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Resonant and Bound States of Charged Defects in Two-Dimensional Semiconductors

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A detailed understanding of charged defects in two-dimensional semiconductors is needed for the development of ultrathin electronic devices. Here, we study negatively charged acceptor impurities in monolayer WS₂ using a combination of scanning tunnelling spectroscopy and large-scale atomistic electronic structure calculations. We observe several localized defect states of hydrogenic wave function character in the vicinity of the valence band edge. Some of these defect states are bound, while others are resonant. The resonant states result from the multi-valley valence band structure of WS₂, whereby localized states originating from the secondary valence band maximum at Γ hybridize with continuum states from the primary valence band maximum at K/K'. Resonant states have important consequences for electron transport as they can trap mobile carriers for several tens of picoseconds.

Impurity doping is the prime technology to control the electrical conductivity of semiconductors over several orders of magnitude [1, 2]. Dopants introduce additional charge carriers that bind to the impurity and can be excited into the delocalized conduction or valence band states. Through this ionization process, the density of mobile charge carriers in the material is increased. However, the Coulomb potential induced by the positively charged donor or negatively charged acceptor scatters mobile charge carriers and thereby reduces carrier mobility. For this reason, it is usually desirable to eliminate any naturally occurring charged defects.

Importantly, the Coulomb potential of charged impurities is screened by the dielectric response of the host semiconductor. The resulting perturbation leads to the formation of localized Bloch states near the band edges [3]. For bulk semiconductors, such as silicon or gallium arsenide, these defect states have hydrogenic wavefunctions [4–7]. Because of the large dielectric constants and small effective masses of traditional bulk semiconductors, the wavefunctions extend over several nanometers and the corresponding binding energies are of the order of a few tenths of an electronvolt [8]. These properties of charged defects are largely independent of their chemical origin and determined only by the long-ranged screened Coulomb potential induced by the impurity.

Many semiconductors exhibit multiple conduction or valence band valleys, i.e. their band structure features multiple extrema in the vicinity of the band gap. Such an electronic structure can result in the formation of *resonant impurity states* [9]. These states occur when a lo-

calized defect level originating from a secondary band extremum lies in the energy range of the continuum states of the primary band extremum [7]. The hybridization with the continuum states leads to an energy shift and a broadening of the localized level into a resonance. Such resonant defect states play an important role in transport properties [10–13] because a mobile charge carrier can be temporarily trapped by the impurity, reducing the carrier mobility [3].

Two-dimensional (2D) semiconductors, such as monolayer transition-metal dichalcogenides (TMDs), have been intensely studied over the last decade as promising candidates for nano-electronic components, optoelectronic devices, and quantum information applications [14, 15]. Understanding the properties of charged defects in these materials is a crucial step toward developing a viable route for impurity doping analogous to traditional (3D) semiconductors. A key difference between 2D and 3D semiconductors is their dielectric response to a charged perturbation. Specifically, the screened potential due to a point charge in 2D is not well-described by the bare Coulomb potential divided by a dielectric constant as in bulk semiconductors. Instead, in 2D semiconductors the dielectric response is relatively weak and highly anisotropic, giving rise to strongly bound excitons that dominate their optical response [16, 17]. Unconventional screening of charged defects in monolayer TMDs with multi-valley band structure and large spin-orbit effects can also give rise to bound defect states with unusual properties, e.g., the prediction that the most strongly bound acceptor state switches from being of K/K'-valley character to Γ -valley character at a critical value of the defect charge [18].

In this Letter, we study the electronic properties of charged impurities in monolayer WS₂ on a graphene/SiC

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substrate using scanning tunneling microscopy / spectroscopy (STM/STS) and atomistic quantum-mechanical calculations. Using STS, we find that negatively charged impurities result in strong upwards band bending and a series of localized defect states near the valence band edge with s - and p -like character. Our calculations suggest that the s -like states are bound defect states originating from either the K/K' or Γ valleys, while the p -like states are resonant defect states resulting from the hybridization of localized states from the Γ valley with continuum states from the K/K' valleys. Finally, we discuss the relevance of these findings for transport devices.

