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Spectroscopic Investigation of Yb-doped Silica Glass for Solid-State Optical Refrigeration

Esmaeil Mobini, Mostafa Peysokhan, Behnam Abaie, Markus P. Hehlen, and Arash Mafi

1 Department of Physics & Astronomy, University of New Mexico, Albuquerque, NM 87131, USA
2 Center for High Technology Materials, University of New Mexico, Albuquerque, NM 87106, USA
3 Space & Remote Sensing (ISR-2), Los Alamos National Laboratory, Los Alamos, NM 87545, USA

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We argue that a high-purity Yb-doped silica glass can potentially be cooled via anti-Stokes fluorescence optical refrigeration. In order to achieve net solid-state optical refrigeration, it is necessary for the cooling efficiency to be positive. This requires the pump wavelength to be greater than the mean fluorescence wavelength, and the internal quantum efficiency as well as the absorption efficiency to be near unity. Our conclusion is reached by showing, using reasonable assumptions for the host material properties, that the non-radiative decay rate of Yb ions can be made substantially smaller than the radiative decay rate. Therefore, an internal quantum efficiency of near unity can be obtained. Moreover, the background absorption coefficient in high-quality silica glass lies in an acceptable range to guarantee a near unity absorption efficiency at room temperature. Using spectral measurements of the fluorescence from a Yb-doped silica optical fiber at different temperatures, we estimate the minimum achievable temperature in Yb-doped silica glass for different values of the internal quantum efficiency.

I. INTRODUCTION

In solid-state optical refrigeration, anti-Stokes fluorescence removes thermal energy from the material, resulting in net cooling. Solid-state optical cooling was first proposed by Pringsheim in 1929 [1] and was put on a solid thermodynamic foundation by Landau in 1946 [2]. Solid-state optical cooling was first experimentally observed in 1995 by Epstein’s group at Los Alamos National Laboratory in Yb-doped ZBLANP glass [3]. Much attention has since been devoted to solid-state optical refrigeration in different materials and geometries due to its interesting basic science properties and potential applications [4]. The quest for solid-state optical cooling in new configurations and materials is on-going [5].

In particular, solid-state optical refrigeration of Yb-doped silica glass, which is extensively used in high-power fiber lasers, is highly desirable. New generations of high power fiber amplifiers and lasers now operate at few kilo-Watt levels [6]. However, the significant heat-load in high-power operation has hindered the efforts to further scale up the power in fiber lasers and amplifiers [6–9]. Different methods have been developed to manage the heat-load in high-power fiber lasers or amplifiers; in particular, solid-state optical refrigeration via anti-Stokes fluorescence has been suggested as a viable path for heat mitigation [10–12]. So far, there is no report of solid-state optical refrigeration in Yb-doped silica; this manuscript is intended to highlight its possibility.

In this context, Radiation-Balanced Lasers (RBL) were first introduced by Bowman in 1999 [10]. In radiation balancing, the heat that originates from the quantum defect of the laser as well as parasitic absorption can be removed by anti-Stokes fluorescence under a very subtle balance condition between different parameters of a laser (or an amplifier) [10, 11, 13]. In other words, the anti-Stokes fluorescence removes the excess heat generated in the medium. Therefore, heat mitigation by radiation-balancing via anti-Stokes fluorescence is highly desirable and will have great practical implications if it can be achieved in Yb-doped silica glass, which is the material of choice for most high-power fiber lasers and amplifiers [6, 14, 15].

Although the first solid-state optical refrigeration was observed in a Yb-doped ZBLAN glass, it soon became apparent that rare-earth-doped crystals can be cooled more efficiently because i) the absorption lines of the Yb ions in crystals have a smaller inhomogeneous broadening, and ii) crystals can accept higher concentrations of rare-earth ions [4, 5]. As such, rare-earth-doped crystals have become the material of choice for solid-state optical refrigeration [16]. Only recently has there been a renewed interest in glasses, mainly because of their potential in RBL fiber lasers and amplifiers [12].

The investigation of solid-state optical refrigeration can be done either directly or indirectly. In a direct investigation, the material is exposed to a laser in a thermally isolated setup, often in a sophisticated vacuum environment [16], and its temperature is measured directly by a thermal camera or similar methods. In an indirect method, the spectroscopic properties of materials at different temperatures are measured to evaluate the possibility of solid-state optical refrigeration [3, 16–18]. In this manuscript, we use the indirect method to argue for the potential of high-quality Yb-doped silica glass for solid-state optical refrigeration and radiation-balancing in lasers and amplifiers.

