Electronic structures, charge transfer, and charge order in twisted transition metal dichalcogenide bilayers

Yang Zhang, Tongtong Liu, and Liang Fu

Electronic structures, charge transfer and charge orders in twisted transition metal dichalcogenide bilayers

Yang Zhang, Tongtong Liu, and Liang Fu

Department of Physics, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139, USA

Moiré superlattices of transition metal dichalcogenide (TMD) bilayers have been shown to host correlated electronic states, which arises from the interplay of emergent moiré potential and long-range Coulomb interactions. Here we theoretically investigate structural relaxation and single-particle electronic properties in moiré superlattices of transition metal dichalcogenide homobilayer and study the ground state charge orders in the effective honeycomb lattice of MX and XM regions. From the large-scale density functional theory calculation and continuum model with layer degrees of freedom, we find that the out-of-plane gating field creates a tunable charge transfer gap and introduces a mass term in the Dirac spectrum. At the flat band limit, we observe a series of charge-ordered insulating states at various fillings $n = 1/4, 1/3, 1/2, 2/3, 1$ with Monte Carlo simulations, and predict that gating field induces a phase transition between different electron crystals at fixed filling $n = 1/2$ or $2/3$. At half-filling n=1, the ground state is a Mott insulator with electronically driven ferroelectricity. Our work demonstrates that transition metal dichalcogenide homobilayer provides a powerful platform for the investigation of tunable charge transfer insulator and charge orders.

Moiré superlattices are a fruitful platform for realizing and controlling correlated electron states, as evidenced by the remarkable success in twisted bilayer graphene (TBG) [1–12] and trilayer graphene-hBN heterostructure[13–16]. Recently a new family of moiré materials based on transition metal dichalcogenides (TMD) [17–29] have attracted great interest. They host an abundance of correlated insulating states at a series of fractional fillings [30–34].

In TMD bilayers, moiré bands are formed from parabolic bands of individual layers. In twisted TMD homobilayers, the moiré bandwidth can be made arbitrarily small by reducing the twist angle, which gives rise to strong correlation without fine tuning. Electrons or holes in these moiré bands are tightly localized in high-symmetry stacking regions, which can be well described by a simple effective tight-binding model. This description offers a convenient starting point for investigating interaction-induced states at finite density. Despite the conceptual simplicity, a quantitative modeling of moiré bands in TMD is highly nontrivial. For example, the moiré bandwidth of TMD heterobilayer WSe_2/WS_2 is only on the order of 10 meV, and depends highly on the lattice relaxation [30, 31, 35, 36].

In this work, using the large-scale density functional theory, continuum model approach and Monte Carlo simulation, we study the effect of structural relaxation and electric field on the moiré band structure in twisted TMD homobilayers and predict novel charge orders at fractional fillings in the strong-coupling regime. We focus on the moiré valence bands originating from the Γ pocket [37–41]. Due to interlayer tunneling and lattice relaxation, these moiré bands are derived from localized orbitals in MX and XM stacking regions that form a honeycomb lattice. We find a pair of massless Dirac fermions at K, K' points of the mini Brillouin zone (BZ), which is protected by the D_3 point group symmetry of the moiré superlattice. Applying an out-of-plane electric field

breaks the sublattice symmetry of the honeycomb lattice and opens a tunable gap Δ at the Dirac point. We introduce a new continuum model for twisted TMD homobilayers, which captures the layer degrees of freedom and the electrically tunable gap.

We further use an extended Hubbard model on the honeycomb lattice and perform Monte Carlo simulations to study the insulating electron crystals in the flat band limit. We find a distinctive set of charge orders at hole fillings $n = 1/4, 1/3, 1/2, 2/3, 1$ on the honeycomb lattice. Interestingly, the charge orders at $n = 1/2$ and $2/3$ both break the rotational symmetry and differ from the proposed states in the WSe_2/WS_2 heterobilayer. And the $n = 1$ insulating state has a spontaneous out-of-plane ferroelectric polarization, which can be switched by the electric field. These symmetry breaking charge orders can be directly probed by the optical anisotropy experiments [34, 42]. Moreover, we predict that phase transitions between distinct charge-ordered states at the same filling can be induced by the electric field, which tunes the charge-transfer gap Δ. Our work shows that twisted homobilayer MoS_2 provides an ideal platform for investigating electrically tunable charge transfer gap and charge orders.

