



CHORUS

This is the accepted manuscript made available via CHORUS. The article has been published as:

Gate-tunable quantum Hall effects in defect-suppressed $\text{Bi}_{\{2\}}\text{Se}_{\{3\}}$ films

Nikesh Koirala, Maryam Salehi, Jisoo Moon, and Seongshik Oh

Phys. Rev. B **100**, 085404 — Published 5 August 2019

DOI: [10.1103/PhysRevB.100.085404](https://doi.org/10.1103/PhysRevB.100.085404)

1 **Gate tunable quantum Hall effects in defect-suppressed Bi₂Se₃ films**

2 Nikesh Koirala^{1†*}, Maryam Salehi², Jisoo Moon¹, Seongshik Oh^{1*}

3 ¹Department of Physics & Astronomy, Rutgers, The State University of New Jersey, Piscataway,
4 New Jersey 08854, U.S.A.

5 ²Department of Materials Science and Engineering, Rutgers, The State University of New Jersey,
6 Piscataway, New Jersey 08854, U.S.A.

7 *Correspondence should be addressed to nikesh@mit.edu and ohsean@physics.rutgers.edu

8 †Present Address: Department of Physics, Massachusetts Institute of Technology, Cambridge,
9 MA 02139 U.S.A.

10 **Despite many years of efforts, attempts to reach the quantum regime of topological surface**
11 **states (TSS) on an electrically tunable topological insulator (TI) platform have so far failed**
12 **on binary TI compounds such as Bi₂Se₃ due to high density of interfacial defects. Here,**
13 **utilizing an optimal buffer layer on a gatable substrate, we demonstrate the first**
14 **electrically tunable quantum Hall effects (QHE) on TSS of Bi₂Se₃. On the *n*-side, well-**
15 **defined QHE shows up, but it diminishes near the charge neutrality point (CNP) and**
16 **completely disappears on the *p*-side. Furthermore, around the CNP the system transitions**
17 **from a metallic to a highly resistive state as the magnetic field is increased, whose**
18 **temperature dependence indicates presence of an insulating ground state at high magnetic**
19 **fields.**

20

21 Key words: Bi₂Se₃, quantum Hall effect, topological insulator, electric field effect, charge
22 neutrality point

23

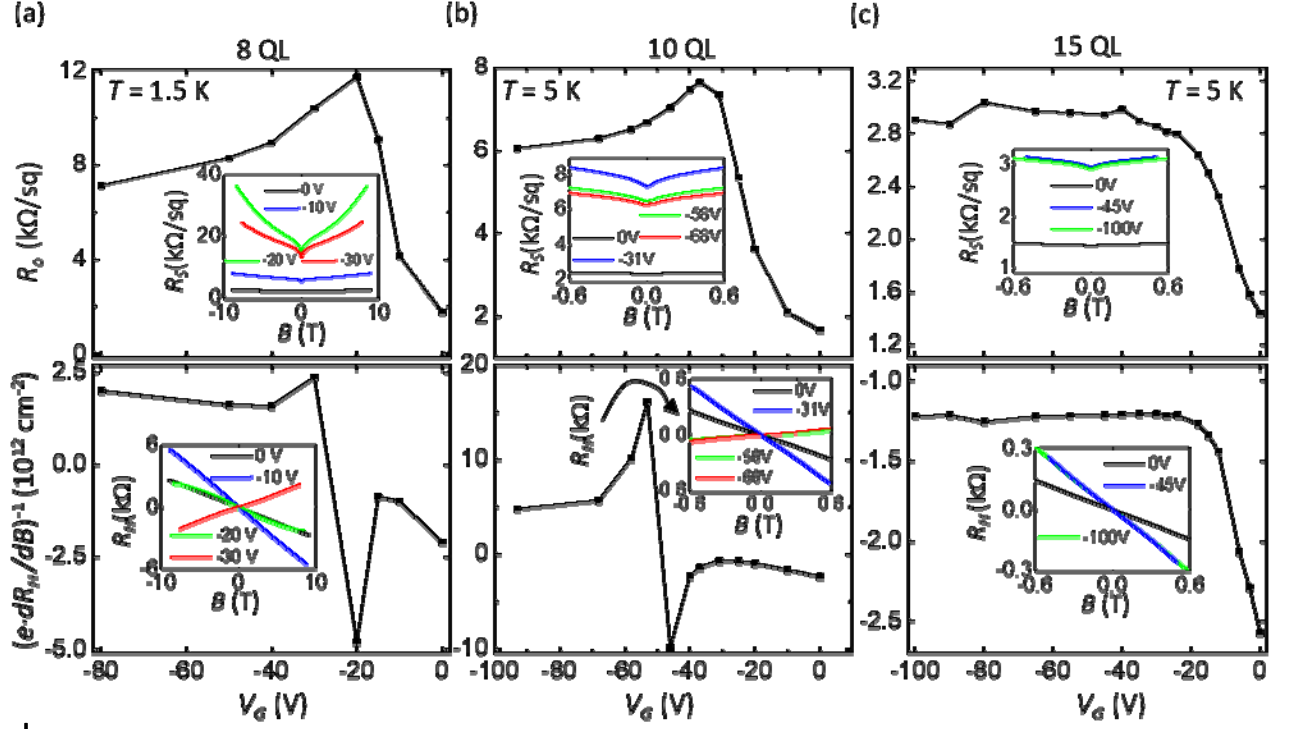
24 TSS provides a rich playground for a number of topological quantum effects such as topological
25 magnetoelectric effect, Majorana fermions and QHE[1-8]. However, due to high level of surface
26 Fermi level originating from unintended native dopants, it has been challenging to access the
27 quantum regime of TSS. In particular, even if QHE is one of the most intensely studied
28 phenomena in 2D systems, gate tuned studies of QHE of TIs has been so far limited only to
29 ternary or quaternary compounds such as $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ and $(\text{Bi}_x\text{Sb}_{2-x}\text{Se}_y\text{Te}_{3-y})$ [4,5,9]. *On the other*
30 *hand, binary compounds can potentially provide cleaner platforms compared with quaternary or*
31 *tertiary solid solutions, allowing better access to the Dirac point, due to increased carrier*
32 *mobilities and reduced electron-hole puddles.* For the binary compounds such as the prototypical
33 TI Bi_2Se_3 , tracking evolution of QHE as a function of gate voltage has not been possible due to
34 high density of bulk and interfacial defects[10,11]. Recently, however, utilizing a structurally
35 and chemically-matched buffer layer that solves the defect problem, QHE was observed in
36 Bi_2Se_3 thin films[12,13]. Here, by adapting this buffer layer scheme to a gatable $\text{SrTiO}_3(111)$
37 substrate[14], we present the first gate-dependent study of QHE in Bi_2Se_3 .