In order to characterize the cooling potential of Yb-doped silica glass, we use the cooling efficiency $\eta_c$ defined as

$$\eta_c = \frac{\text{heat removed}}{\text{heat added}}.$$
as [16, 18]

\[ \eta_c(\lambda_p, T) = \eta_q \eta_{abs}(\lambda_p, T) \frac{\lambda_p}{\lambda_f(T)} - 1. \]  

(1)

In Eq. 1, \( \lambda_f \) is the mean fluorescence wavelength and \( \lambda_p \) is the pump wavelength. \( \eta_q \) is the internal quantum efficiency and \( \eta_{abs} \) is the absorption efficiency; they are defined as

\[ \eta_q = \frac{W_r}{W_{tot}}, \quad W_{tot} = W_r + W_{nr}, \]  

(2)

\[ \eta_{abs}(\lambda_p, T) = \frac{\alpha_r(\lambda_p, T)}{\alpha_r(\lambda_p, T) + \alpha_b}, \]  

(3)

where \( W_r, W_{nr}, \) and \( W_{tot} \) are radiative, non-radiative, and total decay rates of the excited state, respectively. \( \alpha_b \) is the background absorption coefficient, and \( \alpha_r \) is the resonant absorption coefficient. Note that \( \alpha_b \) does not contain the attenuation due to scattering as this process does not lead to heating of the material. We have assumed that due to the small cross-sectional area of optical fibers, the fluorescence escape efficiency to be unity [3, 19]. The mean fluorescence wavelength is defined by

\[ \lambda_f(T) = \frac{\int_{\Delta} \lambda S(\lambda, T) d\lambda}{\int_{\Delta} S(\lambda, T) d\lambda}, \]  

(4)

where \( S(\lambda, T) \) is the fluorescence power spectral density, which is a function of the glass temperature \( T \), and \( \Delta \) is the spectral domain encompassing the relevant emission spectral range [16, 18].

In order to achieve net solid-state optical refrigeration, it is necessary for the cooling efficiency to be positive. Therefore, we must show that \( \eta_c > 0 \) is attainable over a range of \( \lambda_p \) and \( T \) values. It can be seen from Eq. 1 that because \( \lambda_b \) and \( \lambda_f \) are often very close to each other in solid-state optical refrigeration schemes (\( \lambda_p \gtrsim \lambda_f \)), the internal quantum efficiency \( \eta_q \) has to be close to unity (\( \eta_q \approx 1 \)) [3, 16]. There are two main processes that lower the internal quantum efficiency: the multi-phonon non-radiative relaxation and the concentration quenching effect [20–27]. We will argue that the multi-phonon non-radiative relaxation is negligible in Yb-doped silica glass and the concentration quenching process can be prevented if the Yb ion density is kept lower than the characteristic Yb ion quenching concentration.

In order to evaluate the absorption efficiency \( \eta_{abs} \), we need to know the background absorption \( \alpha_b \) and resonant absorption \( \alpha_r \) coefficients. For the background absorption coefficient in Yb-doped silica glass, we will use typical values reported in the literature [28–30]. By measuring the power spectral density of a Yb-doped silica optical fiber from its side at different temperatures, \( S(\lambda, T) \), we can also obtain the resonant absorption coefficient \( \alpha_r \) as well as the mean fluorescence wavelength \( \lambda_f(T) \). Therefore, we will have all the necessary parameters in Eq. 1; using this information we will estimate the cooling efficiency and show that solid-state optical refrigeration is feasible in Yb-doped silica glass.

### II. INTERNAL QUANTUM EFFICIENCY

The internal quantum efficiency is the fraction of the radiative decay versus the total decay of an excited state in a medium; therefore, the presence of non-radiative decay channels characterized by the non-radiative decay rate \( W_{nr} \) in Eq. 2 are responsible for decreasing \( \eta_q \) below unity. The non-radiative decay channels in a typical Yb-doped silica glass can be broken down according to the following equation:

\[ W_{nr} = W_{mp} + W_{OH^-} + W_{Yb} + \sum_{TM} W_{TM} + \sum_{RE} W_{RE}. \]  

(5)

The partial non-radiative decay channels are as follows: \( W_{mp} \) represents the multi-phonon decay of the Yb excited state, \( W_{OH^-} \) accounts for non-radiative decay of the Yb excited state via the high-energy vibrational modes of OH$^-$ impurities, \( W_Yb \) accounts for non-radiative decay in Yb ion clusters, and \( W_{TM} \) and \( W_{RE} \) represent non-radiative decay due to interactions of the excited state with various transition-metal and rare-earth ion impurities, respectively.

