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CRYSTAL STRUCTURE, INCOMMENSURATE MAGNETIC ORDER AND FERROELECTRICITY IN Mn_{1-x}Cu_xWO₄ (x=0 – 0.19)

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We have carried out a systematic study on the effect of Cu doping on nuclear, magnetic, and dielectric properties in $Mn_{1-x}Cu_xWO_4$ for $0 \le x \le 0.19$ by a synergic use of different techniques, viz, heat capacity, magnetization, dielectric, and neutron powder diffraction measurements. Via heat capacity and magnetization measurements we show that with increasing Cu concentration magnetic frustration decreases, which leads to the stabilization of commensurate magnetic ordering. This was further verified by temperature-dependent unit cell volume changes derived from neutron diffraction measurements which was modeled by the Grüneisen approximation. Dielectric measurements show a low temperature phase transition below about 9-10 K. Further more, magnetic refinements reveal no changes below this transition indicating a possible spin-flop transition which is unique to the Cu doped system. From these combined studies we have constructed a magnetoelectric phase diagram of this compound.

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I. INTRODUCTION

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A typical feature of multiferroic materials undergoing a 14 transition to an elliptic spiral ferroelectric phase, is the exis-15 tence of spectacular magnetoelectric effects, such as the po-16 larization flops observed in TbMnO₃ [1] and orthorhombic 17 DyMnO₃ [2] or the sign reversal of electric polarization which 18 is revealed under magnetic field in TbMn₂O₅ [3]. The orienta-19 tion of the applied magnetic field with respect to the magnetic 20 spins influences the stability range of the spiral phase and the 21 electric polarization-flop process. This property was recently 22 illustrated by remarkable magnetic field induced effects ob-23 served in ferroelectric phase of manganese tungstate MnWO₄ 24 in which applied field induces a polarization flop transition [4-25 7]. 26

In most of the recently discovered multiferroics, the ferroelectric polarization can be explained by the inverse Dzyaloshinski-Moriya effect [8–10], where the induced electric polarization of a single pair of spins \vec{S}_i , \vec{S}_j separated by a distance vector $\vec{r}_{i,j}$ is given by [8]

$$\vec{P}_{FE} \propto \vec{r}_{ij} \times \left(\vec{S}_i \times \vec{S}_j\right). \tag{1}$$

The required helical magnetic structure may arise from strong frustration. Since in addition the interaction, equation 1, is only a second order effect, the ferroelectric polarization is rather small in these materials. In the RMnO₃ [1, 11] (R=rare earth) series and in MnWO₄ [5, 12, 13] the electric polarization is about two to three orders of magnitude smaller than in a classical ferroelectric perovskite such as BaTiO₃. As a consequence the observation of electric–field–induced ef-

⁴⁰ fects in the magnetically ordered state is more difficult. Nev-⁴¹ ertheless, it was shown that it is possible in these chiral multi-⁴² ferroics to switch the magnetic order by the application of an ⁴³ electric field at constant temperature [14–16].

The crystal structure of MnWO₄ is monoclinic with space 44 ₄₅ group P2/c, made up of MnO₆ octahedra with high-spin Mn²⁺ ⁴⁶ (d_5) ions and WO₆ octahedra with diamagnetic W⁶⁺ (d_0) ⁴⁷ ions [17]. Recently it was found that MnWO₄ exhibits multi-48 ferroicity in which magnetism causes ferroelectricity, imply-⁴⁹ ing a strong coupling between the two [5, 12, 13]. MnWO₄ is ⁵⁰ one of the prototypical multiferroic materials exhibiting spin-51 current ferroelectricity [5]. It possesses a complex phase di-⁵² agram with 3 antiferromagnetic phases below 14 K namely 53 AF1, AF2 and AF3 at zero magnetic field. AF2 is a ferro-⁵⁴ electric (FE) phase, in which the net polarization is along the $_{55}$ b axis which can be flipped to the a axis with the application of an external magnetic field. This is the first example of 56 the ferroelectric polarization flop induced by magnetic fields 57 58 in transition-metal oxide systems without rare-earth 4f moments. Taniguchi et al. showed that the stability of the mag-60 netoelectric domain walls in a canted magnetic field plays a 61 key role in the directional control of the electric polarization 62 flop phenomenon [7]. From polarized neutron scattering mea-63 surements Sagayama et al. showed that an inverse effect of 64 Dzyaloshinskii-Moriya interaction is the origin of the sponta-⁶⁵ neous electric polarization in the spiral phase of MnWO₄ [6]. 66 From superspace symmetry formalism it was shown that in the 67 AF3 phase, the modulations of two Mn atoms within the unit 68 cell can have a cycloidal component with equal and opposite 69 chiralities canceling their effects and hence no electric polar-⁷⁰ ization is induced. Whereas in the AF2 phase, an additional

72 symmetry relation between the two manganese atoms with 125 cooling down to 5 K, a He bath cryostat was used. chiralities of the same sign which add up to induce macro-73 scopic electric polarization [18]. 74

Recently it was found that the ferroelectric phase is com- 126 75 pletely suppressed in MnWO₄ by doping 10% iron on Mn 76 77 site, which can be again restored with the application of 127 78 ield) in Mn_{0.9}Fe_{0.1}WO₄ is explained by the increase of uni-79 axial spin anisotropy K [19]. Evidence for the increase of K80 with Fe substitution was also derived from neutron scatter-81 ing experiments [20]. It was observed that in $Mn_{1-x} M_x WO_4$ 82 (M=Mg, Zn and $x \le 0.3$), the substitution of the nonmag-83 netic Mg^{2+} ions and Zn^{2+} for the magnetic Mn^{2+} ions result 84 85 ties of MnWO₄ [21]. These substitutions destabilized the non-86 polar magnetic structure AF1 of MnWO₄ but the AF3-to-AF2 87 magnetoelectric phase transition was not affected. This indi-88 cated that the nonmagnetic dopant destroys neither the three-89 dimensional nature of magnetic interactions, nor the spin frus-90 tration within each || c-chain and between || c-chains along 91 $_{92}$ the *a*-direction. In this article we discuss the influence of ⁹³ doping of Cu ions on the nuclear and magnetic structure of 94 MnWO₄.