We study TMD homobilayers with a small twist angle starting from AA stacking, where every metal (M) or chalcogen (X) atom on the top layer is aligned with the same type of atom on the bottom layer [58]. Within a local region of a twisted bilayer, the atom configuration is identical to that of an untwisted bilayer, where one layer is laterally shifted relative to the other layer by a corresponding displacement vector $\boldsymbol{d}_0$. For this reason, the moiré band structures of twisted TMD bilayers can be constructed from a family of untwisted bilayers at various $\boldsymbol{d}_0$, all having 1×1 unit cell. Our analysis thus starts from untwisted bilayers [43].

In particular, $\boldsymbol{d}_0 = 0, -(\boldsymbol{a}_1 + \boldsymbol{a}_2)/3, (\boldsymbol{a}_1 + \boldsymbol{a}_2)/3$, where $\boldsymbol{a}_{1,2}$ is the primitive lattice vector for untwisted bi-

layers, correspond to three high-symmetry stacking configurations of untwisted TMD bilayers, which we refer to as MM, XM, MX. In MM (MX) stacking, the M atom on the top layer is locally aligned with the M (X) atom on the bottom layer, see Fig. 1a. Likewise for XM. The bilayer structure in these stacking configurations is invariant under three-fold rotation around the z axis.

In homobilayer TMD, the spin degenerate Γ pockets in the valence band arise from electron tunneling between the two layers. The $k \cdot p$ Hamiltonian takes the form:

$$\mathcal{H}\left(\boldsymbol{d}_0\right) = \begin{pmatrix} -\frac{\hbar^2 k^2}{2m^*} + \epsilon_b\left(\boldsymbol{d}_0\right) & \Delta_T\left(\boldsymbol{d}_0\right) \\ \Delta_T^\dagger\left(\boldsymbol{d}_0\right) & -\frac{\hbar^2 k^2}{2m^*} + \epsilon_t\left(\boldsymbol{d}_0\right) \end{pmatrix}. \quad (1)$$

Here $m^* = 1.07 m_e$ is the effective mass for the valence band. $\Delta_T\left(\boldsymbol{d}_0\right)$ is the interlayer tunnelling amplitude which depends on the in-plane displacement between the two layers. In contrast to the complex tunneling amplitude for the K pockets [44], here the time-reversal symmetry at Γ pocket enforces $\Delta_T\left(\boldsymbol{d}_0\right)$ to be real. The potential term $\epsilon_{b,t}\left(\boldsymbol{d}_0\right)$ denotes the energy of the valence band maximum in the absence of tunneling, which arises from the unequal layer weight of the wavefunction at MX and XM stacking configuration.

We expand $\Delta_T\left(\boldsymbol{d}_0\right)$ in Fourier components up to the second harmonic term:

$$\Delta_T\left(\boldsymbol{d}_0\right) = w_0 + 2w_1 \sum_{j=1}^3 \cos(\boldsymbol{G}_j \cdot \boldsymbol{d}_0) + 2w_2 \sum_{j=1}^3 \cos(2\boldsymbol{G}_j \cdot \boldsymbol{d}_0), \quad (2)$$

where $G_i (i = 1, 2, 3)$ are the three reciprocal lattice vectorc in monolayer TMD. Due to three-fold rotational symmetry, Δ_T is a local extremum for MM, MX and MX stackings, with $\Delta_T = w_0 + 6w_1 + 6w_2$ for $d_0 = 0$ (MM) and $w_0 - 3w_1 - 3w_2$ for $d_0 = \pm\left(\boldsymbol{a}_1 + \boldsymbol{a}_2\right)/3$ (MX or XM). The zero-momentum-transfer tunneling term w_0 is responsible for the large bonding and antibonding energy splitting for all d_0, while w_1, w_2 capture the variation of the tunneling amplitude at different lateral displacements.

The interlayer tunneling strength depends significantly on the layer spacing d. From the DFT calculation, we find the equilibrium layer spacing of untwisted TMD bilayers in MM, MX and XM stackings: $d_{MM} = 6.63$ Angstroms and $d_{MX} = d_{XM} = 5.97$ Angstroms. The 10% variation of layer spacing is comparable with that in bilayer graphene [45] and strongly impacts the energy splitting of Γ pockets.