38 Low-carrier density Bi_2Se_3 thin films were grown on an electrically insulating buffer
39 layer, which comprises a heterostructure of 5 QL In_2Se_3 – 4 QL $(\text{Bi}_{0.5}\text{In}_{0.5})_2\text{Se}_3$ grown on SrTiO_3
40 (111) substrate following the recipe of ref. 9 [15]. The films were then capped *in situ* either by a
41 100 nm Se or a 50 nm $\text{MoO}_3/50$ nm Se layer to protect against ambient contamination[16]. Then,
42 a ~ 100 nm thick Cu layer was deposited *ex situ* on the back surface of SrTiO_3 substrate as a back
43 gate. The films were then scratched into millimeter sized Hall bars using a metal mask and a
44 tweezer, and indium leads were used to make electrical contacts [15].

45 On these Hall bar patterns, Hall (R_H) and sheet resistance (R_S) were initially measured at
46 magnetic field (B) up to ± 0.6 T in a cryostat at $T = 5$ K. The measured raw data were

47 symmetrized or anti-symmetrized to eliminate mixing of longitudinal and Hall resistances due to
48 imperfection in the measurement geometry [15]. $[e \cdot (dR_H/dB)]^{-1}$, which corresponds to sheet
49 carrier density (n_S) for single-carrier species transport, and mobility (μ) = $(R_o \cdot n_S \cdot e)^{-1}$ were then
50 calculated, where dR_H/dB is the slope of low field linear part of Hall resistance (unless otherwise
51 stated), e is the electronic charge and R_o is the zero-field sheet resistance. These films have n -
52 type carriers with $n_S \approx 5 \times 10^{12} \text{ cm}^{-2}$ and $\mu \approx 1,000 - 3,000 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ [15]. Compared to the films
53 grown directly on SrTiO₃(111), where n_S is typically $\approx 4 \times 10^{13} \text{ cm}^{-2}$, the buffer-layer grown
54 films exhibit an order of magnitude decrease in the defect density[17], which is consistent with
55 our previous report[12]. This low sheet carrier density obtained with the buffer layer was
56 essential for reaching the quantum regime of TSS via gating as we present below.

57 In the rest of the paper, we focus on the gate voltage dependence of R_H and R_S in
58 MoO₃/Se-capped films. As reported previously, MoO₃ capping further reduces the n -type Fermi
59 level of Bi₂Se₃ films toward the charge neutrality point (CNP)[12,18]. We measured films of
60 three different thicknesses: 8, 10 and 15 QL. 8 QL film was measured at $T = 1.5 \text{ K}$ and B up to \pm
61 9 T and 10 QL and 15 QL films were measured at $T = 5 \text{ K}$ and B up to $\pm 0.6 \text{ T}$.



62

63

Figure 1 | Gate-dependent magneto-transport data for Bi_2Se_3 films. (a)-(c), Zero-field sheet

64

resistance (R_o , upper panel) and $(e-dR_H/dB)^{-1}$ (lower panel), which corresponds to sheet carrier density

65

(n_S) for single carrier species transport as a function of back-gate voltage, V_G , for 8, 10 and 15 QL films

66

respectively. Solid black lines are a guide to the eye. The insets show magnetoresistance (R_S , upper panel)

67

and corresponding Hall resistance (R_H , lower panel) as a function of magnetic field, B , taken at several

68

representative back-gate voltage values from which R_o and n_S were extracted. Note the different magnetic

69

field range and temperature for 8 QL compared to 10 and 15 QL films. Ambipolar behavior is observed in

70

(a-b).