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II. EXPERIMENTAL

96 97 98 99 100 101 103 104 small pellets, using a physical property measurement system 105 Quantum Design) in the temperature range 3–300 K. Mag-106 netic measurements were carried out in a commercial physical 107 property measurement system using vibrating sample magne-108 tometer option. To investigate nuclear and magnetic struc-109 ture, time-of-flight (TOF) neutron powder diffraction (NPD) 110 as performed on 8 g of powder samples that were loaded 111 8 mm diameter vanadium cans. Neutron data were col-112 ected at the Spallation Neutron Source (SNS) at Oak Ridge 113 Vational Laboratory on the high resolution neutron powder 114 diffractometer POWGEN [22]. Data were collected for the 115 compositions of Cu, x=0.0, 0.05, 0.1 and 0.19 in the temper-116 ature range 1.5 - 300 K. For each temperature the data was 117 collected using two different center wavelengths, 1.599 Å and 118 .731 Å or 4.797 Å. The crystal and magnetic structure refine-119 nents were carried out from the NPD data using the Rietveld 120 refinement program FullProf [23]. For the dielectric mea-¹²² surements, opposite sides of pressed pellets were covered by 123 silver paint, thus forming a parallel-plate capacitor. The mea-¹²⁴ surements were done using an LCR meter (Agilent 4980). For

III. RESULTS AND DISCUSSION

The thermodynamic signature of the transition between difmagnetic field. The absence of ferroelectricity (at zero 128 ferent phases is usually detected by distinct anomalies in the ¹²⁹ temperature dependence of the heat capacity, $(C_{\rm P})$. Multifer-130 roic materials with several subsequent transitions may show ¹³¹ pronounced anomalies of $C_{\rm P}$. In figure 1 we present the vari-132 ation of C_P with the temperature for $Mn_{1-x}Cu_xWO_4$. For ref-133 erence, the specific heat of a MnWO₄ single crystal is also ¹³⁴ included. All the compositions show two anomalies at $T_{\rm N1}$ in very similar effects on the magnetic and dielectric proper- $_{135}$ and T_{N2} . A third low temperature phase transition, T_{N3} seen ¹³⁶ in the case of MnWO₄ is already missing in the lowest Cu ¹³⁷ doped compound. This is associated with the phase transi-138 tion from the helical AF2 phase to the commensurate AF1 139 phase [4]. This result implies that with Cu doping a quick 140 suppression of the AF1 phase occurs, as a result ferroelectric ¹⁴¹ AF2 phase is extended to the lowest temperature. Similar re-142 sults of quick suppression of the AF1 phase were reported in ¹⁴³ $Mn_{1-x}Co_xWO_4$ and $Mn_{1-x}Zn_xWO_4$ [24, 25].

Dielectric measurements were performed on compositions 144 $_{145}$ x = 0.05, 0.1, and 0.19. The temperature dependence of the dielectric constant (ε'), normalized to the dielectric constant ¹⁴⁷ value at 5 K, is presented in figure 2. To exclude contributions ¹⁴⁸ from electrode polarization or grain boundaries, which can Polycrystalline powders of $Mn_{1-x}Cu_xWO_4$ (x=0.0-0.19) ¹⁴⁹ lead to so-called Maxwell-Wagner relaxations, here we show were prepared by conventional solid state route. Stoichio- 150 the results at a relatively high frequency of 105 kHz [26, 27]. metric amount of precursors, W₂O₃(99.9%), MnO₂(99.9%) ₁₅₁ For x = 0.05 and 0.1, several anomalies in $\varepsilon'(T)$ are found as and CuO(99.99%) were ground well with a mortar and pes- 152 indicated by the arrows in figures 2(a) and (b). Those around tle, pressed into pellets and sintered in a furnace at 900 °C for 153 12 K agree with the findings from specific heat (Fig. 1). With 12 hours in the presence of atmospheric air. This process is 154 increased Cu concentration, these dielectric-constant anomarepeated to achieve homogeneous powder samples. All com- 155 lies become weaker. In addition to the two transitions obpositions were confirmed to be phase pure from x-ray powder 156 served from specific-heat measurements, a third transition is diffraction. Specific heat measurements were carried out on $_{157}$ revealed by the dielectric measurements at $T_x = 9$ and 10 K



FIG. 1. (Colour online) The temperature dependence of specific heat $C_{\rm P}$ of Mn_{1-r}Cu_rWO₄. Different curves are vertically offset by 8 units along $C_{\rm P}$ -axis, zero for each curve is defined by the horizontal dashed lines. Vertical dashed lines are guide to eyes indicating transitions $T_{\rm N1}$ and $T_{\rm N2}$.



FIG. 2. (Colour online) Temperature dependence of dielectric constant at 105 kHz, for $Mn_{1-x}Cu_xWO_4$ with: (a) x=0.05, (b) x=0.1 and (c) x=0.19. The arrows in (a) and (b) indicate the dielectric anomalies and arrows in (c) indicate the anomalies seen in specific heat measurement of the sample with x=0.19.

