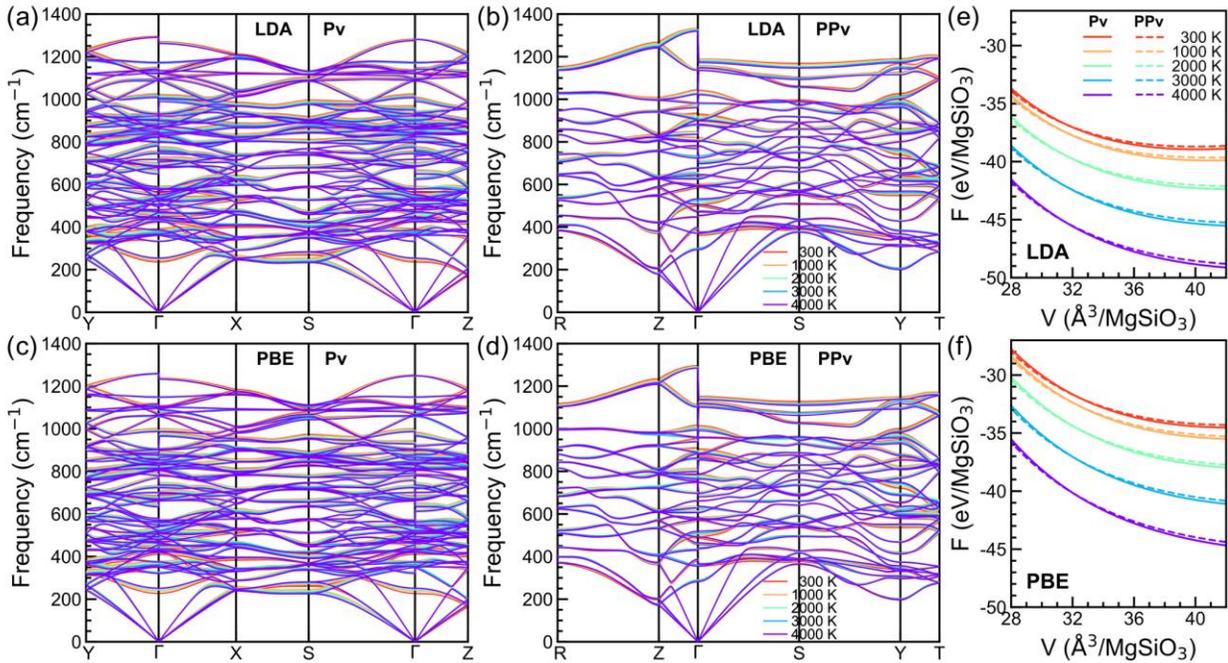
111 We performed AIMD simulations using the projected-augmented wave method [27] as
 112 implemented in VASP [28]. The electron exchange-correlation functional (XC) was treated by
 113 both the LDA [19] and the PBE [20]. Pv and PPv were simulated with 160-atom supercells ($2 \times$
 114 2×2) and 180-atom supercells ($3 \times 3 \times 2$), respectively, both with a Γ \mathbf{k} point sampling. The
 115 supercells are sufficiently large to converge the phonon quasiparticle properties as reported by
 116 previous studies [11,22,29]. For each phase and each XC, MD simulations were carried out in the
 117 NVT ensemble at a series of 5 volumes and a series of 6 temperatures between 300 and 5000 K
 118 controlled by the Nosé thermostat [30,31]. Each MD ran for over 50 ps with a time step of 1 fs. To
 119 approach the thermodynamic limit, anharmonic phonon spectra were evaluated with much denser
 120 \mathbf{q} meshes ($16 \times 16 \times 8$ for Pv and $20 \times 20 \times 10$ for PPv), sufficiently dense to converge the

121 vibrational free energy. Static calculations were conducted using the well-converged $4 \times 4 \times 4$
 122 and $8 \times 8 \times 4$ \mathbf{k} point sampling for Pv and PPv, respectively.

