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The interplay of electronic, magnetic and structural properties of GdB₆ from first principles

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ABSTRACT:

Gadolinium hexaboride (GdB₆) is a well known field emitter material that has been investigated for more than three decades. We perform a systematical density-functional theory (DFT) study of GdB₆ by using the generalized-gradient approximation and considering the electron interaction parameter U. The basic structural and electronic properties are carefully revised, as well as a strong U-value dependence in determining the antiferromagnetic (AFM) magnetic structures of Gd 4felectronic states. We found a small U (0~3eV) showing the most consistent experimental ground-state properties, which gives rise to a magnetic structure with a ground state of C-AFM and a second stable E-AFM. Moreover, we find the distortion modes of boron octahedron play an important role in the interaction between spin and lattice structures in this system. These results will deepen our understanding of the boron-based correlated rare earth compounds.

1. Introduction

The rare earth hexaborides (RB₆) crystallize in the cubic CaB₆-type structure [1], in which individual metal atoms and boron octahedra are arranged in the CsCl structure. They present intriguing physical properties, such as low work function (~2.4-2.6 eV), high thermal conductivity, superconductivity, heavy fermion behavior and high melting points [2-6]. For example, the metallic LaB₆, a well-known thermionic and field electron emission cathode material, becomes superconducting below 0.45 K[6,7]; PrB₆ is believed to be antiferromagnetic (AFM) at low temperatures [8]; NdB₆ has shown favourable field-emission performances [9,10]; EuB₆ is a ferromagnetic (FM) semiconductor with two ferromagnetic transitions [11,12]; CeB₆ shows heavy fermion behavior and is famous as a dense Kondo material [5]; and mixed valence SmB₆ is the first known topological Kondo insulator [13,14] where strong fermions can exhibit topological surface states.

Among the rare earth hexaborides, gadolinium hexaboride (GdB_6) exhibits the lowest work function (~1.5 eV) [15]. However, the magnetic properties of GdB₆ are puzzling as it is not easy to determine the magnetic structure by neutron scattering due to the high neutron absorption of both Gd and B. GdB₆ was thought to have at least two first-order AFM transition with Néel temperature $T_N=15$ K and $T^*=8$ K [16] respectively from the electrical resistivity and magnetic torque studies [17,18]. Galera et al. performed an x-ray scattering experiment and observed a reflection at $(0, 0, \frac{1}{2})$ at the temperature between T_N and T^* . Below T^* , they found a second type of reflection observed at $(\frac{1}{2}, \frac{1}{2}, 0)$ which coexisted with the former reflection [19]. However, they had no idea about whether the reflections of $(\frac{1}{2}, \frac{1}{2}, 0)$ were from a C-AFM or crystallographic distortions. Kasuya believed that the $(\frac{1}{2}, \frac{1}{2}, 0)$ reflection was from a charge dipolar ordering due to a so-called exchange-pair Jahn-Teller effect [20], which called for further examination. In 2004, Luca et al. performed the first neutron diffraction of this system [16], they observed a propagation vector $(\frac{1}{4}, \frac{1}{4}, \frac{1}{2})$ with a basis magnetic moment direction vector $(0, 0, \frac{1}{2})$ at both 2 K and 12 K, namely an E-AFM magnetic phase. However, no abnormalities around T^* were observed in their experiments. One year later, Amara et al. performed a mean-field model investigation combined with their X-ray scattering results indicating the coexistence of magnetic and displacement waves [21]. They supposed that the displacement waves imposed by the magnetic $(\frac{1}{4}, \frac{1}{4}, \frac{1}{2})$ structures are consistent with the $(0, 0, \frac{1}{2})$, $(\frac{1}{2},\frac{1}{2},0)$, and $(\frac{1}{4},\frac{1}{4},\frac{1}{2})$ satellites observed in the X-ray scattering. These complex ordering patterns are characteristic of local moments associated with Gd interacting with metallic conduction electrons that can provide a non-trivial ordering due for example to Fermi surface properties. This calls for a first principles approach to the problem that includes both localized electrons in Gd f states and itinerant electrons on the boron backbone.