¹⁵⁸ for the x=0.05 and 0.1 compounds, respectively. From neu-159 tron diffraction measurements we later show that T_x is not associated with the phase transition from AF2 to AF1 phase as 160 in parent MnWO₄. As indicated by the arrows in figure 2(c), 161 the two broad shoulders in $\varepsilon'(T)$ found for x=0.19 seem to roughly agree with the specific-heat results but a clear deter-163 mination of transition temperatures from the dielectric exper-164 ¹⁶⁵ iments is not possible for this sample.

TABLE I. Structural data for MnWO₄, Mn_{0.9}Cu_{0.1}WO₄ and Mn_{0.81}Cu_{0.19}WO₄ obtained from the NPD data collected at 300 K.

		Cu content (x)		
		x = 0	x = 0.1	x = 0.19
a (Å)		4.8300(5)	4.8120(4)	4.7946(6)
b (Å)		5.7597(6)	5.7645(5)	5.7694(7)
<i>c</i> (Å)		4.9977(5)	4.9838(4)	4.9708(7)
$V(\text{\AA}^3)$		139.009(3)	138.208(2)	137.453(3)
$\beta(^{\circ})$		91.140(7)	91.370(7)	91.579(9)
Atoms				
Mn/Cu	y/b	0.6861(4)	0.6854(6)	0.6875(1)
	B_{iso} (Å ²)	0.5120(47)	0.4970(57)	0.429(11)
W	y/b	0.1815(3)	0.1806(36)	0.1806(3)
	B_{iso} (Å ²)	0.5070(38)	0.5550(38)	0.6620(47)
Global weighted χ^2 (%)		5.73	8.2	8.4

Magnetization measurements of all samples were per- 177 166 167 168 of magnetic susceptibility of the samples at low temperature is 179 above ordering temperatures indicates the presence of short-¹⁶⁹ presented in figure 3(a). From the magnetic susceptibility data ¹⁸⁰ range spin fluctuations above $T_{\rm N}$. Thermal evolution of Curie- $_{170}$ of samples x=0.05, 0.1 and 0.15 only one magnetic ordering $_{181}$ Weiss temperature (Θ_{CW}) and the frustration parameter calcu-



FIG. 3. (Colour online) (a) Susceptibility calculated from the magnetization data measured after a field cooling cycle with applied magnetic field of 1 kOe. (b) Curie-Weiss fit (red lines) to the inverse susceptibilities, a deviation from linear nature is seen below magnetic ordering temperature.(c) The frustration parameter as a function of composition. Inset shows the Curie-Weiss temperature as a function of composition obtained from the Curie-Weiss fits. Straight lines are guide to eyes.

171 temperature is discernible around 14 K and for sample x=0.19 two anomalies are discernible. The thermal evolution of in-172 verse susceptibility obtained from the field-cooled magnetiza-173 tion were fitted with Curie-Weiss law as shown in figure 3(b). 174 Inverse susceptibility of $Mn_{1-x}Cu_xWO_4$ follows Curie-Weiss 175 law down to \sim 75 K, below which it deviates from the fitted 176 curve and shows a marked deviation below ~ 15 K which corformed under magnetic fields of 1 kOe. Thermal evolution $_{178}$ responds to $T_{\rm N1}$. The deviation of inverse susceptibility well



FIG. 4. (Colour online) (a) Temperature evolution of diffraction patterns of Mn_{0.95}Cu_{0.05}WO₄. Magnetic phase transition is discernible with the appearance of additional incommensurate Bragg peaks around 13.5 K (horizontal dashed line). (b) Temperature evolution of part of the diffraction patterns of Mn_{0.9}Cu_{0.1}WO₄. Magnetic phase transition is discernible with the appearance of additional incommensurate Bragg peaks around 14 K. (c) Temperature dependence of diffraction patterns of Mn_{0.8}Cu_{0.19}WO₄. Two magnetic phase transitions are discernible at ~17 K and ~11.5 K.

lated as $f = |\Theta_{CW}|/T_N$ is presented in figure 3(c) as a function 182 of composition. Indeed MnWO₄ has been known to be a mod-183 erately spin frustrated system with the frustration parameter, 184 $f = |\Theta_{\rm CW}|/T_{\rm N} \approx 4.9$, where $\Theta_{\rm CW}$ is approximately -71 K and the Néel temperature $T_{\rm N} = 13.5$ K [12, 28]. From figure 3(c) it is clear that Cu doping increases the Curie-Weiss temperature 187 and reduces frustration. 188

The temperature evolution of diffraction patterns in a 189 selected *d*-space is presented in figure 4. A mag-190 netic phase transition is discernible based on the new in-191 commensurate Bragg peaks below 13.5 K and 14 K in 192 $Mn_{0.95}Cu_{0.05}WO_4$ and $Mn_{0.9}Cu_{0.1}WO_4$, respectively. In the 193 ase of Mn_{0.81}Cu_{0.19}WO₄ two transitions were observed 194 around 17 K and 11.5 K. The fundamental crystal structure 195 of all compositions is monoclinic with space group P2/c. The 200 tion, a decrease in the unit cell volume and increase in the 197 198 in figure 5(a), the refined structure parameters including errors 202 cordance with the Vegard's law, lattice volume decreases as