71

Figures 1 (a)-(c) show R_o (upper panel) and $(e-dR_H/dB)^{-1}$ (lower panel; for single species

72

transport it is equivalent to n_S) as a function of back-gate voltage (V_G) of 8, 10 and 15 QL films,

73

respectively. In all three samples, the n -type carrier density is less than $3 \times 10^{12} \text{ cm}^{-2}$ at $V_G = 0 \text{ V}$,

74

which is well below the maximum carrier density ($\sim 1 \times 10^{13} \text{ cm}^{-2}$) required to make the bulk

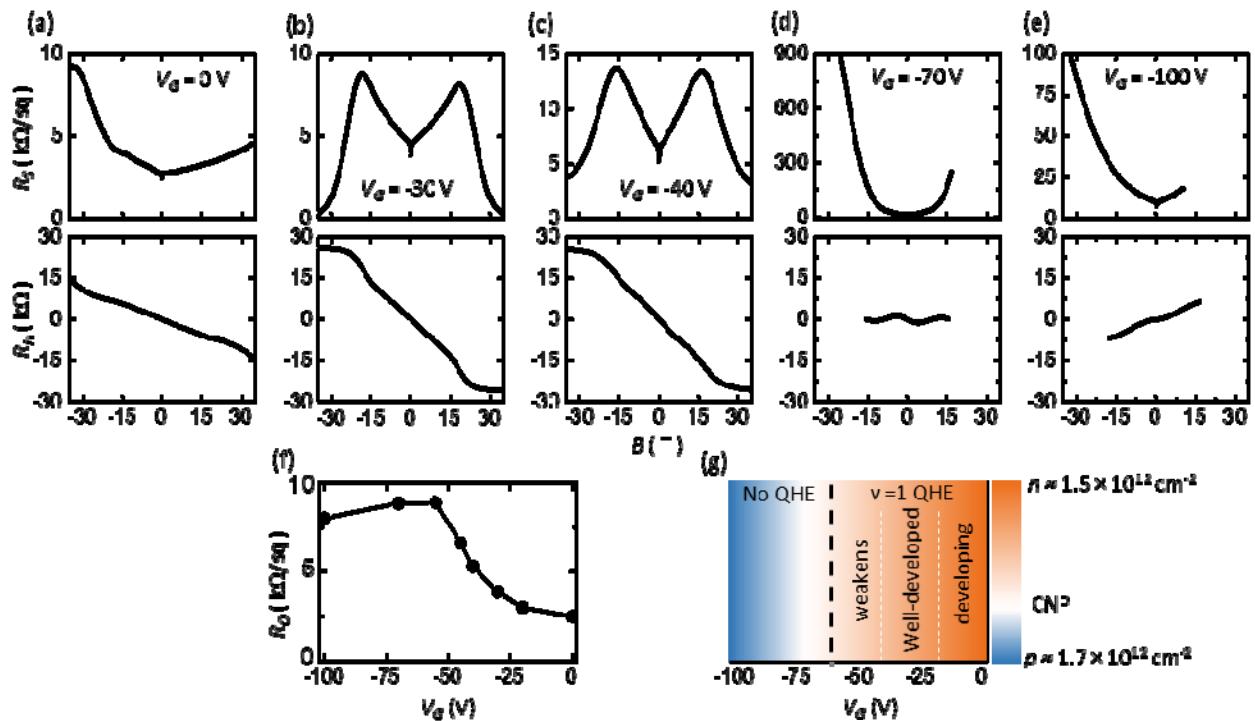
75

state of TIs insulating[19]. For the 15 QL film, we were able to modulate n_S from $2.8 \times 10^{12} \text{ cm}^{-2}$

76

to $1.5 \times 10^{12} \text{ cm}^{-2}$ and R_o from $\sim 1.5 \text{ k}\Omega/\text{sq}$ to $\sim 2.8 \text{ k}\Omega/\text{sq}$, as V_G is tuned from 0 to -20 V.

77 However, ambipolar transport was not observed in this film, presumably because 15 QL is too
 78 thick for its top surface to be electrostatically modulated by the bottom gating. For 8 and 10 QL
 79 films, R_o increases with V_G , reaches a maximum value (for example at $V_G \approx -37$ V for 10 QL
 80 film) and then decreases with further increase in V_G . Concurrently, (*n*-type) n_S decreases with V_G
 81 and eventually changes to *p*-type (for example at $V_G \approx -45$ V for 10 QL film). Such an ambipolar
 82 behavior not only confirms the TSS conduction, but also the tunability of chemical potential
 83 across the CNP[20]. Therefore, we focus on 10 QL films below and carry out more in-depth
 84 measurements up to much higher magnetic fields (34.5 T).