By calculating the work function, we plot in Fig. 1 the band structure of MM and MX-stacked bilayers, with reference energy $E = 0$ chosen to be the absolute vacuum level. Using the relaxed layer spacings, we find the energy splitting in MX (or XM) stacking to be stronger than in MM, as a result of its smaller layer distance. From the different energy splitting at Fig. 1c, we obtain the tunnelling parameters as $w_0 = 338$ meV, $w1 + w2 = -18$ meV. If the same layer spacing were used for both MX and MM bilayers, the opposite (and incorrect) conclusion about the energy splitting would be found, see Fig.1b.
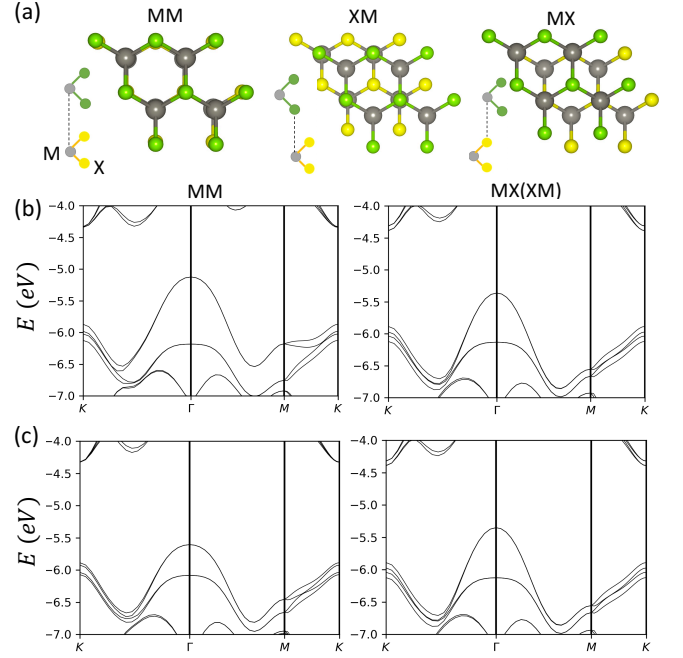

FIG. 1: (a) Lattice structure of MM, MX, XM spots for AA stacking heterobilayer, M stands for metal atom and X stands for chalcogen atom (Green for the top layer, yellow for the bottom layer). DFT band structures of MM and MX(XM) stacking homobilayer in (b) MoS_2/MoS_2 with identical layer spacing; (c) MoS_2/MoS_2 with relaxed layer spacing.

Thus lattice relaxation is crucial for obtaining the correct moiré band structure.

The structure of twisted TMD homobilayers can be described by a lateral shift $\boldsymbol{d}_0$ that varies slowly in space: $\boldsymbol{d}_0 = \theta \hat{z} \times \boldsymbol{r}$. Therefore we construct the following continuum Hamiltonian for the moiré bands from Γ pocket two-band kp model:

$$\mathcal{H} = \begin{pmatrix} -\frac{\hbar^2 k^2}{2m^*} + \epsilon_b(r) & \Delta_T(r) \\ \Delta_T^\dagger(r) & -\frac{\hbar^2 k^2}{2m^*} + \epsilon_t(r) \end{pmatrix} \quad (3)$$

The position-dependent tunneling term is obtained by replacing $\boldsymbol{d}_0$ with $\theta \hat{z} \times \boldsymbol{r}$ in Eq.(2):

$$\Delta_T\left(\boldsymbol{r}\right) = w_0 + 2w_1 \sum_{j=1}^3 \cos(\boldsymbol{G}_j^{\boldsymbol{m}} \cdot \boldsymbol{r}) + 2w_2 \sum_{j=1}^3 \cos(2\boldsymbol{G}_j^{\boldsymbol{m}} \cdot \boldsymbol{r}) \quad (4)$$

Where $\boldsymbol{G}_i^{\boldsymbol{m}} = \boldsymbol{G}_i \theta \times \hat{z} (i = 1, 2, 3)$ are the three reciprocal lattice vectors in moiré superlattice. Likewise, the intralayer potential $\epsilon_{t,b}$ (t, b stand for top and bottom layer, respectively) can be expressed as the first order Fourier expansion over moiré reciprocal lattice vector:

$$\epsilon_{t,b}\left(\boldsymbol{r}\right) = 2V_0 \sum_{j=1,2,3} \cos\left(\boldsymbol{G}_j^{\boldsymbol{m}} \cdot \boldsymbol{r} \pm \phi\right) \quad (5)$$