STM/STS measurements were performed under ultra-high vacuum at low temperature (~ 6 K). WS₂ samples were grown *ex situ* by chemical vapor deposition (CVD) on epitaxial graphene on silicon carbide substrates [19] and subsequent annealing in vacuum at $\sim 200^\circ\text{C}$. A representative overview image of the surface is shown in Fig. 1(a). Even without intentional doping, we observe various point defects, including oxygen-in-sulfur substitutions (O_S), molybdenum-in-tungsten substitutions (Mo_W), and charged defects [20, 21]. Charged defects, which we refer to as “CDs” can be readily distinguished from neutral defects by their characteristic STM contrast as seen in Figs. 1(a,b). CDs are among the most abundant point defects in our samples but their density varies considerably between different CVD preparations. Similar STM contrast has been observed in CVD-grown MoS₂ [22], MOCVD- and MBE-grown WSe₂ [23, 24], MBE-grown MoSe₂ [24] and natural bulk MoS₂(0001) [25]. Some of these reports suggest that the observed defect might be charged [21, 24, 25] but their chemical origin remains unclear.

Using non-contact atomic force microscopy (nc-AFM) with a carbon monoxide (CO) functionalized tip [26] we previously revealed that the charged defect is a chalcogen substitution [21]. However, the chemical origin of the CDs has not yet been conclusively established. Possible candidates include a CH or N substitution at the S site [21]. These defects introduce unoccupied defect states in the band gap close the valence band edge of WS₂ [see the Supplementary Materials (SM) [27]]. Since the Fermi level of the TMD/graphene/SiC heterostructure is pinned by the graphene/SiC substrate, formally unoccupied WS₂ defect states close to the VB will be populated and hence become negatively charged. The graphene/SiC substrate acts like a Fermi sea filling (or depleting) all WS₂ defect states that are below (or above) the Fermi level [28], which resides in the upper half of the WS₂ band gap [29]. As the hybridization between the WS₂ states and the substrate is negligible [30], the impurity charge is an integer multiple of the electron charge.

In constant-current STM imaging, CDs are observed as large depressions at positive sample bias and protrusions at negative sample bias. This is an electronic effect: a negative charge located at the CD site results in local upwards band bending. As shown in the dI/dV spectra taken across a CD in Fig. 1(c), electrons are

pushed to higher energies near the defect, leading to a position-dependent onset of the conduction band minimum (CBM), indicated with a white dashed line. The valence band of WS₂ exhibits two primary maxima at the K and K' points of the Brillouin zone and a secondary maximum at the Γ point [see Fig. 2(a)]. The energies of these band extrema measured away from the defect are indicated by white dotted lines in Fig. 1(c). It should be noted that states originating from the K/K' point give only a weak STS signal. This is due to two factors: their relatively small effective mass results in a small local density of states (LDOS) and their large crystal momentum leads to a small tunnelling matrix element [31]. In contrast, the larger effective mass and smaller crystal momentum at Γ results in a stronger STS signal.

While there are no observed defect states near the CBM, several defect resonances are observed at negative bias in the vicinity of the valence band edge [see Figs. 1(c,d)]. In Fig. 1(e), we have labeled the three dominant resonances as “A”, “B”, and “C”. In addition, there are four very faint resonances observed just above state A [see the Supplementary Materials (SM) [27]]. We find that the localized defect states A and B lie on average 175 mV and 95 mV, respectively, above the onset of the K/K' valence band edge (see Table I), indicating that they are bound states. Resonance C lies 130 mV below the K/K' valence band edge, but 110 mV above the onset of the Γ valence band edge, suggesting that it is a resonant state, i.e., a localized state resonant with the continuum states from the K/K' valleys.