123

124 III. RESULTS AND DISCUSSION

125 Temperature-dependent anharmonic phonon dispersions at constant volume are displayed in
 126 Figs. 1(a)–1(d). Long-range dipole-dipole interaction causing LO-TO splitting was included in the
 127 calculations. Dielectric constants and Born effective charges were computed using density-
 128 functional perturbation theory (DFPT) [32]. The discontinuity in the phonon dispersions at Γ
 129 mainly caused by the directional dependence of the Born effective charge tensors for anisotropic
 130 materials. Temperature-induced frequency shifts are discernible but small for Pv and PPv, meaning
 131 that both phases are weakly anharmonic. Using such anharmonic phonon frequencies interpolated
 132 on a dense \mathbf{q} -mesh, the Helmholtz free energy $F(T)$ at constant volume was computed by
 133 integrating the vibrational entropy via Eqs. (5) and (6). Then, at each temperature, the isothermal
 134 equation of state (EOS) was obtained by fitting $F(V)$ at several volumes to a third-order finite
 135 strain expansion [23,24]. The resulting $F(V, T)$ are shown in Figs. 1(e) and 1(f).

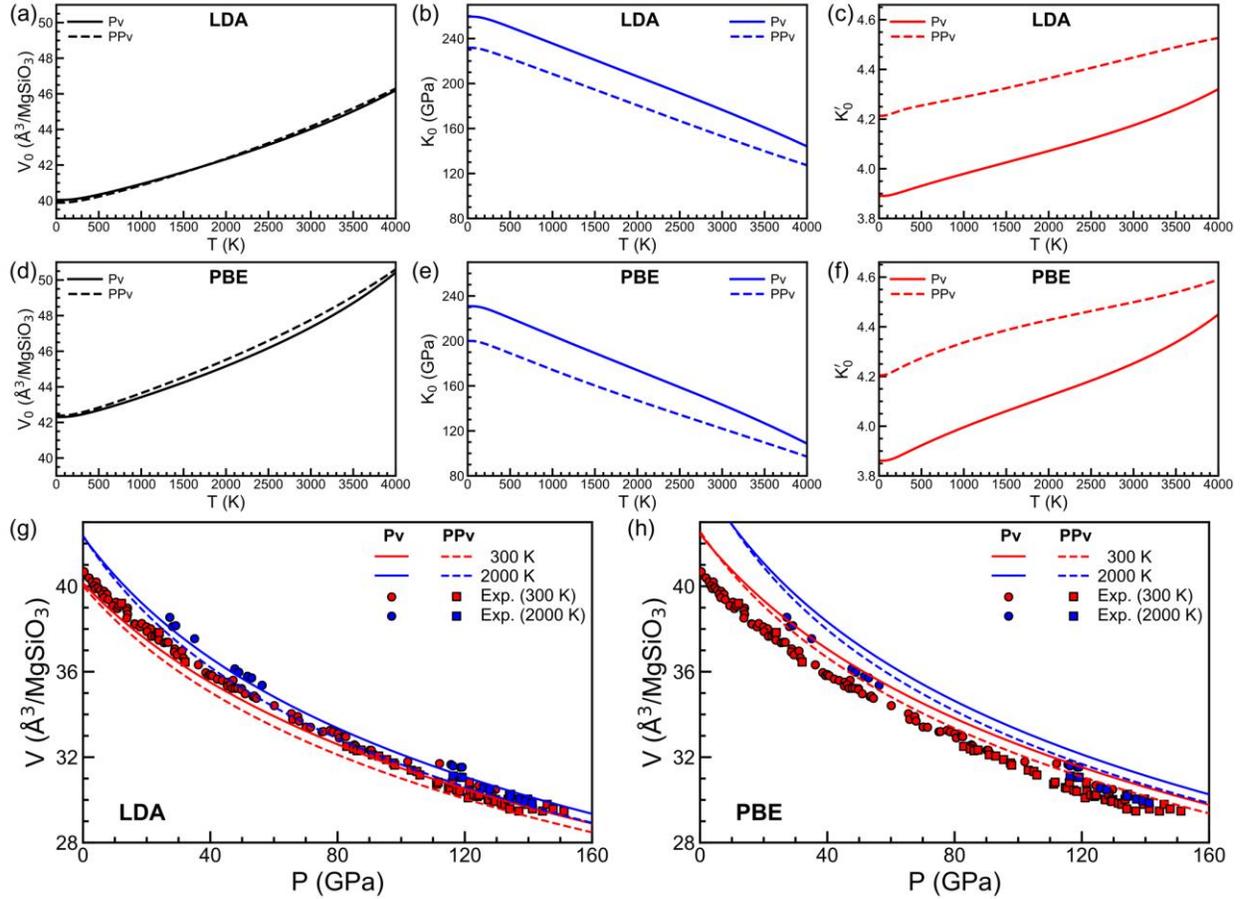


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137 FIG. 1. (a)–(d) Temperature-dependent anharmonic phonon dispersions obtained by the (a)(b)
 138 LDA and (c)(d) PBE for (a)(c) Pv and (b)(d) PPv. Results are shown at their respective volumes
 139 with a static pressure of 120 GPa. (e)(f) Helmholtz free energy $F(V, T)$ vs. volume at different

140 temperatures. Results are obtained by the (e) LDA and (f) PBE for Pv (solid curves) and PPv
141 (dashed curves).