FIG. 5. (Colour online) (a) Unit cell volume and monoclinic angle as a function of Cu concentration at temperature 300 K. lines are only guides. (b)-(d) The temperature dependence of unit cell volume for the Cu compositions x=0.05, 0.1 and 0.19. The curves are fits to equation 2 with the Grüneisen approximation for the zero pressure equation. Insets in (b)-(d) show the deviation of unit cell volume from the fitted data at low temperatures. In the second inset of (d) the unit cell volumes are plotted together, to compare the magnitude of negative thermal expansion close to ordering temperature the y axes of x=0.1 and 0.19 are scaled by adding 0.375 $Å^3$ and 1.155 $Å^3$ respectively to match the value of the sample x=0.05 at 100 K. The lines are guides to eyes.

unit cell volume and monoclinic angle at 300 K is presented 201 monoclinic angle was observed. The volume change is in ac-¹⁹⁹ are tabulated in table (I). With the increase in Cu concentra-²⁰³ x increases, because the ionic radius of Cu²⁺ is smaller than



FIG. 6. (Colour online) Observed and calculated diffractions patterns and their difference for (a) Mn_{0.95}Cu_{0.05}WO₄ at 1.5 K (b) Mn_{0.9}Cu_{0.1}WO₄ at 1.5 K (c) Mn_{0.81}Cu_{0.19}WO₄ at 1.5 K and (d) Mn_{0.81}Cu_{0.19}WO₄ at 13.4 K. Circles are the measured intensities and the curve is the calculated pattern. Top and bottom vertical bars mark the positions of the nuclear and magnetic Bragg reflections respectively. Bottom curve is the difference between the measured and calculated patterns.

 Mn^{2+} [29]. The temperature dependence of unit cell volume $_{229}V_0$ is the volume at T = 0 K in the absence of magnetoelastic 204 here show a negative thermal expansion, an increase in vol- $_{231}$ energy U(T) is given by 206 ume with the decrease in temperature below 50 K. This effect 207 seems to halt around 20 K and the volume starts to shrink be-208 low magnetic ordering temperature. The effect is more pro-209 nounced in lower Cu concentration and diminishes signifi-210 antly with Cu doping as shown in the lower inset of fig-211 ure 5d. The anomalous behavior of the temperature variation 212 of unit cell volume is due to the magneto-elastic effect associ-213 ated with the antiferromagnetic transition at the Néel temper-214 ature. In order to study the spontaneous magnetostriction it is 215 necessary to determine the temperature variation of the lattice 216 arameters and the unit cell volume in the absence of mag-217 etism. One way to determine the background temperature 218 ariation of the lattice parameter and unit cell volume is to 219 extrapolate the paramagnetic high temperature region to low 220 temperature by fitting with a polynomial function [30]. This 221 only an approximation which works in some cases but inolves some uncertainty. Alternatively, we used the Grüneisen 223 approximation for the zero pressure equation of state, in which 224 the effects of thermal expansion are considered to be equiva-225 lent to elastic strain [31]. Within this approach the tempera-226 ²²⁷ ture dependence of the volume can be described by,

$$V(T) = \gamma U(T)/B_0 + V_0 \tag{2}$$

is shown in figure 5(b)-(d). All three compositions presented 200 effect. By adopting the Grüneisen approximation, the internal

$$U(T) = 9Nk_{\rm B}T \left(\frac{T}{\theta_{\rm D}}\right)^3 \int_0^{\theta_{\rm D}/T} \frac{x^3}{{\rm e}^x - 1} {\rm d}x \tag{3}$$

where N is the number of atoms in the unit cell, $k_{\rm B}$ is Boltz-232 $_{233}$ mann's constant and θ_D is the Debye temperature. By fitting the unit cell volume in the paramagnetic state we can 234 get the physical parameters θ_D and V_0 . From the present fit-235 ²³⁶ ting procedure it was not possible to determine γ and B_0 but 237 the ratio of γ/B_0 was set as variable. The result of the fit ²³⁸ is shown as a green solid line in figure 5(b-d). Remarkably 239 the fitted curves deviate from the experimental data at around 50 K much above the magnetic ordering temperature, where 240 the unit cell volume undergoes a negative thermal expansion. 241 Below magnetic ordering temperature $T_{\rm N1}$ the negative vol-242 ume effect seized and the unit cell volume is decreased with 243 temperature. This is inferred as a clear indication of presence 244 ²⁴⁵ of magnetoelastic or magnetovolume effects in these system as well as underlying frustration, though the negative ther-246 mal expansion is significantly small for x=0.19 compound. 247 Temperature evolution of lattice parameters are similar to Co 248 doped compound in which complex magnetic phase diagram is attributed to lattice changes [24]. 250

Representational analysis allows the determination of the 251 where, γ is a Grüneisen parameter, B_0 is the bulk modulus and 252 symmetry-allowed magnetic structures that can result from

TABLE II. The little group of $\mathbf{k} = (\alpha, 1/2, \gamma) = (-0.2183, 1/2, 0.476)$ in P2/c is $G_k = P_c$ (with co-set representatives with respect to the translation group: 1,c). The corresponding IRs are one-dimensional $\Gamma_1(1,c)=(1,\varepsilon)$ and $\Gamma_2(1,c)=(1,-\varepsilon)$, with $-\varepsilon=\exp(\pi i\gamma)$. The basis vectors of the IRs of G_k are given below for the atoms Mn/Cu in the primitive unit cell numbered as: 1(1/2, y, 1/4) and 2(1/2, 1-y, 3/4)related by c-glide plane: c(x, -y+1, z+1/2), respectively.