The sign of phase factor ϕ changes under layer exchange, enforced by C_{2y} symmetry as shown in Fig. 2a. The
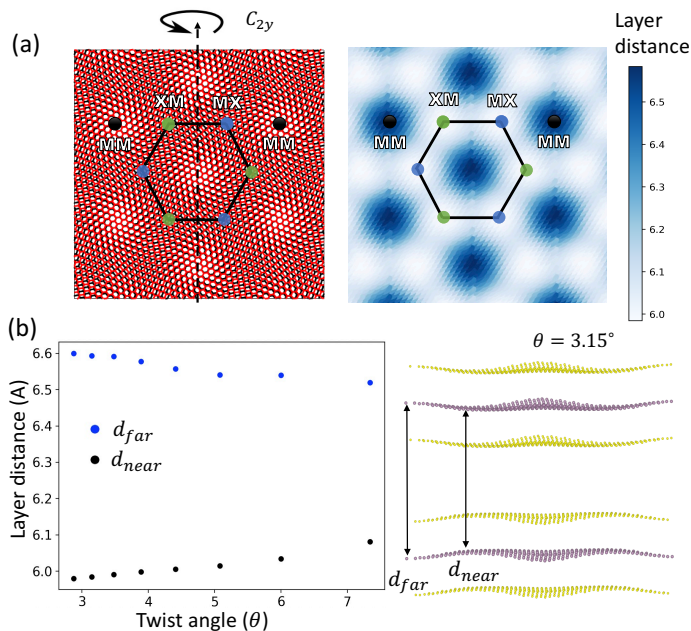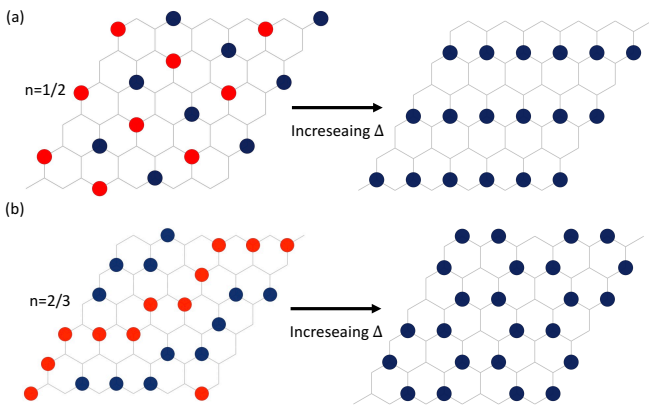
FIG. 4: Ground state charge order at filling (a) n=1/2 with increasing charge transfer gap Δ, (b) n=2/3 with increasing charge transfer gap Δ.

topmost moiré band (see Supplementary Material [49]). A larger field-induced Δ can be achieved in twisted TMD homobilayers with reduced interlayer tunneling (which competes with the layer potential asymmetry). This can be realized by inserting an hBN layer in between the top and bottom TMD layers [24].

In the TMD superlattice, the local minimums of the periodic moiré potential can be viewed as the effective moiré atoms to host charge. Under the harmonic approximation, the size of the Wannier orbital for the topmost moiré band is given by $\xi = \sqrt{\frac{\hbar}{m^*\omega}} = 2\left(\pi\right)^{-\frac{1}{2}}\sqrt{L_m}(\frac{\hbar^2}{m^*V_m})^{\frac{1}{4}}$ [35](V_m is the moiré potential integrated to antibonding orbitals). In homobilayer system without lattice mismatch, the kinetic energy over nearest neighbor interaction (t/V_1) can be tuned arbitrarily small, so that the classical model is well justified at sufficiently small twist angle. The effective extended Hubbard model without kinetic energy is given by:

$$H_0 = \sum_{j\in B}\Delta n_j + \sum_i U n_{i\uparrow}n_{i\downarrow} + \frac{1}{2}\sum_{i\neq j}V_{ij}n_i n_j \tag{6}$$

Here Δ is the charge transfer gap between two sublattice sites A and B, and V_{ij} is the extended interaction between i and j sites.