In the past few decades, density functional theory (DFT) has become an indispensable tool in material sciences and is widely used in determining the electronic and magnetic interactions of materials. However, the available previous DFT studies of GdB_6 mainly focused on its structural, electronic and optical properties. Singh *et al.* calculated the reflectivity and optical conductivity of FM RB_6 using the LSDA+U method [15] but neglected the AFM or paramagnetic state. Furthermore, the selected U value (9 eV) of Gd directly obtained from previous studies on Gd_2O_3 requires some justification [22,23]. Li *et al.* indicated that GdB_6 is an almost perfect near-infrared absorption/reflectance material that could serve as a solar radiation shielding material for windows with high visible light transmittance [24]. They found that the magnetic 4f electrons of Gd are not relevant to the important optical properties of GdB₆, but the neglection of the AFM nature and the Hubbard U effect on 4f electrons needs to be further discussed. It is therefore crucial to perform a systematical DFT study to emphasize the AFM nature of GdB₆, to better understand how to utilize GdB₆, especially determining the magnetic ground state and revealing the interaction between the spin and lattice, and provide reasonable basic parameters for further research.

In this work, we fill these gaps in determining the magnetic structures of GdB_6 from first-principles calculations. The structural properties and electronic structure were carefully investigated, which are in good agreement with experimental value even at the GGA level. However, a strong *U*-value dependence of the magnetic structures and 4*f* electronic states is found and discussed in detail.

2. Computational details

We performed the spin polarized calculations with the projector augmented wave (PAW) approach [25,26] which were implemented in the Vienna ab initio simulation

package (VASP) [27]. The 5p, 6s, 4f, and 5d were taken as valence states for Gd, and 2s and 2p valence states were taken for B respectively. The exchange correlation functional of generalized gradient approximation (GGA) was in the form proposed by Perdew, Burke, and Ernzerhof (PBE) [28], and an additional on-site interaction U [29] were realized in the form introduced by Dudarev et al. The convergence criterion of energy in relaxation was set to be 10^{-6} eV and atomic positions were fully relaxed until the maximum force on each atom was less than 10^{-3} eV/Å. The plane-wave cutoff energy is set to be 600 eV for all calculations. This represents a highly converged basis set. The precision of the total energy is set to be 10^{-8} eV. A $6 \times 6 \times 6 \Gamma$ centered Monkhorst-Pack k-mesh in the Brillouin zone was used here, where the convergence of k-mesh were confirmed to be less 0.1 meV/atom. A $2 \times 2 \times 2$ superlattice (containing 56 atoms) was adopted to construct various magnetic configurations (the schematic diagrams displayed in Fig. S1)[30], which are six AFM orders (A, C, D, E, F, G-AFM) and a FM ordered state. The spin-orbit coupling (SOC) effect on the magnetic structures were examined, although it is expected to be weak due to the half filled f-shell of Gd. The bulk modulus obtained from fitting the Birch-Murnaghan equation of state. Furthermore, the screened hybrid functional proposed by Heyd, Scuseria, and Ernzerhof (HSE06) [31] was applied to investigate electronic properties, which has shown to be successful in a whole range of materials. The mixing of HF:GGA ratio was 0.25:0.75 and a screening parameter of 0.2 \AA^{-1} was adopted.

3. Results and discussion

3.1 Basic structural and electronic properties of cubic GdB₆



Figure 1. (a) The energy values of PBE with U=0 eV, 2 eV, 8 eV versus the volume per formula unit (*f.u.*). The inset shows the cubic structure of GdB₆. (b) The total and projected densities of states (DOS) of C-AFM GdB₆, the upper and lower panels are the PBE and HSE06 results respectively.

The simple cubic structure of GdB₆, space group *Pm m* (No. 221), is displayed in Figure 1a. The boron octahedron is constructed by 6 B atoms and 12 B-B covalent bonds and the B₆ octahedron is located at the center of a cubic Gd lattice. We find that both the PBE calculated volume (69.53 Å³) and bulk modulus (169 GPa) agree very well with the experimental values of 69.58 Å³ and ~170 GPa [32]. Figure 1b shows total and projected the densities of states (PDOS) form PBE and HSE methods, where partially-filled 4*f* orbitals are observed indicating the trivalent nature of Gd³⁺ consistent with previous reflectivity spectra measurement [33]. On the other hand, we find that the PDOS of B atoms has contributions to the total densities of states (DOS) in the whole displayed energy range, and a strong *sp*³ hybridization behavior of the B octahedra is observed (see the charge density map diagrams displayed in Fig. S2)[34], which shows that the B octahedra have strong relative stability.



Figure 2. (a) The *f* orbitals PDOS of GdB₆ under various *U* values. (b) The magnetic moment, volume and d_{ff} versus *U* values are displayed, where d_{ff} is defined as the separation between the peaks of two *f*-orbitals. Here, we use the C-AFM system as a representative.