IR	BV	Atom	BV components					
			$m_{\parallel a}$	$m_{\parallel b}$	$m_{\parallel c}$	$im_{\parallel a}$	$im_{\parallel b}$	$im_{\parallel c}$
Γ_1	ψ_1	1	1	0	0	0	0	0
		2	-0.075	0	0	0.997	0	0
	ψ_2	1	0	1	0	0	0	0
		2	0	0.075	0	0	-0.997	0
	ψ_3	1	0	0	1	0	0	0
		2	0	0	-0.075	0	0	0.997
Γ_2	ψ_4	1	1	0	0	0	0	0
		2	0.075	0	0	-0.997	0	0
	ψ_5	1	0	1	0	0	0	0
		2	0	-0.075	0	0	0.997	0
	ψ_6	1	0	0	1	0	0	0
		2	0	0	0.075	0	0	-0.997

253 a second-order magnetic phase transition, given the crys- 290 tion), but they correspond to different magnetic Wyckoff posi-254 255 256 257 258 259 260 261 agation vector in the temperature range 11.5-17 K was found $_{299} m_z = 1.28(8) \mu_B$ for T=13.5 K. 262 to be $\mathbf{k} = (\frac{1}{2}, 0, 0)$ and below 11.5 K it is $\mathbf{k} = (k_x, \frac{1}{2}, k_z)$. Refined NPD patterns are presented in figure 6(a-d). From 263 264 265 266 267 $_{269}$ metry couplings in the form of irreducible representations and $_{306}$ temperature a distinct change in k_x and k_y is observed. In case ²⁷¹ the crystallographic space group P2/c, for the incommensu- ³⁰⁸ surate with similar wave vector $\mathbf{k} = (-0.214, \frac{1}{2}, 0.457)$. Only ²⁷² rate magnetic propagation vector $\mathbf{k} = (k_x, \frac{1}{2}, k_z)$, is found to ³⁰⁹ AF2 phase with elliptical spin arrangement is ferroelectric 273 have two possible irreducible magnetic representations (IR) 310 which can be explained by spatial inversion symmetry break-²⁷⁴ each having three basis vectors (BV). The computed results ³¹¹ ing spiral spin structure [5]. It should be noted that in the case $_{275}$ for the x=0.05 compound at 1.5 K with the propagation vector $_{312}$ of Mn_{0.95}Cu_{0.05}WO₄ in the whole temperature range the struc-276 278 280 282 ²⁸⁴ C1 and C3 are real coefficients while C2 is a pure imaginary ³²¹ tor along z-direction is nearly commensurate. From our pow-285 coefficient. For x=0.19 in the temperature range 11.5-17 K, 322 der diffraction measurements for x=0.05 and 0.1 compounds with $\mathbf{k} = (\frac{1}{2}, 0, 0)$, four one-dimensional IRs were computed, we don't see any significant change associated with the tran-

TABLE III. Refined unit cell parameters, magnetic propagation vector and coefficients of basis vectors for $Mn_{1-x}Cu_xWO_4$ at 1.5 K.

		Cu content (x)				
		x = 0.05	x = 0.1	<i>x</i> = 0.19		
Unit cell parameters						
	a (Å)	4.8145(6)	4.8045(4)	4.7856(8)		
	<i>b</i> (Å)	5.7565(9)	5.7583(6)	5.7619(8)		
	<i>c</i> (Å)	4.9860(8)	4.9788(5)	4.9653(9)		
	$\beta(^{\circ})$	91.20(1)	91.30(9)	91.49(6)		
Mn/Cu	y/b	0.6841(6)	0.6840(6)	0.6857(2)		
	B_{iso} (Å ²)	0.274(6)	0.184(2)	0.203(1)		
Components of propagation vector $(k_v = \frac{1}{2})$						
	k_x	-0.218(3)	-0.221(6)	-0.223(4)		
	k_z	0.476(1)	0.494(4)	0.526(4)		
Coefficient	s of basis vector	8				
	C1(real)	3.41(1)	3.28(1)	2.79(9)		
	C2(imaginary)	-3.96(2)	-4.08(9)	-3.62(4)		
	C3(real)	2.95(1)	2.91(1)	2.32(3)		

²⁸⁷ for Γ_1 and Γ_2 two BVs are allowed and for Γ_3 and Γ_4 only one 288 BV is allowed. The Shubnikov groups (SG) of each IRs have ²⁸⁹ the same symbol $P_a 2/c$ (in Belov-Neronova-Smirnova notatal structure before the transition and the magnetic propaga- 291 tions and origin shifts [32]. The magnetic moments of the two tion vector (k) of the magnetic ordering. These calculations $_{292}$ atoms (1)(x, y, z) and (2)(-x+1, -y+1, -z+1) in the paramagwere carried out using the program BasIreps included in the 203 netic unit cell have the following configurations for eanch rep-FullProf suite. First, the program k-search, also included in 294 resentation: $\Gamma_1:1(m_x, 0, m_z), 2(-m_x, 0, -m_z); \Gamma_2:1(m_x, 0, m_z), \Gamma_2:1(m_x, 0, m_z), \Gamma_3:1(m_x, 0, m_z), \Gamma_3:1$ the FullProf suite, is used to determine the magnetic propa- $295 2(m_x, 0, m_z)$; $\Gamma_3:1(0, m_y, 0), 2(0, -m_y, 0)$ and $\Gamma_4:1(0, m_y, 0), \Gamma_4:1(0, m_y, 0$ gation vector at different temperatures. For x=0.05 and 0.1 the $_{296}$ 2(0, $-m_y$, 0). Only the representation described by Γ_2 (SGmagnetic propagation vector was found to be $\mathbf{k} = (k_x, \frac{1}{2}, k_z)^{297} P_a 2/c$, Wyckoff site 4f in the doubled unit cell) provides a in whole temperature range. For x=0.19 the magnetic prop- 298 successfull refinement of the data with m_x =1.24(7) $\mu_{\rm B}$ and