	Experiment		Theory	
	K	Γ	K	Γ
A [1s(K/K')]/meV	+175	+415	+95	+355
B [1s(Γ)]/meV	+95	+335	+25	+285
C [2p(Γ)]/meV	-130	+110	-180	+80

TABLE I. Comparison of experimentally determined and theoretically calculated binding energies of three observed resonances: A, B, and C. Values are listed with respect to the K and Γ valence band edges. Positive binding energies indicate a lower energy with respect to the band edge.

Spatially resolved dI/dV maps of the dominant states are shown in Fig. 1(f). A and B have a spherical shape reminiscent of an s -type orbital, whereas C has three lobes and a node at its center, indicative of a trigonally warped $2p$ -type orbital [18]. With a lateral dimension of about 1.5 – 2 nm, the localized defect states are closely confined, as compared to the Rydberg states recently reported for ionized defects in bulk black phosphorus which extend more than ten nanometers [32]. The smaller size of the acceptor states in our experiment is a consequence of the different environmental screening. Specifically, the metallic screening from the doped graphene substrate leads to more localized defect states compared to the states obtained for an insulating substrate.

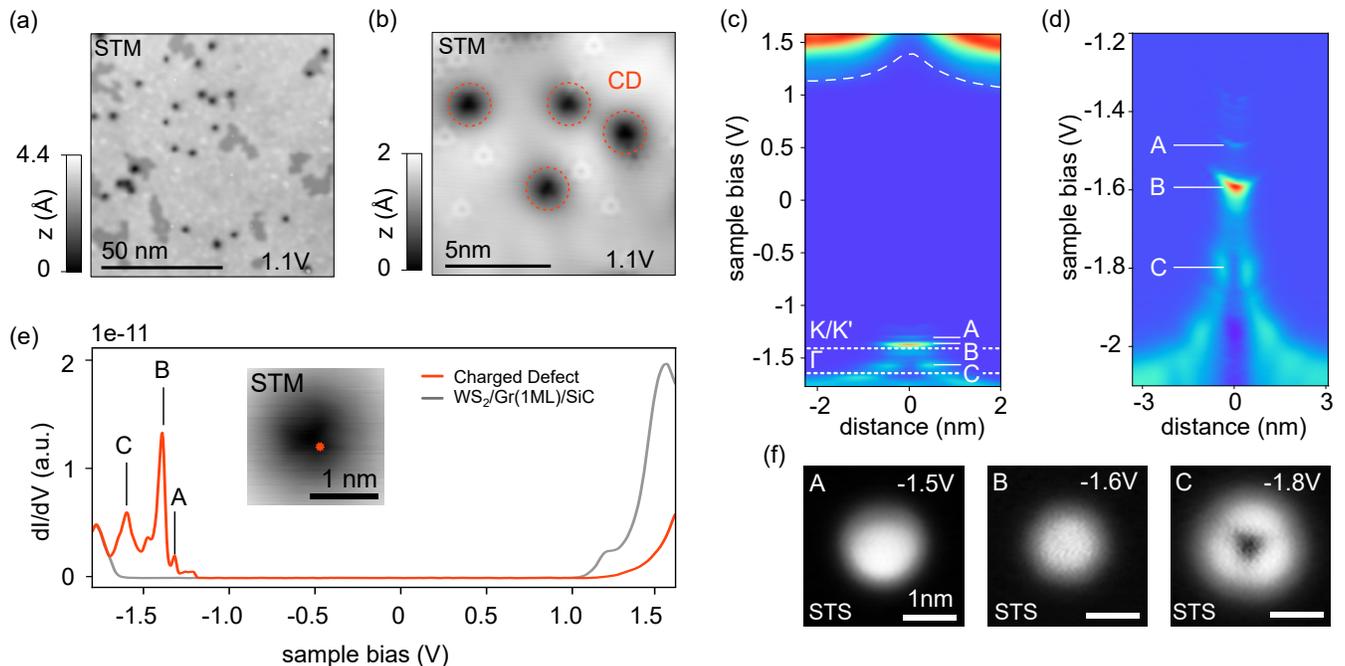


FIG. 1. (a) STM overview of the $\text{WS}_2/\text{graphene}/\text{SiC}$ surface, $V_b = 1.1 \text{ V}$. (b) STM image showing defects in WS_2 . The charged defects (CDs) are indicated with a red circle. (c,d) STS across a CD showing band bending and localized defect states labeled A, B, and C. In (c) the upward bending of the conduction band due to the negatively charged defect is indicated with a dashed white line, and the onset of the K/K' and Γ valleys of pristine WS_2 are indicated with white dotted lines. Note that the spectra in (c) and (d) were taken on different defects. (e) STS of a CD. (f) Constant height dI/dV maps of the defect corresponding to the electronic states A, B, and C.

To understand the experimental observations, we carried out theoretical calculations of acceptor impurities in WS_2 using a recently developed approach that allows the simulation of very large supercells. [18]. In particular, we first calculated the screened potential of the defect, which is modelled as a point charge with $Z = -1$ located $z_d = 2.4 \text{ \AA}$ above the plane of the tungsten atoms, see Fig. 2(b). For this, the dielectric function of the system is computed using the random phase approximation (RPA) and the polarizability is expressed as the sum of contributions from the WS_2 , the doped graphene, and the SiC substrate (see SM for details [27]).