142
143 The obtained temperature-dependent third-order Burch-Murnaghan EOS parameters, i.e.,
144 equilibrium volume (V_0), bulk modulus (K_0), and pressure derivative of the bulk modulus (K'_0) are
145 displayed in Figs. 2(a)–2(f). The shown EOS parameters can be used to reproduce the P - V - T data
146 under LM conditions in this study. The choice of different EOS, e.g., Vinet, may produce different
147 EOS parameters but does not change the reported P - V - T data, thermodynamics, and phase
148 boundary. For both phases and both XC's, as temperature increases, K_0 decrease whereas K'_0
149 increase. For both phases, V_0 obtained by the LDA are smaller than those obtained by the PBE,
150 while K_0 obtained by the LDA are larger than those obtained by the PBE. For both XC's, K_0 of Pv
151 are larger than those of PPv, while K'_0 of Pv are smaller than those of PPv. The calculated EOS
152 parameters at 300 K are summarized and compared to reported experimental measurements in
153 TABLE I. The equilibrium volumes at room temperature predicted by the LDA are smaller than
154 experiments for both Pv and PPv, while results by the PBE are significantly larger than
155 experiments. The bulk moduli at room temperature predicted by the LDA agree well with
156 experiments for both phases, whereas predictions by the PBE are smaller than experiments.
157 Therefore, the LDA reproduces the EOS better than the PBE for both phases. This is further
158 justified by comparing the volumes at higher pressures and temperatures, as shown in Figs. 2(g)
159 and 2(h). The volumes at 300 and 2000 K predicted by the LDA are smaller than experimental
160 measurements. In contrast, the volumes predicted by the PBE are significantly larger than in
161 experiments.



162

163 FIG. 2. (a)–(f) Temperature-dependent third-order Birch-Murnaghan EOS parameters, i.e.,

164 equilibrium volume (V_0), bulk modulus (K_0), and pressure derivative of the bulk modulus (K'_0).

165 Results are obtained with the (a)–(c) LDA and (d)–(f) PBE for Pv (solid curves) and PPv (dashed

166 curves). (g)(h) EOS at 300 (red) and 2000 K (blue) compared with reported experimental

167 measurements (circles for Pv and squares for PPv). Results are obtained with the (g) LDA and (h)

168 PBE for Pv (solid curves) and PPv (dashed curves). Pv experimental data are taken from [33-42]

169 at 300 K and [38,39,41] at 2000 K. PPv experimental data are taken from [2,36,41,43,44] at 300

170 K and [41,44] at 2000 K.

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175 TABLE I. Calculated EOS parameters at 300 K compared to experimental measurements for Pv

176 [34-36,45-50] and PPv [36,43,44].

		V_0 ($\text{\AA}^3/\text{MgSiO}_3$)	K_0 (GPa)	K'_0
Pv	LDA	40.15	255.72	3.91
	PBE	42.44	226.47	3.89
	Exp.	40.58–40.83	246–272	3.65–4.00
PPv	LDA	40.02	227.43	4.24
	PBE	42.56	194.91	4.24
	Exp.	40.55–41.23	219–248	4.00 (fixed)