While the magnetic propagation vector determines the modu- $_{301}$ the magnetic structure refinements Mn_{0.95}Cu_{0.05}WO₄ was lation going from one unit cell to another, magnetic symme- $_{302}$ found to order at ~ 13.5 K, with the incommensurate propatry analysis is needed to determine the coupling between the 303 gation vector $\mathbf{k} = (-0.218, \frac{1}{2}, 0.477)$. The temperature depensymmetry related magnetic sites within one crystallographic 304 dence of components of incommensurate propagation vector unit cell. BasIreps is used to compute all the allowed sym- 305 k = $(k_x, \frac{1}{2}, k_z)$ is presented in figure 7(a). With decrease in their respective basis vectors. The Mn/Cu at the site 2f in 307 of MnWO₄ the magnetic phases AF2 and AF3 are incommen- $\mathbf{k} = (-0.218, \frac{1}{2}, 0.476)$ are presented in table (II). All possible 313 ture is found to be similar to that of AF2 in the parent comcombinations of the two allowed irreducible representations $_{314}$ pound. The magnetic structure in case of Mn_{0.9}Cu_{0.1}WO₄ is were tested against the measured data. Rietveld refinements $_{315}$ similar to that of Mn_{0.95}Cu_{0.05}WO₄ but with a modified propaclearly showed that only with the IR Γ_1 (with real and imagi- $_{316}$ gation vector $\mathbf{k} = (-0.224, \frac{1}{2}, 0.5)$ close to 14 K. The evolution nary components) a successfull refinement of the data can be $_{317}$ of propagation vector with temperature in Mn_{0.9}Cu_{0.1}WO₄ is obtained. The propagation vector and the refined coefficient 318 presented in figure 7(b). Striking change in propagation vecof basis vectors, C1, C2 and C3, for x=0.05, 0.1 and 0.19 $_{319}$ tor close to ordering temperature T_{N1} in this case indicates at 1.5 K is presented in table (III). It should be noted that $_{320}$ that with increased Cu content of x=0.1 the propagation vec-



FIG. 7. (Colour online) Temperature variation of components of incommensurate propagation vector along $x(k_x)$ and $z(k_z)$ axes for (a) $Mn_{0.95}Cu_{0.05}WO_4$, (b) $Mn_{0.9}Cu_{0.1}WO_4$ and (c) $Mn_{0.81}Cu_{0.19}WO_4$. Component of propagation vector along y is $k_y = 0.5$.

sition from T_{N1} (AF3) to T_{N2} (AF2) as seen from specific heat 324 measurements. Considering very narrow temperature range 325 between these two transitions it might be difficult to clarify 326 this with our bulk powder measurements. We expect that for x=0.05-0.15, the magnetic ordering in the temperature range 328 $T_{N2} < T < T_{N1}$ should be collinear incommensurate phase as in 329 ³³⁰ MnWO₄ [5] with magnetic propagation vector similar to AF2 331 phase. Further studies on single crystals with polarized neutron diffraction with smaller temperature steps will be a best 332 ool for the detailed investigation of this structure. 333

334 335 336 337 338 339 ³⁴¹ and the commensurate collinear (AF4) magnetic structure of ³⁹⁷ which requires further scrutiny. ³⁴² Mn_{0.81}Cu_{0.19}WO₄ is presented in figure 8(a) and (b), respectively. The incommensurate structures for lower doping sys-343 tems is quite similar to the one presented in figure 8(a). 344

From our comprehensive study of $Mn_{1-x}Cu_xWO_4$ we are 345

³⁴⁶ able to construct a tentative magnetic phase diagram as shown in figure 9. The magnetic phase diagram of Cu doped com-347 pound found to be much simpler than that of Co doped com-348 pound [24, 33] but very similar to Zn doped compound [25]. 349 With higher doping concentration a collinear magnetic struc-350 351 ture is stabilized at higher temperatures. This is attributed to weakening of spin-phonon coupling and hence lower frustra-352 tion leading to a simpler magnetic ordering. From the neutron 353 diffraction measurement it is clear that the low temperature phase below T_X which is observed from dielectric measure-355 356 ments is incommensurate cycloidal phase. Magnetic structure 357 refinements confirmed that the magnetic phase below T_X (region marked with gray lines in figure 9) is not associated with 358 the transition from cycloidal structure with magnetic vectors 359 $\mathbf{k} = (k_x, 0.5, k_z)$ to collinear structure with magnetic vector $\mathbf{k} = (0.25, 0.5, -0.5)$ as seen in MnWO₄ [12]. This leads to the inference that below T_X the magnetic structure undergoes a temperature induced spin flip transition with similar magnetic propagation vectors which is indistinguishable from powder diffraction measurements. The suppression of low temperature collinear phase can be attributed to extremely sensitive 366 exchange coupling and anisotropy constants with respect to 367 perturbations [25, 34]. In the present case chemical doping plays the role of perturbations. In a recent report based on magnetization, specific heat and ferroelectric polarization 370 measurements, Liang et al showed that by the substitution of 371 lower spin (1/2) Cu^{2+} for the higher spin (5/2) Mn^{2+} ion the multiferroic phases of MnWO₄ are strongly affected [38]. The Cu substitution will introduce a low spin with different ex-374 375 change coupling and anisotropy constants affecting the magnetic and ferroelectric states. This leads to the stabilization 376 377 of ferroeletric spin spiral phase at low temperatures with in-378 creasing Cu content. The microscopic exchange interactions can be obtained through inelastic neutron scattering (INS) ex-380 periments investigating the magnetic excitations. According 381 to INS experiments on MnWO₄, to explain the magnetic ex-382 citation spectrum, up to 11 different exchange pathways were 383 required to fit the data proving the three dimensional char-³⁸⁴ acter of magnetic fluctuations [35]. This three dimensional ³⁸⁵ nature explains the robustness of cycloidal spiral order in Cu-³⁸⁶ doped MnWO₄, since percolation threshold for site dilution is ³⁸⁷ much lower than for two-dimensional systems [36]. Based on ³⁸⁸ a semiphenomenological Landau theory, authors in [37] clar-389 ified the effect of different dopants on the phase diagram of With further increase in doping in case of $Mn_{0.81}Fe_{0.19}WO_4$ 300 $Mn_{1-r}M_rWO_4$ (*M*=Fe,Zn,Mg). The origin of complex phase we observe a commensurate magnetic (CM) structure at 391 diagrams in these compounds is attributed to competition be- $T_{\rm N1}=17$ K with $\mathbf{k} = (\frac{1}{2}, 0, 0)$ which is similar to AF4 phase 392 tween different superexchange interactions with contrasting in parent compound. Below 11.5 K it undergoes another mag- 393 behavior of doping with different ions. We expect that the netic phase transition to AF2 phase with propagation vector 394 theoretical analysis presented in [37] should be compatible = $(-0.225, \frac{1}{2}, 0.531)$ which is modulated with temperature ₃₉₅ for Mn_{1-x}Cu_xWO₄ as well. The temperature induced spinas shown in figure 7c. The incommensurate cycloidal (AF2) 386 reorientation remains to be unique to the present compound