Near the Fermi level, we find that both PBE and HSE show partially occupied Gd_d orbitals, which hybridize with the B states. These are the active electronic states for transport. In addition, two localized Gd_f peaks are observed in two specific energy ranges, namely the occupied one with PBE: -4.5~-5 eV (HSE: -9~-9.5 eV) and the unoccupied one with PBE: 0~0.6 eV (HSE: 2.3~2.9 eV) respectively. These are the exchange split majority and minority spin Gd *f* states. Note that already at the PBE level the exchange splitting is sufficient to fully polarize the Gd *f* shell. The HSE gives an extremely large and unreasonable energy splitting about 12 eV of two *f*-orbitals which is more than two times of the PBE results. This overestimation is due to the simplistic treatment of the exchange functional rooted in the original Fock exchange energy, which is also found in Ce₂O₃ system[35] where the PBE0 (HSE06) overestimate up to 45% (15%) of the Ce 4*f*-5*d*4*f* energy gap.

To deal with the well-known errors in the DFT, we also performed the GGA+U calculations for this system. The additional U increases the energy separation between the minority and majority f states. Figure 2a shows that the Hubbard U in particular pushes unoccupied f orbitals away from the Fermi level, while at the same time increasing the binding energy of the majority spin states as illustrated in Figure 2b.

The U value not only changes the f states dramatically, but also alters the GdB_6 structure, electronic properties and magnetic properties. We find that the volume reduces monotonically when the U value increases, while the magnetic moment increases with the U value. The magnetic moment of Gd^{3+} obtained by us is within an integration sphere as in standard VASP calculations. We also note that the quoted magnetic moments are spin moments, since the half full shell of Gd does not allow significant orbital components. When the U = 2 eV, it agrees best with the experimental value 6.9 $\mu_{\rm B}$ [16] which was derived with a Bragg R-factor = 2.7 and a magnetic R-factor = 15. Hence, the small U value range (0 eV \leq U \leq 2 eV) might be more favorable in the prediction of the structure, bulk modulus and its magnetic moment. A small $U_{\rm eff} \approx 2.0$ eV value for GGA was also found to give the best overall description for the energetics, lattice constants, and magnetic ordering in Ce_2O_3 system[35], instead of the calculated value of 4.50 eV from the linear-response approach [36] as emphasized recently by us that it would lead to poor ground state properties in the simple materials like Fe [37]. Another reason for using smaller U values is that the recent constrained random phase approximation calculations [38] of many rare-earth nickelates [39] tend to favor smaller U values toward the f-orbitals of these systems.



3.2 Low temperature magnetic structures

Figure 3. The relative energy of various AFM configurations from (a) GGA+U and (b) GGA+U+SOC calculations, where the E_0 , E_1 , E_2 and E_3 represent the ground state

energy, the first, second and third higher energy state energies, respectively, which are represented by different colors and symbols. The corresponding magnetic structures are also indicated beside each symbol.

Figure 3a shows the GGA+U results of the relative energy of various AFM orderings, where we find C-AFM maintains as the ground magnetic structure until the U is larger than 5 eV. Then after an intermediate F-AFM at U = 6 eV, the ground magnetic structure becomes A-AFM in the wide U value range. We also examine the spin-orbit coupling (SOC) effect on the magnetic structures in Figure 3b. The SOC has no change in the ground magnetic structure, and only slightly shifts the relative energy of D-AFM and F-AFM within the calculational error. As mentioned, this weak SOC effect is consistent with the fully polarized f shell, which precludes orbital moments. On the other hand, the sub-ground magnetic structure is more dependent on the chosen U values. The sub-ground magnetic structure favors the E-AFM in a small U value range (U≤3 eV), while favors F-AFM or D-AFM in the large U value range.

As discussed above, a small U value ($U \le 2 \text{ eV}$) allows us to closely reproduce the experimental volume, bulk modulus and magnetic moment. We also find the sub-ground magnetic structure E-AFM is consistent with the experimental satellite observed in $(\frac{1}{4}, \frac{1}{4}, \frac{1}{2})$ at 12 K. Furthermore, the relative energy difference between E-AFM and C-AFM is from 1 meV/atom (U = 0 eV) to 0.6 meV/atom (U = 2 eV), corresponding to about 12-6 K in the temperature scale. Therefore, combining our DFT calculations with the small U value range and previous experiments, we think that there may indeed be a lower-energy magnetic state than E-AFM, namely the C-AFM. Such C-AFM ground structure is consistent with the $(\frac{1}{2}, \frac{1}{2}, 0)$ satellites observed in scattering experiments. The occurance of this state is connected with a relatively low value of U in our calculations. Such low values of U favor increased interaction between itinerant electrons and the Gd f local moments. On the other hand, higher values of U, which lead to stronger localization of the f states, would lead to the A-AFM state. Therefore the combination of spectroscopic and magnetic

measurements may provide insights into the magnetic behavior and its origins in this compound.