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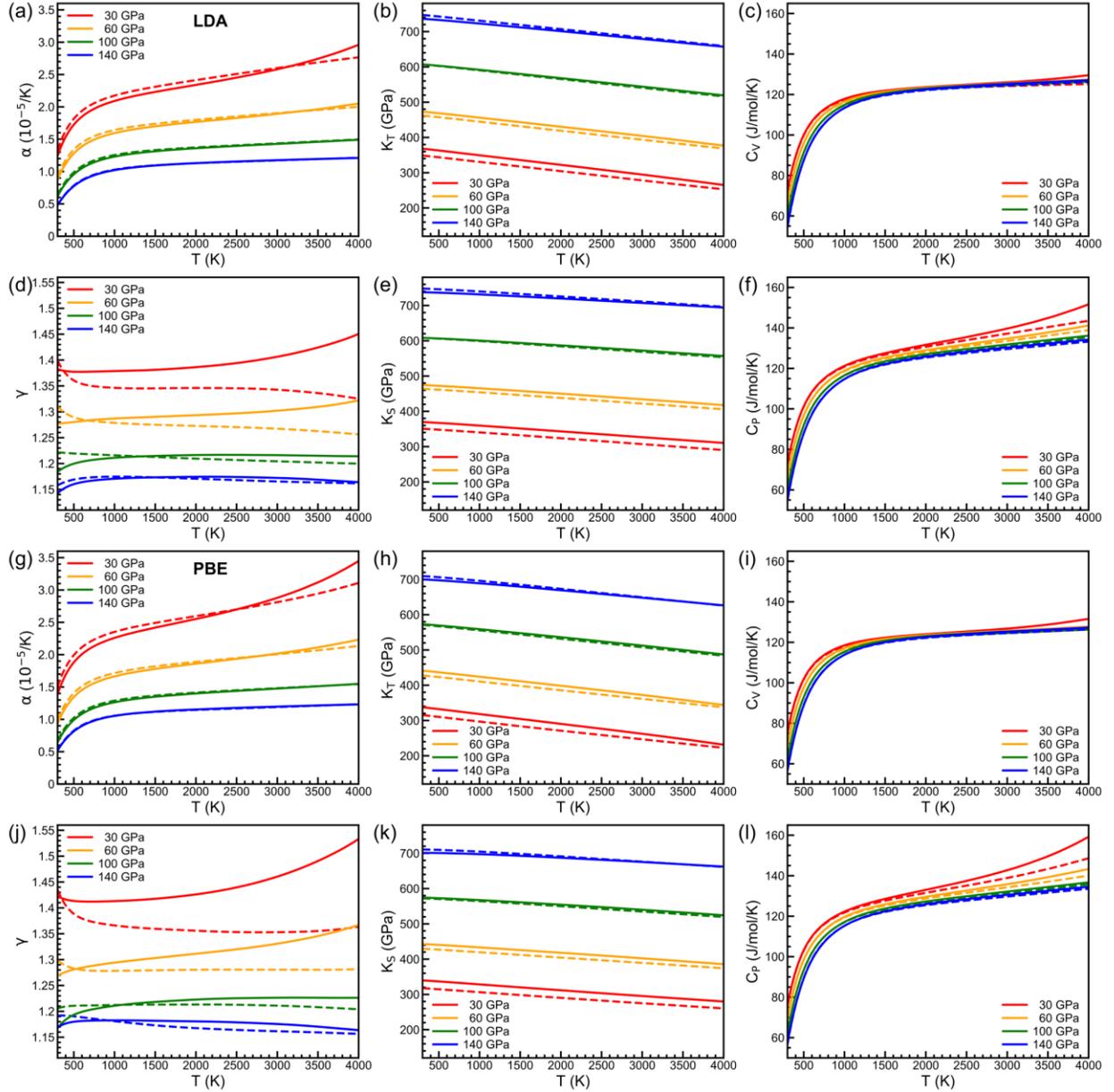
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With fully anharmonic $F(V, T)$, anharmonic thermodynamic quantities are readily calculated. Pressure, thermal expansivity (α), isothermal bulk modulus (K_T), isochoric heat capacity (C_V), thermodynamic Grüneisen parameter (γ), adiabatic bulk modulus (K_S), and isobaric heat capacity (C_P) are obtained from $P = -\left(\frac{\partial F}{\partial V}\right)_T$, $\alpha = \frac{1}{V}\left(\frac{\partial V}{\partial T}\right)_P$, $K_T = -V\left(\frac{\partial P}{\partial V}\right)_T$, $C_V = T\left(\frac{\partial S}{\partial T}\right)_V$, $\gamma = \frac{V\alpha K_T}{C_V}$, $K_S = K_T(1 + \gamma\alpha T)$, and $C_P = C_V(1 + \gamma\alpha T)$, respectively. Systematic results on the anharmonic thermodynamic properties for both phases using both XCs are shown in Fig. 3. For both XCs, γ and C_P of Pv are generally larger than those of PPv, especially at high temperatures and low pressures. For both phases, the LDA yields smaller α but larger K_T and K_S than the PBE. All reported quantities are essential in geodynamic modeling. In particular, α and γ are helpful indicators of the relative importance of anharmonicity compared to the QHA results [51-54]. K_T and K_S , along with other thermoelastic properties, are essential in interpreting seismic tomography [55]. C_V and C_P are needed in theory and experiments to evaluate the lattice thermal conductivity [56,57].