FIG. 8. (Colour online) (a) ICM structure of $Mn_{0.81}Cu_{0.19}WO_4$ with propagation vector $\mathbf{k} = (-0.225, \frac{1}{2}, 0.531)$, the number of visible unit cells along a, b and c direction are, 5, 2 and 20, respectively. (b) High temperature CM structure of Mn_{0.81}Cu_{0.19}WO₄ with propagation vector (0.5, 0, 0), two unit cells along all three axes are shown. Gray box indicate one unit cell.

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IV. CONCLUSIONS

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399 $\leq x \leq 0.19$ we have shown that substitution of Cu for 400 0 Mn results in a reduction of the frustration. Also a reduc-401 ⁴⁰² tion in negative thermal expansion with the increased Cu doping was observed which hints to a reduction in spin-phonon 403 404 coupling with the higher Cu content. Temperature and dop-405 ing dependence of lattice parameters establish a strong depen-406 dence of magnetic structure on lattice changes. Both $T_{\rm N1}$ and 417 $_{407}$ T_{N2} increased with higher Cu content. This is in contrast to ⁴⁰⁸ Mn_{1-x}Zn_xWO₄ [25]. The presence of third transition T_X is 409 unique to the present compound. We note again, our NPD 410 data confirms that the origin of T_X is not ICM to CM observed



FIG. 9. Tentative phase diagram of $Mn_{1-x}Cu_xWO_4$ with phase boundaries obtained from specific heat (asterisk), NPD (inverted triangle) and dielectric (circle) measurements. The symbols in black, red and green corresponds to the transitions T_{N1} , T_{N2} and T_X , respectively. Blue asterisk indicates the transition from AF2 to AF1 phase in MnWO₄. (PM-paramagnet, CM-commensurate magnet, ICMincommensurate magnet, FE-ferroelectric and PE-paraelectric)

⁴¹¹ in MnWO₄ at T_{N3} . A possible origin of this phase transition ⁴¹² is the temperature induced spin reorientation. Further single From our detailed investigation of the $Mn_{1-x}Cu_xWO_4$ for ⁴¹³ crystal neutron diffraction and electric polarization measure-⁴¹⁴ ments are desirable to shed light on the nature of ferroelectric 415 and magnetic ordering below $T_{\rm X}$ and in the region between 416 $T_{\rm N1}$ and $T_{\rm N2}$.

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- T. Kimura, T. Goto, H. Shintani, K. Ishizaka, T. Arima, and [1] Y. Tokura, Nature 426, 55 (2003).
- [2] J. Strempfer, B. Bohnenbuck, M. Mostovoy, N. Aliouane, D. N. Argyriou, F. Schrettle, J. Hemberger, A. Krimmel, and M. Zimmermann, Phys. Rev. B 75, 212402 (2007).
- [3] N. Hur, S. Park, P. A. Sharma, J. S. Ahn, S. Guha, and S. Cheong, Nature 429, 392 (2004).
- G. Lautenschläger, H. Weitzel, T. Vogt, R. Hock, A. Böhm, [4] M. Bonnet, and H. Fuess, Phys. Rev. B 48, 6087 (1993).
- K. Taniguchi, N. Abe, T. Takenobu, Y. Iwasa, and T. Arima, [5] Phys. Rev. Lett. 97, 097203 (2006).