In twisted homobilayer MoS$_2$, the gating field introduces a charge transfer gap Δ. We first discuss the situation with large Δ. At filling $n < 1$, the effective tight-binding model reduces to a triangular lattice model, as in the case of WSe$_2$/WS$_2$, and exhibits similar charge orders. Various insulating states have been observed at fractional fillings $n = 1/4, 1/3, 2/5, 1/2, 3/5, 2/3$[30–32]. Due to the strong on-site Coulomb repulsion $U \gg \Delta$, the system at $n = 1$ should be regarded as a charge transfer insulator [35]. When doped to a higher filling $n > 1$, additional charges transfer to the other sublattice/layer.

Here we further study the charge orders of honeycomb lattice with small Δ including $\Delta = 0$ in flat band limit. We perform classical Monte Carlo simulation up to 120×120 sites with periodic boundary condition for the

extended Hubbard model with different gate distances from $d = L_m/2$ to $d = 10L_m$. The distance dependent interaction strength is plotted in Fig. S2 up to V_{100}, and the interaction cutoff is chosen as $0.1\%V_1$. We identify a series of charge orders at $n = 1/4, 1/3, 1/2, 2/3, 1$. For $n < 1/2$, moiré electrons are all filled to one sublattice, exhibiting similar charge orders as the reduced triangular lattice in WSe2$_2$/WS$_2$ heterobilayer (see Supplementary Materials [49]).

Interestingly, for small Δ, charge transfer involving two sublattices already takes place for filling $n \geq 1/2$. At filling factor $n = 1/2$, we find an emerging rectangular lattice with $\sqrt{3} \times 2$ periodicity. This state breaks the three-fold rotational symmetry and can be viewed as the combination of the stripe states on both sublattices, each at $1/4$ filling. This rectangular electron crystal is energetically favorable compared to the enlarged 2×2 honeycomb crystal at all gate screening distances. In contrast, at large Δ, the ground state becomes a simple stripe state on the triangular sublattice with lower on-site potential, as shown in Fig. 4. We find the critical charge transfer gap is $\Delta_c = 2(V_2 - V_3 - V_4 + 2V_6 - V_9 + V_{12} + ...)$. For $d = L_m = 9.1$ nm, $\Delta_c = 0.12\frac{e^2}{\epsilon L_m} \sim 3.8$ meV can be reached by realistic gating field. We note the critical Δ_c can be further lowered by increasing moiré wavelength.

At filling factor $n = 2/3$, the charges form a zigzag stripe order with 6×6 periodicity breaking the C_3 rotational symmetry. This zigzag type charge configuration is energetically favored compared to a linear stripe at screening distances from $d = 1/2L_m$ to $d = 10L_m$. As Δ increases, the zigzag charge stripe transitions to the $\sqrt{3} \times \sqrt{3}$ crystal that occupies one sublattice sites only, as shown in Fig. 4. We find the critical charge transfer gap is $\Delta_c = V_2 - V_3 - \frac{10}{3}V_4 + \frac{14}{3}V_5.... = 0.04\frac{e^2}{\epsilon L_m} \sim 1.3$ meV at $d = L_m = 9.1$ nm.

The transition between distinct electron crystals at the same filling is first-order. This should lead to a kink in the sublattice/layer charge imbalance as a function of the gating field. This prediction, which is a main result of our work, can be tested in MoSe$_2$/hBN/MoSe$_2$ heterostructure [24], where the gating field induced charge transfer between the top and bottom layers has already been observed at relatively high temperature.

For $n = 1$, we find that even at $\Delta = 0$, the ground state is a fully sublattice polarized Mott insulating state, which spontaneously breaks the honeycomb lattice symmetry. As the two sublattice sites MX and XM have different layer weight, the Mott insulating state at n=1 develops a finite out-of-plane ferroelectric polarization, which can be switched by the electric field. The ferroelectricity driven by the Mott physics in TMD moiré systems goes beyond the conventional ferroelectricity and enable the fast switching due to electronic origin [50]. For filling $n > 1$, charge-2e trimer can be the lowest energy excitation when tuning the charge transfer gap Δ, providing a platform to design unconventional superconductivity [51].