192

193 FIG. 3. Thermodynamic quantities, i.e., thermal expansivity (α), isothermal bulk modulus (K_T),
 194 isochoric heat capacity (C_V), thermodynamic Grüneisen parameter (γ), adiabatic bulk modulus
 195 (K_S), and isobaric heat capacity (C_P) vs. temperature at different pressures. Results are obtained
 196 by the (a)–(f) LDA and (g)–(l) PBE for Pv (solid curves) and PPv (dashed curves).

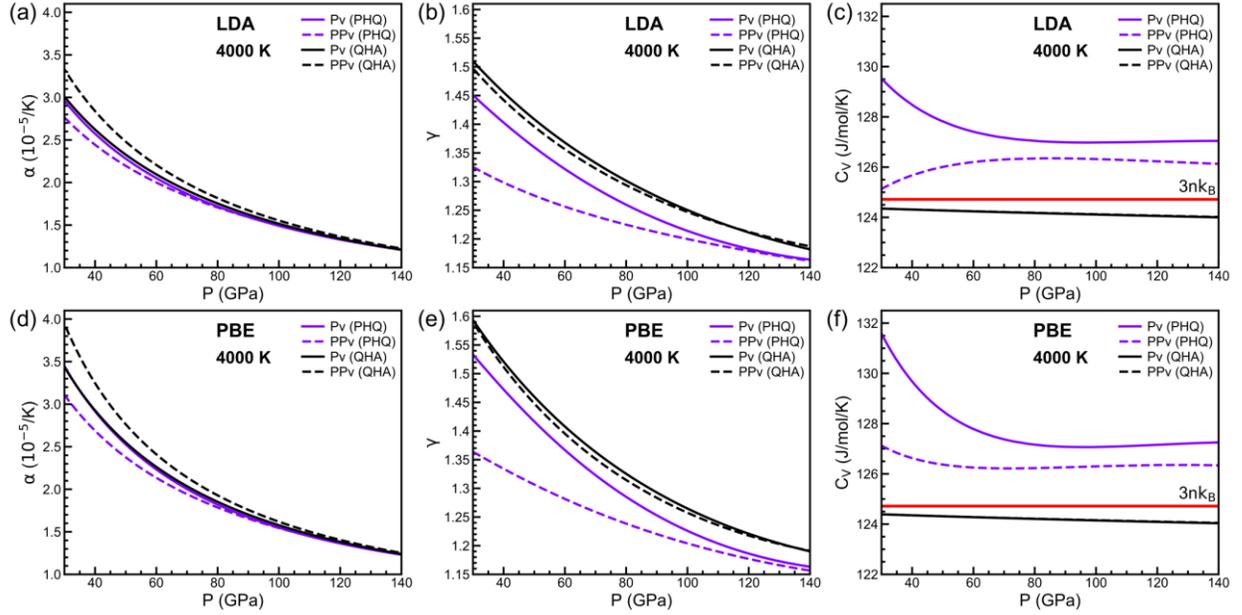
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198 Anharmonic effects on the thermodynamic properties are further analyzed by comparing α , γ ,
 199 and C_V at high temperatures, e.g., 4000 K with results obtained by the QHA, as displayed in Fig.
 200 4. It is known that the overestimation of α compared to experiments, especially at high

201 temperatures and low pressures, is the fingerprint of the QHA's shortcomings [51-54]. For Pv, α
202 predicted by the QHA agrees very well with that by the PHQ, even down to 30 GPa at 4000 K.
203 This means the QHA works quite well for Pv. For PPv, in contrast, the QHA overestimates α
204 compared to the PHQ results, and the overestimation becomes more and more significant at low
205 pressures. Note that in the QHA, the temperature effects are accounted for by extrinsic volumetric
206 effects only, i.e., quasi-harmonic thermal expansivity [58]. Whereas in the PHQ, anharmonicity is
207 treated as the intrinsic temperature dependence of phonon frequencies and is, in principle, fully
208 included by the method. Our direct comparison between the QHA and PHQ results justifies the
209 statement about α as an indicator of the QHA's shortcomings [51-54].