85
 86 **Figure 2 |** QHE as a function of magnetic field at several gate voltage values for a 10 QL film. (a)-
 87 (e) sheet resistance (R_S , upper panel) and Hall resistance (R_H , lower panel) up to $|B| = 34.5$ T at $V_G = 0, -$
 88 $30, -40, -70$ and -100 V respectively. Change in the sign of Hall effect and corresponding maxima in (f)
 89 zero field sheet resistance R_0 at $V_G \approx -70$ V indicates that *p*-type carriers dominate the transport for $V_G < -$
 90 70 V. For *n*-type carriers, $\nu = 1$ QHE is clearly observed at high magnetic fields. (g) Evolution of QHE

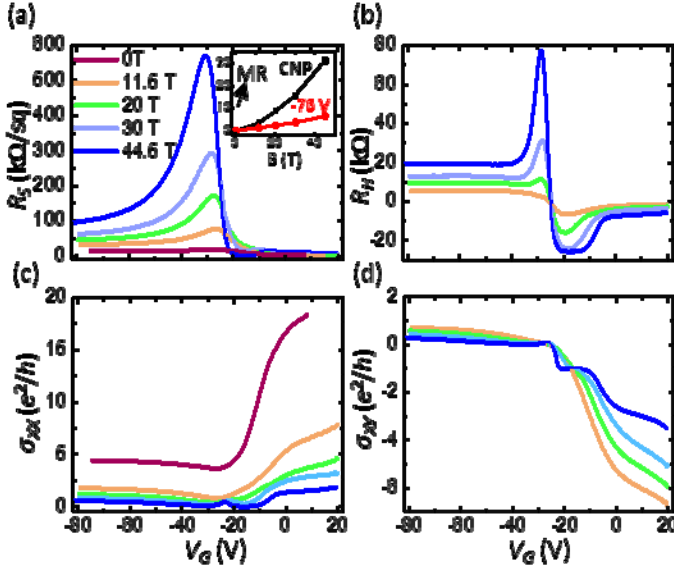
91 with carrier density, where QHE disappears when carriers change from n - to p -type. p -type carrier density
92 for $V_G = -100$ V was estimated from the average slope of Hall effect in (e).

93 Figures 2(a)-(e) show R_S (upper panel) and R_H (lower panel) of another identically
94 prepared 10 QL film as a function of B up to ± 34.5 T for various V_G values at $T = 0.35$ K [15].
95 At low magnetic fields, magnitude of the negative slope of R_H increases as V_G changes from 0 to
96 -40 V, indicating decreasing n -type carrier density from $1.5 \times 10^{12} \text{ cm}^{-2}$ to $5.9 \times 10^{11} \text{ cm}^{-2}$. At V_G
97 $= -70$ V, R_H fluctuates strongly around zero indicating mixed transport from electrons and holes
98 and at $V_G = -100$ V the slope becomes positive albeit non-linear. The nonlinear Hall effect
99 possibly indicates multi-carrier transport likely due to the effect of electron-hole puddles or due
100 to residual n -type carriers on the top surface[5,21]. However, the overall positive slope of R_H
101 indicates that conduction is now dominated by p -type carriers. Figure 2(f) shows corresponding
102 change in R_o , where it increases till $V_G = -55$ V, shows a maximum at $-70 \text{ V} \leq V_G \leq -55 \text{ V}$ and
103 then decreases below $V_G = -70$ V. This indicates that the film goes through CNP at $V_G \approx -70$ V.

104 At high magnetic fields, we observe increasingly developed dips in R_S and plateaus at
105 h/e^2 in R_H for $-30 \text{ V} \leq V_G \leq 0 \text{ V}$ indicating $\nu = 1$ QH state. Additionally, we observe developing
106 plateau-like features at $R_H \approx h/3e^2$ in the same voltage range, which is consistent with top and
107 bottom surfaces having similar carrier density in this gate voltage range [15]. At $V_G = -40$ V both
108 the dip in R_S and the plateau in R_H are less well developed than at $V_G = -30$ V indicating that the
109 $\nu = 1$ QH state weakens as the Fermi level is lowered toward CNP. For $V_G = -70$ V (-100 V), QH
110 signature is completely gone and R_S increases monotonically with B and reaches ~ 878 k Ω /sq
111 (~ 90 k Ω /sq) at $B = 25$ T (32 T), corresponding to $\sim 10,000\%$ ($\sim 1000\%$) of magnetoresistance as
112 defined by $\text{MR}\% = \frac{R_S(B) - R_S(0)}{R_S(0)} \times 100 \%$. In Fig. 2(g), we summarize our observation, where $\nu =$
113 1 QHE emerges with decreasing n -type carrier density until $V_G \approx -30$ V, then diminishes while

114 approaching CNP and gives way to a highly resistive state in the p -regime. The color plot in Fig.
115 2(g) was obtained by smoothly interpolating between the carrier densities measured at $V_G = 0$ V
116 and at $V_G = -100$ V. We note that the p -type carrier density at $V_G = -100$ V in Fig. 2(g) was
117 estimated by taking the average, rather than the low-field, slope of the Hall resistance.