In homobilayer WSe$_2$, the valence band maximum is

located at K with weak interlayer tunneling amplitude and intralayer potential both on the order of 10 meV. The complex tunneling term between two layers brings further complications for the theoretical and experimental investigation of the insulating states [26, 28, 44]. In the case of heterobilayer WSe_2/WS_2, the existence of secondary moiré potential minima is proposed [35]. However, due to the large band offset between WSe_2 and WS_2, charges are localized at the one layer, which limits the tunability of potential difference between two moiré regions. The fractional filling insulating states in moiré systems especially WSe_2/WS_2 heterobilayer have been investigated under the generalized Wigner crystal picture [30, 52, 53], and it is fully consistent with our charge order picture driven by long range Coulomb interaction. Beyond the previous triangular lattice realized in WSe_2/WS_2 heterobilayer, we introduce an underlying effective honeycomb (moiré) lattice, which gives rise to the electrically tunable phase diagram.

In conclusion, we present a combined study of lattice relaxation, single-particle electronic structure, and ground state charge orders on the twisted homobilayer MoS_2. Unlike the previous moiré charge transfer insulator in $WSe2_2/WS_2$ heterobilayer, here out-of-plane gat-

ing field breaks C_{2y} symmetry and induces a controllable charge transfer gap. With Monte Carlo simulation, we predict additional stripe-type charge orders at fillings $n = 1/2, 2/3$ in the emergent honeycomb lattice with $\Delta = 0$. When increasing Δ, these electron crystals transit to fully sublattice polarized states. We further predict the ferroelectricity at the $n = 1$ Mott insulating state, which enables the ultrafast switching of electronic polarization. Our work demonstrates that the interplay between two moiré regions leads to the charge transfer insulator [35, 54] and serves as a platform for creating novel correlated states, such as unconventional density wave [55, 56], charge stripes [34], spin superfluid[57] and superconductivity[51].

[1] Y. Cao, V. Fatemi, A. Demir, S. Fang, S. L. Tomarken, J. Y. Luo, J. D. Sanchez-Yamagishi, K. Watanabe, T. Taniguchi, E. Kaxiras, et al., Nature **556**, 80 (2018).
[2] Y. Cao, V. Fatemi, S. Fang, K. Watanabe, T. Taniguchi, E. Kaxiras, and P. Jarillo-Herrero, Nature **556**, 43 (2018).
[3] X. Lu, P. Stepanov, W. Yang, M. Xie, M. A. Aamir, I. Das, C. Urgell, K. Watanabe, T. Taniguchi, G. Zhang, et al., Nature **574**, 653 (2019).
[4] A. Kerelsky, L. J. McGilly, D. M. Kennes, L. Xian, M. Yankowitz, S. Chen, K. Watanabe, T. Taniguchi, J. Hone, C. Dean, et al., Nature **572**, 95 (2019).
[5] Y. Jiang, X. Lai, K. Watanabe, T. Taniguchi, K. Haule, J. Mao, and E. Y. Andrei, Nature **573**, 91 (2019).
[6] Y. Xie, B. Lian, B. Jäck, X. Liu, C.-L. Chiu, K. Watanabe, T. Taniguchi, B. A. Bernevig, and A. Yazdani, Nature **572**, 101 (2019).
[7] Y. Choi, J. Kemmer, Y. Peng, A. Thomson, H. Arora, R. Polski, Y. Zhang, H. Ren, J. Alicea, G. Refael, et al., Nature Physics **15**, 1174 (2019).
[8] M. Yankowitz, S. Chen, H. Polshyn, Y. Zhang, K. Watanabe, T. Taniguchi, D. Graf, A. F. Young, and C. R. Dean, Science **363**, 1059 (2019).
[9] E. Codecido, Q. Wang, R. Koester, S. Che, H. Tian, R. Lv, S. Tran, K. Watanabe, T. Taniguchi, F. Zhang, et al., Science Advances **5**, eaaw9770 (2019).
[10] A. L. Sharpe, E. J. Fox, A. W. Barnard, J. Finney, K. Watanabe, T. Taniguchi, M. Kastner, and D. Goldhaber-Gordon, Science **365**, 605 (2019).
[11] S. Tomarken, Y. Cao, A. Demir, K. Watanabe, T. Taniguchi, P. Jarillo-Herrero, and R. Ashoori, Physical review letters **123**, 046601 (2019).
[12] U. Zondiner, A. Rozen, D. Rodan-Legrain, Y. Cao, R. Queiroz, T. Taniguchi, K. Watanabe, Y. Oreg, F. von Oppen, A. Stern, et al., arXiv preprint arXiv:1912.06150 (2019).
[13] G. Chen, L. Jiang, S. Wu, B. Lyu, H. Li, B. L. Chittari, K. Watanabe, T. Taniguchi, Z. Shi, J. Jung, et al., Nature Physics **15**, 237 (2019).
[14] G. Chen, A. L. Sharpe, P. Gallagher, I. T. Rosen, E. J. Fox, L. Jiang, B. Lyu, H. Li, K. Watanabe, T. Taniguchi, et al., Nature **572**, 215 (2019), URL https://doi.org/10.1038/s41586-019-1393-y.
[15] M. Serlin, C. Tschirhart, H. Polshyn, Y. Zhang, J. Zhu, K. Watanabe, T. Taniguchi, L. Balents, and A. Young, Science (2019).
[16] X. Liu, Z. Hao, E. Khalaf, J. Y. Lee, K. Watanabe, T. Taniguchi, A. Vishwanath, and P. Kim, arXiv preprint arXiv:1903.08130 (2019).
[17] C. Zhang, C.-P. Chuu, X. Ren, M.-Y. Li, L.-J. Li, C. Jin, M.-Y. Chou, and C.-K. Shih, Science advances **3**, e1601459 (2017).
[18] H. Yu, G.-B. Liu, J. Tang, X. Xu, and W. Yao, Science advances **3**, e1701696 (2017).
[19] K. L. Seyler, P. Rivera, H. Yu, N. P. Wilson, E. L. Ray, D. G. Mandrus, J. Yan, W. Yao, and X. Xu, Nature **567**, 66 (2019).
[20] K. Tran, G. Moody, F. Wu, X. Lu, J. Choi, K. Kim, A. Rai, D. A. Sanchez, J. Quan, A. Singh, et al., Nature **567**, 71 (2019).
[21] L. Yuan, B. Zheng, J. Kunstmann, T. Brumme, A. B. Kuc, C. Ma, S. Deng, D. Blach, A. Pan, and L. Huang, Nature Materials pp. 1–7 (2020).
[22] W. Li, X. Lu, S. Dubey, L. Devenica, and A. Srivastava, Nature Materials pp. 1–6 (2020).
[23] M. Brotons-Gisbert, H. Baek, A. Molina-Sánchez,