210 γ is another important indicator of the importance of anharmonicity [52]. The inadequacy of
211 the QHA is also seen in its overestimation of γ compared to experiments for other minerals [52,54].
212 For Pv, the QHA overestimates γ compared to the PHQ, yet the difference remains roughly
213 constant as pressure decreases. For PPv, however, the overestimation of γ by the QHA becomes
214 significant at low pressures. Such observation is in line with that for α , both of which lead to the
215 conclusion that the QHA works better for Pv than PPv and the QHA becomes inadequate at high
216 temperatures and low pressures. In addition, the QHA γ of both phases show positive temperature
217 dependence at constant pressure and above ~ 300 K (classical regime) [59,60]. Whereas the PHQ
218 γ of both phases are nearly temperature-independent and that of PPv even shows slightly negative
219 temperature dependence (see Figs. 3(d) and 3(j)) under similar conditions. Such features were
220 observed by experiments for other minerals, in contrast to the strong positive temperature
221 dependence produced by the QHA [52]. These features in the PHQ calculations also demonstrate
222 the effectiveness of the approach.

223 The insufficiency of the QHA and the anharmonic effects captured by the PHQ is further seen
224 in the comparison of C_V . At high temperatures, C_V obtained using the QHA converges to an upper
225 limit of $3nk_B$. This is because $3nk_B$ is the high-temperature limit within the Debye model [61] for
226 harmonic crystals with temperature-independent frequencies. In contrast, C_V obtained using the
227 temperature-dependent frequencies from the PHQ can exceed this limit at high temperatures
228 [23,62,63]. Note that C_V obtained by the QHA at high temperatures is subject to the mathematical
229 limit. The relatively larger difference between the QHA and PHQ results for Pv's C_V does not
230 contradict the previous conclusion that the QHA works better for Pv than PPv.



231
 232 FIG. 4. Comparison of (a)(d) α , (b)(e) γ , and (c)(f) C_V computed with the PHQ approach (purple)
 233 and the QHA (black) vs. pressure at 4000 K. Results are obtained by the (a)–(c) LDA and (d)–(f)
 234 PBE for Pv (solid curves) and PPv (dashed curves). The red horizontal lines label the high-
 235 temperature limit, $3nk_B$, within the Debye model.