- 441 442 M. Matsuura, and K. Hirota, Phys. Rev. B 77, 220407 (2008).
- [7] K. Taniguchi, N. Abe, H. Umetsu, H. A. Katori, and T. Arima, 477 443 Phys. Rev. Lett. 101, 207205 (2008). 444
- H. Katsura, A. Balatsky, and N. Nagaosa, Phys. Rev. Lett. 98, 479 [8] 445 27203 (2007). 446
- M. Mostovoy, Phys. Rev. Lett. 96, 067601 (2006). [9] 447
- [10] T. Kimura, G. Lawes, T. Goto, Y. Tokura, and A. P. Ramirez, 448 Phys. Rev. B 71, 224425 (2005). 449
- 450 [11] T. Goto, T. Kimura, G. Lawes, A. P. Ramirez, and Y. Tokura, Phys. Rev. Lett. 92, 257201 (2004). 451
- 452 [12] A. H. Arkenbout, T. T. M. Palstra, T. Siegrist, and T. Kimura, Phys. Rev. B 74, 184431 (2006). 453
- 454 [13] O. Heyer, N. Hollmann, I. Klassen, S. Jodlauk, L. Bohaty, P. Becker, J. A. Mydosh, T. Lorenz, and D. Khomskii, J. Phys. 455 Condens. Matter 18, L471 (2006). 456
- 457 [14] T. Lottermoser, T. Lonkai, U. Amann, D. Hohlwein, J. Ihringer, and M. Fiebig, Nature 430, 541 (2004). 458
- T. Finger, D. Senff, K. Schmalzl, W. Schmidt, L. P. Regnault, 459 [15] P. Becker, L. Bohatỳ, and M. Braden, Phys. Rev. B 81, 054430 460
- 461 (2010).16] A. Poole, P. J. Brown, and A. S. Wills, J. Phys.: Conf. Ser. 145, 462
- 012074 (2009). 463
- [17] H. Weitzel, Solid State Commun. 7, 1249 (1969). 464
- [18] I. Urcelay-Olabarria, J. Perez-Mato, J. Ribeiro, J. García-465 Muñoz, E. Ressouche, V. Skumryev, and A. Mukhin, Physical 466 Review B 87, 014419 (2013). 467
- R. P. Chaudhury, B. Lorenz, Y. Q. Wang, Y. Y. Sun, and C. W. 91 468 Chu, Physical Review B 77, 104406 (2008). 469
- [20] E. García-Matres, N. Stüßer, M. Hofmann, and M. Reehuis, 470 Eur. Phys. J. B 32, 35 (2003). 471
- 472 [21] L. Meddar, M. Josse, P. Deniard, C. La, G. Andre, F. Damay,
- V. Petricek, S. Jobic, M. H. Whangbo, M. Maglione, et al., 473
- Chem. Mater. 21, 5203 (2009). 474

- [6] H. Sagayama, K. Taniguchi, N. Abe, T. Arima, M. Soda, 475 [22] A. Huq, J. P. Hodges, O. Gourdon, and L. Heroux, Z. Kristal-476 logr. Proc. 1, 127 (2011).
 - J. Rodriguez-Carvajal, Physica B 192, 55 (1993). [23]
 - [24] Y.-S. Song, J.-H. Chung, J. M. S. Park, and Y.-N. Choi, Phys. 478 Rev. B 79, 224415 (2009).
 - R. P. Chaudhury, F. Ye, J. A. Fernandez-Baca, B. Lorenz, Y. Q. [25] 480 Wang, Y. Y. Sun, H. A. Mook, and C. W. Chu, Phys. Rev. B 83, 481 014401 (2011). 482
 - P. Lunkenheimer, V. Bobnar, A. V. Pronin, A. I. Ritus, A. A. [26] 483 Volkov, and A. Loidl, Phys. Rev. B 66, 052105 (2002). 484
 - P. Lunkenheimer, S. Krohns, S. Riegg, S. Ebbinghaus, 485 [27] A. Reller, and A. Loidl, Eur. Phys. J. Spec. Top. 180, 61 (2010). 486
 - H. Dachs, Solid State Commun. 7, 1015 (1969). 487 [28]
 - [29] R. D. Shannon, Acta Crystallogr., Sect. A: Found. Crystallogr. 488 32, 751 (1976), ISSN 0567-7394. 489
 - 490 [30] T. Chatterji, B. Ouladdiaf, and D. Bhattacharya, J. Phys. Condens. Matter 21, 306001 (2009). 491
 - D. C. Wallace, Thermodynamics of crystals (Dover Publica-492 [31] tions, New York, 1998). 493
 - [32] J. M. Perez-Mato, S. V. Gallego, E. S. Tasci, L. Elcoro, G. de la 494 495 Flor, and M. I. Aroyo, Annu. Rev. Mater. Res. 45, 13.1 (2015).
 - F. Ye, S. Chi, J. A. Fernandez-Baca, H. Cao, K.-C. Liang, [33] 496 Y. Wang, B. Lorenz, and C. W. Chu, Phys. Rev. B 86, 094429 497 (2012).498
 - [34] M. Kenzelmann, A. Harris, A. Aharony, O. Entin-Wohlman, 499 500 T. Yildirim, Q. Huang, S. Park, G. Lawes, C. Broholm, N. Rogado, et al., Phys. Rev. B 74, 014429 (2006). 501
 - [35] H. Ehrenberg, H. Weitzel, H. Fuess, and B. Hennion, J. Phys.: 502 Condens. Matter 11, 2649 (1999). 503
 - [36] P. Kharel, A. Kumarasiri, A. Dixit, N. Rogado, R. J. Cava, and 504 G. Lawes, Philos. Mag. 89, 1923 (2009). 505
 - 506 [37] S. Matityahu, A. Aharony, and O. Entin-Wohlman, Phys. Rev. B 85, 174408 (2012). 507
 - [38] K.-C. Liang, Y. Wang, Y. Sun, B. Lorenz, and C. Chu, Ferro-508 electrics 470, 43 (2014). 509