To calculate the LDOS of a charged impurity in WS_2 , we carry out large-scale tight-binding calculations using a three-band tight-binding model [33] with the screened defect potential included as an on-site energy [18]. This allows us to obtain converged results for a 45×45 WS_2 supercell. This supercell, which would be extremely challenging to model with *ab initio* techniques, is sufficiently large to capture the decay of the shallow impurity states and obtain accurate defect state energies. While this approach allows us to describe states induced by the long-ranged screened defect potential, the model does not capture short-ranged chemical interactions that could give rise to additional defect states.

Figures 3(a) and (b) show the calculated LDOS of an acceptor impurity in WS_2 . In agreement with the measured dI/dV map [Fig. 1(c)], we observe a significant

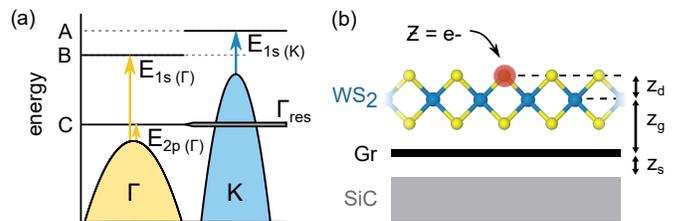


FIG. 2. (a) Schematic of the electronic structure of a negatively charged defect in WS_2 . Shown are the primary valence band maximum at K (a second degenerate maximum exists at K') and the secondary maximum at Γ as well as the bound and resonant defect states. The states labelled $1s(\text{K})$, $1s(\Gamma)$ and $2p(\Gamma)$ correspond to the experimental features A, B and C in Fig. 1, respectively. (b) Schematic of the system consisting of a WS_2 layer on graphene on a silicon carbide substrate with a substitutional impurity with charge Z . Distances z_d , z_g , and z_s are indicated (see SM for details [27]).

defect-induced upward bending of the conduction band edge. In the vicinity of the valence band edge, a series of defect states are found. Fig. 3(c) shows the simulated STS images of the three most strongly bound defect states that give rise to significant peaks in the LDOS (see SM for details [27]). The two lowest energy states appear similar to experimental states A and B as circular bright spots with a similar size [Fig. 1(f)]. The energy