In the following, we will discuss the various non-radiative decay channels in Eq. 5 and show that they can be made sufficiently small to allow for a near-unity internal quantum efficiency value (\( \eta_q \approx 1 \)). We first begin with the multi-phonon relaxation that originates from the coupling of the excited state with the vibrational wavefunctions of the ground state. Using the energy-gap law [20, 22, 24, 31], we can calculate the decay rate from

\[ W_{mp} = W_0 e^{-\alpha_b(E_g-E_p)}, \]  

(6)

where \( E_p \) is the maximum phonon energy of the host material, and \( E_g \) is the energy gap of the dopant ion (Yb). \( W_0 \) and \( \alpha_b \) are phenomenological parameters, whose values strongly depend on the host-material [20, 22, 24, 31]. Figure 1 shows the multi-phonon non-radiative decay rates of silica and ZBLAN glasses versus the energy gaps of the doped ions at \( T = 300 \) K, using the parameters shown in Table I.

<table>
<thead>
<tr>
<th>Host</th>
<th>( W_0 ) (s$^{-1}$)</th>
<th>( \alpha_b ) (cm)</th>
<th>( E_g ) (cm$^{-1}$)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Silica</td>
<td>7.8 $\times$ 10$^7$</td>
<td>4.7 $\times$ 10$^{-3}$</td>
<td>1.10 $\times$ 10$^3$</td>
</tr>
<tr>
<td>ZBLAN</td>
<td>1.7 $\times$ 10$^4$</td>
<td>2.1 $\times$ 10$^{-3}$</td>
<td>0.58 $\times$ 10$^3$</td>
</tr>
</tbody>
</table>

The vertical solid line in Fig. 1 marks the energy gap of a Yb$^{3+}$ ion. It is evident that for Yb-doped silica glass, the non-radiative decay rate is around \( W_{mp} \text{silica} \approx 10^{-8} \) s$^{-1}$, which is much smaller than the Yb-doped ZBLAN glass multi-phonon decay rate \( W_{mp} \text{ZBLAN} \approx 7.8 \times 10^{7} \) s$^{-1}$. Therefore, we can calculate the decay rate from

\[ W_{mp} = W_0 e^{-\alpha_b(E_g-E_p)}, \]  

(6)

where \( E_p \) is the maximum phonon energy of the host material, and \( E_g \) is the energy gap of the dopant ion (Yb). \( W_0 \) and \( \alpha_b \) are phenomenological parameters, whose values strongly depend on the host-material [20, 22, 24, 31]. Figure 1 shows the multi-phonon non-radiative decay rates of silica and ZBLAN glasses versus the energy gaps of the doped ions at \( T = 300 \) K, using the parameters shown in Table I.

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$10^{-4} \text{ s}^{-1}$. This comparison suggests that with respect to Yb multi-phonon relaxation, silica glass is a more suitable choice for solid-state optical refrigeration than ZBLAN glass.

FIG. 1. Multi-phonon non-radiative decay rate ($W_{nr}$) of Yb-doped ZBLAN and silica glasses versus energy gap ($E_g$) calculated from Eq. 6 and the parameters listed in Table I.

Considering the advances in materials synthesis of fiber preforms, the term $W_{OH}$ in Eq. 5 can be made very small (see e.g. dry fiber technology [32]); therefore, it can be neglected [23]. It has also been shown by Auzel et al. [25] that the total effect of the last three terms in Eq. 5, $W_{Yb} + \Sigma W_{TM} + \Sigma W_{RE}$, can be described by a phenomenological equation based on a limited diffusion process, modeled as a non-radiative dipole-dipole interaction between the ions and impurities [25, 26]. This concentration quenching process can be prevented if the Yb ion density is lower than the critical quenching concentration of the Yb-doped silica glass, which exists because there are impurities. Therefore, the critical quenching concentration is generally a sample specific quantity. That is, it would be higher for lower impurity concentrations. For a Yb ion density smaller than the critical quenching concentration, the internal quantum efficiency can approach $\eta_q \approx 1$ [25–27]. It must be noted that an internal quantum efficiency of $\eta_q = 0.95$ is reported in [33] for Yb-doped silica, which is consistent with our claim that $W_{nr}$ can be made quite small in Yb-doped silica.