118 In order to observe continuous evolution of transport with V_G , we have measured R_S and
119 R_H as a function of V_G at various magnetic fields B from 0 to 44.5 T and various temperatures T
120 from 0.35 K to 9 K on another 10 QL thick film. Figures 3(a) and 3(b) show R_S and R_H ,
121 respectively, at different magnetic fields and at $T = 0.35$ K [15]. For all magnetic fields we
122 observe a peak in R_S and a change in the sign of R_H from negative to positive at -30 V $\lesssim V_G \lesssim -$
123 26 V, indicating that p -type carriers dominate the transport below these gate voltages. For higher
124 magnetic fields ($B > 23$ T), we observe a developing dip in R_S and a plateau in $R_H \approx -25.8$ k Ω for
125 -20 V $\lesssim V_G \lesssim -15$ V indicative of $\nu = 1$ QH state for n -type carriers. Near CNP and for p -type
126 carriers, we observe neither the dips in R_S nor the plateaus in R_H in magnetic field up to 44.5 T.
127 Inset of Fig. 3(a) shows the magnetic field dependence of R_S at CNP (R_{CNP}) and at $V_G = -76$ V ($R_{$
128 $-76V}$, where p -type carriers dominate) indicating that R_S increases monotonically with B for both
129 CNP and p -type carriers, reaching as high as ~ 250 k Ω (44 k Ω) and ~ 44 k Ω (18 k Ω) at $B = 44.5$
130 T (11.5 T), respectively, which correspond to $\sim 3,000\%$ (600%) and 650% (100%) of
131 magnetoresistance.



132

133 **Figure 3 | Gate voltage dependent transport properties of a 10 QL film at $T=0.35$ K and B field up**
 134 **to 44.5 T. (a) sheet resistance (R_S) and (b) Hall resistance (R_H) at $T=0.35$ K as a function of V_G at several**
 135 **B values from 0 to 44.5 T. Corresponding (c) sheet (σ_{XX}) and (d) Hall (σ_{XY}) conductance. $\nu = 1$ QHE is**
 136 **observed at $V_G \approx -15$ V to -20 V and non-saturating magnetoresistance (MR) = ————— with B is**
 137 **observed for $V_G \lesssim -21$ V as plotted in inset of (a) for CNP and $V_G = -76$ V.**