A. Campbell, E. Scerri, D. White, K. Watanabe, T. Taniguchi, C. Bonato, and B. D. Gerardot, Nature Materials pp. 1–7 (2020).

[24] Y. Shimazaki, I. Schwartz, K. Watanabe, T. Taniguchi, M. Kroner, and A. Imamoğlu, Nature **580**, 472 (2020).

[25] Y. Bai, L. Zhou, J. Wang, W. Wu, L. J. McGilly, D. Halbertal, C. F. B. Lo, F. Liu, J. Ardelean, P. Rivera, et al., Nature materials pp. 1–6 (2020).

[26] L. Wang, E.-M. Shih, A. Ghiotto, L. Xian, D. A. Rhodes, C. Tan, M. Claassen, D. M. Kennes, Y. Bai, B. Kim, et al., Nature materials pp. 1–6 (2020).

[27] L. J. McGilly, A. Kerelsky, N. R. Finney, K. Shapovalov, E.-M. Shih, A. Ghiotto, Y. Zeng, S. L. Moore, W. Wu, Y. Bai, et al., Nature Nanotechnology **15**, 580 (2020).

[28] Z. Zhang, Y. Wang, K. Watanabe, T. Taniguchi, K. Ueno, E. Tutuc, and B. J. LeRoy, Nature Physics pp. 1–4 (2020).

[29] A. Weston, Y. Zou, V. Enaldiev, A. Summerfield, N. Clark, V. Zólyomi, A. Graham, C. Yelgel, S. Magorrian, M. Zhou, et al., Nature Nanotechnology pp. 1–6 (2020).

[30] E. C. Regan, D. Wang, C. Jin, M. I. B. Utama, B. Gao, X. Wei, S. Zhao, W. Zhao, Z. Zhang, K. Yumigeta, et al., Nature **579**, 359 (2020).