3.3 The influence of magnetic structure on the electronic structure and lattice distortion

Table 1. The input structures for AMPLIMODES analysis are evaluated within a symmetry tolerance of 10^{-4} Å. Mode amplitudes in distorted structures with different multiplication of their primitive unit cell are not directly comparable. The amplitudes are normalized with respect to the primitive unit cell of the high-symmetry structure by a normalization factor.

Туре	Mode	k-vector	Amplitude (10 ⁻⁴ Å)	Norm. factor	Global distortion (10 ⁻⁴ Å)
А	Γ_1^+	(0,0,0)	0	1	4
	Γ_3^+	(0,0,0)	4		
С	Γ_1^+	(0,0,0)	0	1	3
	Γ_3^+	(0,0,0)	3		
D	Γ_1^+	(0,0,0)	0	/2	8
	Γ_3^+	(0,0,0)	2	NΖ	
Е	M_1^+	(1/2,1/2,1/2)	5		
	Γ_1^+	(0,0,0)	0	2	19
	Γ_3^+	(0,0,0)	0		
	X_1^+	(0,1/2,0)	3		
F	X_2^+	(0,1/2,0)	1		
	M_1^+	(1/2,1/2,1/2)	9		
	Γ_1^+	(0,0,0)	0	/2	8
	Γ_3^+	(0,0,0)	2	νZ	
	M_1^+	(1/2,1/2,1/2)	5		



Figure 4. Schematic illustration of the distortion modes of GdB_6 . The red arrows represent the direction of the displacements. The magnitudes of the distortion modes are listed the Table 1.

The electronic DOS for various magnetic structures are shown in Figure 5. These are very similar with each other below the Fermi level. However, the unoccupied states show quite different distributions, for instance, the paramagnetic (PM) and G-AFM configurations tend to have sharper peaks in the 3-3.5 eV energy range. These different unoccupied states are mainly derived from 5*d* states, hybridized with the boron backbone.



Figure 5. The projected density of states of PBE calculations for paramagnetic (PM), FM and six AFM phases of GdB₆.

Band structures for various magnetic structures are given in Figure 6. The dispersive bands below Fermi level come from the mixing of Gd d and B states, showing small effective mass for all AFM configurations. The low effective mass emphasizes the itinerancy of this system, as does the experimentally observed conductivity. We also notice the dispersions of these bands are sensitive to the particular AFM order. This is consistent with the fact that charge transport anisotropy is observed in recent low temperature experiments [40] even though the crystal symmetry without magnetism would preclude anisotropy.



Figure 6. The projected band structures of GGA calculations for six AFM phases of GdB₆.

In Table 2, we list the calculated volume and bulk modulus for the different magnetic configurations of GdB₆. Similar to the electronic structures, the lattice parameters are sensitive to the magnetic order, though weakly so. This is shown in Figure 4. It is to be noted that the FM, PM and G-AFM systems maintain the cubic crystal phase, while the other AFM orders investigated here lead to tetragonal crystal phase with space group P4/mmm (No. 123).

Table 2. The lattice volume and bulk modulus of the GdB₆ calculated by using GGA, GGA+U (U=2eV) and (U=8eV).

	Volume (Å ³)							B(GPa)
Туре	А	С	D	Е	F	G	FM	
GGA	69.55	69.53	69.51	69.55	69.53	69.54	69.53	169

U=2 eV	69.42	69.42	69.42	69.42	69.41	69.42	69.41	171
U=8 eV	69.18	69.17	69.18	69.18	69.19	69.17	69.18	173
Exp				69.19[4	1]			170 [32]