138

139

140

141

142

143

144

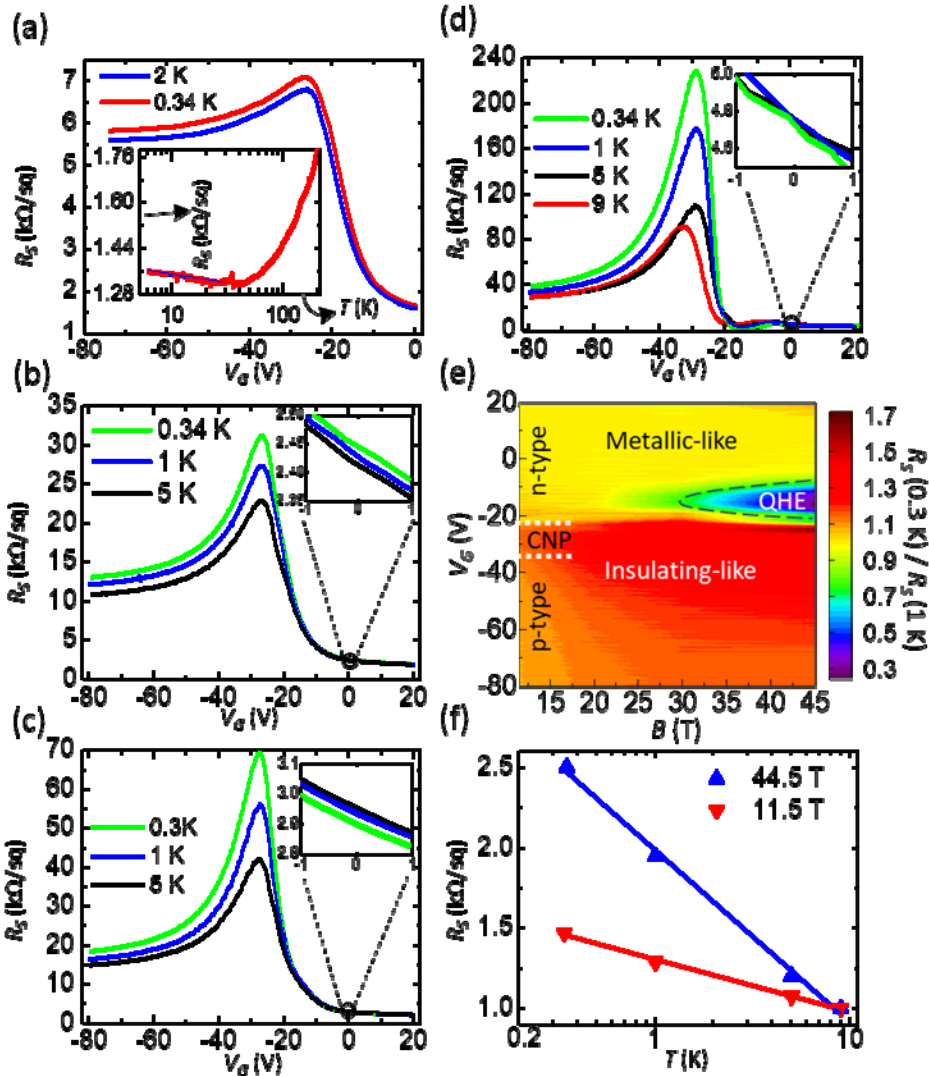
145

146

In order to get an additional perspective, we plot sheet conductance $\sigma_{XX} = R_S / (R_S^2 + R_H^2)$ and Hall conductance $\sigma_{XY} = R_H / (R_S^2 + R_H^2)$ in Fig. 3(c) and 3(d), respectively. Consistent with resistance plots, we observe a σ_{XY} plateau at $\sim h/e^2$ and a minimum in σ_{XX} at $-20 \text{ V} \lesssim V_G \lesssim -15 \text{ V}$ indicative of $\nu = 1$ QHE. Apart from a plateau-like feature at $\sigma_{XY} \approx 0$ and corresponding minimum in σ_{XX} for $-35 \text{ V} \lesssim V_G \lesssim -30 \text{ V}$, which can possibly indicate the $\nu = 0$ state, no features resembling QHE are observed in conductance plots for $V_G \lesssim -30 \text{ V}$ [4,5,9,22], implying that QHE is lost when p -type carriers dominate the transport. The lack of QHE on the p -side is consistent with both the recent transport result on compensation-doped Bi_2Se_3 films and the lack of Landau levels (LL) on the p -side of Bi_2Se_3 in scanning tunneling spectroscopy measurements[13,23,24].

147 It can be explained by the proximity of the Dirac point to the bulk valence band and the much
148 broader surface band on the p -side of Bi_2Se_3 [25,26]. This is in marked contrast with Sb-based
149 TIs, which exhibit QHE and LLs for both n - and p -sides due to relatively symmetric surface
150 bands with a well-exposed Dirac point[4,27,28,27].

151 Next, we discuss temperature dependence of R_S versus V_G at different magnetic fields in
152 order to understand the behavior of p -type carries in Bi_2Se_3 . As shown in Fig. 4(a), zero-field R_S
153 is $< 7.5 \text{ k}\Omega/\text{sq}$, which is much lower than the quantum resistance ($25.8 \text{ k}\Omega$), and increases only
154 slightly ($\sim 4\%$ at CNP) at $T = 0.35 \text{ K}$ compared to $T = 2 \text{ K}$ for all V_G . At higher temperatures ($T \approx$
155 6 to 12 K), we observe similarly small variation in R_S with temperature in a different but
156 identically prepared 10 QL film [15]. In the inset of Fig. 4(a) we show R_S versus T at $V_G = 0 \text{ V}$,
157 where small upturn at low temperatures is observed, consistent with previous studies on TI films.
158 In addition, we observe weak anti-localization at low fields for all V_G values and in all samples
159 (see Fig. 2 and 3). Both of these observations are consistent with gapless Dirac band transport in
160 the presence of disorder[4,29].



161

162 **Figure 4 | Temperature dependence of resistance at high magnetic field.** Sheet resistance (R_S) at $B =$

163 **(a)** 0 T **(b)** 11.5 T **(c)** 20 T and **(d)** 44.5 T as a function of V_G at several different temperatures. Note that

164 the film was not measured at 9 K in (c). Inset in (a) shows semi-log plot of R_S versus T at $V_G = 0$ V

165 indicating a small upturn at low temperatures, which fits $\log(T)$ dependence as indicated by the blue line.

166 Insets in (b-d) show behavior of R_S around $V_G = 0$ V in greater detail. **(e)** 2D plot of ratio

167 $R_S(0.3\text{ K})/R_S(1\text{ K})$ plotted as a function of V_G and B . Metallic-like behavior is observed for $V_G > -21$ V while, for V_G

168 $\lesssim -21$ V insulating-like behavior is observed. **(f)** Semi-log plot of temperature dependence of normalized

169 R_{CNP} at $B = 11.5$ T and 44.5 T, where larger slope at 44.5 T indicates stronger insulating tendency: the
170 solid lines are least-square fits using $R_{CNP} \sim \log(T)$.

171 In order to see how transport changes at higher fields, we plot R_S versus V_G at 11.5 T, 20
172 T and 44.5 T for $T = 0.35$ K – 9 K in Fig. 4 (b)-(d), respectively [15]. A slight shift in V_G
173 corresponding to CNP at $T = 9$ K compared to lower temperatures is observed, but does not
174 affect our analysis. In Fig. 4 (b)-(d), $R_S(T_1) \approx R_S(T_2)$ for $V_G > -21$ V indicating metallic
175 behavior, while for $V_G \lesssim -21$ V, $R_S(T_1) \gg R_S(T_2)$ suggesting an insulating behavior for these
176 gate voltages for all three magnetic fields, where $T_1 < T_2$ are temperatures. Figure 4(e)
177 summarizes this observation, where we show a 2D plot of the ratio of R_S at $T = 0.3$ K to that at T
178 $= 1$ K ($R_{0.3\text{ K}}/R_{1\text{ K}}$) as a function of V_G and B . n -type carrier region shows metallic-like behavior
179 (i.e. $R_{0.3\text{ K}}/R_{1\text{ K}} \approx 1$) along with QHE at high magnetic fields, while an insulator-like highly
180 resistive state (i.e. $R_{0.3\text{ K}}/R_{1\text{ K}} > 1$) is observed near CNP and for p -type carrier region at high
181 fields.