III. ABSORPTION EFFICIENCY AND MEAN FLUORESCENCE WAVELENGTH

In order to calculate the cooling efficiency, we still need to obtain the resonant absorption coefficient and the mean fluorescence wavelength, both of which can be obtained from a spectroscopic investigation. The resonant absorption coefficient is used in conjunction with Eq. 3 to determine the absorption efficiency. The setup implemented in our experiment is shown in Fig. 2 and consists of a single-mode Yb-doped silica fiber (DF-1100, from Newport Corporation) that is pumped by a Ti:Sapphire laser at $\lambda = 900 \text{ nm}$. The fiber is mounted on a plate whose temperature is changed from nearly 180 K up to 360 K. The fluorescence of the Yb-doped silica fiber is captured by a multimode fiber from the side of the Yb-doped silica fiber and is sent to an Optical Spectrum Analyzer (Yokogawa-AQ6319). Figure 3 shows the measured fluorescence spectra (power spectral density $S(\lambda, T)$), normalized to their peak values at $\lambda_{peak} \approx 976 \text{ nm}$, at different temperatures.

FIG. 2. The experimental setup is shown. A Ti:Sapphire laser at $\lambda = 900 \text{ nm}$ is used to pump a single-mode Yb-doped silica fiber using a 20X objective lens. The fiber is mounted on a plate whose temperature is changed from nearly 180 K up to 360 K. The temperature of the plate is monitored by a temperature sensor. The fluorescence of the Yb-doped silica fiber is captured by a multimode fiber from the side of the Yb-doped silica fiber and is sent to an Optical Spectrum Analyzer.

FIG. 3. Measured peak normalized emission spectra (fluorescence power spectral density) of DF-1100 Yb-doped silica fiber at different temperatures.

By inserting the measured fluorescence spectra into
Eq. 4 and considering $\Delta \in \{905\text{nm}, 1150\text{nm}\}$, the dependence of the mean fluorescence wavelength on temperature is obtained. The mean fluorescence wavelength follows approximately the following function:

$$
\lambda_f(T) \approx 999 \text{ (nm)} + b \times T^{-1}, \quad b = 2735 \pm 31 \text{nm/K}.
$$

(7)

This behavior at temperatures above 245 K to 360 K is nearly linear, which is similar to that reported in other host materials, such as ZBLAN [17, 20].

In order to calculate the resonant absorption coefficient $\alpha_r$, we first calculate the emission cross section $\sigma_e$, and then use the McCumber relation to obtain the absorption cross section $\sigma_a$ and then the resonant absorption coefficient $\alpha_r$ [34–36]. The emission cross section is obtained from the measured fluorescence power spectral density $S(\lambda, T)$ via the Füchtbauer-Ladenburg equation [36, 37]:

$$
\sigma_e(\lambda, T) = \frac{\lambda^5}{8\pi n^3 c \tau_r(T)} \times \int_\lambda S(\lambda, T)d\lambda,
$$

(8)

where $n$ is the refractive index of the fiber core, $c$ is the speed of light in vacuum, and $\tau_r = W_r^{-1}$ is the radiative lifetime.

In order to apply Eq. 8, the radiative lifetime at each temperature needs to be measured. In high-quality samples for which the non-radiative decay rates are negligible compared to the radiative decay rates, the fluorescence lifetimes are comparable to the radiative lifetimes ($\tau_f \approx \tau_r$); therefore, we measured the fluorescence lifetimes at different temperatures from the side of the fiber [38]. Using this assumption, the emission cross sections at different temperatures are calculated and are shown in Fig. 4. The absorption cross sections can be readily obtained using the McCumber relation:

$$
\sigma_a(\lambda, T) = \sigma_e(\lambda, T) \times Z(\lambda, T),
$$

(9)

$$
Z(\lambda, T) = \exp \left[ \frac{hc}{k_bT} \left( \frac{1}{\lambda} - \frac{1}{\lambda_0} \right) \right],
$$

where $k_b$ is the Boltzmann constant, $h$ is the Planck constant and $\lambda_0 = 976 \text{nm}$ is the wavelength corresponding to the zero-line phonon energy [14, 18, 35]. The resonant absorption coefficient can be calculated from $\sigma_a(\lambda, T)$ in Eq. 9 (and Fig. 4) using

$$
\alpha_r(\lambda, T) = \sigma_a(\lambda, T) \times N.
$$

(10)