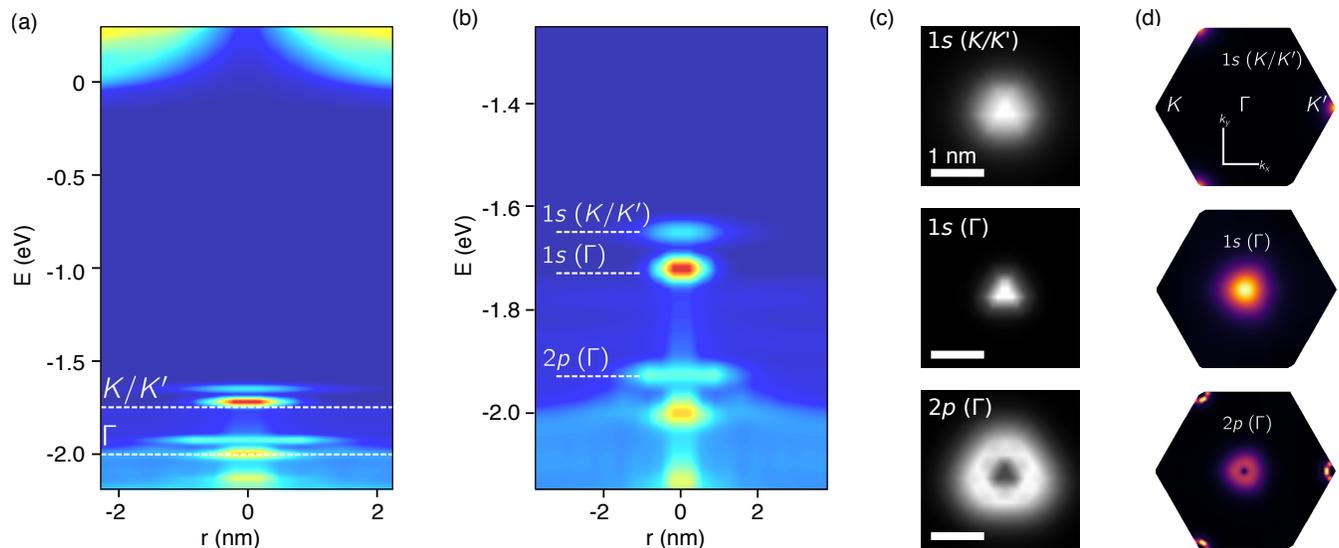


FIG. 3. (a), (b) Calculated local density of states of a negatively charged defect in WS₂ on graphene/SiC substrate. In (a), the onset of the K/K' and Γ valleys are indicated by dashed lines. In (b), the defect states are labeled by their hydrogenic quantum numbers and valley. (c) Simulated STS images of defect states. (d) Projections of defect wavefunctions onto unperturbed WS₂ states.

differences with respect to the valence band maximum at K/K' of these states are 95 meV and 25 meV, respectively. While these values are somewhat smaller than the experimental ones, see Table I, their difference (70 meV) is in good agreement with experiment (80 meV). As these states exhibit symmetries similar to the eigenstates of the two-dimensional hydrogen atom, we label them as $1s$ states. Counterintuitively, we find that the lower energy $1s$ state is less localized than the other $1s$ state. At higher energies, a $2p$ -like defect state is found that is characterized by a node at its center. This state resembles the experimental state C in Fig. 1(f). Its energy with respect to the primary valence band maximum at K is -180 meV indicating that it lies between Γ and K/K'. Again, we find that the energy difference between this state and the less strongly bound $1s$ state is in good agreement with experiment.

To understand which band extrema these defect states originate from, we decompose the defect wave functions into contributions from unperturbed WS₂ states. Fig. 3(d) shows that the lowest energy defect state originates from the K valley (a second degenerate state with opposite spin originates from the K' valley) and is labelled as $1s$ (K), while the other $1s$ state originates from the Γ valley. Even though the $1s$ (Γ) state resides less deep in the band gap, its effective binding energy with respect to the relevant Γ band edge is greater than that of the $1s$ (K) state [see Fig. 2(a)]. Consequently, the wavefunction of the $1s$ (Γ) state is more confined, reestablishing the general anti-correlation between binding energy and spatial extent. The larger binding energies associated with states originating from the Γ valley is a direct consequence of its larger effective mass. The analysis of