182 In order to further understand the nature of the highly-resistive state, we have plotted
183 $R_{CNP}(T)/R_{CNP}(9\text{ K})$ as a function of temperature at $B = 11.5$ T and 44.5 T in Fig. 4(f). R_{CNP}
184 $(T)/R_{CNP}(9\text{ K})$ increases logarithmically with decreasing temperature for both $B = 11.5$ T and
185 44.5 T, with stronger insulating behavior observed at higher field as indicated by greater slope of
186 R_{CNP} vs. $\log(T)$ for $B = 44.5$ T. For comparison, insets of Fig. 4 (b)-(d) show an enlarged view of
187 R_S versus V_G at $V_G \approx 0$ V, where R_S either decreases or does not increase significantly with
188 decreasing temperature indicating a metallic behavior. Such an increasingly insulating behavior
189 near CNP at higher magnetic fields indicates presence of a magnetic-field-induced insulator
190 phase, whose origin remains unknown at present. Local and non-local measurements at lower

191 temperatures and higher magnetic fields could elucidate the nature of this ground state, which we
192 leave for future work.

193 In conclusion, we have studied gate-dependent QHE on low-carrier density Bi₂Se₃ thin
194 films for the first time by employing a novel buffer layer growth method on gate-amenable
195 SrTiO₃ substrates. At low fields we observe ambipolar transport for thinner films, and at high
196 fields we observe $\nu = 1$ QHE for *n*-type carriers. On the other hand, for CNP and *p*-type carriers
197 we observe non-saturating magnetoresistance up to $B = 44.5$ T, whose temperature dependence
198 point to the existence of a magnetic-field-induced insulating state. Further experimental and
199 theoretical efforts are necessary to clarify its origin.

200 **Acknowledgements:**

201 This work is supported by the Gordon and Betty Moore Foundation's EPIQS Initiative
202 (GBMF4418) and National Science Foundation (NSF) grant EFMA-1542798. A portion of this
203 work was performed at the National High Magnetic Field Laboratory which is supported by NSF
204 Cooperative Agreement No. DMR-1644779 and the State of Florida.

205 **Author contributions:**

206 N.K. and S.O. conceived the experiment. N.K., M.S. and J.M. synthesized the samples
207 and performed transport measurements. N.K. and S.O. wrote the manuscript with inputs from all
208 the authors. All authors contributed to the scientific analysis and manuscript revisions.

209

210 **References:**

211 ¹ A.M. Essin, J.E. Moore, and D. Vanderbilt, Phys. Rev. Lett. **102**, 146805 (2009).

212 ² L. Wu, M. Salehi, N. Koirala, J. Moon, S. Oh, and N.P. Armitage, Science **354**, 1124 (2016).

213 ³ L. Fu and C.L. Kane, Phys. Rev. Lett. **100**, 96407 (2008).

214 ⁴ R. Yoshimi, A. Tsukazaki, Y. Kozuka, J. Falson, K.S. Takahashi, J.G. Checkelsky, N. Nagaosa,
215 M. Kawasaki, and Y. Tokura, *Nat. Commun.* **6**, 6627 (2015).

216 ⁵ Y. Xu, I. Miotkowski, C. Liu, J. Tian, H. Nam, N. Alidoust, J. Hu, C.-K. Shih, M.Z. Hasan, and
217 Y.P. Chen, *Nat. Phys.* **10**, 956 (2014).

218 ⁶ Q.L. He, L. Pan, A.L. Stern, E.C. Burks, X. Che, G. Yin, J. Wang, B. Lian, Q. Zhou, E.S. Choi,
219 K. Murata, X. Kou, Z. Chen, T. Nie, Q. Shao, Y. Fan, S.-C. Zhang, K. Liu, J. Xia, and K.L.
220 Wang, *Science* **357**, 294 (2017).

221 ⁷ D. Xiao, J. Jiang, J.-H. Shin, W. Wang, F. Wang, Y.-F. Zhao, C. Liu, W. Wu, M.H.W. Chan,
222 N. Samarth, and C.-Z. Chang, *Phys. Rev. Lett.* **120**, 56801 (2018).

223 ⁸ C.-Z. Chang, J. Zhang, X. Feng, J. Shen, Z. Zhang, M. Guo, K. Li, Y. Ou, P. Wei, L.-L. Wang,
224 Z.-Q. Ji, Y. Feng, S. Ji, X. Chen, J. Jia, X. Dai, Z. Fang, S.-C. Zhang, K. He, Y. Wang, L. Lu,
225 X.-C. Ma, and Q.-K. Xue, *Science* **340**, 167 (2013).