In order to further study magnetoelastic distortions we combined our DFT results with the AMPLIMODES[42] analysis. We show the detailed distortion modes in Figure 4 and list their normalized magnitudes in Table S2. We find that the magnetic structures (A, C, D, E and F-AFM) affect both the amplitudes and distortion modes, where the B octahedron is found to play an important role in the rich relationship between the spin and lattice. The A-AFM and C-AFM configurations only undergo two modes, namely the Γ_1^+ and Γ_3^+ , while the Γ_3^+ is the primary mode. These two modes are also observed in D, E and F-AFM systems but with smaller magnitudes than other modes. As shown in Figure 4, the Γ_1^+ mode displays the movements of all B atoms to the center of the B-octahedron, while in the Γ_3^+ mode four B atoms in the *ab* plane move towards the center whereas two B atoms along the *c* axis move away from the center of the B-octahedron. The D-AFM and F-AFM are similar with each other, as they undergo another primary mode M_1^+ together with the relatively small Γ_1^+ and Γ_3^+ modes. This M_1^+ mode is only contributed by the B atoms in the *ab* plane. The E-AFM configuration seems to be most complicated, where two extra X_1^+ and X_2^+ modes are found in addition to the Γ_1^+ , Γ_3^+ and M_1^+ modes. The in-plane B atoms displacements have similar patterns for the $X_1^{\scriptscriptstyle +}$ and $X_2^{\scriptscriptstyle +}$ modes, but the two apical B atoms show relatively opposite movements for the two modes. In summary, our results of distorted structures strongly suggest that the GdB₆ crystal should present tetragonal symmetry as the ground state induced by magnetic ordering, although it is cubic at room temperature [18]. This is presumably the origin of the transport anisotropy.

4. Summary and Conclusions

In summary, we have carried out first-principles calculations to investigate the geometric, electronic and magnetic structure properties of GdB_6 with GGA and GGA + U methods. The 4f electronic states, structural and magnetic properties are strongly

dependent on the U value. We have found good agreement with experimental results of magnetic moments, volume and bulk modulus values in a U value range ($0 \text{ eV} \le U \le$ 3 eV), in which the ground magnetic structure is C-AFM and the first higher-energy magnetic structure is E-AFM. Whereas, in a wide U value range ($5 \text{ eV} \le U \le 10 \text{ eV}$), the ground magnetic structure is found to be A-AFM. We find non-trivial spin-lattice coupling involving distortions of the B octahedra. Importantly, we also find a relationship between the magnetic order of the ground state and the strength of the on-site Coulomb interaction. It will be of interest to examine this in future experiments, for example combinations of scattering and photoemission spectroscopy.

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References

[1] H. Hacher Jr and M. Lin, Solid State Commun. 6, 37 (1968).

[2] B. Matthias, T. Geballe, K. Andres, E. Corenzwit, G. Hull, and J. Maita, Science 159, 530 (1968).

[3] G. Schell, H. Winter, H. Rietschel, and F. Gompf, Physical Review B 25, 1589 (1982).

[4] N. Xu, C. E. Matt, E. Pomjakushina, J. H. Dil, G. Landolt, J.-Z. Ma, X. Shi, R. S. Dhaka, N.

C. Plumb, M. Radović, V. N. Strocov, T. K. Kim, M. Hoesch, K. Conder, J. Mesot, H. Ding and M. Shi arXiv preprint arXiv:1405.0165 (2014).

[5] T. Komatsubara, N. Sato, S. Kunii, I. Oguro, Y. Furukawa, Y. Ōnuki, and T. Kasuya, Journal of Magnetism and Magnetic Materials **31**, 368 (1983).

[6] H. Hagiwara, H. Hiraoka, R. Terasaki, M. Ishii, and R. Shimizu, Scanning Electron Microsc., 473 (1982).

[7] K. Qi, Z. Lin, W. Chen, G. Cao, J. Cheng, and X. Sun, Applied Physics Letters **93**, 093503 (2008).

[8] Y. Onuki, A. Umezawa, W. Kwok, G. Crabtree, M. Nishihara, T. Yamazaki, T. Omi, and T. Komatsubara, Physical Review B 40, 11195 (1989).

[9] J. Xu, G. Hou, Takao Mori, H. Li, Y. Wang, Y. Chang, Y. Luo, B. Yu, Y. Ma and T. Zhai, Advanced Functional Materials 23, 5038 (2013).

[10] C. McCarthy and C. Tompson, Journal of Physics and Chemistry of Solids **41**, 1319 (1980).

[11] S. Patil, G. Adhikary, G. Balakrishnan, and K. Maiti, Solid State Communications **151**, 326 (2011).

[12] M. Aronson, J. Sarrao, Z. Fisk, M. Whitton, and B. Brandt, Physical Review B **59**, 4720 (1999).