the $2p$ resonance reveals that this impurity level mostly originates from unperturbed states in the vicinity of Γ , but also has contributions from a few states near K (note that mixing only occurs between states with the same spin preventing any hybridization with states from K'). The valley hybridization results from the resonant nature of the $2p$ defect state and can be understood within the framework of effective mass theory [3]. In this approach, one first neglects the coupling between different valleys and obtains a set of bound defect states for each valley. When the coupling between valleys is turned on, the bound states from the secondary valence band extremum at Γ that lie in the same energy range as continuum states from the K and K' valleys can mix with the continuum states. The resulting hybridized states can be expressed as

$$\psi_{2p(\Gamma)}(\mathbf{r}) = c_{2p(\Gamma)}\phi_{2p(\Gamma)}(\mathbf{r}) + \sum_{n \in \{K, K'\}} \int_{\Omega_n} d\mathbf{k} c_n(\mathbf{k})\phi_{n\mathbf{k}}(\mathbf{r}), \quad (1)$$

where $\phi_{2p(\Gamma)}$ and $\phi_{n\mathbf{k}}$ denote the bound and continuum states obtained for decoupled valleys, respectively, (with n labelling the valley), $c_{2p(\Gamma)}$ and $c_n(\mathbf{k})$ are complex coefficients, and Ω_n denotes the k -space region corresponding to the n^{th} valley.

Finally, we discuss the effect of resonant impurities on electron transport. If a hole near a primary band extrema (i.e., in the K or K' valley) in an external electric field reaches the energy of the resonant impurity state $E_{2p(\Gamma)}$, it does not participate in conduction for a time $\tau_{\text{res}} = \hbar\Gamma_{\text{res}}^{-1}$. Here, Γ_{res} denotes the width of the resonant

impurity level given by

$$\Gamma_{\text{res}} = \pi \sum_{n \in \{K, K'\}} \int_{\Omega_n} d\mathbf{k} |V_{2p(\Gamma), n\mathbf{k}}|^2 \delta(E_{2p(\Gamma)} - \epsilon_{n\mathbf{k}}), \quad (2)$$

where $\epsilon_{n\mathbf{k}}$ denotes the energy of a continuum state with crystal momentum \mathbf{k} in valley n and $V_{2p(\Gamma), n\mathbf{k}}$ is the matrix element of the Bloch-transformed screened impurity potential between $\phi_{2p(\Gamma)}(\mathbf{r})$ and $\phi_{n\mathbf{k}}(\mathbf{r})$ [3]. Evaluating this expression with our first-principles defect potential and tight-binding wave functions yields $\Gamma_{\text{res}} = 0.02$ meV and $\tau_{\text{res}} = 30$ ps. This result shows that resonant defect states exhibit extremely small linewidths corresponding to long trapping times. Thus, we predict that resonant states should lead to a drastic reduction in conductivity when the Fermi level reaches the resonant impurity level. Note that in a WS_2 transport device, graphene could not be used as substrate as most carriers would propagate through the graphene. Without graphene, the screening of the impurity charge would be reduced, resulting in a resonant impurity level that lies closer to the VBM at K/K' . The trapping time, however, would not change significantly because of the flat density of state of WS_2 near the VBM.

In this Letter, we examined the localized acceptor states associated with negatively charged point defects in WS_2 using scanning tunneling microscopy and spectroscopy. We identify these states as resonant and bound hydrogenic states of the screened Coulomb potential, which are to a large extent independent of the chemical impurity species. Our large-scale atomistic electronic structure calculations reproduce the band bending and binding energies of the defect states as well as their wave function shapes and sizes as observed in STS measurements. The hydrogenic states exhibit s - and p -type character and originate from both the K/K' and Γ valence

band valleys. Despite the hybridization with the primary valence band around K and K' , a resonant state emerging from Γ is clearly resolved. We further predict that resonant states can trap mobile carriers for up to 30 picoseconds, impairing the mobility of transport devices. These insights provide a framework to understand the physics of hydrogenic bound defect states associated with charged defects naturally present in TMDs, guiding future efforts in chemical doping of TMDs and other two-dimensional semiconductors.

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