[31] Y. Tang, L. Li, T. Li, Y. Xu, S. Liu, K. Barmak, K. Watanabe, T. Taniguchi, A. H. MacDonald, J. Shan, et al., Nature **579**, 353 (2020).

[32] Y. Xu, S. Liu, D. A. Rhodes, K. Watanabe, T. Taniguchi, J. Hone, V. Elser, K. F. Mak, and J. Shan, arXiv preprint arXiv:2007.11128 (2020).

[33] X. Huang, T. Wang, S. Miao, C. Wang, Z. Li, Z. Lian, T. Taniguchi, K. Watanabe, S. Okamoto, D. Xiao, et al., arXiv preprint arXiv:2007.11155 (2020).

[34] C. Jin, Z. Tao, T. Li, Y. Xu, Y. Tang, J. Zhu, S. Liu, K. Watanabe, T. Taniguchi, J. C. Hone, et al., arXiv preprint arXiv:2007.12068 (2020).

[35] Y. Zhang, N. F. Yuan, and L. Fu, Physical Review B **102**, 201115 (2020).

[36] H. Li, S. Li, M. H. Naik, J. Xie, X. Li, J. Wang, E. Regan, D. Wang, W. Zhao, S. Zhao, et al., arXiv preprint arXiv:2007.06113 (2020).

[37] M. H. Naik and M. Jain, Physical review letters **121**, 266401 (2018).

[38] L. Xian, M. Claassen, D. Kiese, M. M. Scherer, S. Trebst, D. M. Kennes, and A. Rubio, arXiv preprint arXiv:2004.02964 (2020).

[39] M. Angeli and A. MacDonald, arXiv preprint arXiv:2008.01735 (2020).

[40] Z. Zhan, Y. Zhang, G. Yu, F. G. J. A. Silva-Guillen, and S. Yuan, arXiv preprint arXiv:2005.13868 (2020).

[41] S. Venkateswarlu, A. Honecker, and G. T. de Laissardière, arXiv preprint arXiv:2005.13054 (2020).

[42] Y. Shimazaki, C. Kuhlenkamp, I. Schwartz, T. Smolenski, K. Watanabe, T. Taniguchi, M. Kroner, R. Schmidt, M. Knap, and A. Imamoglu, arXiv preprint arXiv:2008.04156 (2020).

[43] S. McDonnell, A. Azcatl, R. Addou, C. Gong, C. Battaglia, S. Chuang, K. Cho, A. Javey, and R. M. Wallace, ACS nano **8**, 6265 (2014).

[44] F. Wu, T. Lovorn, E. Tutuc, I. Martin, and A. MacDonald, Physical review letters **122**, 086402 (2019).

[45] K. Uchida, S. Furuya, J.-I. Iwata, and A. Oshiyama, Physical Review B **90**, 155451 (2014).

[46] J. Klimeš, D. R. Bowler, and A. Michaelides, Physical Review B **83**, 195131 (2011).

[47] G. Kresse and J. Furthmüller, Computational materials science **6**, 15 (1996).

[48] R. Bistritzer and A. H. MacDonald, Proceedings of the National Academy of Sciences **108**, 12233 (2011).

[49] See Supplemental Material at [URL] for the construction of commensurate structures, details of the ab-initio calculation and continuum model, and additional charge orders .

[50] Z. Zheng, Q. Ma, Z. Bi, S. de la Barrera, M.-H. Liu, N. Mao, Y. Zhang, N. Kiper, K. Watanabe, T. Taniguchi, et al., Nature **588**, 71 (2020).

[51] K. Slagle and L. Fu, arXiv preprint arXiv:2003.13690 (2020).

[52] B. Padhi, C. Setty, and P. W. Phillips, Nano letters **18**, 6175 (2018).

[53] B. Padhi, R. Chitra, and P. W. Phillips, Physical Review B **103**, 125146 (2021).

[54] J. Zaanen, G. Sawatzky, and J. Allen, Physical Review Letters **55**, 418 (1985).

[55] H. Pan, F. Wu, and S. D. Sarma, arXiv preprint arXiv:2004.04168 (2020).

[56] H. Pan, F. Wu, and S. D. Sarma, arXiv preprint arXiv:2008.08998 (2020).

[57] Z. Bi and L. Fu, arXiv preprint arXiv:1911.04493 (2019).

[58] AB stacking can be viewed as a 180° rotation of top layer