[13] M. Sundermann, H. Yava, s, K. Chen, D.J. Kim, Z. Fisk, D. Kasinathan, M. W. Haverkort,
P. Thalmeier, A. Severing and L. H. Tjeng, Physical review letters 120, 016402 (2018).

[14] F. Lu, J. Zhao, H. Weng, Z. Fang, and X. Dai, Physical review letters 110, 096401 (2013).

[15] N. Singh, S. M. Saini, T. Nautiyal, and S. Auluck, Journal of Physics: Condensed Matter19, 346226 (2007).

[16] S. Luca, M. Amara, R. Galéra, F. Givord, S. Granovsky, O. Isnard, and B. Beneu, PhysicaB: Condensed Matter **350**, E39 (2004).

[17] T. Tanaka, R. Nishitani, C. Oshima, E. Bannai, and S. Kawai, Journal of Applied Physics51, 3877 (1980).

[18] H. Nozaki, T. Tanaka, and Y. Ishizawa, Journal of Physics C: Solid State Physics **13**, 2751 (1980).

[19] R. Galera, D. Osterman, J. Axe, S. Kunii, and T. Kasuya, Journal of Applied Physics 63, 3580 (1988).

[20] T. Kasuya, Journal of magnetism and magnetic materials **174**, L28 (1997).

[21] M. Amara, S. E. Luca, R.-M. Galéra, F. Givord, C. Detlefs, and S. Kunii, Physical Review B 72, 064447 (2005).

[22] H. Jamnezhad and M. Jafari, Journal of Computational Electronics 16, 272 (2017).

[23] P. Larson, W. R. Lambrecht, A. Chantis, and M. van Schilfgaarde, Physical Review B 75, 045114 (2007).

[24] L. Xiao, Y. Su, J. Ran, Y. Liu, W. Qiu, J. Wu, F. Lu, F. Shao, D.Tang and P.Peng, Journal

of Applied Physics **119**, 164903 (2016).

[25] C. J. Först, C. R. Ashman, K. Schwarz, and P. E. Blöchl, Nature 427, 53 (2004).

[26] G. Kresse and D. Joubert, Physical Review B 59, 1758 (1999).

[27] G. Kresse, Phys. Rev. B 54, 11169 (1996).

[28] J. P. Perdew, Phys. Rev. Lett. 77, 3865 (1996).

[29] V. I. Anisimov, J. Zaanen, and O. K. Andersen, Physical Review B 44, 943 (1991).

[30] SeeSupplementalMaterialathttp://link.aps.org/supplemental/10.1103/PhysRevB.97.085401for detailed information aboutSchematic of six antiferromagnetic structures (A, C, D, E, F, and G-AFM) in a 2×2×2superlattice.

[31] J. Heyd, G. E. Scuseria, and M. Ernzerhof, The Journal of Chemical Physics **118**, 8207 (2003).

[32] G. Grechnev, A. Logosha, A. Panfilov, and N. Y. Shitsevalova, Journal of Alloys and Compounds **511**, 5 (2012).

[33] S.-i. Kimura, T. Nanba, S. Kunii, and T. Kasuya, Physical Review B 50, 1406 (1994).

[34] See	Supplemental	Material	at

http://link.aps.org/supplemental/10.1103/PhysRevB.97.085401 for detailed information about

the Charge density distribution of GdB₆.

[35] J. L. Da Silva, M. V. Ganduglia-Pirovano, J. Sauer, V. Bayer, and G. Kresse, Physical Review B **75**, 045121 (2007).

[36] M. Cococcioni and S. De Gironcoli, Physical Review B 71, 035105 (2005).

[37] Y. Fu and D. J. Singh, arXiv preprint arXiv:1904.11825 (2019).

[38] F. Aryasetiawan, M. Imada, A. Georges, G. Kotliar, S. Biermann, and A. Lichtenstein, Physical Review B 70, 195104 (2004).

[39] A. Hampel, P. Liu, C. Franchini, and C. Ederer, npj Quantum Materials 4, 5 (2019).

[40] M. Anisimov, V. Glushkov, A. Bogach, S. Demishev, N. Samarin, A. Samarin, N.

Shitsevalova, A. Levchenko, V. Filipov, S. Gábani, K. Flachbart and N. Sluchanko, Acta

Physica Polonica, A. 131 (2017).

[41] T. Y. Kosolapova, *Handbook of high temperature compounds: properties, production, applications* (CRC Press, 1990).

[42] D. Orobengoa, C. Capillas, M. I. Aroyo, and J. M. Perez-Mato, Journal of Applied Crystallography 42, 820 (2009).