Here, we will assume a typical Yb ion density of $N = 5 \times 10^{25} \text{ m}^{-3}$. We now have all the necessary ingredients to calculate the cooling efficiency $\eta_c$ in Eq. 1. We only need to provide a value for the background absorption coefficient in Eq. 3 to determine the absorption efficiency $\eta_{abs}$. Here, we assume a background absorption coefficient of $\alpha_b = 10 \text{dB/km} \approx 2.3 \times 10^{-3} \text{m}$, which is a typical value for commercial grade Yb-doped silica fibers. Using this information, we present a contour plot of the cooling efficiency $\eta_c$ in Fig. 5 as a function of the pump wavelength and temperature, assuming that $\eta_q = 1$. Note that we only know the values of $\alpha_r(\lambda, T)$ at discrete values of temperature $T$ for which our measurements were performed in Fig. 3; the density plot in Fig. 5 is an interpolation of the measured values. It is seen in Fig. 5 that the cooling efficiency decreases with decreasing temperature; this behavior is due to the red-shift of the mean fluorescence wavelength and the decrease in the resonant absorption coefficient with decreasing temperature [16, 20].

In practice, it is impossible to achieve an internal quantum efficiency of unity; therefore, in Fig. 6 we investigate the effect of a non-ideal internal quantum efficiency on the cooling efficiency, for $\lambda_p = 1030 \text{nm}$, as a function of the temperature. The discrete points in Fig. 6 signify the values of $\eta_c$ obtained for the assumed $\eta_q$ at the particular
measured temperatures reported in Fig. 3. The apparent difference between the cooling efficiency obtained for \( \eta_q = 1 \) versus \( \eta_q = 0.98 \) highlights the importance of having a high-quality glass for radiative cooling. While the discrete points in Fig. 6 reveal the main expected behavior of \( \eta_q \) versus the temperature, it is helpful to estimate the minimum achievable temperature for solid-state optical refrigeration in Yb-doped silica glass, subject to the assumptions made about \( \eta_q \), \( N \), and \( \alpha_b \). In order to do so, we next present an analytical fitting to the discrete points in Fig. 6 that can be used to estimate the minimum achievable temperature. The analytical fitting, which is described in the next paragraph, is used in conjunction with Eq. 1 to plot the colored lines for each value of \( \eta_q \) in Fig. 6 and is in reasonable agreement with the experimentally measured discrete data. From the discussions above and Eqs. 8, 9, and 10, we note that \( \alpha_r(\lambda_p, T) \) (at the pump wavelength) can be expressed as:

\[
\alpha_r(\lambda_p, T) \propto \frac{1}{c} \frac{\lambda_p^{5}}{\tau_r(T)} \times \frac{S(\lambda_p, T)}{\int_{\lambda} S(\lambda, T) d\lambda} \times Z(\lambda_p, T).
\]

(11)

In Ref. [38], we performed fluorescence lifetime measurements in Yb-silica. Here, we present a fitting of \( \tau_r(T) \) to an analytical form that is based on a two-level excited state:

\[
\tau_r(T) = \frac{1}{\tau_1^{-1} + \tau_2^{-1}} \exp\left(-\frac{\delta E}{k_b T}\right).
\]

(12)

\( \tau_1 = 798 \pm 2 \mu s \), and \( \tau_2 = 576 \pm 27 \mu s \) are the lifetimes of the first and second energy levels of the excited state, respectively, and \( \delta E = 506 \pm 56 \text{ cm}^{-1} \) is the energy difference between these two levels [36, 39]. We also present the following approximation:

\[
\frac{\lambda_p^{5}}{\int_{\lambda} S(\lambda, T) d\lambda} \approx 7.4 + \left(\frac{d}{T}\right)^{5}, \quad d = 205.9 \pm 2.4 \text{ K}.
\]

(13)

Using Eqs. 12 and 13, we can approximate \( \alpha_r(\lambda_p, T) \) [Eq. 11] with the following mathematical form:

\[
\alpha_r(\lambda_p, T) \approx \frac{\alpha_{r,0}}{c \tau_r(T)} \times \left(7.4 + \left(\frac{d}{T}\right)^{5}\right) \times Z(\lambda_p, T).
\]