226 ⁹ Y. Xu, I. Miotkowski, and Y.P. Chen, *Nat. Commun.* **7**, 11434 (2016).

227 ¹⁰ H. Zhang, C.-X. Liu, X.-L. Qi, X. Dai, Z. Fang, and S.-C. Zhang, *Nat. Phys.* **5**, 438 (2009).

228 ¹¹ H. Cao, J. Tian, I. Miotkowski, T. Shen, J. Hu, S. Qiao, and Y.P. Chen, *Phys. Rev. Lett.* **108**,
229 216803 (2012).

230 ¹² N. Koirala, M. Brahlek, M. Salehi, L. Wu, J. Dai, J. Waugh, T. Nummy, M.-G. Han, J. Moon,
231 Y. Zhu, D. Dessau, W. Wu, N.P. Armitage, and S. Oh, *Nano Lett.* **15**, 8245 (2015).

232 ¹³ J. Moon, N. Koirala, M. Salehi, W. Zhang, W. Wu, and S. Oh, *Nano Lett.* **18**, 820 (2018).

233 ¹⁴ R.C. Neville, B. Hoeneisen, and C.A. Mead, *J. Appl. Phys.* **43**, 2124 (1972).

234 ¹⁵ See Supplemental Material.

235 ¹⁶ M. Salehi, M. Brahlek, N. Koirala, J. Moon, L. Wu, N.P. Armitage, and S. Oh, *APL Mater.* **3**,
236 91101 (2015).

237 ¹⁷ J. Chen, H.J. Qin, F. Yang, J. Liu, T. Guan, F.M. Qu, G.H. Zhang, J.R. Shi, X.C. Xie, C.L.
238 Yang, K.H. Wu, Y.Q. Li, and L. Lu, *Phys. Rev. Lett.* **105**, 176602 (2010).

239 ¹⁸ M.T. Edmonds, J.T. Hellerstedt, A. Tadich, A. Schenk, K.M. O'Donnell, J. Tosado, N.P.
240 Butch, P. Syers, J. Paglione, and M.S. Fuhrer, *ACS Nano* **8**, 6400 (2014).

241 ¹⁹ M. Brahlek, N. Koirala, M. Salehi, N. Bansal, and S. Oh, *Phys. Rev. Lett.* **113**, 026801 (2014).

242 ²⁰ D. Kim, S. Cho, N.P. Butch, P. Syers, K. Kirshenbaum, S. Adam, J. Paglione, and M.S.
243 Fuhrer, *Nat. Phys.* **8**, 459 (2012).

244 ²¹ H. Beidenkopf, P. Roushan, J. Seo, L. Gorman, I. Drozdov, Y.S. Hor, R.J. Cava, and A.
245 Yazdani, *Nat. Phys.* **7**, 939 (2011).

246 ²² T. Morimoto, A. Furusaki, and N. Nagaosa, *Phys. Rev. Lett.* **114**, 146803 (2015).

247 ²³ T. Hanaguri, K. Igarashi, M. Kawamura, H. Takagi, and T. Sasagawa, *Phys. Rev. B* **82**,
248 81305(R) (2010).

249 ²⁴ P. Cheng, C. Song, T. Zhang, Y. Zhang, Y. Wang, J.-F. Jia, J. Wang, Y. Wang, B.-F. Zhu, X.
250 Chen, X. Ma, K. He, L. Wang, X. Dai, Z. Fang, X. Xie, X.-L. Qi, C.-X. Liu, S.-C. Zhang, and
251 Q.-K. Xue, *Phys. Rev. Lett.* **105**, 76801 (2010).

252 ²⁵ Y. Xia, D. Qian, D. Hsieh, L. Wray, A. Pal, H. Lin, A. Bansil, D. Grauer, Y.S. Hor, R.J. Cava,
253 and M.Z. Hasan, *Nat. Phys.* **5**, 398 (2009).

254 ²⁶ T. V Menshchikova, M.M. Otrokov, S.S. Tsirkin, D.A. Samorokov, V. V Bebneva, A. Ernst,
255 V.M. Kuznetsov, and E. V Chulkov, *Nano Lett.* **13**, 6064 (2013).

256 ²⁷ J. Zhang, C.-Z. Chang, Z. Zhang, J. Wen, X. Feng, K. Li, M. Liu, K. He, L. Wang, X. Chen,
257 Q.-K. Xue, X. Ma, and Y. Wang, *Nat. Commun.* **2**, 574 (2011).

258 ²⁸ C. Pauly, C. Saunus, M. Liebmann, and M. Morgenstern, *Phys. Rev. B* **92**, 85140 (2015).

259 ²⁹ M. Lang, L. He, X. Kou, P. Upadhyaya, Y. Fan, H. Chu, Y. Jiang, J.H. Bardarson, W. Jiang,

260 E.S. Choi, Y. Wang, N.-C. Yeh, J. Moore, and K.L. Wang, *Nano Lett.* **13**, 48 (2013).

261