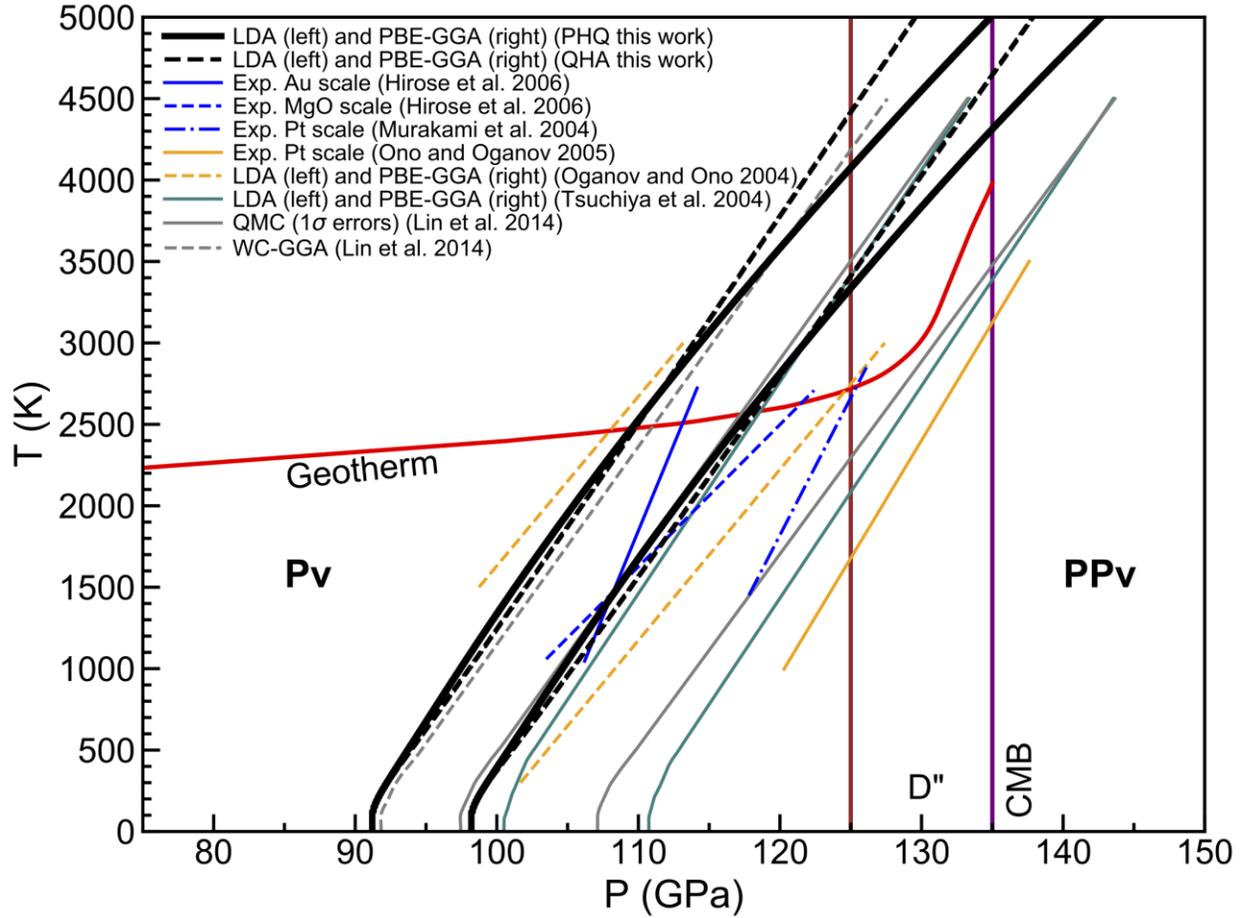
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237 The Helmholtz free energy is then converted into the Gibbs free energy, $G = F + PV$. By
 238 comparing G of Pv and PPv, phase boundaries using both XC's are obtained and shown in Fig. 5.
 239 QHA results are also displayed to demonstrate anharmonic effects on the phase boundary.
 240 Previously reported experimental measurements and theoretical predictions are also shown. The
 241 discrepancies among the reported experimental results are significant, with transition pressures
 242 ranging from ~ 113 – 131 GPa at 2500 K and Clapeyron slopes ranging from ~ 4.7 – 11.5 MPa/K
 243 [2,3,5,8,9]. This is mainly caused by the different pressure scales used in the measurements [5,8],
 244 e.g., Au scales [64–67], MgO scale [68], and Pt scale [69]. The discrepancies among the reported
 245 theoretical results are also significant, with transition pressures ranging from ~ 108 – 128 GPa at
 246 2500 K and Clapeyron slopes ranging from ~ 7.5 – 9.9 MPa/K [3,4,9,10]. Although the LDA gives
 247 better EOS than the GGA, the GGA (PBE-GGA [3,9,58] or WC-GGA [10]) has been reported to
 248 reproduce better polymorphic phase boundaries than the LDA [3,9,10,58].

249 We obtain transition pressures to be 110 and 117 GPa at 2500 K by the LDA and the PBE-
 250 GGA, respectively. LDA and PBE-GGA phase boundaries in this study agree relatively well and

251 fall in between the two boundaries predicted by the same functionals by Oganov and Ono [3,9].
252 Our transition pressures are considerably smaller than those reported by Tsuchiya *et al.* [4] for the
253 same functionals. This may be caused by either the use of an under-converged $4 \times 4 \times 2$ **k** point
254 sampling for PPv or the use of different software, i.e., Quantum ESPRESSO [70] by Tsuchiya *et*
255 *al.* [4]. Our calculated PBE-GGA phase boundary agrees better with experiments [2,3,5,8,9] than
256 the LDA, which is in line with previous reports [3,9,10,58]. All previous calculations [3,4,9,10]
257 obtained thermodynamic properties using the QHA, and the predicted phase boundaries above
258 ~ 500 K have zero curvature. The PHQ phase boundary shows a curvature and deviates more and
259 more from the QHA boundary with increasing temperature. The calculated Clapeyron slopes by
260 both methods and both XCs are summarized in TABLE II. The QHA gives a temperature-
261 independent 8.0 MPa/K by averaging the LDA and PBE results, in agreement with previous
262 calculations [3,4,10]. The PHQ Clapeyron slope increases with temperature. The average values
263 of the slopes by the two XCs are 7.6 MPa/K at 1000 K, 9.0 MPa/K at 2500 K, and 10.3 MPa/K at
264 4000 K. The large values of the Clapeyron slope at high temperatures are in better agreement with
265 the measurements by Hirose *et al.* [5,8] using the MgO scale [68].