(14)

Fitting the analytical in Eq. 14 to the discrete points in Fig. 6, we find the dimensionless coefficient \( \alpha_{r,0} = (0.95 \pm 0.01) \times 10^{6} \). The fitted lines in Fig. 6 show that the minimum achievable temperature can reach down to \( T_{\text{min}} = 138 \text{ K} \) for \( \eta_q = 1 \), \( T_{\text{min}} = 175 \text{ K} \) for \( \eta_q = 0.99 \), and \( T_{\text{min}} = 290 \text{ K} \) for \( \eta_q = 0.98 \). Figure 6 also shows that the maximum cooling efficiency for Yb-silica glass is around \( \eta_c^{\text{max}} \approx 2\% \) at room temperature for \( \lambda_p = 1030 \text{ nm} \). Setting the background absorption to zero (\( \alpha_b = 0 \)) increases this value to \( \eta_c^{\text{max}} \approx 2.2\% \). In order to increase the cooling efficiency, the background absorption must be minimized, the internal quantum efficiency has to be close to unity, and the ion dopant density \( N \) must be increased to enhance the resonant absorption coefficient. We note that these requirements are not necessarily compatible with each other; e.g., increasing \( N \) can potentially decrease \( \eta_q \) due to quenching. Therefore, a compromise determined by careful measurements must be obtained.

IV. DISCUSSION AND CONCLUSION

It must be noted that by taking \( N = 5 \times 10^{25} \text{ m}^{-3} \) in this manuscript, we have implicitly assumed that the silica glass host is co-doped with some modifiers like Al\(_2\)O\(_3\), in order to shift the quenching concentration to higher values to reduce clustering and ensure an adequate cooling efficiency [40, 41]. For pure silica, applying the model developed by Auzel et al. [25] to the experimental data from Ref. [23], it can be shown that \( N = 0.7 \times 10^{25} \text{ m}^{-3} \) can guarantee a near unity internal quantum efficiency [40, 41]. Using \( N = 0.7 \times 10^{25} \text{ m}^{-3} \) in pure silica, we have calculated the minimum achievable temperature to be \( T_{\text{min}} = 216 \text{ K} \) for \( \eta_q = 1 \), \( T_{\text{min}} = 262 \text{ K} \) for \( \eta_q = 0.99 \). For \( \eta_q = 0.98, T_{\text{min}} \) is higher than the room temperature. As expected, a decrease in ion density results in a lower cooling efficiency.

In conclusion, we have argued that a high-purity Yb-doped silica glass can potentially be cooled via anti-Stokes fluorescence. In order to achieve net solid-state optical refrigeration, it is necessary for the cooling efficiency to be positive. This requires the pump wavelength to be greater than the mean fluorescence wavelength, and the internal quantum efficiency as well as the absorption efficiency to be near unity. For the internal quantum efficiency we have shown that, in principle, the non-radiative decay rate \( W_{nr} \) can be made substantially smaller than the radiative decay rate \( W_r \). Therefore, an internal quantum efficiency of near unity can be obtained, making Yb-doped silica glass suitable for solid-state optical refrigeration. Our assessment is based on reasonable assumptions for material properties, e.g. we have assumed a typical
background absorption coefficient of $\alpha_b = 10 \text{dB/km}$ and an internal quantum efficiency of larger than $\eta_b = 0.98$. We have made spectral measurements of the fluorescence from a Yb-doped silica optical fiber at different temperatures. Using these measurements, we have reported the temperature dependence of the mean fluorescence wavelength, and have estimated the minimum achievable temperature in Yb-doped silica glass.

Our analysis highlights the potential for Yb-doped silica glass to be used as the gain medium for radiation-balanced high-power fiber lasers and amplifiers. In order to observe the solid-state optical refrigeration of silica, high-quality samples with adequate Yb doping concentrations are required. It is preferable to co-dope silica with modifiers such as Al$_2$O$_3$ to increase the solubility of Yb ions and consequently increase the cooling efficiency. Of course, proper thermal isolation in solid-state optical refrigeration is essential. Such experiments are quite delicate [16] and Yb-doped silica will be no exception.

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Er$^{3+}$, and Ho$^{3+}$ doped Y$_2$O$_3$,” Optical Materials 24, 103–109 (2003).