266 The last column of TABLE II summarizes the calculated transition temperatures at the core-
267 mantle boundary (CMB) by both methods and both XCs. The QHA LDA phase boundary beyond
268 5000 K was linearly extrapolated. The solid-liquid phase transition of MgSiO_3 [71-74] was not
269 considered and is not in the scope of this study. By averaging the LDA and PBE results, the PHQ
270 gives a transition temperature of 4660 K at the CMB. The accumulated errors in the QHA lead to
271 an average overestimation of 510 K, i.e., 5170 K at the CMB. Our PHQ result is in fairly good
272 agreement with a recently reported experimental transition temperature of 4800 K at the CMB [75].
273 Such temperatures are much higher than the present-day CMB temperature of ~ 4000 K (3600–
274 4300 K [76-78]). Therefore, the geotherm [79] crosses the calculated phase boundary only once at
275 ~ 2500 K. The double-crossing hypothesis [80] states that the back transformation from PPv to Pv
276 may occur within the D'' layer above the CMB. Our results show the double-crossing hypothesis
277 does not hold for pure MgSiO_3 . Nevertheless, anharmonicity bends the phase boundary to lower
278 temperatures at high pressures, and anharmonic effects facilitate the double-crossing scenario [80].
279 Additional shifts [81,82] and broadening [81-87] of the phase boundary in mantle occurring alloys
280 among MgSiO_3 , Al_2O_3 , FeSiO_3 , and Fe_2O_3 still need to be carefully and consistently addressed to
281 shed more light on the double-crossing issue.



282
 283 FIG. 5. Pv-PPv phase boundaries obtained in this work with the PHQ (solid black curves) and the
 284 QHA (dashed black curves) under the LM conditions are compared with reported experimental
 285 measurements [2,3,5,8,9] and previous theoretical predictions [3,4,9,10] in pure MgSiO₃. The two
 286 solid gray curves by the quantum Monte Carlo (QMC) show the boundaries of a 1 σ error band
 287 [10]. The red curve shows the geotherm [79]. The vertical brown and purple lines show the onset
 288 of the D'' region and the CMB, respectively.

289
 290
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 292
 293
 294 TABLE II. Calculated Clapeyron slopes at different temperatures and transition temperatures at
 295 the CMB.

		Clapeyron slope (MPa/K)			Transition temperature (K)
		at 1000 K	at 2500 K	at 4000 K	at the CMB
PHQ	LDA	7.6	9.0	10.3	5010
	PBE	7.7	9.0	10.4	4310
QHA	LDA	7.9	7.9	7.9	5690
	PBE	8.1	8.1	8.1	4650

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298 IV. CONCLUSIONS

299 Using the phonon quasiparticle (PHQ) approach [11,12], we computed temperature-dependent
300 anharmonic phonon quasiparticle dispersions of MgSiO₃ perovskite (Pv) and post-perovskite
301 (PPv). Both the LDA [19] and PBE [20] were used to provide confident constraints on the
302 thermodynamic properties and phase boundary for the Pv-PPv transition. Fully anharmonic free
303 energies were calculated in the thermodynamic limit ($N \rightarrow \infty$) within the phonon gas model
304 [14,15]. Thermal equation of state, thermal expansivity (α), thermodynamic Grüneisen parameter
305 (γ), isothermal (K_T) and adiabatic (K_S) bulk moduli, isochoric (C_V) and isobaric (C_P) heat
306 capacities, and phase boundary were then obtained. Comparing PHQ and QHA results, we see that
307 α and γ are overestimated by the QHA, especially for PPv at high temperatures and low pressures.
308 At high temperatures, QHA C_V converges to the high-temperature limit of $3nk_B$, while the PHQ
309 C_V can exceed this limit. The PPv phase transition pressure is 114 GPa at 2500 K by averaging the
310 LDA and PBE results. Anharmonic effects produce a phase boundary with non-zero curvature
311 above 500 K. The Clapeyron slope (dP/dT) increases with temperature, e.g., 7.6 MPa/K at 1000
312 K, 9.0 MPa/K at 2500 K, and 10.3 MPa/K at 4000 K. The transition temperature at the core-mantle
313 boundary (CMB) is 4660 K, which is 510 K lower than the QHA prediction, yet still much higher
314 than the expected present-day CMB temperature. Hence, the geotherm [79] crosses the phase
315 boundary only once at ~ 2500 K, and the double-crossing phenomenon [80] should not happen in
316 pure MgSiO₃. Additional Al and Fe alloying effects on the Pv-PPv transition must still be
317 considered before the geotherm's double-crossing of the phase boundary in natural mantle
318 composition can be addressed [81-87].

319

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